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# Phase equilibria in the Tl–Tll–Te system and thermodynamic properties of the $Tl_5Te_{3-x}I_x$ solid solution

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#### 1. Introduction

Complex chalcogenides and chalcogen-halides of heavy *p* metals are considered perspective functional materials. Many of them exhibit semiconductor, thermoelectric, photoelectric, and magnetic properties [1,2]. According to recent investigations, Tl<sub>6</sub>Sel<sub>4</sub> and Tl<sub>5</sub>Se<sub>2</sub>I are promising materials for efficient X-ray and  $\gamma$ -ray detection, outperforming the current state-of-the art material for room temperature operation, CdZnTe (CZT) [3,4]. Given the promising performance of thallium-based chalcogenides and chalcogenide-halide, these compounds continue to attract attention despite of toxicity of thallium derivatives.

The synthesis and growth of the single crystals from the melt requires the knowledge of the respective phase diagrams, which makes the investigation of phase equilibria very important in course of the material development. This is particularly true for quaternary phases, which frequently present wide homogeneity fields based on complex substitutional patterns. Typically, their functional properties can be tuned by varying the composition, which in turn calls for the development of a synthetic approach

#### ABSTRACT

The Tl-TlI-Te ternary system was investigated by using the DTA and XRD analyses, microhardness and EMF measurements, leading to the construction of important polythermal and isothermal sections as well as the projection of the liquidus surface. Determined were the fields of primary crystallization and the types and coordinates of non- and monovariant equilibria. The system features the formation of continuous solid solutions ( $\delta$ -phase) between the Tl<sub>5</sub>Te<sub>3</sub> and Tl<sub>5</sub>Te<sub>2</sub>I compounds. The homogeneity area of the  $\delta$ -phase completely covers the Tl<sub>5</sub>Te<sub>2</sub>I-Tl<sub>5</sub>Te<sub>3</sub>-Tl<sub>2</sub>Te subsystem. The partial molar thermodynamic functions ( $\Delta \overline{\Delta}, \Delta \overline{H}, \Delta \overline{S}$ ) of thallium as well as the standard integral thermodynamic functions of the Tl<sub>5</sub>Te<sub>3-x</sub>I<sub>x</sub> solid solution (for various *x*) were calculated based on the results of the EMF measurements. © 2013 Elsevier B.V. All rights reserved.

toward the preparation of a required stoichiometry within a homogeneity range.

In our previous works, phase equilibria and thermodynamic properties of the systems Tl–TlCl–S [5], Tl–TlBr–S [6], Tl–TlL–S [7], Tl–TlCl–Se [8], Tl–TlBr–Se [9], Tl–TlCl–Te [10], and Tl–TlBr–Te [11] were investigated. A number of polythermal and isothermal sections as well as projections of liquidus surfaces were constructed. The homogeneity fields of the intermediate phases were fixed and their fundamental thermodynamic functions were calculated.

In this work, we report the results of the complete investigation of phase equilibria and thermodynamic properties of the Tl–Tll–Te system. The layout of this paper is as follows. We start with reviewing the literature data on this system and compounds in it, follow with the description of the experimental procedure, and then discuss our results on the phase equilibria, polythermal sections, liquidus surface, and thermodynamic properties of the  $Tl_5Te_{3-x}I_x$  solid solution.

#### 1.1. Review of the literature data

The ternary system Tl–Te–I was investigated along various polythermal sections [12–15].

For the Tl–Te boundary system, 4 compounds have been reported; they are  $Tl_2Te$ ,  $Tl_5Te_3$ , TlTe, and  $Tl_2Te_3$ . The former two





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melt congruently at 698 and 726 K, respectively, whereas the latter two decompose upon melting at 573 and 511 K, respectively [16,17]. The quasi-binary system TII–Tl<sub>2</sub>Te is characterized by the formation of a ternary compound Tl<sub>5</sub>Te<sub>2</sub>I that melts by a synthectic reaction at 775 K [12]. The immiscibility area occupies the 20– 80 mol% TII concentration interval at a synthectic temperature. Eutectic is degenerated near TII and crystallizes at 713 K. Authors of [12] assume the presence of the peritectic equilibrium between Tl<sub>5</sub>Te<sub>2</sub>I and Tl<sub>2</sub>Te at ~730 K. The homogeneity region of Tl<sub>5</sub>Te<sub>2</sub>I at ~600 K is in the range ~65–745 mol% Tl<sub>2</sub>Te. The homogeneity areas of TII and Tl<sub>2</sub>Te were not determined in that work.

