



Electrical characteristics of thermal CVD B-doped Si films on highly strained Si epitaxially grown on Ge(100) by plasma CVD without substrate heating

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ABSTRACT

Using an 84% relaxed Ge(100) buffer layer formed on Si(100) by electron cyclotron resonance (ECR) plasma enhanced chemical vapor deposition (CVD), influence of strain upon electrical characteristics of B-doped Si film epitaxially grown on the Ge buffer have been investigated. For the thinner B-doped Si film, surface strain amount is larger than that of the thicker film, for example, strain amount reaches 2.0% for the thickness of 2.2 nm. It is found that the hole mobility is enhanced by the introduction of strain to Si, and the maximum enhancement of about 3 is obtained. This value is higher than that of the usually reported mobility enhancement by strain using $\text{Si}_{1-x}\text{Ge}_x$ buffer. Therefore, introduction of strain using relaxed Ge film formed by ECR plasma enhanced CVD is useful to improve future Si-based device performance.

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1. Introduction

Atomically controlled heterostructure growth of group IV semiconductor is important to improve Si-based device performance [1,2], because nanometer-order thick film with atomically flatness and uniformity are necessary to achieve highly strained heterostructures. For example, strained Si is an attractive material to enhance channel mobility in Si MOSFET [3], and it is obtained by lattice matching of Si to $\text{Si}_{1-x}\text{Ge}_x$ buffer using heteroepitaxial growth. Using $\text{Si}_{1-x}\text{Ge}_x$ ($x < 0.8$) buffer layer, hole mobility enhancement was observed [6]. In our previous work, we have reported that, using electron cyclotron resonance (ECR) plasma enhanced chemical vapor deposition (CVD) without substrate heating, epitaxial growth of relaxed Ge on Si(100) [4] and undoped highly strained Si on 84% relaxed Ge/Si(100) (u-Si/rGe/Si(100)) [5] with atomically flat surface was achieved by suppression of incident ion energy. In this work, using a relaxed Ge buffer layer formed by ECR plasma enhanced CVD, introduction of higher strain to epitaxially grown B-doped Si and mobility enhancement in B-doped strained Si (B-sSi) was investigated.

2. Experimental

Substrate used was p-type Si(100)-on-insulator (SOI) wafer. The resistivity and thickness of SOI are 20–30 $\Omega \text{ cm}$ and 50–70 nm, respectively. Therefore, sheet resistance of SOI was about 4 $\text{M}\Omega/\square$. Epitaxial growth of u-Si/rGe on the SOI was performed by ECR plasma CVD system shown in Fig. 1 [4,5]. The relaxed Ge and undoped strained Si were grown by GeH_4 and SiH_4 reaction, respectively, under

ECR Ar plasma irradiation from a plasma generating chamber. For the 53 nm-thick relaxed Ge deposition, ion flux density and peak energy of the incident Ar ions were about $2 \times 10^{16} \text{ cm}^{-2} \text{ s}^{-1}$ and 3 eV,

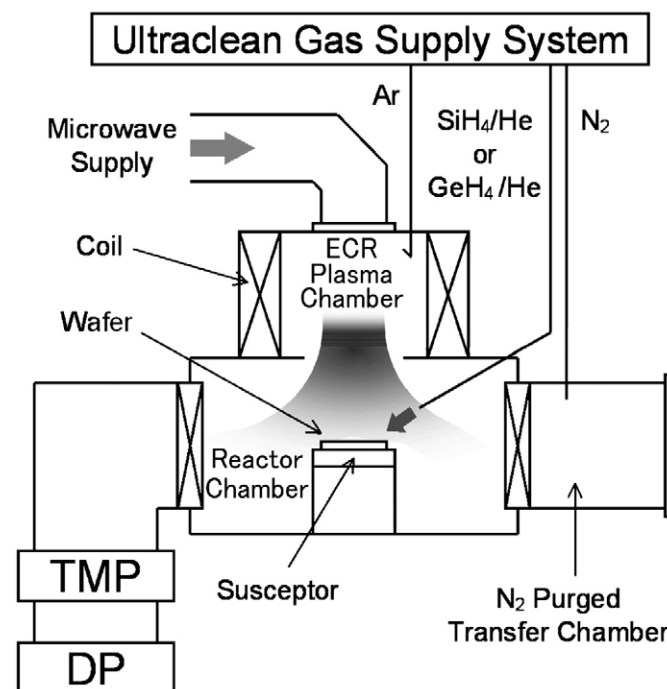


Fig. 1. Schematic of ultraclean ECR Ar plasma enhanced CVD system for epitaxial growth of u-Si/rGe on SOI. Vacuum pumping was done by a turbo-molecular pump (TMP) and a dry pump (DP).

