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# Fast and selective $Cu_2O$ nanorod growth into anodic alumina templates via electrodeposition

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## 1. Introduction

The synthesis of nanostructures, in the ranges of only a few hundred nanometers, such as nanowires, -rods, and -tubes, has generated great interest owing to their distinct properties and unique applications, as compared to bulk materials [1,2]. For such nano-objects, template-based approaches are one of the most common fabrication methods for mass production and alignment [3]. In particular, anodic aluminum oxide (AAO) templates have been widely employed in the production of nanostructures due to their relatively uniform pore distribution, as well as their thermal and chemical stability [4,5]. Electrochemical deposition is generally used for the growth of metals and conducting metal oxides because of the following advantages: (i) the thickness and morphology of the nanostructure can be precisely controlled by adjusting the electrochemical parameters, (ii) relatively uniform and compact deposits can be synthesized in template-based structures, (iii) higher deposition rates are obtained, and (iv) the equipment is inexpensive due to the non-requirements of either a high vacuum or a high reaction temperature [6,7]. Note that our previous studies have demonstrated the fabrication of nanostructures such as Ni [8,9], Bi<sub>2</sub>Te<sub>3</sub> [10], Pb–CeO<sub>2</sub> [11] nanowires, and V<sub>2</sub>O<sub>5</sub> agglomeration

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# ABSTRACT

The fast and selective growth of cuprous oxide ( $Cu_2O$ ) nanorods into anodic aluminum oxide (AAO) templates is achieved under optimized alkaline conditions via electrochemical deposition. The growth rate of  $Cu_2O$  nanorods at room temperature reached 360 nm/min, the fastest rate reported to date. The synthesis of  $Cu_2O$  nanorods by applying a constant current by using  $Cu_2O$  nanotubes as a transition state is extensively discussed; a Pt pottery-shaped layer played a key role as a seed layer for the fast  $Cu_2O$  growth. We report here the existence of regions of nanostructured  $Cu_2O$  based on our studies and previous relevant works, which include potential-pH curves for  $Cu^{2+}$ -lactate solutions.

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[12]. Thus, it is expected that the preparation of nanosized materials via electrodeposition into an AAO template can be a powerful fabrication method.

Copper (I) oxide (Cu<sub>2</sub>O) is one of the best photovoltaic p-type semiconducting materials, with a direct band gap of ca. 2.0 eV [13]. Furthermore, Cu<sub>2</sub>O has been attracting considerable attention not only due to its potential applications for splitting water into H<sub>2</sub> and O<sub>2</sub> upon visible light irradiation, but also for its application in cost-effective solar cell production and non-toxic devices [14–16].

Recently, we have extensively reported the preparation of  $Cu_2O$  nanowires or -rods via electrodeposition into AAO templates [17–20]. The electrodeposition of  $Cu_2O$  is available from an alkaline solution of copper lactate, according to the following reactions [20]:

$$2Cu^{2+} + 2e^{-} \rightarrow 2Cu^{+} \tag{1}$$

$$2Cu^{+} + 2OH^{-} \rightarrow 2Cu(OH)_{s}$$
<sup>(2)</sup>

$$2Cu(OH)_{s} \rightarrow Cu_{2}O + H_{2}O \tag{3}$$

however, such reactions typically take place under optimized electrochemical preparative conditions, which includes the presence of sufficient  $OH^-$  ions (pH > 8) and a solution temperature of ca. 65 °C. If the solution is unsuitable for this optimized condition, the formation of Cu metal could be the dominant reaction into the porous template [19,21]:



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**Fig. 1.** SEM images of AAO template and Pt pottery-shaped layer. Top view of AAO template: (a) the ordered side and (b) the solution side. Cross-sectional view of Pt pottery-shaped layer after the selective dissolution of AAO templates in 1 M NaOH for 5 h: (c–i) remaining Pt pottery-shaped layer, (c-ii) bottom of the Pt layer film, and (d) EDX analysis of Pt pottery-shaped layer. Source: H. Ju et al.

$$Cu^{2+} + 2e^{-} \rightarrow Cu \tag{4}$$

In particular, a Pourbaix diagram shows that the Cu<sub>2</sub>O domain exists in a very narrow potential range at all pH levels [22]. Moreover, Switzer and co-workers [23–25] have already reported that the co-deposition of Cu and Cu<sub>2</sub>O nanostructures was also observed due to the mass transfer limitation of OH<sup>-</sup> ions and the increase of resistance. For this reason, previous studies have shown that the codeposition of Cu<sub>2</sub>O/CuO or Cu<sub>2</sub>O/Cu into AAO templates or onto thin films [26–28]. In addition, a few reports demonstrated the fabrication of selective Cu<sub>2</sub>O nanorods using AAO templates, but with a very low deposition rate; between 33.3 mm/min [20] and 10.3 nm/min [29], at a deposited layer-by-layer through a filling of pores in the AAO template via electrodeposition because most templates are insulators [30]. Thus, to overcome the low yield of the deposition rate, it is necessary to sufficiently increase the deposition interface.

In this study,  $Cu_2O$  nanorods and nanotubes were electrochemically deposited into AAO templates at a 10–20 times faster growth rate with high selectivity at room temperature by providing the  $Cu_2O$  nucleation site with a seed layer. Furthermore, our results provide a new insight into the mechanism affecting fast electrochemical  $Cu_2O$  deposition, based on a pE–pH diagram.

### 2. Experimental

A platinum metal layer with a thickness of a few hundreds of nanometers was sputtered onto one side of a commercial AAO membrane (Whatman, Anodisc 25) having a nominal pore size of 200 nm by direct current (DC) sputtering (GS NanoTech Inc.). The Ptsputtered layers acted as a working electrode to provide nucleation sites of Cu<sub>2</sub>O, as shown in Fig. 1. During sputtering, a Pt potteryshaped layer was formed at the bottom of the AAO pores and along the wall surfaces of the AAO channels. However, the non-Pt pottery-shaped layer had a potential effect that was unsuitable for shortcut formation due to the breakdown of metal oxides, which commonly leads to a disturbed growth rate [20]. Therefore, having a Pt pottery-shaped layer as a starting seed metal layer was able to increase the electroactive surface. This phenomenon suggests that metal ions favorably migrate to a conduction substrate placed onto the boundary of AAO template during the initial stages of electrochemical growth of Cu and Ni [31,32].

Here, Ag/AgCl, 3 M KCl and Pt wires were employed as the reference and counter electrodes, respectively. The electrolyte consisted of 45 g of 99.99%  $CuSO_4$  (Aldrich) and 75 mL of 88% lactic acid (Aldrich). By complexing with lactate ions,  $Cu^{2+}$  ions were stabilized and the pH was raised to alkaline values [13]. An



**Fig. 2.** Corresponding potential profiles during electrochemical deposition performed at room temperature and pH 9 at various applied current densities for 500 s: (i)  $-0.5 \text{ mA/cm}^2$ , (ii)  $-1 \text{ mA/cm}^2$ , (iii)  $-3 \text{ mA/cm}^2$ , (iv)  $-5 \text{ mA/cm}^2$ , and (v)  $-10 \text{ mA/cm}^2$ , cm<sup>2</sup>. Source: H. Ju et al.

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