

# Influence of the reactive N<sub>2</sub> gas flow on the properties of rf-sputtered ZnO thin films

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## Abstract

Nitrogen (N)-doped ZnO thin films were RF sputtered with different N<sub>2</sub> volume (ranging from 10% to 100%) on sapphire (001) substrates. The influence of N<sub>2</sub> vol.% on the properties of ZnO films was analyzed by various characterization techniques. The X-ray diffraction studies showed that the films grow along the preferential (002) crystallographic plane and the crystallinity varied with varying N<sub>2</sub> vol.%. The films sputtered with 25 vol.% N<sub>2</sub> showed better crystallinity. The transmittance was decreased with increasing N<sub>2</sub> volume until 25% and was almost constant above 25%. A maximum optical band gap (2.08 eV) obtained for 10 vol.% N<sub>2</sub> decreased with increasing N<sub>2</sub> volume to reach a minimum of 1.53 eV at 100%. The compositional analysis confirmed the incorporation of N into ZnO films, and its concentration increased with increasing N<sub>2</sub> volume to reach a maximum of  $\sim 3.7 \times 10^{21}$  atom/cm<sup>3</sup> at 75% but then decreased slightly to  $3.42 \times 10^{21}$  atoms/cm<sup>3</sup>. The sign of Hall coefficient confirmed that the films sputtered with  $\leq 25$  vol.% N<sub>2</sub> possess p-type conductivity which changes to n-type for  $> 25$  vol.% N<sub>2</sub>.

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## 1. Introduction

Due to wide direct-band-gap (3.37 eV) and large exciton binding energy ( $\sim 60$  meV), the importance of ZnO thin films in opto-electronic devices have attracted more attention. Many growth techniques such as sputtering, molecular beam epitaxy, metal–organic chemical vapor deposition, pulsed laser deposition, etc., have been employed to study ZnO in detail [1–4]. Among these techniques, sputtering has been widely used to study ZnO films owing its low cost, simplicity, low processing temperature and easy doping. The difficulty in obtaining p-type ZnO, which is a barrier for their applications in short-wavelength light-emitting or laser diodes, is still being explored. In order to realize p-type ZnO films, various group-I and -V elements in the periodic table have been employed [1,2,5–7] and some research groups have claimed that p-type ZnO has been obtained. However, few available reports [2,6,8] including a recent report on the RT electroluminescence from

ZnO p–n homo-junction do not show both high hole concentration and mobility. A theoretical study shows that nitrogen (N) dopant is the optimal candidate for obtaining p-type ZnO [9]. Further, many N sources such as N<sub>2</sub>, NO, N<sub>2</sub>O and NH<sub>3</sub> have also been studied [2,8,10–13]. Among the available N sources, an easy getting, economic and non-toxic N<sub>2</sub> has been widely used in sputtering techniques [12,13]. Hitherto, a systematic investigation on the influence of nitrogen volume percentage (N<sub>2</sub> vol.%) on the properties of ZnO was not carried out. In this context, N-doped ZnO films have been RF sputtered with different N<sub>2</sub> vol.% at room temperature. The influence of N<sub>2</sub> vol.% on the structural, morphological, optical, compositional and electrical properties of ZnO films was studied in detail.

## 2. Experimental details

ZnO films were RF sputtered on sapphire (001) substrate at room temperature from a ceramic ZnO target. The films were studied as a function of N<sub>2</sub> vol.%, which was estimated from the N<sub>2</sub> and Ar gas flows during sputtering. The ratio of N<sub>2</sub> and Ar

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gas flows was maintained at 4:36, 10:30, 20:20, 30:10 and 40:0 sccm for obtaining the N<sub>2</sub> vol.% of 10, 25, 50, 75 and 100, respectively. In each case, the total flow of the gas was fixed at 40 sccm. The chamber was evacuated initially to  $\sim 2 \times 10^{-6}$  mbar. All the films were sputtered with a constant power of 100 W at a total pressure of  $1.2 \times 10^{-2}$  mbar for 60 min. A radiofrequency power generator (13.6 MHz) from Advanced Energy (Model: RFX 2500) was used for sputtering the target. The thickness of the films was measured using a surface profilometer (Dektak3). The crystal structure of the films was confirmed using an X-ray diffractometer (DMAX-III C from Rigaku; sealed tube, Cu K<sub>α</sub> radiation) in Bragg–Brentano geometry ( $\theta/2\theta$  coupled). Optical transmittance ( $T$ ) was measured using a double-beam spectrophotometer (Shimadzu UV-3100). The surface morphology was analyzed using scanning electron microscopy (SEM). The nature of chemical surface was detected through X-ray photoelectron spectroscopy (XPS). The profile of the composition Vs depth was determined by secondary-ion mass spectrometry (SIMS) using Cs<sup>+</sup> primary beam. The electrical parameters were estimated using a Hall measurements setup (Bio-Rad HL5500 Hall system) with a permanent magnet of 5 kG in van der Pauw configuration.

