

Structural, electrical and optical properties of sputter-deposited Nb-doped TiO₂ (TNO) polycrystalline films

Naomi Yamada^{a,*}, Taro Hitosugi^{a,b}, Ngoc Lam Huong Hoang^{a,b}, Yutaka Furubayashi^a, Yasushi Hirose^a, Seiji Konuma^a, Toshihiro Shimada^{a,b}, Tetsuya Hasegawa^{a,b}

^a Kanagawa Academy of Science and Technology (KAST), Kawasaki 213-0012, Japan

^b Department of Chemistry, University of Tokyo, Tokyo 113-0033, Japan

Available online 14 October 2007

Abstract

Transparent conductive oxide (TCO) films of polycrystalline Nb-doped TiO₂ (TNO) were fabricated by post-annealing reactively sputtered amorphous films under H₂ atmosphere. Carrier transport properties of the H₂-annealed films were found to be strongly dependent on substrate temperature T_s and oxygen content in sputtering atmosphere. A minimum resistivity (ρ) of $9.5 \times 10^{-4} \Omega \text{ cm}$ and an average visible transmittance of $\sim 75\%$ were obtained at $T_s = RT$ and oxygen content of 10%. This ρ value is of the same order as those of epitaxial TNO films, indicating that polycrystalline TNO has sufficient potential as a practical TCO suitable for large-area applications.

© 2007 Elsevier B.V. All rights reserved.

Keywords: Transparent conductive oxide; Nb-doped TiO₂; TNO; Anatase; Sputtering; Polycrystalline films; Post-annealing

1. Introduction

Use of transparent conductive oxides (TCOs) has been explosively expanded owing to the rapid growth of flat panel display (FPD) market [1]. In particular, sputter-deposited Sn-doped In₂O₃ (ITO) has been established as a practical TCO material because of its excellent resistivity ($\rho \sim 2 \times 10^{-4} \Omega \text{ cm}$) and visible transmittance (80–90%) [2]. However, indium has a shortage problem, stimulating researchers to seek new TCO materials free from indium.

Recently, we found that Nb- or Ta-doped TiO₂ epitaxial films grown on various single crystal substrates by pulsed laser deposition (PLD) exhibit low ρ ($< 3 \times 10^{-4} \Omega \text{ cm}$) and excellent internal transmittance ($> 90\%$) in the visible region [3–6]. This was followed by PLD growth of polycrystalline Nb-doped TiO₂ (TNO) films on glass, indicating ρ of $1.5 \times 10^{-3} \Omega \text{ cm}$ and transmittance T_r of 60–80% [7]. Lower ρ value down to $5 \times 10^{-4} \Omega \text{ cm}$ was attained by post-annealing PLD-grown amorphous films under pure H₂ atmosphere [8]. These are good demonstrations that TNO has sufficient potential as an ITO-alternative TCO material.

Practically, TCO films, including ITO, have been mainly fabricated by sputtering technique, which is suitable for low-cost and uniform coatings on large-area substrates. In order to establish TNO as a practical TCO material, thus, it is highly desirable to develop sputtering-based procedures for high-quality TNO films. Gillispie et al. succeeded in sputter-deposition of TNO epitaxial films on SrTiO₃ with ρ of $\sim 3 \times 10^{-4} \Omega \text{ cm}$ and T_r of $> 80\%$, which are comparable to those of epitaxial PLD films [9]. Moreover, we have achieved $\rho \sim 1 \times 10^{-3} \Omega \text{ cm}$ and $T_r = 60\text{--}80\%$ [10], by applying the above-mentioned post-annealing technique to sputter-deposited TNO films.

In this paper, we report electrical, optical and structural properties of TNO polycrystalline films, prepared by post-annealing sputter-deposited amorphous films, as functions of sputtering conditions, such as substrate temperature T_s and oxygen partial pressure during amorphous deposition. By comparing these properties between sputter- and PLD-grown films, material parameters that govern the transparent conductivity of TNO will be discussed.

2. Experimental details

Polycrystalline TNO films were fabricated by post-annealing sputtered amorphous films, as follows. The amorphous TNO

* Corresponding author. Tel.: +81 44 819 2081; fax: +81 44 819 2083.

E-mail address: n-yamada@ksp.or.jp (N. Yamada).

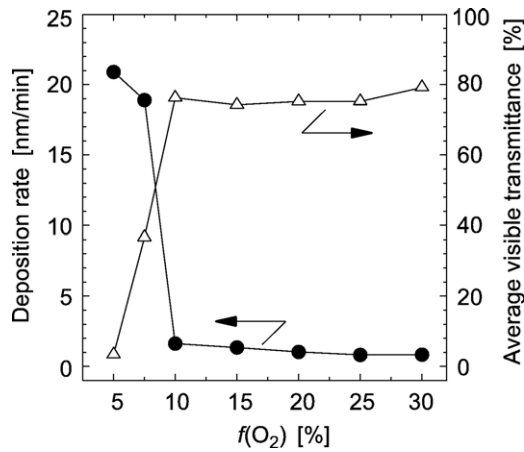


Fig. 1. Deposition rate (closed circle) and average visible transmittance (open triangle) of as-deposited TNO film (T_s =RT) as functions of $O_2/(Ar+O_2)$ flow ratio $f(O_2)$ during deposition of amorphous films.

