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Magnetism and magnetocaloric effect in YNi_4Si -type RNi_4Si (R=Ce, Gd, Tb and Dy) compounds



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A.V. Morozkin^{a,*}, R. Nirmala^b, S.K. Malik^c

^a Department of Chemistry, Moscow State University, Leninskie Gory, House 1, Building 3, GSP-2, Moscow 119992, Russia

^b Indian Institute of Technology Madras, Chennai 600036, India

^c Departamento de Física Teórica e Experimental, Universidade Federal do Rio Grande do Norte, Natal 59082-970, Brazil

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ABSTRACT

Magnetic properties and magnetocaloric effect of YNi₄Si-type RNi₄Si (R=Ce, Gd, Tb and Dy) compounds have been investigated. Magnetic measurements indicate the intermediate valence state of cerium in YNi₄Si-type CeNi₄Si. The magnetocaloric effect of GdNi₄Si, TbNi₄Si and DyNi₄Si are calculated in terms of isothermal magnetic entropy change and they reach maximum values of -22.9 J/kg K, -13.5 J/kg K and -15.6 J/kg K for a field change of 140 kOe near ~ 28 K, 47 K and 27 K and they show maximum values of -12.7 J/kg K, -7.8 J/kg K and -9.3 J/kg K for a field change of 50 kOe near ~ 28 K, 42 K and 22 K, respectively. In contrast to GdNi₄Si, the magnetization-field isotherms for TbNi₄Si and DyNi₄Si exhibit hysteresis loop at 2 K due to strong magnetocrystalline anisotropy.

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1. Introduction

Recently, an orthorhombic derivative of CaCu₅-type structure was detected in the YNi₄Si-type *R*Ni₄Si compounds (*R*=Y, La, Ce, Sm, Gd–Ho) [1]. The YNi₄Si-type {Gd, Tb, Dy}Ni₄Si compounds show ferromagnetic-type ordering at 25 K, 37 K and 19 K, respectively. Below the magnetic ordering temperature, TbNi₄Si exhibits the *b*-collinear magnetic ordering of Tb magnetic sublattice with **Cmm'm** magnetic space group and wave vector **K**=[0, 0, 0] (here **Cmm'm**={1, m_x, 2_y, m_z] × {1,1′ × i} × {1, 1/[1/2, 1/2, 0]}). The Tb magnetic moment reaches a value of 8.66 μ_B /Tb in zero applied filed while no magnetic ordering was detected for Ni sublattice of TbNi₄Si [1,2]. Perhaps, the orthorhombic distortion of initial CaCu₅-type compounds is a prospective way to optimize the magnetic properties [3,4]. For these reasons, the magnetic properties of YNi₄Si-type {Ce, Gd–Dy}Ni₄Si have been investigated in detail in the present work.

The valence state and magnetic properties of CaCu₅-type CeNi₄Si [5] and magnetic properties and magnetocaloric effect of CaCu₅-type GdNi₄Si [6] and GdNi₅[7] and magnetic properties of DyNi₄Si [8] and {Tb, Dy}Ni₅[9,10] were investigated earlier. This work aims to understand some of the specific features of magnetic properties of the YNi₄Si-type *R*Ni₄Si compounds as the orthorhombic derivatives of CaCu₅-type compounds.

2. Materials and methods

The RNi₄Si sample were prepared by arc-furnace melting of the stoichiometric amounts of Ce, Gd, Tb and Dy (99.9 wt%), Ni (99.95 wt%) and Si (99.99 wt%). The samples were annealed at 1070 K for 200 h in an argon atmosphere and subsequently quenched in ice-cold water. The structure, phase purity and composition of the polycrystalline samples were evaluated using powder X-ray diffraction (XRD) and electron microprobe analysis. The X-ray data were obtained on a Rigaku D/MAX-2500 diffractometer (Cu K_{α 1} radiation, 2 θ =10–80°, step 0.02°, 1 s per step). An INCA-Energy-350 X-ray EDS spectrometer (Oxford Instruments) on the Jeol JSM-6480LV scanning electron microscope (20 kV accelerating voltage, beam current 0.7 nA and beam diameter 50 µm) was employed to perform the microprobe analyzes of the samples. Signals averaged over three points per phase estimated standard deviations of 1 at% for rare earth elements (measured by L-series lines), 1 at% for Ni and 1 at% for Si (measured by K-series lines).

