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Fabrication of TiO₂ nanocrystals/nanorods composites thin film electrode: Enhanced performance of dye-sensitized solar cells

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1. Introduction

Among various metal oxide semiconductors, TiO₂ was widely studied for its low cost, nontoxic peculiarities and chemical stability [1]. Due to its large exciton binding energy and electron mobility, TiO₂ thin films play an important role in photovoltaic devices, including dye sensitized solar cells (DSCs) [1], perovskitesensitized solar cells [2], polymer-inorganic hybrid solar cells [3], quantum dot-sensitized solar cells [4] and inorganic solid-state solar cells [5]. In order to improve the photovoltaic properties of DSCs, the investigation on the charge transport and recombination in TiO₂ thin film becomes very important [6]. The mesoporous TiO₂ thin film prepared from TiO₂ nanocrystals (NCs) via a screenprinting method [7] can serve as both a substrate for the dye and the electron conductor in photoanode of DSCs. Generally, the electrodes of DSCs based on NCs can provide larger surface area. However, the relatively poor electron transport property of TiO₂ NCs makes it difficult for further improving the photovoltaic properties of DSCs. In recent years, one-dimensional nanomaterials, including nanotubes [8], nanorods [9] and nanofibers [10], have received great attention in fabricating DSCs due to their superior electron transport properties. In our previous work, we reported that the DSCs fabricated with seaweed-like TiO₂ nanoarrays have much lower recombination rate and longer electron life time [11]. In addition, the combination of TiO_2 NCs and TiO_2

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ABSTRACT

The TiO₂ thin film (CTTF) composed of nanocrystals (NCs) and one-dimensional (1D) single crystalline nanorods (NRs) was prepared and then used as electrode to fabricate dye sensitized solar cells (DSCs). The kinetic parameters of electron transport in CTTF were investigated. Results indicated that due to the direct path for fast electron transport in NRs, the fabricated CTTF electrodes-based DSCs have much lower recombination rate, longer electron life time and electron diffusion length than that based on the pure TiO₂ NCs electrodes.

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nanorods (NRs) was reported an effective way to enhance electron transport properties [12]. However, the dynamic mechanisms of DSCs were rarely reported. In this paper, DSCs based on combined TiO₂ NRs and NCs were fabricated and the influence of NRs on the electron transport properties of DCSs was investigated.

2. Experimental

The preparation of 1D single crystalline TiO₂ nanorods (NRs) were synthesized similarly as the process in Ref. [13]. TiO₂ powder (Aldrich) was firstly added into 10 M NaOH aqueous solution. Then the mixture solution was transferred into a Teflon-lined autoclave and heated at 170 °C for 24 h. After the autoclave was cool down to room temperature naturally, the product was washed with deionized water for several times to obtain the intermediates. The intermediates were then mixed with concentrated nitric acid and maintain at pH=2 for 6 h. After that, they were washed with deionized water for several times until the pH value of the solution became neutral. The obtained one-dimensional micron-sized titanate materials were transferred into Teflon-lined autoclave and filled with the nitric acid liquor of pH=2 at 170 °C for 24 h. The cooled product was dispersed ultrasonically, centrifuged to remove nitric acid, and finally washed with de-ionized water and ethanol three times to obtained TiO₂ NRs.

The processes for preparing TiO_2 paste and fabricating solar cells were similar as Ref. [7]. Three kinds of pastes containing mixed TiO_2 NCs and NRs were prepared with NRs content of





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0, 5 and 10 wt% (labeled as pastes A–C), respectively. The CTTFs with the thickness of 7.4 μ m were prepared by screen printing method with above three different pastes. Then, they were fabricated into DSCs marking with devices A–C after sensitized-treating the CTTFs with dye C106.

3. Results and discussion

The transmission electron microscopy (TEM) images showed in Fig. 1 were performed on a JEOL JEM-2100 microscopy, which indicates that the TiO₂ NRs have smooth surface and two triangular protuberances at two ends. The TiO₂ NRs show a diameter of 27–41 nm and length range from 64 to 206 nm, and the average diameter and length of TiO₂ NRs are about 32 and 137 nm, respectively. The clear and regular lattice fringes in the HRTEM image (the inset) reveal the highly crystalline nature of TiO₂ NRs. The measured distance between two adjacent fringes is 0.33 nm, being agreement with the (110) plane of rutile TiO₂ [14].

