Journal of Alloys and Compounds 596 (2014) 58–62

Contents lists available at ScienceDirect

Journal of Alloys and Compounds

journal homepage: www.elsevier.com/locate/jalcom

Magnetic and magnetocaloric properties of equiatomic alloys *R*Al (*R* = Ho and Er)

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ARTICLE INFO

Article history:
Received 26 September 2013
Received in revised form 27 January 2014
Accepted 27 January 2014
Available online 4 February 2014

Keywords:
DyAl-type structure
Magnetocaloric effect
Metamagnetic transition

ABSTRACT

Magnetic and magnetocaloric properties of binary equiatomic alloys *R*Al (*R* = Ho and Er) are investigated in this work. HoAl compound undergoes antiferromagnetic (AFM)–ferromagnetic (FM) transition at T_1 = 13 K, and FM-paramagnetic (PM) transition at the Curie temperature T_C = 20 K, with temperature increasing. ErAl compound exhibits AFM-PM transition at the Néel temperature T_N = 9 K. The maximal values of magnetic entropy change (ΔS_M) are −22.5 J kg^{-1} K^{-1} for HoAl and −16.4 J kg^{-1} K^{-1} for ErAl for a magnetic field change of 0–5 T. The values of refrigerant capacity (RC) are 379.5 J kg^{-1} and 259.7 J kg^{-1} for HoAl and ErAl respectively. The large magnetocaloric effect makes *R*Al (*R* = Ho and Er) compounds promising candidates for low-temperature magnetic refrigerants.

1. Introduction

The magnetocaloric effect (MCE) is the thermal response of a magnetic material when subjected to magnetic field variation owing to the magneto-thermal coupling. It is characterized by the adiabatic temperature change ΔT_{ad} and the isothermal entropy change ΔS_M [1,2]. Based on this effect, magnetic refrigeration is one of the promising technologies, for its energy saving and environmental friendliness compared with the conventional vapor-compression technology, and consequently attracts lots of attention [1,3–5]. For practical application, intense investigations have been dedicated to room-temperature magnetic cooling materials, such as $Gd_5Si_2Ge_2$ [1,3], $La(Fe,Si)_{13}$ [6–8], MnAs [9–11], $MnFeP_{1-x}As_x$ [5,12,13], and Heusler alloys [14–16]. These materials undergo a first-order phase transition and exhibit giant magnetocaloric effect. At the same time, materials for low-temperature magnetic refrigeration also attracts a great deal of interest for the potential application in the refrigeration of gas (nitrogen, hydrogen, and helium) liquefaction, such as *R*FeSi (*R* = Tb, Dy [17], and Er [18]), some Tm based intermetallic compounds (TmGa [19], TmCuAl [20], and Tm_3Co [21]), and $ErMn_2Si_2$ [22].

Except materials mentioned above, we search and develop new magnetocaloric materials and pay attention to rare-earth based intermetallic compounds for large atomic moments of rare-earth elements. There are few articles on binary equiatomic alloys *R*Al (*R* = Gd, Tb, Dy, Ho, and Er), especially on magnetocaloric effect of them [23–27]. The *R*Al compounds crystallize in the DyAl-type orthorhombic structure with *Pbcm* space group. It is formed from aluminum chains and trigonal prism similar to a half-cell of CsCl [28]. The *R*Al compounds are antiferromagnetic with T_N of 42 K, 72 K, 20 K, and 10 K for *R* = Gd, Tb, Dy, and Er, respectively. However, HoAl is ferromagnetic with T_C of 26 K [24,28]. In present work, we carried out a further study on the magnetic and magnetocaloric properties of *R*Al (*R* = Ho and Er) compounds.

