

## Study on effects of swift heavy ion irradiation on the crystal structure in CeO<sub>2</sub> doped with Gd<sub>2</sub>O<sub>3</sub>

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### ABSTRACT

To simulate the effects of Gd<sub>2</sub>O<sub>3</sub>-doping and high-energy fission products in UO<sub>2</sub>, Gd<sub>2</sub>O<sub>3</sub>-doped CeO<sub>2</sub> pellets were irradiated with 200-MeV Xe<sup>14+</sup> ions. Doping and irradiation effects were analyzed using X-ray diffraction (XRD) and extended X-ray absorption fine structure (EXAFS). The lattice constant of CeO<sub>2</sub> decreases and the local structure is disordered with increased doping levels. However, the irradiation induces an expansion of the lattice and a disordering of atomic arrangement near the Gd atoms. The effects of the irradiation become more pronounced with increasing Gd<sub>2</sub>O<sub>3</sub>-dopant levels. Our results are compared with those of a study involving Er<sub>2</sub>O<sub>3</sub>-doped CeO<sub>2</sub>.

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### 1. Introduction

In current light-water nuclear power plants, to achieve a stable supply of energy and reduce the amount of spent fission fuel (UO<sub>2</sub>), a high-burn-up extension for the fission fuel is an effective option. In high-burn-up fission fuel, however, the initial reactivity of the fuel is promoted, because of the use of enriched fission fuels. To control the initial reaction, gadolinium trioxide, which has a high neutron-absorption cross section, has been doped into UO<sub>2</sub> fuels as a burnable poison [1]. UO<sub>2</sub> fuels are exposed to irradiation from high-energy fission products (FPs) which have energies around 100 MeV. The high-energy FPs induce radiation damage in the UO<sub>2</sub> fuel through elastic collisions and high-density electronic excitation [2]. We have suspected that the Gd<sub>2</sub>O<sub>3</sub> doping and the energetic FP irradiation affect the structure of UO<sub>2</sub>. Therefore, it is important to study these effects on the structure of UO<sub>2</sub> fuels to allow for the safe operation of nuclear power plants.

In this study, CeO<sub>2</sub> pellets were used to simulate UO<sub>2</sub>. CeO<sub>2</sub> has the same fluorite structure as that of UO<sub>2</sub>, and has properties such as lattice constant and thermal conductivity that are similar to those of UO<sub>2</sub>. Therefore, CeO<sub>2</sub> has been widely used to simulate the effects of FPs on UO<sub>2</sub> [3,4]. Hence, we doped CeO<sub>2</sub> pellets with Gd<sub>2</sub>O<sub>3</sub> and irradiated them with 200-MeV Xe<sup>14+</sup> ions to simulate the effects of high-energy FPs. To characterize the effects of Gd<sub>2</sub>O<sub>3</sub>-doping and ion irradiation, X-ray diffraction (XRD) and extended X-ray absorption fine structure (EXAFS) measurements

were used. The result obtained in this study was compared those of a previous study that used Er<sub>2</sub>O<sub>3</sub> as a dopant [5].

### 2. Experimental procedure

The specimens used in this study were Gd<sub>2</sub>O<sub>3</sub>-doped CeO<sub>2</sub> bulk pellets. CeO<sub>2</sub> and Gd<sub>2</sub>O<sub>3</sub> (1, 5, 10 mol%) powders which were 99.9% pure were homogeneously ground and mixed for 24 h using a ball mill. Then the mixtures were compacted into pellets by uniaxial and hydrostatic pressing. The pellets were sintered at 1673 K for 12 h in air.

Gd<sub>2</sub>O<sub>3</sub>-doped CeO<sub>2</sub> pellets were irradiated at room temperature with 200-MeV Xe<sup>14+</sup> ions using a high-energy ion accelerator at the Japan Atomic Energy Agency (JAEA-Tokai). The ion fluences were  $1 \times 10^{12}$ ,  $5 \times 10^{12}$ ,  $1 \times 10^{13}$ , and  $2 \times 10^{13}$  cm<sup>-2</sup>. Some specimens were not irradiated to serve as control specimens.

