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The influence of alloy chemistry on the cutting performance and deformation kinetics of titanium alloys during turning

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Abstract

Machining trials were undertaken to study how alloy chemistry influences the relative cutting performance and resulting subsurface deformation for a series of commercially available titanium alloys of increasing β content. Using an experimental orthogonal machining operation, this project focuses on studying what factors influence how efficiently a cutting insert can become embedded into a workpiece and how these factors further influence the overall cutting process. © 2016 The Authors. Published by Elsevier B.V. This is an open access article under the CC BY-NC-ND license

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1. Introduction

There is a growing body of evidence that demonstrates how subsurface deformation generated during machining is sensitive to both the microstructure of the workpiece and machining parameters, such as cutting speed and feed rate [1-3]. There is also awareness that the imparted deformation can be regarded as 'damage', since its presence has been shown to be deleterious [4]. This investigation studies the primary factors that influence the forces during machining and subsequent microstructural deformation. Cutting forces are important to machinists, as the level of force can affect vibration limits, imparted residual stresses and tool fracture.

2. Nomenclature

K _c	Tangential cutting coefficients
$K_{\rm f}$	Feed cutting coefficients
P_1	Cutting edge coefficients
P ₂	Feed edge coefficients

3. Experimental process

3.1. Materials tested

In order to investigate the role of alloy chemistry on

cutting force coefficients (CFCs) and machining induced deformation, the following commercial titanium alloys were supplied by Timet Ltd, as shown in Table 1. The range of α + β titanium alloys demonstrates an increase in β stability, with Ti-6242 being a near α and Ti-5553 being a near β titanium alloy. This can be illustrated using the alloys' molybdenum equivalency number, [5] this factor is an indicator of quantity of β stabilising elements and as such is indicative of how much β phase is contained within the material's microstructure.

Table 1 Generic compositions, room temperature ultimate tensile strength
(UTS) and molybdenum equivalency for the experimental alloys

Material	Composition wt.%	Mo	UTS
	_	Eqv.	MPa
Ti-6242	Al-6.5, Sn-2.25, Zr-4.5, Mo-6.5, Fe-	2.4	1004
	0.15, O-0.15, C-0.04, N-0.04		
Ti-54M	Al-4.7, V-4, Mo-0.68, Fe-0.53,	4.7	850
ELI	O-0.09, C-0.007		
Ti-575	Al-5.3, V-7.7, Fe-0.2, Si-0.5, O-0.2, C-	5.7	1180
	0.05		
Ti-17	Al-5, Sn-2, Zr-2, Mo-4, Cr-4,	9.0	1172
	O-0.16, C-0.05, N-0.13		
Ti-5553	Al-5.12, V-4.7, Mo-4.7, Fe-0.31, Fe-	12.1	1159
	2.77, Si-0.05, O-0.1282,		
	C-0.009, N-0.0055		

The relationship between the resultant cutting forces and the machining feed rate have been commonly used as a measure of a material's machining performance, with

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researchers adopting the Oxley model, which can be used to gain cutting force coefficients (CFCs) from the plotted data.

For this investigation each of the alloys were provided by Timet Ltd in billet form, measuring >150 mm in diameter and >200 mm in length. The material specimens were supplied in the as-forged condition, whereby the billet has been allowed to cool slowly following hot working. Following this cool, the material has not been subjected to any further thermomechanical processing or heat treatments.

3.2. Machining trials

In order to benchmark a material's relative cutting force performance, the following experimental procedure was used to capture machining cutting force coefficients (CFCs). The experimental machining arrangement employs an orthogonal cutting condition, comprising of a 4 mm thick ring cut into the face of each of the titanium billets, with an outer diameter approximately equal to the original diameter of the supplied workpiece. During each cutting trial the cutting insert is brought into contact with the ring such that the direction of feed is parallel to the billet axis, and such that the cutting edge is perpendicular to the direction of feed. In order to obtain the materials' CFCs, the average steady state forces are obtained for a series of variable feed rate trials, with each cut undertaken using a fixed surface speed (40 m.min⁻¹) and depth of cut (equal to ring thickness, 4 mm). For each successive trial, the feed rate is increased in 0.05 mm.rev⁻¹ increments between 0.1 mm.rev⁻¹ and 0.25 mm.rev⁻¹, with individual feed rate trials conducted with a maximum of 3 repeats. A new cutting edge was used for each test. Throughout the trials the force feedback response was recorded using a Kistler 9121 dynamometer that held the tool holder. The output signals were processed through a Kistler 5070 amplifier and the data was recorded using at an acquisition rate of 20 kHz. Table 2 details the tools used during average force measurement investigations. In addition to the standard CFC trial inserts, two additional insert types were chosen in order to investigate the role of edge preparation on the resulting cutting forces and their subsequent effect on microstructural deformation.





For all operations; Hocut 795B, 5% concentration, delivered at 13 l.min⁻¹

An Alicona InfiniteFocus SL high-resolution microscope was used to assess the edge radius of each insert. Figure 1a shows a typical point cloud image, with Figure 1b illustrating the corresponding edge profile and average edge radius determined using a circle best-fit function.



Figure 1 (a) Alicona InfiniteFocus 3D point cloud, (b) application of a profile scan and circle best-fit profile.

4. Results and discussion

4.1. Cutting force coefficients

The data in Figure 2 and Table 3 show the resulting cutting force coefficients for both the cutting and feed components of force. The cutting force (K_e) and feed force (K_f) coefficients corresponds to the gradient of the linear relationship. The cutting (P_1) and feed (P_2) edge coefficients are determined by extrapolating the linear relationship to the y-intercept.



Figure 2 CFC plots for the cutting and feed components.

Table 3 Material CFC components (units shown to 3 significant figures).

Material	Kc	K _f	P1	P_2	
Ti - 6242	4660	1270	380	538	
Ti - 54M ELI	6490	1680	130	465	
Ti - 574	6460	2110	290	573	
Ti - 17	7610	3570	200	456	
Ti - 5553	9970	6670	22.0	170	

The CFC data illustrates how as the molybdenum equivalency of the alloys is increased; the force coefficients (K_c and K_f) also increase. This suggests that there is a potential dependency with alloy chemistry and, in turn, the material mechanical properties. This observation may align with the attributed increase in room temperature UTS with increasing molybdenum equivalency, as previously shown in Table 1. The edge coefficients (P_1 and P_2) do not however seem to display a sensitivity to alloy chemistry.

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