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Fabrication of wrinkled Nb-doped TiO_2 nanofibres via electrospinning

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ABSTRACT

Nb-doped TiO_2 (NTO) nanofibres (NFs), ranging from cylindrical NFs to wrinkled NFs, were synthesised using an electrospinning method. Mechanisms leading to the formation of structures with different properties were demonstrated. In order to synthesise wrinkled NTO NFs, the mole ratios of the Nb precursor to the Ti precursor were controlled to be 0, 0.09, 0.23, 0.38, and 0.59. The NTO NFs prepared from the mole ratio of 0.59 had an excellent wrinkled structure with a high specific surface area and phase transformation from an anatase structure to a rutile structure.

1. Introduction

Recently, one-dimensional (1-D) nanostructures have attracted great attention because of their peculiar optical, electrical, chemical, and electrochemical properties, as well as their uses in various applications such as biodiagnostics, electrochemistry, optics, and electronics [1,2]. Much effort has been made to fabricate 1-D nanostructures with various morphologies, such as multi-segment nanowires (NWs), nanorods, hollow NWs, core–shell NWs, and branched NWs [1,3–5]. Various synthetic methods, such as template-directed synthesis, lithography, chemical vapour deposition, vapour phase synthesis, and electrospinning, have been developed in order to obtain such 1-D nanostructures [5–7]. Among these methods, electrospinning, which produces very fine fibres in the range 20–1000 nm, is a very attractive technique for making 1-D nanostructures with various morphologies. This method is used in bio-pharmaceuticals, filters, sensors, electronics/photovoltaics, and energy-conversion systems because of its advantages such as cost effectiveness, good repeatability, control of fibre dimensions, and large-scale production [8]. The morphologies of 1-D nanostructures can have direct effects in the above-mentioned applications and, by extension, the morphological modification of 1-D nanostructures by electrospinning has become an important issue. For example, Li et al. reported the fabrication of TiO_2 hollow nanofibres (NFs) through co-electrospinning using two immiscible liquids [9]. Kurban et al. controlled the morphologies to give porous fibres and cylindrical fibres via co-electrospinning together with a solvent-selection system, for use in hydrogen-storage materials [10]. An et al. recently reported the synthesis of carbon NF (CNF) composites consisting of SnO_2 and SiO_2 nanoparticles by co-electrospinning; the CNF nanocomposites had a rough and dense surface, which could be controlled by varying the weight percentage ratios of SnO_2 and SiO_2 nanoparticles [11]. In addition, among various transparent conductive oxides (TCOs) such as Sn-doped In_2O_3 (ITO) and F-doped SnO_2 (FTO), Nb-doped TiO_2 (NTO) materials have been receiving much attention as promising TCO materials because of their low resistivities (9.5×10^{-4} and $2–3\times10^{-4}\ \Omega$ cm) and high transmittances (60–85% and 77–97% in the visible region) using sputtering and pulsed laser deposition [12,13]. However, wrinkled NTO NFs have not yet been reported.

In this study, we successfully synthesised wrinkled NTO NFs by an electrospinning technique. We discuss the mechanisms leading to the formation of different morphologies, ranging from cylindrical NFs to wrinkled NFs.

2. Experiments

Titanium (IV) isopropoxide ($Ti[OCH(CH_3)_2]_4$, Aldrich, 97%), poly (vinylpyrrolidone) ($(C_6H_9NO)n$, $M_w=1\ 300\ 000$ g/mol, Aldrich), and acetic acid (CH_3CO_2H, Aldrich, 99.7%) were dissolved in N,N-dimethylformamide ($HCON(CH_3)_2$, Aldrich, 99%) by stirring for 1 h. Niobium (V) ethoxide ($Nb(OCH_2CH_3)_5$, Aldrich, 99.9%) was then added to the Ti precursor solution. In order to obtain wrinkled NTO NFs, the relative mole ratios of the Nb to the Ti precursor were controlled to be 0, 0.09, 0.23, 0.38, and 0.59 (referred to as single TiO_2 NFs, and samples A–D). The as-prepared solution was loaded into a 12 mL standard syringe equipped with a 23-gauge stainless-steel needle. The feeding rate was fixed at 0.04 mL/h, using a syringe pump. The voltage of the power supply was ~14.5 kV. The distance between the needle tip and the collector was maintained at ~20 cm. All of the as-spun samples were heat-treated at 500 °C

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for 5 h in air. We obtained NTO NFs with a range of morphologies, such as cylindrical NFs and wrinkled NFs.

The structures and crystallinities of the samples were examined using X-ray diffraction (XRD, Rigaku Rint 2500) with Cu K_α radiation in the range of 10–90°. The morphological changes and structural properties of the samples were investigated using field-emission scanning electron microscopy (FESEM, Hitachi S-4700) and transmission electron microscopy (TEM, JEOL-2100F, KBSI Suncheon Center) operated at 200 kV. The surface areas of the NFs were measured using the Brunauer–Emmet–Teller (BET, Micromeritics ASAP2010) method.