The structure of the specimens was investigated using two kinds of measurements. The lattice structure of the specimens was characterized using a conventional Cu-K $\alpha$  X-ray diffractometer. To study the local structure surrounding the Gd atoms, EXAFS measurements near the Gd L3-edge (7.249 keV) were performed at room temperature on beam line, BL27B, of the photon factory at the High Energy Accelerator Research Organization (KEK-PF).

### 3. Results and discussion

First, the effects of Gd<sub>2</sub>O<sub>3</sub> doping are discussed. Fig. 1a shows the widely-scanned XRD spectra for Gd<sub>2</sub>O<sub>3</sub>-doped CeO<sub>2</sub>

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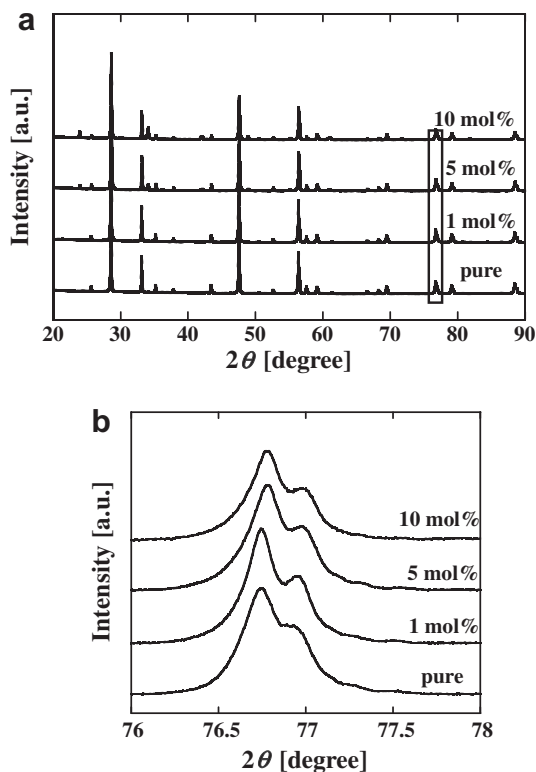


Fig. 1. XRD spectra for the  $Gd_2O_3$ -doped specimens; (a) widely-scanned spectra and (b) the (331) peaks.

and undoped  $CeO_2$ . In the figure, no new phases can be seen and the lattice structure remains unchanged after doping. This result suggests that Gd atoms occupy regular Ce lattice sites in the Ce-crystal structure in  $Gd_2O_3$ -doped  $CeO_2$ . However, when analyzing diffraction peaks at each diffraction angle, we observe a peak shift to higher angles with increasing amounts of  $Gd_2O_3$ -dopant. For example, Fig. 1b shows the shift of the (331) diffraction peaks caused by  $Gd_2O_3$ -doping. This result means the lattice constant decreases through  $Gd_2O_3$ -doping. In this figure, two peaks, which correspond to  $Cu-K\alpha_1$  and  $Cu-K\alpha_2$  X-rays, can be observed. Note that in the data analysis in the current experiment, the peak components of  $K\alpha_2$  were subtracted out of the data and the peaks corresponding to  $K\alpha_1$  were used to calculate the lattice constant and the peak width. Fig. 2 shows the lattice constant and full width at half maximum (FWHM) for  $Gd_2O_3$ -doped  $CeO_2$  as a function of the amount of  $Gd_2O_3$  dopant, which was calculated from the (331) peaks. The lattice constant decreases with increasing amounts of  $Gd_2O_3$ . This can be explained from the difference in atomic size between the Ce atoms and Gd atoms. In the lanthanide series, the atomic size decreases with increasing atomic number. Therefore, the atomic size of Gd is smaller than that of Ce, and the doping of  $Gd_2O_3$  into  $CeO_2$  causes the shrinkage of the lattice. On the other hand, the FWHM changes little after  $Gd_2O_3$ -doping.

