



# Polarization dependence of the optical response in SnO<sub>2</sub> and the effects from heavily F doping

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## ABSTRACT

The optical properties of intrinsic SnO<sub>2</sub> (TO) and fluorine doped (FTO) are characterized in terms of the dielectric function  $\varepsilon(\hbar\omega) = \varepsilon_1(\hbar\omega) + i\varepsilon_2(\hbar\omega)$  by electronic structure calculations. The intrinsic TO shows intriguing absorption characteristics in the 3.0–8.0 eV region: (i) the low energy region of the fundamental band gap ( $3.2 < \hbar\omega < 3.9$  eV), the optical transitions  $\Gamma_3^+ \rightarrow \Gamma_1^+$  (valence-band maximum to conduction-band minimum) is symmetry forbidden, and the band-edge absorption is therefore extremely weak. (ii) In the higher energy region ( $3.9 < \hbar\omega < 5.1$  eV) the  $\Gamma_5^- \rightarrow \Gamma_1^+$  transitions (from the second uppermost valence band) is strongly polarized perpendicular to the main *c* axis. (iii) Transitions with polarization axis parallel to *c* axis are generated from  $\Gamma_2^- \rightarrow \Gamma_1^+$  transitions (from the third uppermost valence bands), and dominates at high energies ( $5.1 < \hbar\omega$  eV). Heavily F doped TO (FTO) with doping concentrations  $n_F = 4 \times 10^{20} \text{ cm}^{-3}$  changes the absorption significantly: (iv) Substitutional F<sub>O</sub> generates strong inter-conduction band absorption at 0.8, 2.2, and 3.8 eV which affects also the high frequency dielectric constant  $\varepsilon_{\infty}$ . (v) Interstitial F<sub>i</sub> is inactive as a single dopant, but act as a compensating acceptor in highly *n*-type FTO. This explains the measured non-linear dependence of the resistivity with respect to F concentration.

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## 1. Introduction

Cassiterite tin oxide SnO<sub>2</sub> (TO) is an important metal-oxide for efficient dielectrics, electric resistors, catalysis, optical sensor devices, coating oxide technologies [1], and also in gas sensor technologies [2]. Fluorine doped TO (FTO) thin films are inexpensive (compared to indium tin oxide) with conducting-transparent properties of the oxide, which opens for a broad range of low-cost photovoltaics and optoelectronics [3]. FTO can easily be heavily F doped in the order of  $10^{20}$ – $10^{21} \text{ cm}^{-3}$  showing intriguing resistivity characteristic with respect to the F concentration  $n_F$ . Resistivity has been found to decrease for moderate doping concentrations  $0 < n_F < 10^{20} \text{ cm}^{-3}$ , but increases for doping concentrations  $n_F > (1-3) \times 10^{21} \text{ cm}^{-3}$  [4,5]. Optical properties of TO and FTO have been studied in details both experimentally and theoretically [4–12]. However, still the fundamental band-gap energy  $E_g$  is not well established ( $E_g \approx 3.2$ – $4.7$  eV [8,9,11,13–18]), and the impact on the resistivity and optical response due to heavily F doping is not fully understood.

In this work, we use density-functional theory to investigate the linear optical response of TO and FTO in terms of the complex dielectric function  $\varepsilon(\hbar\omega) = \varepsilon_1(\hbar\omega) + i\varepsilon_2(\hbar\omega)$  with respect to the photon energy  $\hbar\omega$ . We employ the projector augmented wave (PAW) method [19]

within the local density approximation (LDA) and with a modeled on-site self-interaction-like correction (SIC) potential within the LDA + U<sup>SIC</sup>. We analyze the calculated dielectric function  $\varepsilon(\hbar\omega)$ , refractive index  $\tilde{n}(\hbar\omega) = n(\hbar\omega) + ik(\hbar\omega)$  and the absorption coefficient  $\alpha(\hbar\omega)$  for TO and FTO, and from this modeling we better understand the resistivity with respect to high F doping of TO. For instance, as a complement to earlier suggestions that the high resistivity at very high F content is due to lattice distortion, we demonstrate that this effect is a consequence of the site preference F dopants. That is, whereas moderate F doping generates *n*-type donors with F mainly on the oxygen site (F<sub>O</sub>), the heavily F doped FTO generates also interstitial dopants (F<sub>i</sub>) that forms neutralizing F<sub>O</sub>–F<sub>i</sub> complexes. Thereby, F<sub>i</sub> which is a rather inactive atom in the low F concentration, acts as compensating acceptor when FTO is heavily *n*-type doped. Therefore, we suggest that the increased resistivity in the highly doped FTO is explained by the presence of the F<sub>i</sub> dopants. This is an intriguing situation where the material generates a self-compensation of an otherwise to strong *n*-type character.

Moreover, the variation in measured  $E_g$  of TO is explained by our theoretical analysis of the valence- to conduction-band optical transitions. We find that the low energy band-edge absorption at  $3.2 < \hbar\omega < 3.9$  eV is symmetry forbidden, and is thereby difficult to detect in the absorption measurements, explaining the difficulties to establish the band-gap value. Finally, we report a very strong polarization dependence of the optical response. We suggest that this can be beneficially utilized in future technological applications where polarization is a sensitive detector parameter.