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respectively, and for the 1.7 nm-thick undoped strained Si deposition, these values were about $4 \times 10^{15} \text{ cm}^{-2} \text{ s}^{-1}$ and 0.5 eV, respectively [5]. After u-sSi/rGe deposition, the samples were keeping in N_2 atmosphere due to minimization of the surface oxidation before B-doped strained Si growth. Epitaxial growth of B-doped strained Si was performed by ultraclean low-pressure CVD (LPCVD) in a $\text{SiH}_4\text{--B}_2\text{H}_6\text{--H}_2$ gas mixture at 500°C [7]. The samples were dipped in 2% diluted HF and carried in

LPCVD system through the N_2 purged transfer chamber at room temperature. Then, a reactor chamber was heated up to 500°C . At 500°C , strain relaxation of undoped strained Si and intermixing at heterointerface in u-sSi/rGe/Si structure were scarcely expected for the heat-treatment [5]. For all of B-doped strained Si deposition, partial pressure of H_2 , SiH_4 and B_2H_6 were fixed at 24, 6 and $2 \times 10^{-4} \text{ Pa}$, respectively.

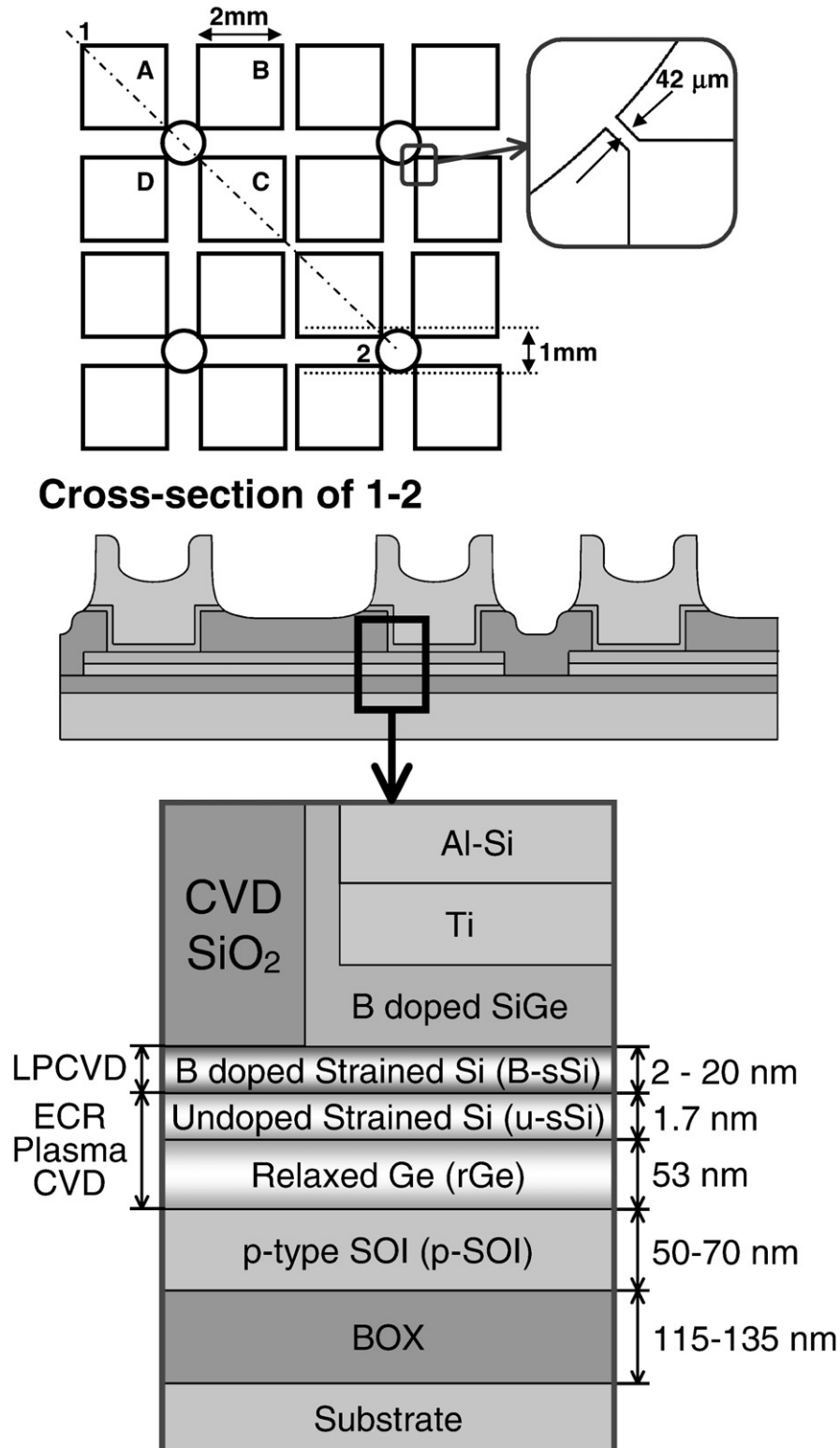


Fig. 2. Schematic of sample structure measured by van der Pauw method for Hall effects measurement.

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