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Journal of Alloys and Compounds

journal homepage: www.elsevier.com/locate/jallcom



A study on microstructure and luminescent properties of oxyfluoride silicate glass-ceramics with (Ho³⁺,Yb³⁺):NaYF₄ crystallites

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ARTICLE INFO

Article history:
Received 1 July 2011
Received in revised form 7 September 2011
Accepted 8 September 2011
Available online 16 September 2011

Keywords: Glass-ceramics NaYF₄ nanocrystals (Ho³⁺/Yb³⁺) luminescence Excited-state dynamics

ABSTRACT

Glass-ceramics containing NaYF $_4$ nanocrystals were prepared by heat-treatment from oxyfluoride silicate-based glass doped with Ho $^{3+}$ and Yb $^{3+}$ ions. The formation of crystalline fluoride phase was confirmed by X-ray diffraction and transmission electron microscopy. Absorption and emission spectra revealed that a fraction of Ho $^{3+}$ and Yb $^{3+}$ ions is incorporated into the NaYF $_4$ ordered lattice influencing spectroscopic features of glass-ceramics in comparison with those of precursor glass. Green up-conversion emission (545 nm) originating in the 5 F $_2$ level in glass-ceramics and up-converted red emission (650 nm) originating in the 5 F $_5$ level in as-melted glass were observed under optical pumping into Yb $^{3+}$ absorption band and analyzed. Although both emissions in both materials are achieved by two-photon excitations, the relation between green and red emission intensity in glass-ceramics and glass implies that processes relevant to up-conversion phenomena are different. Based on a careful analysis of relaxation dynamics of Ho $^{3+}$ and Yb $^{3+}$ excited states, the mechanisms involved in conversion of the infrared radiation into the visible emission in these materials are proposed and discussed.

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1. Introduction

Oxyfluoride glass-ceramics systems doped with rare earth ions have been researched widely in the last decade since they possess not only higher chemical and mechanical stability than fluoride glasses but also lower phonon energy than oxide ones. It has been reported in numerous papers that nano-sized crystalline phase like PbF_2 , $Pb_{1-x}Cd_xF_2$, MeF_2 (Me = alkaline earth cations), LaF_3 or $NaYF_4$ can be created in glasses by controlled thermal treatment without loss of transparency. An interest in luminescent properties of trivalent holmium ion is related to laser emission achieved in the ${}^{5}I_{7} \rightarrow {}^{5}I_{8}$ transition around 2.2 μm (eye-save spectral region) and to the visible emission achieved in up-converted processes. Since a direct excitation of Ho³⁺ with diode lasers is inefficient, Ho³⁺ ion is usually sensitized with Yb3+ that shows a very strong and broad absorption band at around 1.0 µm, easy available for commercial laser diodes. Energy exchange from Yb³⁺ to Ho³⁺ allows to pump holmium luminescent states for the infrared and visible emissions. Numerous published paper were devoted to luminescence properties of Ho³⁺ and (Ho³⁺,Yb³⁺) in various multicomponent glasses [1–4] and glass-ceramics containing LaF_3 , PbF_2 or $Pb_{1-x}Cd_xF_2$ crystal phase [5-7]. However, to the best of our knowledge, optical studies of Ho³⁺ sensitized with Yb³⁺ in glass-ceramics systems with

the nanocrystalline phase of NaYF₄ have not been reported. The NaYF₄ has been recognized as very efficient up-conversion matrix [8] especially for $\rm Er^{3+}$ or $\rm Tm^{3+}$ with Yb³⁺ as a codopant [9,10]. These reported results were motivation to investigate spectroscopic features of silicate glass-ceramics with (Ho³⁺,Yb³⁺):NaYF₄ crystallites.

In this paper, we report the morphology of silica-based glass-ceramics with NaYF4 nanocrystals, single- and double-doped with Ho³+ and Yb³+, and emission properties of Ho³+–Yb³+ co-doped glass and glass-ceramics materials under optical pumping into Ho³+ or Yb³+ absorption bands. Particular attention has been directed to relaxation dynamics of holmium and ytterbium excited states in order to explain mechanisms responsible for up-converted green emission originating in the $^5\mathrm{S}_2$ level in the glass-ceramics and red emission from the $^5\mathrm{F}_5$ state in glass.

2. Experimental

The mixture of high purity (4 N, Sigma–Aldrich) reagents were melted to obtain the glass samples with the batch composition in mol%: $40 \text{SiO}_2 - 25 \text{Al}_2 \text{O}_3 - 18 \text{Na}_2 \text{CO}_3 - 7 \text{Na} \text{F} - (10 - x - y) \text{YF}_3 - x \text{HoF}_3 - y \text{YbF}_3 \ (x = 0, 0.5 \text{ and } y = 0, 1, 4)$. The anhydrous Na₂CO₃, NaF, YF₃ and LnF₃ (Ln=Ho, Yb) were used. Owing to a fluorine loss that may occur in the preparation procedure, a 10 mol% excess of YF₃ was used. The 20g of starting reagents, thoroughly mixed and put in a covered corundum crucible in dry box, were introduced into a resistance furnace at 1450 °C and melted for one hour and fifteen minutes in normal atmosphere. The glassy liquid was poured into copper plate. To obtain glass-ceramics samples, the precursor (as-melted) glass ingots were cut into pieces and heat-treated at temperature ranging from 610 °C to 640 °C and time ranging from 2 to 4h to find optimal crystallization conditions.

