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The formation of $ZnTe:Cu$ and Cu_xTe double layer back contacts for CdTe solar cells

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ABSTRACT

Processes to form a Cu_xTe layer on a ZnTe:Cu layer were investigated by various deposition methods and heat treatment. These were to make ZnTe:Cu/Cu_xTe double layer contacts for CdTe solar cells and to have benefit from both the well-matched interfacial property of ZnTe:Cu and low-resistivity Cu_xTe . The method to form Cu_xTe compounds was annealing the bi-layer of copper and tellurium which were deposited by sputtering. At an adequate annealing condition, Cu_xTe alloy was formed while maintaining underlying ZnTe without intermixing. Cu_xTe was also formed using co-deposition method. However, in terms of crystallinity, the bi-layer deposition method was considered to be better. According to this experiments, $ZnTe:Cu/Cu_xTe$ double layer contact was successfully formed. This double layer contact is expected to be a good candidate for high efficiency CdTe solar cell.

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1. Introduction

CdTe has been regarded as an attractive material for solar cell because of its optimum band gap (1.45 eV) and high optical adsorption coefficient (over $10^5\,\mathrm{cm}^{-1}$ in optical range). Recently CdTe solar cells have been growing rapidly at photovoltaic markets because of its reasonably high module efficiency of 10% and cost competitiveness of under 1\$/W [\[1\]](#page--1-0).

However, CdTe solar cells still need higher efficiency to lower the module cost and to increase the market share. There are several challenging points to improve the efficiency such as high-transparency window, low resistance contacts, low-defect density junction and so on. Among these points, back contact is an important issue that affects both efficiency and stability [\[2\].](#page--1-0) High work function of CdTe (5.9 eV) makes it difficult to form low resistance ohmic contacts. General method to make an ohmic contact is heavily doping the CdTe surface. The shallow highly doped region facilitates the carrier tunneling. Copper doped ZnTe (ZnTe:Cu) [\[3–5\]](#page--1-0) and Cu–Te alloy (Cu_xTe) have been popular contact materials [\[6–10\].](#page--1-0)

ZnTe:Cu is an advantageous contact material which has wellmatched interfacial properties. The lattice parameter of ZnTe and CdTe is very similar. Additionally, the valence band of ZnTe:Cu is well aligned with that of CdTe [\[3\]](#page--1-0). This well-matched interface makes the movement of carriers across the interface easier. However, the carrier density of ZnTe:Cu is not high enough. It has been reported that the contact resistivity between ZnTe:Cu and elec-

* Corresponding author. E-mail addresses: [donghwan@korea.ac.kr,](mailto:donghwan@korea.ac.kr) solar@korea.ac.kr (D.H. Kim). trode metal is high and the contact resistivity largely depends on the kind of the metal electrode [\[4\].](#page--1-0) Cu_xTe has been another popular contact material. The interfacial property of Cu_xTe does not match with CdTe as nicely as ZnTe: Cu. However, Cu_xTe is a degenerated ptype semiconductor. Thus, ohmic contact can be easily formed using common conductors such as Aluminum [\[9,10\].](#page--1-0) Considering the material properties mentioned above, it is expected that the combination of CuxTe and ZnTe:Cu can make a back contact layer which has both low resistivity ohmic contact and good interfacial property with CdTe.

Conventionally, ZnTe:Cu has been deposited by sputtering method using an alloy target. Cu_xTe has been formed by making CdTe surface tellurium-rich and then depositing copper. After that, annealing follows to make the Cu_xTe alloy [\[8\].](#page--1-0) However, this method cannot be used to make $ZnTe:Cu$ and Cu_xTe bi-layer back contact because Cu_xTe has to be formed on $ZnTe$: Cu, not on CdTe. So we need to form Cu_xTe layer using deposition of both copper and tellurium. Although Yun et al. reported the deposition of Cu_xTe contact on CdTe using thermal evaporation of $Cu₂Te$ powder [\[6\],](#page--1-0) formation of Cu_xTe contacts using deposition of both copper and tellurium has been relatively less studied than reacting copper with tellurium-rich CdTe surface. Additionally, it is necessary to form Cu_xTe while maintaining the underlying $ZnTe:Cu$ layer. If the ZnTe:Cu layer is intermixed with the Cu_xTe layer, back contact layer will lose its beneficial property.

In this paper, various processes to form a Cu_xTe layer on a ZnTe:Cu layer were investigated. Especially, process parameters were controlled so that the Cu_xTe formation process did not damage the existing ZnTe:Cu layer.