2. Experimental details

*R*Al (*R* = Ho and Er) compounds were prepared by arc melting appropriate amounts of high-purity elements in argon atmosphere. The cast samples were sealed in an evacuated quartz tube and annealed at 1173 K for 30 days, then quenched in liquid nitrogen. The crystalline structures and the crystal lattice parameters of compounds were determined by the room-temperature X-ray powder diffraction and the Rietveld refinement. The isothermal and isofield magnetic properties were measured in a MPMS SQUID VSM magnetometer from Quantum Design Inc.

3. Results and discussion

Fig. 1 shows the room-temperature X-ray diffraction (XRD) patterns and Rietveld refinement of HoAl and ErAl compounds. The results indicate that both HoAl and ErAl crystallize in orthorhombic DyAl-type structure. The calculated lattice parameters are a = 5.8065 Å, b = 11.3357 Å, and c = 5.5852 Å for HoAl compound

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http://dx.doi.org/10.1016/j.jallcom.2014.01.202
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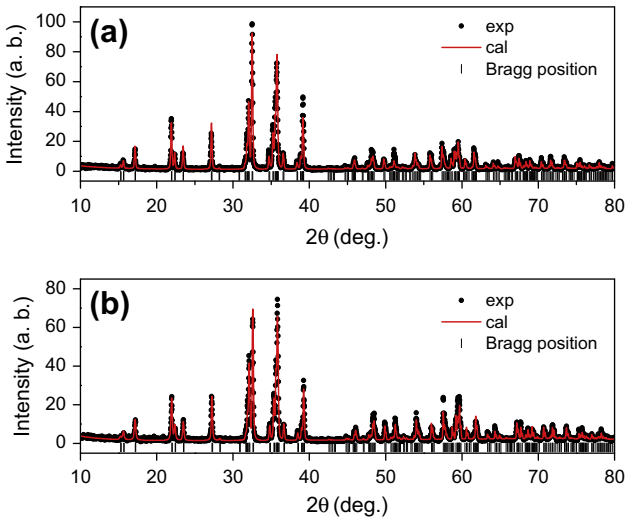

Fig. 1. Room-temperature XRD and Rietveld refinement of *R*Al (*R* = Ho and Er) compounds.

and $a = 5.7832$ Å, $b = 11.2799$ Å, and $c = 5.5644$ Å for ErAl compound, respectively. These values are consistent with the values reported before [26,27].

Fig. 2(a) and (b) displays the zero-field-cooling magnetization curves (M–T) of HoAl and ErAl under a field of 0.01 T, respectively. From Fig. 2(a), it is observed that HoAl compound undergoes two successive transitions at 13 K and 20 K respectively. This result is consistent with the previous report. Bécle et al. [28] reported that HoAl was ferromagnetic under $T_C = 26$ K. Meanwhile, he pointed out that the neutron diffraction pattern at 4.2 K exhibited the characteristic of the coexistence of an antiferromagnetic and ferromagnetic arrangement. Consequently, the observed two phase transitions could be corresponding to AFM–FM transition at $T_1 = 13$ K and FM–PM transition at $T_C = 20$ K, respectively. This analysis can be confirmed by the isothermal magnetization curves and magnetic field dependence of magnetic entropy change. Fig. 2(b) presents the M–T curve of ErAl compound. A typical magnetic transition from AFM to PM at $T_N = 9$ K is observed. This result is in agreement with the result reported before. The insets of Fig. 2(a) and (b) show the Curie–Weiss fits of HoAl and ErAl in the PM region, respectively. The effective paramagnetic moments per R atom (R = Ho and Er) are $11.0\mu_B$ and $9.8\mu_B$, close to the free ion values ($10.6\mu_B$ for Ho^{3+} and $9.6\mu_B$ for Er^{3+}).