Magnetization measurements on polycrystalline samples were carried out using a vibrating sample magnetometer (VSM attachment on PPMS Dynacool System, Quantum Design, USA) in the temperature range of 2–300 K and in magnetic fields up to 140 kOe. Magnetization data were obtained in zero-field-cooled (zfc) and field-cooled (fc) states in 100 Oe applied field to determine the magnetic ordering temperatures. Magnetization as a function of temperature was measured in 5 kOe field in zero-field-cooled state to obtain effective paramagnetic moments and paramagnetic Curie temperatures. Magnetization-field hysteresis curve

^{*} Corresponding author. E-mail address: morozkin@tech.chem.msu.ru (A.V. Morozkin).

was recorded at 2 K to obtain saturation magnetic moments and coercive field data. Magnetization-field isotherms were obtained at various temperatures ranging from 2 K to 102 K with a temperature step of 5 K and a field step of 2.5 kOe to calculate isothermal magnetic entropy change near the magnetic transition.

The unit cell data were derived from powder XRD using the Rietan-program [11,12] in the isotropic approximation at room temperature. The paramagnetic susceptibility was fitted to Curie-Weiss law and the effective magnetic moments and paramagnetic Curie temperatures were obtained [13]. Magnetocaloric effect (MCE) is calculated in terms of the isothermal magnetic entropy change, ΔS_m , using the magnetization vs. field data obtained near the magnetic transition using the thermodynamic Maxwell equation [14].

3. Results

3.1. Crystal structure

Both microprobe and X-ray powder analyzes show that RNi_4Si (R=Ce, Gd, Tb and Dy) are single-phase samples with composition of ternary phase of $R_{16(1)}Ni_{68(1)}Si_{16(1)}$ having YNi₄Si-type structure (Table 1). As a rule, the formation of CaCu₅-type RNi_4Si solid solution leads to a small expansion of the unit cell along the *c* axis and an increase of unit cell volume *V* relative to that of the initial CaCu₅-type RNi_4Si compounds. However, the formation of YNi₄Si-type RNi_4Si compounds leads to an expansion of the unit cell along the *a* axis, but a contraction along *b* and *c* axes with still an increase of the unit cell volume *V* relative to that of the initial unit cell of CaCu₅-type RNi_5 and RNi_4Si compound as shown in Table 1 (initial CaCu₅-type RNi_5 and RNi_4Si compounds are given in term of YNi₄Si-type structure in Table 1 for comparison).

3.2. CeNi₄Si

Magnetization and inverse magnetic susceptibility of YNi₄Si-type CeNi₄Si vs. temperature in an applied field of 5 kOe are shown in Fig. **1a**. The experimental data were fitted with the modified Curie–Weiss law $\chi(T)=\chi_0+C/(T-\Theta_P)$. The obtained

 $\chi_0 = 5.84 \times 10^{-4}$ emu/mol, $\Theta_P = -4.8$ K and $C = 2.9 \times 10^{-2}$ emu K/mol and the effective magnetic moment $M_{eff} = 0.48 \mu_B/f.u.$ derived from the Curie constant *C* are close to that observed in CaCu₅-type CeNi₄Si [5] (Fig. 1b). Since, the magnetic moment of Ce⁴⁺ is zero (and Ce³⁺ ion value $M_{eff}/Ce = 2.54 \mu_B[15]$), the reduction of the moment in YNi₄Si-type CeNi₄Si down to $M_{eff} = 0.48 \mu_B$ may be due to the Ce intermediate-valence, as in CaCu₅-type CeNi₄Si with $M_{eff} = 0.52 \mu_B/fu$ [5]. According to the magnetization data, the CeNi₄Si does not show magnetic ordering down to 2 K in an applied field of 5 kOe (Fig. 1a).

The ferromagnetic-like behavior in the *M*–*H* isotherm and a small value of magnetic moment of 0.04 μ_B /fu in 140 kOe at 2 K of YNi₄Si-type CeNi₄Si (Fig. 1c) is close to that observed in CaCu₅-type CeNi₄Si with *M*_{sat}=0.05 μ_B /fu in field of 90 kOe at 1.7 K [5].