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The X-ray powder diffraction spectra were recorded with X'Pert PRO X-ray diffractometer (Cu, $K\alpha_1$ radiation: 1.54060 Å), in the range of $2\Theta = 5-80^\circ$. Morphological studies encompassing transmission electron microscopy (TEM), and selected area electron diffraction (SAED) measurements were performed using a Philips Transmission Electron Microscope CM-20 Super Twin with 0.24 nm resolution at 200 kV. Absorption spectra were recorded using a Varian 5 UV-vis-NIR spectrophotometer. Low-resolution emission spectra were performed by microspectrometer Ocean Optics Model USB 200. Upconversion spectra were excited by an Apollo Instruments diode laser emitting continuous wave (cw) radiation at 980 nm with a maximum power up to 3 W. Resulting luminescence was dispersed by an Optron DongWoo DM711 monochromator with 750-mm focal length and detected by a Hamamatsu R-955 photomultiplier in UV-vis and by PbS detector in NIR spectral range. Luminescence decay curves were recorded upon a pulsed excitation delivered by a Continuum Surelite optical parametric oscillator (OPO) pumped with the third harmonic of Nd:YAG laser. The decays were measured with Hamamatsu R-955 photomultiplier or a cooled InSb Janson I10D detector (depending on the spectral range) connected to a Tektronix Model TDS 3052 digital oscilloscope. All measurements were carried out at room temperature.

3. Results and discussion

3.1. Morphological characteristics

Fig. 1 presents the X-ray diffractograms of samples single-doped with $\mathrm{Ho^{3+}}$ (0.5% of $\mathrm{HoF_3}$) and $\mathrm{Yb^{3+}}$ (1% of $\mathrm{YbF_3}$), and double-doped with $\mathrm{Ho^{3+}}$ and $\mathrm{Yb^{3+}}$ (0.5% $\mathrm{HoF_3}$ –4% $\mathrm{YbF_3}$) that were heat-treated at temperatures from 610 to 630 °C. Sharp and intensive diffraction peaks are observed in the spectra of heat-treated glasses co-doped with $\mathrm{Ho^{3+}}$ and $\mathrm{Yb^{3+}}$ and single doped with $\mathrm{Yb^{3+}}$ ions whereas, the $\mathrm{Ho^{3+}}$ -doped glass not so easy undergoes structural modifications in the same thermal and temporal conditions regardless of holmium concentration (data not presented here). It seems that a tendency to crystallization may be controlled by ionic radii of lanthanide impurities. Ions from the beginning ($\mathrm{Nd^{3+}}$, $\mathrm{Pr^{3+}}$) and the end ($\mathrm{Yb^{3+}}$) of lanthanide series were found as nucleating agents that promote evidently the crystallization of $\mathrm{NaYF_4}$ phase in the systems [11,12]. The crystalline phase formed in the glass-ceramics has been

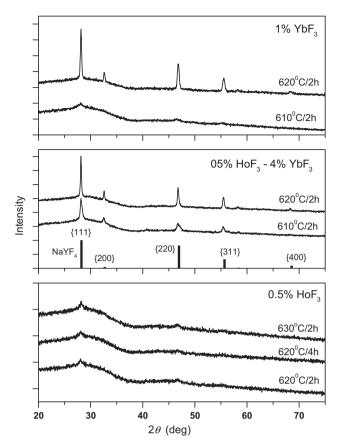


Fig. 1. X-ray diffraction patterns recorded for samples thermal treated at different temperatures. The cubic NaYF₄ pattern with 2Θ and $\{h\,k\,l\}$ values are presented for the comparison.

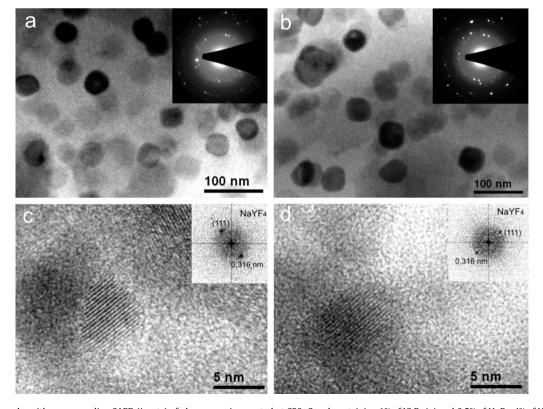


Fig. 2. TEM micrographs with corresponding SAED (insets) of glass-ceramics created at $620\,^{\circ}$ C and containing 1% of YbF₃ (a) and 0.5% of HoF₃-4% of YbF₃ (b). The HRTM images with corresponding fast Fourier transforms (FFT – insets) of the samples with 1%YbF₃ (c) and 0.5%HoF₃ (d) heat-treated at $610\,^{\circ}$ C and $620\,^{\circ}$ C, respectively.

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