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2. Experiment

ZnTe:Cu layer was deposited at 200 \degree C on corning glass by R.F. sputtering. ZnTe compound doped with 6 at.% copper was used as the target. It was deposited at an elevated temperature for good uniformity and electrical properties. The layer thickness was fixed to 150 nm. Cu_xTe was deposited on the ZnTe:Cu layer. It was deposited by either thermal evaporation or sputtering. In thermal evaporation, $Cu₂Te$ powder was used as the evaporation source. In sputtering, copper and tellurium were used as the targets. In sputtering, two methods were investigated – one was to deposit copper and tellurium bi-layer and then anneal the layers to make alloy phase of Cu_xTe , the other was to co-sputter copper and tellurium at the same time. In the former process, Cu was deposited with various thicknesses for phase control while Te layer was fixed to 140 nm. After bi-layer deposition, the samples were annealed in a vacuum furnace (${\sim}1{\times}10^{-3}$ torr) at 150 °C and 250 °C for 30 min, respectively. After the Cu_xTe formation, the films were characterized. Phase variation was characterized using XRD, composition change across the film thickness was investigated using AES, and the resistivity was measured using 4-point probe method.

3. Results

ZnTe:Cu was easily sputtered from the alloy target. When it was deposited at room temperature, the film resistivity was high and the copper concentration was not uniform. However, when the film was deposited at 200 \degree C, copper concentration became uniform throughout the film area and the resistivity decreased from 570 to 16 m Ω cm. The results reported by other researchers were the same [\[5\].](#page--1-0) When the film is deposited at an elevated temperature, the diffusion of elements enhances, and as the result, the composition of the film gets uniform and closer to stoichiometric value.

Deposition of Cu_xTe alloy is a more challenging work than ZnTe:Cu. Firstly, Cu_xTe layer was deposited by thermal evaporation of $Cu₂Te$ powder. But it was very difficult to form the reproducible Cu_xTe layer using this method. The film resistivity of the deposited Cu_xTe film varied one experiment to another in this experiment due to the difference of composition. This can be explained according to Eq. (1) . Cu₂Te alloy dissociates into copper and tellurium in the evaporation source. The dissociated copper remains with $Cu₂Te$ while Te vaporizes [\[7\]](#page--1-0). As the result, the evaporation source becomes more copper-rich as the deposition goes on

 $3Cu₂Te = (2Cu₂Te)CuTe + Cu$ (1)

 $(2Cu₂Te)CuTe = 2Cu₂Te + Cu + Te(vap).$

In Yun's experiment, a fixed amount of $Cu₂Te$ powder was loaded in a tungsten boat and all of the powder was consumed during each deposition [\[6\]](#page--1-0). In this way, the total amount of evaporated copper and tellurium could be fixed in every deposition. This method may be a solution of the reproducibility problem, but it is not suitable for mass production.

Sputtering method was investigated to form Cu_xTe layer. In this paper, copper and tellurium targets were used separately. Two approaches were investigated in this experiment – copper and tellurium bi-layer deposition followed by thermal annealing and codeposition of copper and tellurium at the same time. Fig. 1 shows the phase variation results of the bi-layer approach after annealing at 150 °C and 250 °C. The thickness of copper and tellurium layers were 100 nm and 140 nm, respectively. This thickness gives the atomic ratio of copper and tellurium to make $Cu₂Te$. The sample before annealing showed separate phases of copper, tellurium

Fig. 1. XRD patterns of the samples annealed at different temperature: (a) asdeposition, (b) 150 °C, and (c) 250 °C.

and ZnTe, as expected. When the bi-layer was annealed at 150 \degree C, the formation of Cu–Te alloy was observed. When copper and tellurium reacted at higher temperature, for example at 250 °C, the phase rarely changed from that formed at 150 °C.

In order to make $ZnTe:Cu/Cu_xTe$ contact, $ZnTe:Cu$ layer should not be damaged after Cu_xTe formation. At high process temperature, ZnTe and Cu_xTe layers may intermix with each other. In Fig. 2, the distribution of copper, tellurium and zinc after annealing was observed using AES. After the annealing at 150° C, the zinc concentration profile did not change from that without annealing. However, after annealing at 250 °C, ZnTe:Cu and Cu_xTe layers intermixed with each other, and it was possible to observe diffusion of copper and tellurium atoms toward the ZnTe:Cu layer and diffusion of the zinc atom toward the Cu_xTe layer. Thus, it was possible to conclude that the heat treatment at 150° C for 30 min was an adequate condition at which Cu_xTe alloy can be formed without damaging the underlying ZnTe:Cu layer.

The phase of the final Cu_xTe alloy is determined by the ratio of the reacting copper and tellurium. In order to control the phase of

Fig. 2. AES analysis of the samples annealed at different temperature.

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