This compound forms a continuous solid solution with  $Tl_5Te_3$ [13]. The polythermal sections TII–TITe and TII–Tl\_2Te\_3 [14] are stable in subsolidus, i.e., consist of heterogeneous mixtures of primary compounds, but they are non-quasi-binary because of the peritectic decomposition of TITe and  $Tl_2Te_3$ . There are wide areas of immiscibility in the phase diagrams of both systems. The phase diagram of the quasi-binary system TII–Te is characterized by monotectic and eutectic equilibria [15]. The *T*–*x* diagrams of the boundary systems TI–TII [18] and TII–Te [15] are of the monotectic type. The binary compound TII constituting one of the corners of the ternary system TI–TII–Te melts congruently at 715 K and undergoes the TII<sub>I</sub>  $\Leftrightarrow$  TII<sub>II</sub> polymorph transformation at 451 K [18]. Also, the preparation, stability regions, and properties of Tl<sub>2</sub>Tel<sub>6</sub> were reported [19].

The following structural information is available from the literature regarding the binary and ternary compounds of the Tl–Te–I system (see Table 1). The low-temperature modification of Tll crystallizes in the orthorhombic symmetry and has a molecular-like crystal structure, whereas the high-temperature modification belongs to the CsCl structure type [20,21].

Tl<sub>2</sub>Te crystallizes in the monoclinic space group  $C_2/c$  with 44 formulas per unit cell and has an unusually complex structure composed of Tl<sup>1+</sup> and Te<sup>2–</sup> [17]. TlTe and Tl<sub>2</sub>Te<sub>3</sub> possess less complex crystal structures [22,23], however, they exhibit systems of homonuclear Tl–Tl or Te–Te bonds that make their structure somewhat similar to cluster halides of bismuth and tellurium [24,25]. Tl<sub>5</sub>Te<sub>3</sub> and Tl<sub>5</sub>Te<sub>2</sub>I belong to a structural family of compounds crystallizing in the In<sub>5</sub>Bi<sub>3</sub> structure type and its less symmetric analogs [26,27]. Crystal structures of other thallium selenahalides and tellurohalides of the 5:2:1 stoichiometry also belong to this structure family [8–11,28,29].

Finally, the crystal structure of  $Tl_2Tel_6$  features  $Te^{4+}$  cations in a distorted octahedral environment of iodine atoms [30]. Note that this compound does not belong to the ternary Tl–Tll–Te system studied in this work.

#### 2. Experimental

#### 2.1. Synthesis

Elemental thallium (Tl-99.999%, Alfa Aesar), tellurium (Te-99.999%, Alfa Aesar) and iodine (I-99% resublimed pearls, PA-ACS) were used as received. At the first step, they were used for the preparation of binary compounds  $Tl_2Te$ ,  $Tl_5Te_3$ , TITe,  $Tl_2Te_3$ , TII, and  $Tl_5Te_2I$ .

#### Table 1

Structural data for compounds of the TI-TII-Te system.

Congruently melting compounds Tl<sub>2</sub>Te and Tl<sub>5</sub>Te<sub>3</sub> were prepared from the stoic chiometric amounts of the corresponding elements by one-step melting in vacuumsealed (~10<sup>-2</sup> Pa) silica ampoules at 800 K, followed by cooling in a switched-off furnace. For incongruently melting TITe and Tl<sub>2</sub>Te<sub>3</sub> after fusing at 800 K the ampoules were slowly cooled to  $560 \pm 5$  K (TITe) or  $500 \pm 5$  K (Tl<sub>2</sub>Te<sub>3</sub>), annealed for 500 h, and then slowly cooled in a switched-off furnace.

Tll was synthesized from the elements following a specially designed procedure, which takes into account high volatility of iodine, describes in detail elsewhere [27].

Ternary compound  $Tl_5Te_2I$  was synthesized by melting appropriate amounts of the synthesized TII and  $Tl_2Te$  in a vacuum-sealed quartz ampoule. This compound decomposes by a synthectic reaction upon melting [12]. Accordingly, the ampoule after fusing at 800 K was cooled slowly to 730 K and held at this temperature for ~800 h.

