



Short communication

# Flexible graphite film with laser drilling pores as novel integrated anode free of metal current collector for sodium ion battery

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## ABSTRACT

In this work, flexible polyimide graphite films with laser drilling pores were prepared by laser drilling pore technique. Then a full cell was built based on the porous graphite film anode,  $\text{Na}_3\text{V}_2(\text{PO}_4)_3$  cathode, and  $\text{NaPF}_6$  in diglyme electrolyte. This unique battery system makes the best of co-intercalation mechanism of  $\text{Na}^+$ -solvent into porous graphite film host. Moreover, the porous graphite film is directly served as an integrated anode with no electrochemically inactive components, such as binders, conductive agents and metallic current collectors. The above merits allow remarkable progress in electrochemical performance, especially the high energy density and much improved cycle life, which will significantly boost the high energy batteries. Our sodium-ion system is potentially promising power sources for promoting the substantial use of low-cost energy storage systems.

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## 1. Introduction

Over the past few decades, lithium-ion batteries (LIBs), by virtue of their high energy density and long cycle life, have been widely applied in portable electronic devices, electric vehicles and large-scale electrical energy storage systems [1–3]. However, for further development of LIBs, one of serious challenges still to be solved is the high cost of limited lithium resources [4]. Therefore, development of novel electrochemical storage devices is not only desirable but also necessary. In this background, sodium-ion batteries (NIBs) are regarded as a promising alternative since sodium is considered as natural abundance. Moreover, Li and Na share similar physical and chemical properties as alkali metals [5]. The remarkable cost-effective advantage endows the NIBs with great potentials in renewable energy fields.

The exploration of anode materials for NIBs started with carbonaceous materials, such as graphitic carbon and non-graphitic carbon materials [6–11]. It was generally considered that graphite anode, commonly used in LIB systems, could barely intercalate sodium in common electrolyte systems by forming binary graphite intercalation compounds (*b*-GICs), because sodium ions are too large to insert into the graphite interlayers [12,13]. Fortunately, the situation changed when Adelhelm's group found that sodium can be reversibly intercalated into artificial graphite by taking advantage of co-intercalation mechanism in the form of ternary graphite intercalation compounds (*t*-GICs)

in a diglyme-based electrolyte, which milestone graphite materials towards anode of NIBs [14]. More recently, Kang's group also investigated the  $\text{Na}^+$ -solvent co-intercalation combined with partial pseudocapacitive behaviors in natural graphite [15]. However, the capacity retention and the energy density of the full cell should be further promoted.

Flexible polyimide graphite films possess a highly in-plane oriented structure [16], which endows the graphite film with excellent planar electrical conductivity and considerable mechanical strength. By laser drilling pores on the graphite films, more intercalation channels were obtained to improve the kinetics. In comparison with conventional multicomponent electrodes that are coated on current collector forming a mixture of binder and active material, a porous graphite film anode with no binder, conductive agents and current collector could improve the energy density and simplify the production process of a full cell [17]. Besides, a porous graphite film anode can also enlarge the contact area with electrolyte which might be favorable for co-intercalation process. To the best of our knowledge, a novel NIB full cell consisting of a porous graphite film anode and a  $\text{Na}_3\text{V}_2(\text{PO}_4)_3$  cathode in diglyme-based electrolyte, which exhibits much improved energy density and prolonged cycling behavior, has not been reported previously in the literature. Therefore, it is fundamentally significant to develop such a novel NIB anode system. We expect that it would strike a chord and make a contribution to the cost-effective and environmentally friendly NIBs.

## 2. Experimental

The Kapton-type polyimide films of 50  $\mu\text{m}$  thickness manufactured by Dupont Co., Ltd. were used as starting materials. They were

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carbonized step by step at 600, 1600 and 2200 °C, respectively. Then the carbonized films were further graphitized at 2800 °C for 3 h in an argon atmosphere, by being sandwiched between polished graphite plates. After graphitization, the graphite films were pressed to obtain 30  $\mu\text{m}$  thickness. Pores of 50  $\mu\text{m}$  in diameter were obtained by laser drilling technique (YLP-1-100-20-20, IPG (Beijing) Fiber Laser Technology Co., Ltd.). The distance between the pores was 240  $\mu\text{m}$ . The depth of pores is also 30  $\mu\text{m}$  and can be controlled.

The  $\text{Na}_3\text{V}_2(\text{PO}_4)_3$  powders was synthesized according to previously reported literature [18]. Typically,  $\text{NaCO}_3$ ,  $\text{NH}_4\text{H}_2\text{PO}_4$ ,  $\text{V}_2\text{O}_5$  and glucose were firstly mixed in stoichiometric ratio and preheated at 400 °C for 6 h. Then, the mixture was further heated at 800 °C in Ar atmosphere for 12 h. Excess glucose was added in order to obtain absolutely pure  $\text{Na}_3\text{V}_2(\text{PO}_4)_3$  product.

