



Effects of deformation on hydrogen absorption and desorption properties of titanium

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ABSTRACT

In this study, we analyzed the effects of deformation on hydrogen absorption and desorption properties of titanium to improve such properties. Hydrogen was introduced into commercially pure (99.5%) titanium by the electrochemical method. The amount and existing state of hydrogen were examined using hydrogen desorption curves obtained by thermal desorption spectroscopy. Hydrogen absorption was promoted by applying tensile deformation prior to charging, which leads to hydride formation within a short charging time. The amount of hydrogen absorbed decreased when the volume fraction of deformation twins exceeded about 0.2. It was considered that hydrogen was mainly trapped by dislocations forming hydride while a large fraction of deformation twins hindered dislocation motion, thus reducing dislocation density leading to a decrease in the amount of absorbed hydrogen. Almost half the charged hydrogen was released when in-plane compressive stress was applied to a charged plate specimen at room temperature due to hydride decomposition under compressive stress.

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1. Introduction

Titanium absorbs a large amount of hydrogen as hydride (gravimetric density: 4 mass%, volumetric density: 148 kg/m³) owing to the negative enthalpy of formation of TiH₂ (−131 kJ/mole H₂), that requires high temperatures for the release of hydrogen under atmospheric pressure [1]. To make titanium more attractive as a hydrogen storage material, it is desirable to explore processes for hydrogen release under ambient conditions while accelerating absorption. In this study, deformation was applied to titanium as a method of improving its hydrogen absorption and desorption properties.

There have been many reports on the formation kinetics and configuration of hydride in hcp materials including titanium, especially under the influence of stress [2–4]. However, there was little study on the behaviour of hydrogen in deformed titanium and on the kinetics of hydride formation in connection with defects, which can serve as trapping sites of hydrogen and become preferential sites of hydride formation [5,6]. In hcp metals, mechanical twins are formed under deformation and their number increases linearly with applied deformation [7–10]. The contribution of twins to total deformation is small [11,12], because the relative atomic movement is limited. Most of the plastic flow occurs by the motion of dislocations, even in hcp metals.

However, the presence of a number of twins and their interactions were considered to hinder the motion of dislocations [13]. Therefore, deformation twins and twin boundaries play a role in hydrogen absorption by affecting dislocation density. The role of twin in hydrogen absorption properties was analyzed in this study.

Liu and Nakasa reported the result of an experiment on anomalous hydrogen absorption behavior during electrochemical charging that may be caused by hydride decomposition under compressive stress [14]. We previously measured hydrogen release from hydrogenated titanium during compressive or tensile deformation using a tensile testing machine fitted with a vacuum chamber and a quadropole mass spectrometer [15]. The results showed that the release of hydrogen was observed only under compressive stress. On the basis of these findings, hydrogen release from hydrogenated titanium under compressive stress was measured and analyzed.

2. Materials and experimental procedure

Commercially pure (99.5%) titanium of 1 mm or 0.1 mm thickness was used in the experiments. Anode electrochemical charging was conducted at a potential of −1.5 V (vs. a Ag/AgCl reference electrode) in a 323 K aqueous H₂SO₄ solution of pH 2.0, which corresponds to the conditions in the corrosion region of the Pourbaix diagram of titanium [16].

Three-pole electrochemical charging was employed in hydrogen charging. A specimen, a platinum plate serving as a counter electrode, and a Ag/AgCl reference electrode were immersed in H₂SO₄ solution under a constant potential by using a potentiostat. Hydrogen desorption curves were measured by thermal desorption analysis to assess the amount of hydrogen together with its existing state in the material. The specimen was heated in high-purity argon gas at a heating rate of

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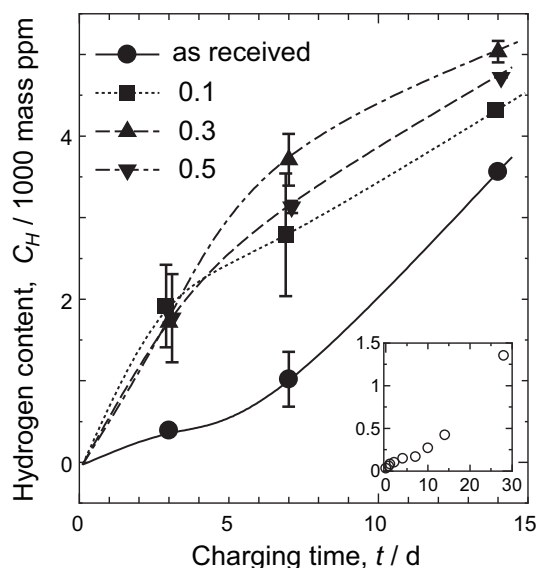


Fig. 1. Dependences of amount of hydrogen absorbed by titanium on charging time and tensile strain (0.1, 0.3, or 0.5) applied prior to charging. Inserted graph shows hydrogen absorption properties measured with the same material under a charging condition of a potential of -1.0 V (vs. a Ag/AgCl reference electrode) in 303 K aqueous H_2SO_4 solution of $\text{pH}3.0$.