thin films were deposited on heated ($T_s=150^\circ\text{C}$) or unheated (T_s =RT) alkali-free glass substrates (Corning #1737) by reactive DC magnetron sputtering. A $Ti_{0.94}Nb_{0.06}$ alloy disk of 50 mm was used as the target. Base pressure below 5×10^{-4} Pa was established, prior to each deposition run. Film deposition was conducted in a mixture of Ar and O_2 with various $O_2/(Ar+O_2)$ flow ratio, $f(O_2)$, ranging from 5 to 30% under a total pressure of 1.0 Pa. The DC power applied to the target was kept constant at 180 W during sputtering. Before the film deposition, the target surface was sputter-cleaned using pure Ar for 15 min in order to remove surface oxide layers and contamination, and was subsequently pre-sputtered for 10 min under the same condition as the film deposition. The typical deposition time and film thickness were 120 min and 120–150 nm, respectively. The as-deposited films were post-annealed in pure H_2 under the pressure of 1×10^5 Pa for 60 min in a rapid thermal annealing furnace, where the annealing temperature was raised to 600°C at a rate of $10^\circ\text{C}/\text{min}$.

Carrier transport properties, including resistivity (ρ), carrier density (n_e), and Hall mobility (μ_{H}), of the TNO thin films were determined using the van der Pauw method at room temperature. Structural properties were characterized by X-ray diffraction (XRD) using a diffractometer equipped with a two-dimensional detector (Bruker D8 Discover) and cross-sectional transmission electron microscopy (TEM). Surface morphologies of as-deposited and annealed films were observed by using an atomic force microscopy (AFM). Optical measurements were performed using a UV–VIS–NIR spectrophotometer in a wavelength region of 300–2300 nm.

3. Experimental results and discussions

3.1. Sputtering mode

Deposition rate (T_s =RT) and average visible transmittance ($\lambda=400\text{--}800$ nm) as functions of $f(O_2)$ are plotted in Fig. 1. The average visible transmittance T_{ave} over wavelengths

ranging from 400 to 800 nm is defined by the following formula,

$$T_{\text{ave}} = \frac{\int_{400}^{800} T_r(\lambda) d\lambda}{400}, \quad (1)$$

where, $T_r(\lambda)$ is a transmittance at a wavelength λ (in nm unit). Increase of $f(O_2)$ from 5 to 10% causes steep decrease of deposition rate and sudden improvement of transmittance from 0 to 75%. This is due to the transition of sputtering mode from metallic sputtering mode to the compound sputtering one [11]. We henceforth discuss the properties of transparent TNO films deposited at $f(O_2) \geq 10\%$.

3.2. Structural properties

θ – 2θ XRD patterns of TNO films prepared at $f(O_2)=10\%$ on unheated (T_s =RT) and heated ($T_s=150^\circ\text{C}$) substrates are

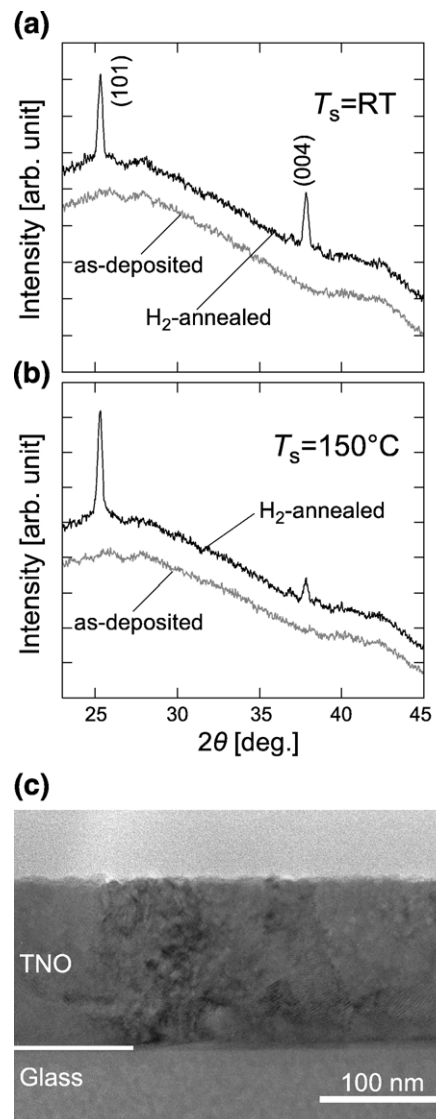


Fig. 2. X-ray diffraction patterns of as-deposited (gray lines) and H_2 -annealed TNO films (black lines) deposited at $f(O_2)=10\%$ on (a) unheated (T_s =RT) and (b) heated ($T_s=150^\circ\text{C}$) substrates. (c) Cross-sectional TEM image of H_2 -annealed films presented in (a).

Download English Version:

<https://daneshyari.com/en/article/1674582>

Download Persian Version:

<https://daneshyari.com/article/1674582>

[Daneshyari.com](https://daneshyari.com)