Thus, the YNi₄Si-type CeNi₄Si also shows mixed valence of Ce^{3+} - Ce^{4+} , as in the CaCu₅-type CeNi₄Si [5].

3.3. GdNi₄Si.

The paramagnetic susceptibility of GdNi₄Si in the temperature range of ~30–300 K in 5 kOe field follows Curie–Weiss law (inset in Fig. 2a and b). The fit to the Curie–Weiss law yields a positive paramagnetic Weiss temperature Θ_p =24.7 K and an effective magnetic moment per formula unit (M_{eff} /fu) of 8.12 μ_B for GdNi₄Si. This M_{eff} /fu yields an effective magnetic moments per Ni of ~0.9 μ_B in GdNi₄Si (assuming that Gd has the theoretical Gd³⁺ effective moment of 7.94 μ_B [15]) which is slightly more than the theoretical magnetic moment of pure Ni (0.616 μ_B) [15]. However, such effective magnetic moment from free ion value without invoking the moment on Ni.

The low field zfc and fc magnetization data obtained in 100 Oe field indicate a ferromagnetic ordering at 26 K for the GdNi₄Si compound (Fig. 2a).

The saturation behavior of magnetization at 2 K and the value of magnetic moment of 7.2 μ_B/fu in 140 kOe field at 2 K suggests a collinearly ordered ferromagnetic state of GdNi_4Si (the theoretical ordered state moment of Gd^{3+} ion is 7 $\mu_B[15]$) (Fig. 2b). In fields of \sim 10 kOe the magnetization of GdNi_4Si orders completely (Fig. 2c).

Table 1

Unit cell data of CaCu₅-type *R*Ni₄Si and *Y*Ni₄Si-type *R*Ni₄Si compounds (*R*=Ce, Gd, Tb and Dy) (space group *Cmmm*, N 65, oC12). The unit cell of CaCu₅-type *R*Ni₅ and *R*Ni₄Si compounds are given in term of YNi₄Si-type unit cell.

N	Compound	Type structure	<i>a</i> (nm)	<i>b</i> (nm)	c (nm) ^b	y _{Ni1}	$3^{1/2\bullet}a/b$	V (nm ³)	Refs.
1	CeNi₅ CeNi₄Si CeNi₄Si	CaCu₅ CaCu₅ YNi₄Si	0.4874 0.4852 0.50424	0.84420 0.84039 0.82018	0.4004 0.40536 0.40108	1/3 1/3 0.3379	1 1 1.06485	0.164750 0.165289 0.165874	[17,18] [5] [1]
2	GdNi₅ GdNi₄Si GdNi₄Si	CaCu₅ CaCu₅ YNi₄Si	0.4902 0.4904 0.50769	0.84905 0.84940 0.82353	0.3964 0.3977 0.39525	1/3 1/3 0.3396	1 1 1.06778	0.164984 0.165660 0.165253	[17,18] [6] [1]
3	TbNi₅ TbNi₄Si TbNi₄Siª	CaCu₅ CaCu₅ YNi₄Si	0.4894 0.50626(1)	0.84767 0.82189(1)	0.3966 0.39516(1)	1/3 1/3 0.3396(2)	1 1 1.06689	0.164529 0.164420	[17,18] [8] This work[1]
4	DyNi₅ DyNi₄Si DyNi₄Si	CaCu₅ CaCu₅ YNi₄Si	0.48756 0.50479	0.84448 0.82005	0.39673 0.39499	1/3 1/3 0.3415	1 1 1.06619	0.163347 0.163505	[17,18] [8] [1]

^a TbNi₄Si: Tb 2a, [0, 0, 0] Ni1 4i [0, y_{Ni}, 0], Ni2 4f [1/4, 1/4, 1/2], Si 2c [1/2, 0, 1/2], β_{11} =0.009754, β_{22} =0.003701, β_{33} =0.016010 (β_{11} =[1 [A]/2a]², β_{22} =[1 [A]/2b]², β_{33} =[1 [A]/2c]²), R_F =2.7%. The crystallographic data of TbNi₄Si used with permission– JCPDS – International Center for Diffraction Data. ^b Shortest *R*-2*R* interatomic distance is same to the *c* cell parameter of the CaCu₅-type and YNi₄Si-type compound. Download English Version:

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