100 K/h . The amount of released gas was measured using a gas chromatograph every 5 min .

In the experiment examining the effect of tensile deformation on hydrogen absorption properties, tensile deformation was applied to a specimen of 1 mm thickness and 20 mm gauge length at a displacement rate of 0.1 mm/min at room temperature prior to electrochemical charging. The specimen was annealed at 973 K for 3 h before deformation for full anneal heat treatment [17]. The strain applied was chosen to be 0.1 , 0.3 , or 0.5 . Strain was obtained by dividing the displacement of a cross-head of a testing machine by the gauge length of the specimen. After tensile deformation, the surface of the specimen was polished with #800 emery paper to minimize the effects of surface changes caused by deformation.

In the experiment examining the effect of compressive stress on hydrogen desorption properties, specimens of $20\text{ mm} \times 5\text{ mm}$ and 0.1 mm thickness annealed at 973 K for 30 min were charged electrochemically at 323 K for 3 days . Under the charging conditions used in this study, uniform hydride phase was formed up to approximately $30\text{ }\mu\text{m}$ from the surface of the specimen and precipitated hydride is observed up to $50\text{ }\mu\text{m}$ from the surface in the metallographic examination. Therefore, the specimen of 0.1 mm thickness used in the experiments contains hydride throughout the thickness.

3. Results and discussion

3.1. Effect of tensile deformation on hydrogen absorption properties

The relationship between charging time and the amount of hydrogen absorbed by the specimens that was deformed uniformly with different strains is shown in Fig. 1. The inserted graph in Fig. 1 shows hydrogen absorption properties measured with the same material under a charging condition of a potential of -1.0 V (vs. a Ag/AgCl reference electrode) in a 303 K aqueous H_2SO_4 solution of $\text{pH}3.0$. This data was used as a reference for hydrogen absorption property of the material, and the effect of deformation on the property is measured and compared at three charging time. At the start of charging, the amount of hydrogen increases linearly with absorption time, then the relationship becomes parabolic due to hydride formation [18,19]. In the as-received material, the linear relation at the early stage is not apparent. This may be related to hydride decomposition on the surface due to the compressive stress caused by the formation of a hydride phase on the surface [14]. The amount of hydrogen at a certain charging time increases with deformation, except at a strain of 0.5 . The reason for this will

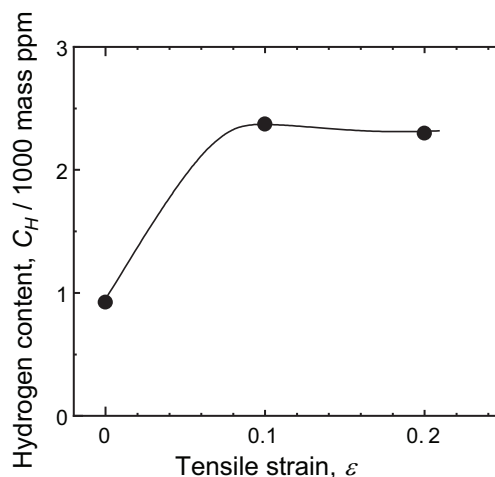


Fig. 2. Dependence of amount of hydrogen absorbed by titanium on tensile strain applied at 373 K prior to charging.

be discussed later. Results of X-ray diffraction (XRD) measurement show that the hydride phase of $\text{TiH}_{1.924}$ appears within a shorter charging time with increasing deformation. Therefore the effect of tensile deformation prior to hydrogen charging is the promotion of hydride formation.

The role of dislocations and deformation twinning in hydrogen absorption was analyzed experimentally by controlling the number of twins under a fixed strain. Since deformation twinning is strongly dependent on temperature [9,11], a fixed strain was applied to the specimen at temperatures from 77 to 373 K . The volume fraction of twins was measured by the point count method. A grid of $12.5\text{ }\mu\text{m}$ square was overlaid on an $800\text{ }\mu\text{m} \times 600\text{ }\mu\text{m}$ optical micrograph. The test points at the intersections of the grid that were located on a twin were counted, and the ratio of their number to the total number of test points was defined as the volume fraction of deformation twins. The measurement was conducted in at least 10 micrographs to obtain the average volume fraction.

Fig. 2 shows the relationship between tensile strain and the amount of hydrogen absorbed by the specimens deformed at 373 K . At this temperature, the volume fraction of deformation twins is approximately 0.01 in the strain range applied; therefore only the effect of dislocation on absorption of hydrogen is shown. Hydride is formed in the material because the amount of absorbed hydrogen readily exceeds the solubility limit of titanium at room temperature [20]. The amount of absorbed hydrogen increases rapidly with strain when the applied strain is small, then gradually becomes almost constant. This suggests that hydride formation is strongly related to dislocations. Although the binding energy of hydrogen to various types of defects in titanium has not yet been reported, dislocations are considered to be major trapping sites for hydrogen compared with other defects such as twin boundaries [13].

Fig. 3 shows the relationship between the volume fraction of deformation twins and the amount of absorbed hydrogen under a constant tensile strain of 0.1 . The amount of hydrogen is not affected by the volume fraction of twins until it reached about 0.2 . As mentioned earlier, deformation is mainly caused by dislocation motion. On the other hand, with increasing number of twins, their contribution to total deformation increases; furthermore secondary twins occur, thereby reducing the equivalent size of grains. Both factors cause a decrease in dislocation density that leads to a smaller number of trapping sites for hydrogen and hydride, thus reducing the amount of absorbed hydrogen. At a strain of 0.5 , the volume fraction of deformation twins that increases linearly with applied deformation exceeds the threshold value mentioned above,

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