Fig. 3a shows the normalized EXAFS spectra near the Gd L3-edge for  $Gd_2O_3$ -doped  $CeO_2$ . EXAFS oscillations above the Gd edge are clearly observed in each spectrum. Fig. 3b shows the  $k^3$ -weighted Fourier transforms (FT) corresponding to the EXAFS spectra in Fig. 3a. The first peak at 2 Å for each spectrum corresponds to the first coordination of the Gd atoms for the fluorite structure (i.e. O atoms) and the second peak at 3.5 Å for each spectrum corresponds to the second coordination of the Gd atoms for the fluorite structure (i.e. primarily Ce atoms). The intensity of the peaks decreases and their widths increase with increasing dopant levels. Ohashi et al., have reported the similar effects from  $Gd_2O_3$ -doping

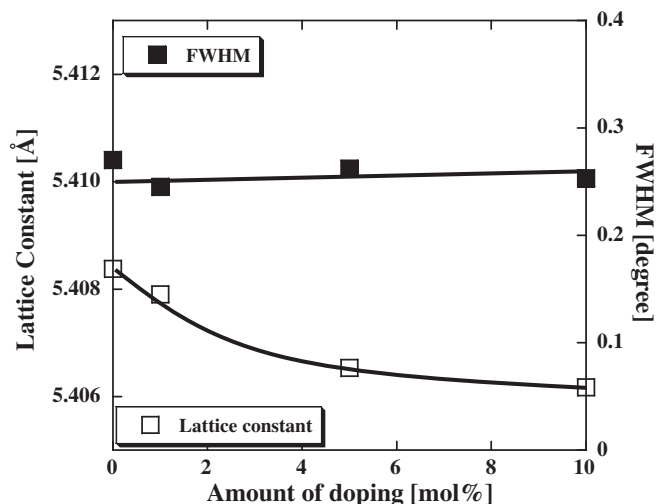


Fig. 2. Lattice constant and FWHM of the (331) peaks of the  $Gd_2O_3$ -doped specimens as a function of the amount of  $Gd_2O_3$ -dopant.

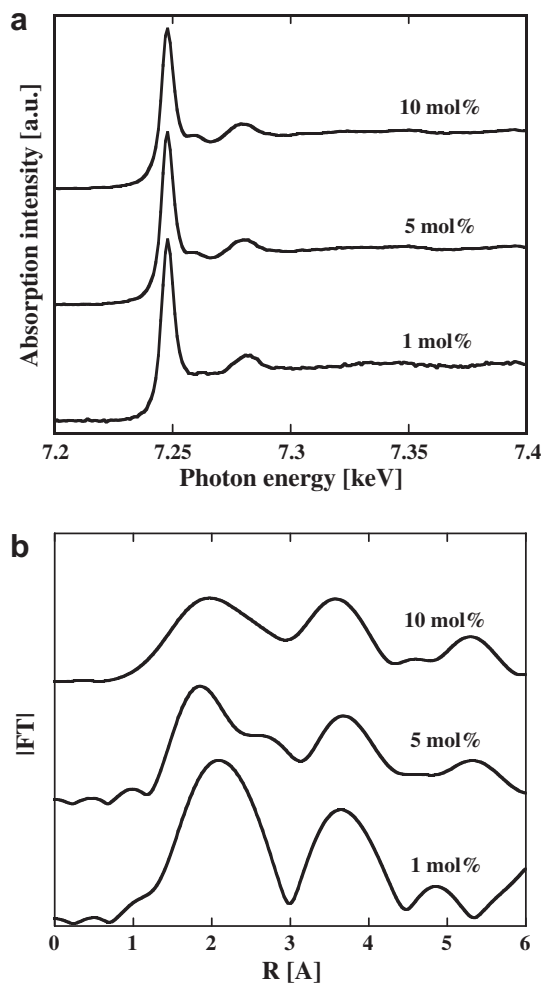


Fig. 3. Gd L3-edge EXAFS spectra for the  $Gd_2O_3$ -doped specimens; (a) normalized spectra and (b) corresponding Fourier transforms.

in  $CeO_2$  based on EXAFS analysis [6]. This means that the local structure surrounding the Gd atoms is more disordered with increasing amounts of dopant. This result suggests that the differ-

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