The *I–V* characteristics of device A–C in Fig. 2 reflect the performance of devices, and the detailed photovoltaic parameters are listed in the inset table. The measurements were carried out by using Keithley model 2602 A digital source meter, and A BOS-X-1000G solar simulator was used to give irradiance of 100 mW cm^{-2} . When the TiO₂ NRs content in CTTFs increased from 0 to 10 wt%, an obvious increase in the short-circuit current densities (J_{sc}) and conversion efficiency (η) were observed, and open-circuit voltage (V_{oc}) and fill factor (FF) had a slight change. The device C based on 10 wt% TiO₂ NRs CTTF electrode, whose J_{sc} , V_{oc} , FF and η were 17.72 mA cm⁻², 726 mV, 0.617 and 7.95%, respectively, exhibited the highest performance. Generally, the enhanced performance of DSCs can be mainly ascribed to two reasons, including the increasing ability of dye absorption and light harvesting of photoanode, and the improvements in electrical transport property. As shows in Fig. 3(a) the UV-vis absorption spectrums of CTTFs with different composites were monitored on a TU-1810PCS spectrophotometer to investigate (not-sensitized films as reference) the light-harvesting capacity. The peaks near 550 nm correspond to the characteristic absorption of C106 dye [15]. It can be seen that the light harvesting efficiency of CTTFs decreases slightly with the increasing of TiO₂ NRs content. Such result can be explained by the reduction of the absorbed dve molecules due to the decrease of surface area of the CTTF when the content of TiO₂ NRs increased. On the contrary, with the increasing of TiO₂ NRs content, the performance of devices enhanced. Thus, the electrical transport property enhanced by



Fig. 1. TEM and HRTEM (inset) images of the TiO₂ nanorods.



Fig. 2. *J*-*V* characteristics of devices measured under AM1.5G sunlight (inset Table: Detailed photovoltaic parameters).

the TiO_2 NRs may play a key role in improving the performance of DSCs.

The charge recombination rate constant (K_r) , density of states (DOS) and chemical capacitance (C_{μ}) of devices A–C were studied by transient photoelectrical decay measurement [15]. As shown in Fig. 3(b), the K_r values for all the three devices increase exponentially with the increasing of $V_{\rm oc}$. Moreover, device A shows the highest K_r value followed by device B and C. For example, at $V_{\rm oc}$ =616 mV, the $K_{\rm r}$ values of devices A–C are 59.3, 47.2 and 31.9 s⁻¹, respectively. The higher K_r value indicates quicker charge recombination at TiO₂/electrolyte interfaces and more wastage of photogenerated electrons, which finally leads to the poor photovoltaic performance of device A. The TiO₂ NRs can provide direct electron transport path for photoelectron and then promote the electron transport. As a result, the K_r in TiO₂/electrolyte interface for device C was reduced, and the photoelectron can be transported more quickly to external circuit. The curves of DOS and C_{μ} vs V_{oc} for the DSCs with different electrodes are showed in Fig. 3(c). The DOS and C_{μ} increase exponentially with the increasing of V_{oc} , and the DOS and C_{μ} values of device C is maximum and that of device A is minimum under the same $V_{\rm oc}$ condition. The higher DOS value of device C probably due to more electrons gather in the TiO_2 /electrolyte interface ascribing its lower K_r value than other devices. Under the certain light, the TiO₂ surface states and trap states are increased with the increasing of DOS. It may be noted that the higher DOS of device C should give a lower V_{oc} . However, this does not occur because the TiO₂ NRs also act to strongly reduce the $K_{\rm r}$.

In order to further confirm the TiO₂ NRs content-dependent enhancement of the electron transport ability of the TiO2 NCs/NRs based thin film electrode, the electrochemical impedance measurement was performed on the electrochemical workstation (IM6ex, Germany, Zahner) in the frequency range of 1 mHz-00 kHz. A Z-view data processing software was applied to calculate the kinetic parameters of electronic transport [16]. Fig. 4 shows the relations between electron lifetime ($\tau_{\rm n}$), electron diffusion coefficient (D_n) and electron diffusion length (L_n) vs applied bias voltage ($V_{\rm b}$). It can be seen that under the same $V_{\rm b}$, the values of τ_n and L_n increase and D_n slightly decrease with the increasing of TiO₂ NRs content. When the $V_{\rm b}$ is 500 mV, the $\tau_{\rm n}$ of device C based on 10 wt% TiO₂ NRs CTTF electrode was 12.50 ms, which is about 8 times to pure TiO₂ NCs electrode (1.45 ms). High τ_n value means an effective inhibition for the charge recombination reaction, which is consistent with the variation of K_r . The L_n in device has been estimated by measuring D_n and τ_n : $L_n = (D_n \tau_n)^{1/2}$ [16]. The $L_{\rm n}$ of device C is larger than that of device A and B. The electron

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