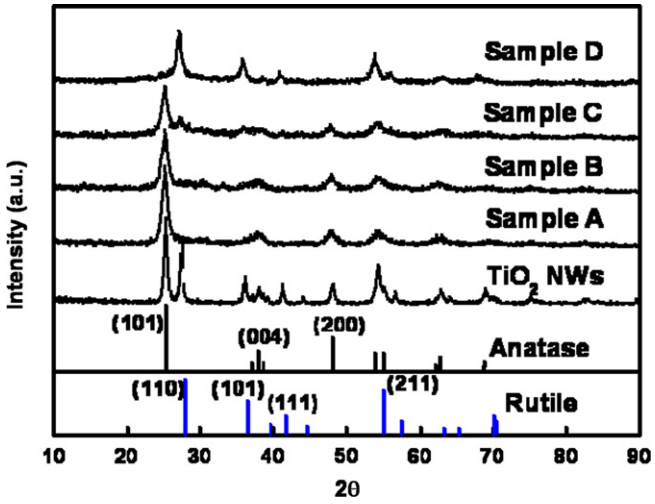

Fig. 1. XRD data of single TiO_2 NFs, and samples A–D after calcination.

3. Results and discussion

Fig. 1 shows the XRD patterns of single TiO_2 NFs, and of samples A–D, obtained after calcination. In the case of the single TiO_2 NFs, XRD peaks were observed at 25.3°, 37.8°, 48.0°, 53.9°, 62.7°, and 68.8°, corresponding to the (101), (004), (200), (105), (204), and (116) planes of anatase TiO_2 phases (JCPDS no. 841286) and at 27.9°, 36.4°, 41.7°, and 55.1°, corresponding to the (110), (101), (111), and (211) planes of rutile TiO_2 phases (JCPDS no. 881175). For samples A–C, the XRD peaks indicate that the anatase TiO_2 phases decreased gradually as the amount of Nb doping increased, and the diffraction peaks relative to the (101) planes of samples A–C shifted gradually to lower 2θ values with increasing amounts of Nb doping; this is because the ionic radius of Nb^{5+} (0.62 Å) is bigger than that of Ti^{4+} (0.56 Å), and can be explained using the Bragg equation ($\lambda = 2d \sin\theta$). In particular, sample D shows that at a critical level of Nb doping, phase transformation from anatase TiO_2 to rutile TiO_2 occurred, as shown in Fig. 1. This phenomenon could be explained as grain growth suppression in the NFs as a result of the critical amount of Nb doping, which implies preferential growth orientation of the NFs [14].

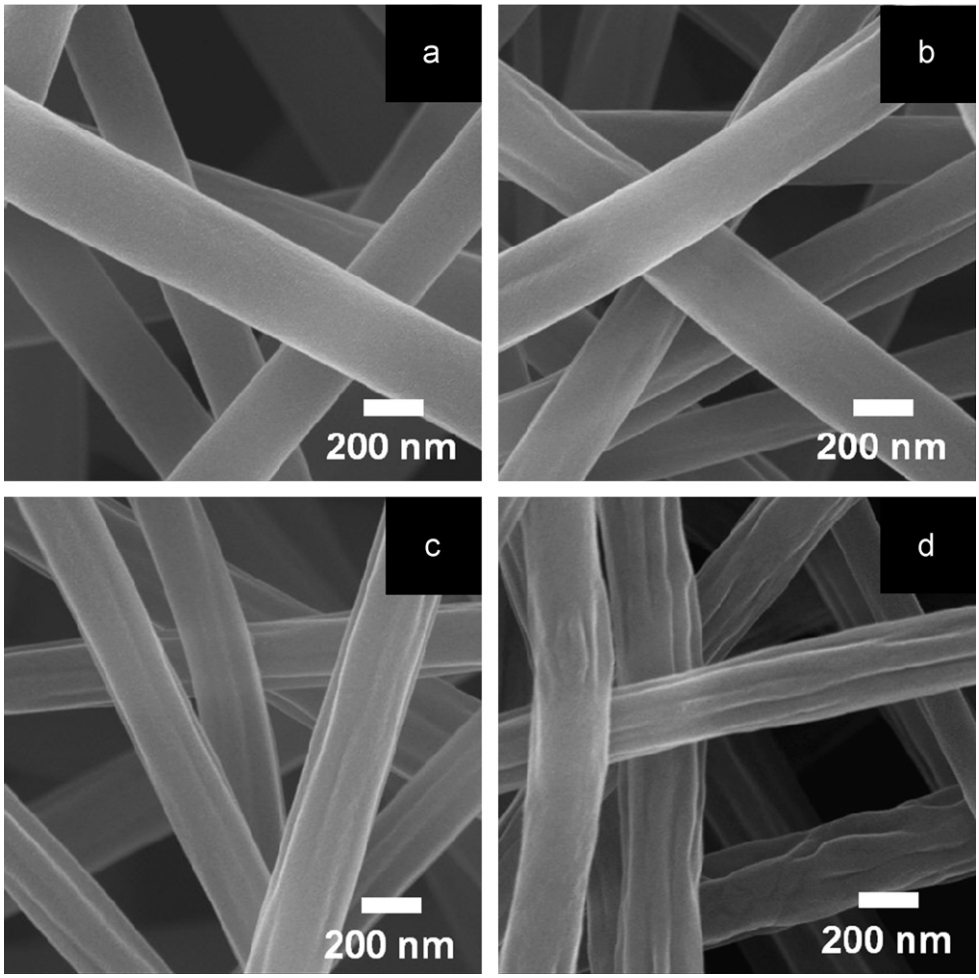

Fig. 2. SEM images of as-spun samples A–D (a–d) before calcination.