Samples for the investigation of phase equilibria and thermodynamic properties were prepared from  $Tl_2Te$ ,  $Tl_5Te_3$ , TITe,  $Tl_2Te_3$ , TII, and  $Tl_5Te_2I$ . The total mass was 1 g. In most cases, after determining the solidus temperature, samples were annealed at 20–30 K below the solidus for 800–1000 h.

#### 2.2. Analysis

X-ray powder diffraction (XRD), differential thermal analysis (DTA), measurement of microhardness, and electromotive force were used to characterize the samples.

The XRD analysis was performed on a Bruker D8 ADVANCE diffractometer with the Cu K $\alpha_1$  radiation. The lattice parameters were refined using the Topas V3.0 software. XRD confirmed that the pre-synthesized binary and ternary compounds were phase-pure, and that the unit cell parameters perfectly matched the literature data for binary compounds. The powder XRD pattern for Tl<sub>5</sub>Te<sub>2</sub>I was indexed using Topas V3.0 software. It was confirmed that Tl<sub>5</sub>Te<sub>2</sub>I crystallizes in the tetragonal Tl<sub>5</sub>Te<sub>3</sub> structure type with the space group *I4/mcm* and the following unit cell parameters: *a* = 9.026(1), *c* = 13.324(3) Å, *z* = 4, which perfectly matches the literature data (see Table 1).

DTA was carried out with a Termoskan-2 device. The temperature was monitored by a Chromel–Alumel thermocouple. The ramp rate was 5 K min<sup>-1</sup>. Temperatures of thermal effects were taken mainly from the heating curves. The melting point of the ternary compound  $Tl_5Te_2l$  was found to be in agreement with the literature data [12,27].

Microhardness was measured with a PMT-3 tester, the typical loading being 20 g.  $\,$ 

For the electromotive force (EMF) measurements, the following concentration chains were assembled:

$$(-)Tl(solid)/glycerin + KI + TlI/[Tl - Te - I](solid)(+)$$

$$(1)$$

In the chains of type (1), metallic thallium was used as the left electrode, while equilibrium alloys of the system Tl–Te–I were exploited as the right electrode. A saturated glycerin solution of Kl with the addition of 0.1 mass% Tll was used as an electrolyte. The electrodes were prepared by pressing the powdered alloys in the form of pellets (0.5 g) in a molybdenum wire. The temperature was stabilized at 350 K for 40–50 h. EMF was measured by the compensation method in the temperature range of 300–400 K with the accuracy of  $\pm 0.1$  mV, using a high-resistance universal B7–34A digital voltmeter. The detailed description of the methods of assembling the electrochemical cell and carrying out the EMF measurements are given in [31,32].

#### 3. Results and discussion

To obtain complete a picture of phase equilibria in the system Tl–Tll–Te we have prepared and investigated a number of samples in the sections  $Tl_2Te$ –Tll,  $[Tl_2I]$ –Tl<sub>2</sub>Te, and  $Tl_5Te_2I$ –Te(Tl), and also several additional samples out of the given sections. The results are given in Figs. 1–8 and Tables 2–6.

Compound	Space group	Unit cell parameters	Ref.
TII	Cmcm (LT) Pm-3m (HT)	<i>a</i> = 4.57, <i>b</i> = 12.92, <i>c</i> = 5.24 Å <i>a</i> = 4.201 Å	[20,21]
Tl <sub>2</sub> Te	$C_2/c$	$a = 15.662, b = 8.987, c = 31.196$ Å, $\beta = 100.76^{\circ}$	[17]
Tl <sub>5</sub> Te <sub>3</sub>	I4/mcm	<i>a</i> = 8.929, <i>c</i> = 12.620 Å	[26]
TlTe	$P4_2/nmc$	a = 18.229, c = 6.157 Å	[22]
Tl <sub>2</sub> Te <sub>3</sub>	Сс	$a = 17.413, b = 6.552, c = 7.910$ Å, $\beta = 133.6^{\circ}$	[23]
Tl <sub>5</sub> Te <sub>2</sub> I	I4/mcm	<i>a</i> = 9.026; <i>c</i> = 13.324 Å	[27]
Tl <sub>2</sub> Tel <sub>6</sub> <sup>a</sup>	$P2_1/c$	<i>a</i> = 7.765, <i>b</i> = 8.174, <i>c</i> = 13.756 Å, $\beta$ = 124.2°	[30]

<sup>a</sup> Composition out of the TI-TITe-I system; see text.

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