

# Comparison of hydrogen sulfide gas and sulfur powder for synthesis of molybdenum disulfide nanosheets



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## ABSTRACT

Layered transition metal dichalcogenides (TMDCs) have the potential to be used as an alternative to graphene in nano applications, because of their unique opto-electric properties. However, the large-scale synthesis of TMDCs has not been thoroughly investigated with different sulfurization agents, nor have the effects of defects and vacancies on such synthesis been determined. This study therefore looks at the synthesis of MoS<sub>2</sub> nanosheets from a thin Mo film to compare the effects of using H<sub>2</sub>S gas or sulfur powder as the sulfurization agent. In either case, a four-layered nanosheet is obtained; however, the good stoichiometry of MoS<sub>2</sub>, smallest defects, and fewest vacancies are obtained using H<sub>2</sub>S.

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## 1. Introduction

Layered transition metal dichalcogenides (TMDCs) have recently drawn significant attention for use as next generation nano-electronics [1], thin film transistors [2], logic circuits [3], sensors [4], catalysts [5,6], and optoelectronic devices [7,8]. The physical, electrical, and optical properties of TMDCs have so far mostly been studied through physical exfoliation of nanosheets and their transfer onto suitable substrates [9–11]. However, nanosheets with a large surface area can only be achieved through synthesis directly onto a substrate. Recently, the layered MoS<sub>2</sub> was most famous in TMDCs because of the exotic properties of optical and electrical. The early stage method of synthesizing MoS<sub>2</sub> is the sulfurization of metal and metal-oxide films using a sulfurization agent [12–16], and good quality MoS<sub>2</sub> and WS<sub>2</sub> nanosheets have been achieved through chemical vapor deposition (CVD) [17–20].

There is a potential problem in the use of sulfur powder as a sulfurization agent, namely the fact that its vaporization requires temperatures above 200 °C, and that its vapor has a S<sub>8</sub> crown ring

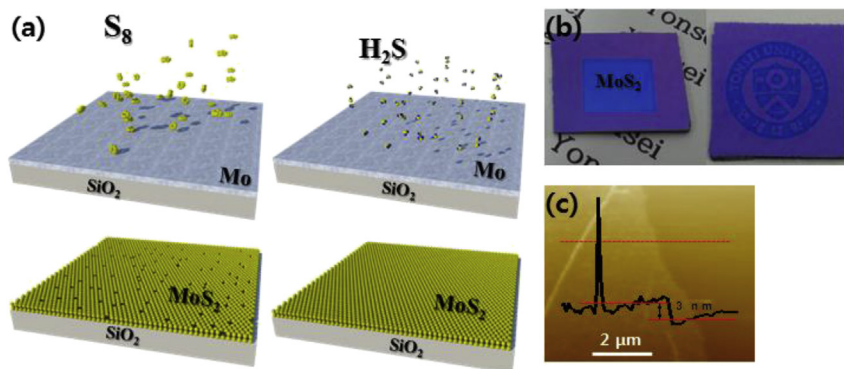
structure that leads to inhomogeneous sulfurization [21]. In contrast, H<sub>2</sub>S gas has a simple molecular structure with a low decomposition energy, ensuring a reliable and homogeneous reaction [22,23]. The synthesis of MoS<sub>2</sub> nanostructures has been previously achieved using H<sub>2</sub>S gas [24, 25], but was found to result in S vacancies forming near nanocluster edges, grain boundaries, and defect sites [26, 27, 28] that greatly affected the optical and electrical properties [27, 28]. However, as there has been no comparative study of different sulfurization agents, this study compares the synthesis of MoS<sub>2</sub> using H<sub>2</sub>S gas with that using sulfur powder at temperatures of 1000 and 750 °C. The intrinsic properties of the MoS<sub>2</sub> nanosheets obtained are also investigated through pump probe measurements, though the ease with which MoS<sub>2</sub> readily changes its electrical properties prevented these from being included.

## 2. Experimental section

The single-crystal SiO<sub>2</sub> and MgO substrates were prepared by ultrasonic cleaning in acetone, IPA, and DI water for 10 min each. For metal sulfurization, 5 or 10 nm Mo metal films were deposited by sputtering for 5 or 10 s, respectively. Fig. 1 (a) shows the

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**Fig. 1.** (a) Schematic illustration of the sulfurization process. (b) Optical images of MoS<sub>2</sub> layers grown on a SiO<sub>2</sub> substrate. (c) AFM image of a four-layer MoS<sub>2</sub> nanosheet ~4 nm in thickness on a SiO<sub>2</sub>/Si substrate.

differences in the sulfurization process depending on whether initial sulfurization in a tube type furnace at 500 °C was carried out using 5 sccm of H<sub>2</sub>S gas or 3 g of sulfur powder (introduced at 120 °C) under a 50 sccm flow of Ar. Following this initial sulfurization, the temperature was increased to 750 or 1000 °C and held for 10 min to enhance the quality of the MoS<sub>2</sub> nanosheet. The sample preparation conditions are summarized in Fig. 2.

To measure the thickness of the MoS<sub>2</sub> nanosheets produced, each was measured by atomic force microscopy (AFM). Differences between the two different sulfurization methods were assessed by characterizing the different nanosheets using Raman spectroscopy, X-ray photoemission spectroscopy (XPS), tunneling electron microscopy (TEM), and pump probe measurement. The XPS measurements were performed at the 09A beam line of the National Synchrotron Radiation Research Center (NSRRC) in Hsinchu, Taiwan, with all data being obtained at a base pressure of less than  $1 \times 10^{-9}$  mbar.