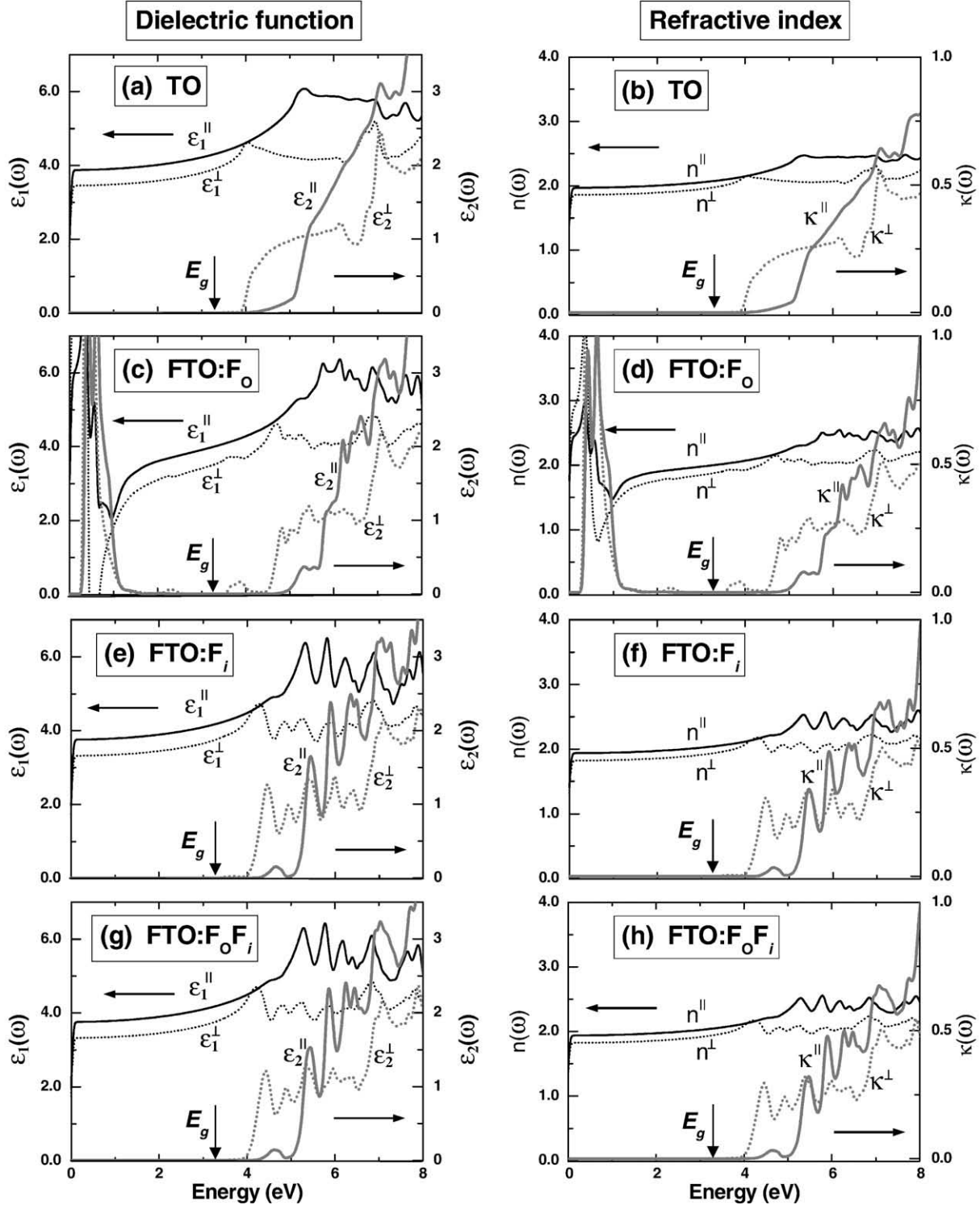
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## 2. Computational details

TO crystallizes in space group  $D_{4h}^{14}$ . F substituting an O site (i.e.,  $F_O$ ) lowers the crystallographic symmetry  $C_{2v}$  and F on an interstitial site [i.e.,  $F_i$  at  $(1/2, 0, 1/2)$  or  $(0, 1/2, 1/2)$ ] to  $C_{2h}$ . The theoretical analysis is performed with the PAW/LDA +  $U_{\text{SIC}}$  method [19] using  $U_d(\text{cation}) = 10$  eV and  $U_s(\text{anion}) = -6$  eV which corrects the LDA band gap and

improve the localization of cation  $d$ -states [20,21]. Doping is modeled by  $3 \times 3 \times 4$  supercells of originally 216 atoms, considering the experimental lattice parameters of TO:  $a = 4.737$  Å,  $c = 3.185$  Å, and  $u = 0.307$  [13]. Three different types of heavily F doped systems are investigated: TO with a F substituting oxygen (FTO: $F_O$ ;  $n_F = 3.9 \times 10^{20} \text{ cm}^{-3}$ ), TO with a F interstitial (FTO: $F_i$ ;  $n_F = 3.9 \times 10^{20} \text{ cm}^{-3}$ ), and TO with  $F_O$  plus  $F_i$  (hFTO: $F_O + F_i$ ;  $n_F = 7.8 \times 10^{20} \text{ cm}^{-3}$ ). The ions of FTO are relaxed by means of



**Fig. 1.** The dielectric function  $\varepsilon(\hbar\omega) = \varepsilon_1(\hbar\omega) + i\varepsilon_2(\hbar\omega)$  (left column) and refractive index  $\tilde{n}(\hbar\omega) = n(\hbar\omega) + i\kappa(\hbar\omega)$  (right column) for TO and FTO show strong polarization dependence. The fundamental band gap is  $E_g \approx 3.2$  eV for all both TO and FTO but onset to optical absorption across the gap occurs at around 4 eV with predominantly perpendicular polarization.  $F_O$  is a donor (c–d) changing the optical response. Single  $F_i$  is inactive dopant (e–f) but acts as a compensating acceptor in highly n-type FTO (g–h). A Lorentzian broadening of 40 meV was used for  $\varepsilon(\hbar\omega)$  and  $\tilde{n}(\hbar\omega)$ .

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