### 3. Results and discussion

The average value of film thickness found varying between 1.5 and 2.2  $\mu\text{m}$ . A minimum thickness (1.5  $\mu\text{m}$ ) obtained for 10 vol.% N<sub>2</sub> increased with increasing N<sub>2</sub> volume and reached a maximum of 2.2  $\mu\text{m}$  at 100%. X-ray diffraction (XRD) patterns of the films were recorded in the  $2\theta$  ranging  $30^\circ$ – $75^\circ$ . A strong diffraction peak from the sapphire substrate at  $2\theta$  around  $42^\circ$  concealed the visibility of other peaks to noise levels. Hence, the samples were scanned at two discrete  $2\theta$  ranges of  $30^\circ$ – $40^\circ$

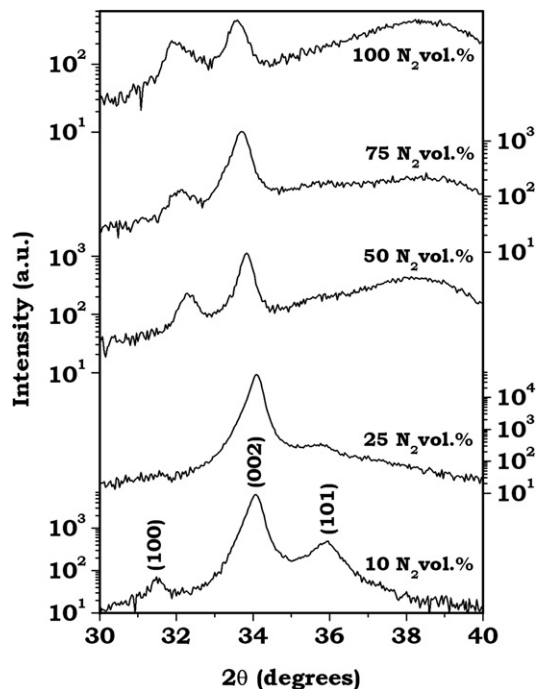


Fig. 1. XRD patterns of the N-doped ZnO films as a function of N<sub>2</sub> vol.%.

Table 1

Comparison of structural and electrical parameters of N-doped ZnO films

N <sub>2</sub> (vol.%)	Data obtained from (002) diffraction peak			Data from Hall measurements			
	Angle, $2\theta$ (deg)	$c$ (Å)	FWHM (deg)	Crystallite size (nm)	$\rho$ ( $\Omega$ cm)	$\mu$ ( $\text{cm}^2/\text{V s}$ )	$n$ ( $\text{cm}^{-3}$ )
10	34.06	5.26	0.328	25.05	$1.90 \times 10^2$	257	$+1.27 \times 10^{14}$
25	34.08	5.26	0.223	36.85	$5.57 \times 10^0$	22	$+5.18 \times 10^{16}$
50	33.84	5.29	0.325	25.27	$3.29 \times 10^{-1}$	17	$-1.10 \times 10^{18}$
75	33.72	5.31	0.389	21.10	$2.06 \times 10^{-1}$	15	$-2.01 \times 10^{18}$
100	33.59	5.33	0.524	15.66	$1.30 \times 10^{-1}$	19	$-2.48 \times 10^{18}$

$c$  – lattice parameter; FWHM – full-width at half-maximum;  $\rho$  – bulk resistivity;  $\mu$  – Hall mobility;  $n$  – carrier concentration.

and  $45^\circ$ – $75^\circ$ , respectively. In the  $45^\circ$ – $75^\circ$   $2\theta$  range, a very weak peak obtained at around  $72^\circ$  (10 vol.% N<sub>2</sub>) was confirmed as a secondary diffraction peak from (002) by matching with a standard hexagonal ZnO (ICDD card no. 36-1451). This peak was disappeared when the N<sub>2</sub> vol.% increased to 25. To authenticate the influence of N<sub>2</sub> vol.% on the samples, the XRD patterns in the  $30^\circ$ – $40^\circ$   $2\theta$  range are shown in Fig. 1. It may be noteworthy that the  $y$ -axis is given in logarithmic scale due to the very high intensity (several thousands arbitrary units) of (002) diffraction peak. For 10 vol.% N<sub>2</sub> films, a strong peak was obtained from ZnO (002) plane showing a preferential orientation. Additionally, two other peaks from (100) and (101) planes were obtained. When the N<sub>2</sub> vol.% increased to 25, the intensity of (002) peak increased by about an order of magnitude and the (100) and (101) diffraction peaks disappear. The increased intensity of preferred <002> orientation substantiates the improvement in crystallinity. The intensity of (002) peak decreases with increasing N<sub>2</sub> volume above 25% and the peak perceptibly shifts towards lower  $2\theta$  angle side, which probably suggests that the crystallinity decreases due to the increase in N defects concentration and tensile stress in the films. In addition, at high N<sub>2</sub> volume (>25%), two peaks around  $32^\circ$  and  $38^\circ$  (broad peak) were obtained. These additional peaks were not matched either with the standard data or with previous reports. The XRD data are summarized in Table 1 together with the electrical properties. The lattice parameter (5.26 Å) obtained for  $\leq 25$  vol.% N<sub>2</sub> increased with increasing N<sub>2</sub> volume to a maximum of 5.33 Å at 100 vol.% N<sub>2</sub>. The crystallite size calculated using peak width increased from 25.05 to 36.85 nm for the increase in N<sub>2</sub> volume from 10% to 25%, but then decreased with increasing N<sub>2</sub> volume and reached a minimum of 15.66 nm at 100%. The variation in crystallite size probably indicates that the low N<sub>2</sub> volume ( $\leq 25\%$ ) promotes the grain growth and reduces the tensile stress. This may probably due to the effect of compensation of N defects by O vacancies at low N<sub>2</sub> concentration. When the N<sub>2</sub> volume increased above 50%, the grain growth is suppressed by the increased tensile stress [13].

The surface morphology obtained from SEM studies is comparatively shown in Fig. 2 for various N<sub>2</sub> volume. For 10 vol.% N<sub>2</sub> films, spherical shaped grains with the size varying between  $\sim 15$  and 40 nm are seen on the surface. The grains are

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