The electrical characteristics and physical properties of the MoS<sub>2</sub> nanosheet were analyzed using ultrafast pump-probe methods so as to prevent any contamination or damage. These measurements were carried out with a visible probe range based on a 250 KHz Ti:sapphire laser system (Coherent RegA 9050) producing 50 fs pulses at 800 nm. A white-light continuum was created by focusing these 800 nm pulses into a sapphire disk, with 400 nm pulses being generated by a second harmonic.

### 3. Results and discussion

As shown in Fig. 1(b), large-area MoS<sub>2</sub> nanosheet with a uniform color was selectively grown using H<sub>2</sub>S gas. After transferring it onto a clean SiO<sub>2</sub> substrate, an AFM line profile (Fig. 1(c)) was used to determine its thickness as 3 nm, which equates to four layers of MoS<sub>2</sub> nanosheet.

Fig. 2(a) shows the Raman spectra of MoS<sub>2</sub> nanosheets, prepared under various conditions (Fig. 2), from which the width of each peak can be used to determine the crystallinity of the sample [29, 30]. That is, the 24 cm<sup>-1</sup> gap between the E<sub>12g</sub> and A<sub>1g</sub> peaks corresponds to four layers of the MoS<sub>2</sub> nanosheet [30], and therefore confirms the AFM-measured thickness. Note also that the Si peak is normalized to allow such comparison between the intensities and widths of the E<sub>12g</sub> and A<sub>1g</sub> peaks. Fig. 2(b) shows the peak distance as a function of the Mo film thickness, suggesting a strong relation between the thickness of the MoS<sub>2</sub> nanosheets and the Mo films. There is, however, a poor conformity to the Raman spectrum when the number of layers exceeds four, with the larger intensity ratio of sample H10 indicating that a good crystallinity can be obtained by annealing at 1000 °C with H<sub>2</sub>S gas. The full-width at half maximum

(FWHM) values of the Raman peak given in Fig. 2(c) reveal that the width of the E<sub>12g</sub> peak relative to that of the A<sub>1g</sub> peak is lower in the synthesized MoS<sub>2</sub> nanosheet than in mechanically exfoliated-MoS<sub>2</sub> (the black dotted line), which can be attributed to planar vibration (E<sub>12g</sub>) being strongly related to the randomly distributed grains found in CVD-grown MoS<sub>2</sub> nanosheets [13]. The FWHM values and peak intensities are also much higher at 750 °C than at 1000 °C, especially when H<sub>2</sub>S gas is used for sulfurization.

The quality and crystallinity of the MoS<sub>2</sub> nanosheet were assessed by TEM, with Fig. 3(a,b) showing the nanosheet prior to sulfurization. The use of H<sub>2</sub>S (Fig. 3(d,f)) was found to produce an enhanced crystallinity when compared to using sulfur powder at the same annealing temperature (Fig. 3(c,e)). The diffraction patterns and FWHM in the insets of Fig. 3(c–f) show evidence of a six-fold-symmetry following sulfurization [13, 31], with the ring pattern of sample H10 indicating a polycrystalline structure.

The XPS results for the Mo 3d and S 2p core-levels of samples S7, H7, S10, and H10 are shown in Fig. 4, from which the atomic ratio (stoichiometry) of each sample was obtained based on the ratio of the Mo 3d peaks intensity to that of the S 2p peak. As shown in Fig. 2, samples prepared in H<sub>2</sub>S gas had a comparable atomic concentration to bulk MoS<sub>2</sub>, whereas the use of sulfur powder resulted in concentrations either under (750 °C) or over (1000 °C) stoichiometry. This non-stoichiometry is believed to be caused by vacancies and defects generated during synthesis [26, 27, 28]. With either sulfurization agent, the width of the Mo 3d and S 2p core-level peaks roughly doubled; however, with only sulfur powder, the Mo 3d core levels exhibited a peak suggesting oxidation to MoO<sub>3</sub> at a higher binding energy. This phenomenon most likely originates from the substrate, with the oxygen species produced believed to interrupt the crystal structure and degrade the electrical properties of the thin film. This suggests that the use of H<sub>2</sub>S gas is the more suitable option for the synthesis of MoS<sub>2</sub> nanosheets.

To further confirm the superiority of H<sub>2</sub>S gas over sulfur powder, ultrafast optical-pump-probe measurements of the carrier relaxation time were performed. For this, four different samples of MoS<sub>2</sub> nanosheet were prepared on MgO substrates under conditions of: H<sub>2</sub>S at 750 °C (Sample I), sulfur powder at 750 °C (Sample II), H<sub>2</sub>S gas at 1000 °C (Sample III), and sulfur powder at 1000 °C (Sample IV). A MgO (100) substrate was used as this is not excited by probing measurements with a large band gap of 7.8 eV, as transparent substrates need a band gap larger than the photon energy (3.01 eV) of the pump pulses used for measurement.

Prior to pump-probe measurement, the ground state absorption was measured to determine an appropriate pump-probe wavelength. As can be seen in Fig. 5 (a), the A and B excitons are

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