The porous graphite films were punched into 10 mm diameter disc electrode and used as anode directly. The mass was 3 mg. The cathode was prepared by mixing  $\text{Na}_3\text{V}_2(\text{PO}_4)_3$  (80 wt.%) with polyvinylidene fluoride binder (10 wt.%) and conductive carbon black (10 wt.%) in N-methyl-2-pyrrolidone solvent. Then the slurry was pasted onto an aluminum foil followed by drying in vacuum at 120 °C for 2 h. After roll-pressing, 10 mm diameter disc electrodes were punched. The loading mass of the active material on the electrode was 2.4 mg. A Teflon Swagelok® type test cell was assembled with a sodium metal counter electrode and a glassy fibrous separator in Argon-filled glove-box. Similarly, the full cell was fabricated using a porous graphite film anode and  $\text{Na}_3\text{V}_2(\text{PO}_4)_3$  cathode. All the electrochemical performances were tested in 1 M  $\text{NaPF}_6$  in diglyme electrolyte. The  $\text{H}_2\text{O}$  content in the electrolyte is <20 ppm. Before assembling the cells, the  $\text{NaPF}_6$  salt was maintained in a vacuum oven at 180 °C for 72 h. Cycle voltammetry (CV) of the cells was performed on an electrochemical workstation (ZAHNER-Elektrik GmbH & Co. KG, Germany). Galvanostatic charge and discharge behaviors were tested on a battery testing system (Wuhan LAND electronics Co., Ltd., China). Morphological and structural information was obtained from field emission scanning electron

microscopy (SEM, HITACHI S-4800) and high-resolution transmission electron microscopy (HRTEM, JEOL 2100F). X-ray diffraction (XRD) patterns were recorded in a Bruker-AXS Micro-diffractometer (D8 ADVANCE) with Cu K $\alpha$  radiation ( $\lambda = 1.5406 \text{ nm}$ ).

### 3. Results and discussion

Fig. 1 demonstrates the XRD patterns and SEM images of the graphite film and  $\text{Na}_3\text{V}_2(\text{PO}_4)_3$  powder. As shown in Fig. 1a, the sharp and strong peak around  $2\theta = 26.5^\circ$  is typical of layered graphite structure. The HRTEM image gives a more direct evidence of highly in-plane oriented structure (the inset in Fig. 1a). The parallel basal planes are well stacked with an average distance of 0.3355 nm. In order to increase the intercalation channels of the sodium ions, the uniform and orderly pores with diameter of 50  $\mu\text{m}$  are obtained by laser drilling technique on the surface of the graphite films, as can be seen in Fig. 1b. All diffraction peaks of  $\text{Na}_3\text{V}_2(\text{PO}_4)_3$  powders are in good agreement with the NASICON framework (Fig. 1c), which can be well in agreement with the typical XRD patterns of  $\text{Na}_3\text{V}_2(\text{PO}_4)_3$  powders with no peaks of any other phases or impurities [19]. Fig. 1 suggests that the micro or nano sized particles could be favorable for ion diffusion because the transport length and the chemical diffusion resistance are decreased [20].

The electrochemical properties of the porous graphite film and  $\text{Na}_3\text{V}_2(\text{PO}_4)_3$  electrodes against Na metal were evaluated in 1 M  $\text{NaPF}_6$ /diglyme electrolyte. Fig. 2a, shows the CV curves of the graphite film with and without drilling pores. Obviously, the graphite film without drilling pores exhibits negligible electrochemical activity. Interestingly, when the drilling pores are formed on graphite films, which provide a large number of electrochemical active sites, reversible sodium ion intercalation/deintercalation peaks are observed, showing similar electrochemical behaviors with natural or artificial graphite electrode in the same electrolyte [14,15]. The charge/discharge profiles deliver a reversible capacity of 127.1  $\text{mAh g}^{-1}$  (Fig. 2b). Interestingly,

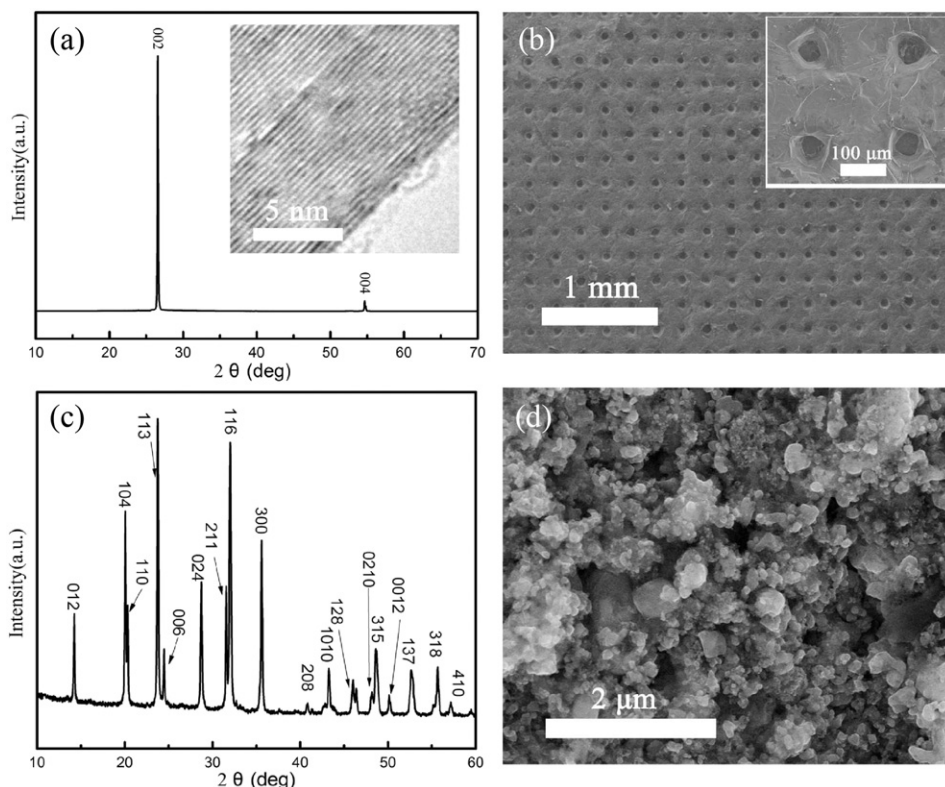


Fig. 1. (a) Typical XRD pattern and (b) SEM images of the porous graphite film, the inset in (a) is its HRTEM; (c) XRD pattern and (d) SEM image of  $\text{Na}_3\text{V}_2(\text{PO}_4)_3$  powder.

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