The isothermal magnetization curves (M–H) of HoAl and ErAl are measured under applied fields up to 5 T in a vide temperature range. Fig. 3(a) shows the M–H curve of HoAl compound. There is not obvious magnetic hysteresis, which is important for practical application of magnetic refrigerant materials. Fig. 3(b) presents the M–H curves in a temperature range from 5 to 15 K at low magnetic fields. The intersections among these curves are observed, which are caused by field-induced metamagnetic transition from AFM to FM state. The observed metamagnetic transition demonstrates the existence of antiferromagnetic state at low temperatures, which is in agreement with the result of M–T curve. Fig. 4(a) displays the M–H curves of ErAl compound. As an antiferromagnet, ErAl compound is expected to exhibit a metamagnetic transition below $T_N = 9$ K in the M–H curve. However, it is interesting that two successive field-induced transitions are observed in the temperature range of 5–9 K (shown in Fig. 4(b)). This result resembles the result of ErTiSi [29], in which a field-induced AFM–AFM transition and a subsequent AFM–FM metamagnetic transition were observed with increasing magnetic field at low temperatures. From Fig. 4(b), ErAl compound firstly undergoes a possible field-induced AFM-AFM transition at about 0.3 T and a succedent AFM–FM metamagnetic transition at about 1.2 T with increasing magnetic field at 5 K.

Associated with the rich transition behaviors, magnetic entropy changes ΔS_M are calculated by Maxwell's relation $\Delta S_M = \int_0^H \left(\frac{\partial M}{\partial T}\right) dH$. Fig. 5(a) displays the temperature dependence of $-\Delta S_M$ for different magnetic field changes for HoAl compound. The observed two peaks centered at about 8 K and 20 K are corresponding to the field-induced metamagnetic transition at T_1 and FM–PM transition at T_C, respectively. For a field change of 0–5 T, the maximal value of $-\Delta S_M$ for HoAl reaches 22.5 J kg^{-1} K^{-1} at 20 K. Fig. 5(b) presents the temperature dependence of $-\Delta S_M$ for ErAl compound. It is seen clearly that the value of $-\Delta S_M$ are negative at low temperatures under low fields. With increasing temperature and magnetic field,

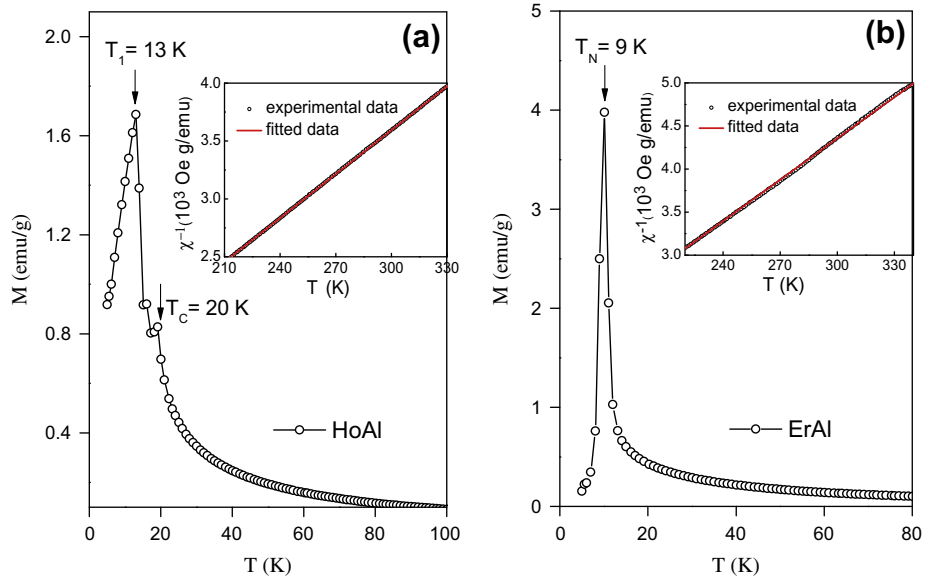

Fig. 2. Temperature dependences of zero-field-cooling magnetization for HoAl (a) and ErAl (b) under 0.01 T. The insets of (a) and (b) are the Curie–Weiss fits for HoAl and ErAl, respectively.