



Review

Effect of Cr substitution on magnetic and magnetic entropy change of $\text{La}_{0.65}\text{Eu}_{0.05}\text{Sr}_{0.3}\text{Mn}_{1-x}\text{Cr}_x\text{O}_3$ ($0.05 \leq x \leq 0.15$) rhombohedral nanocrystalline near room temperature

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ARTICLE INFO

Article history:

Received 25 August 2014

Received in revised form

25 September 2014

Accepted 27 September 2014

Available online 1 October 2014

Keywords:

Magnetocaloric effect

Magnetic property

X-ray diffraction

Nanocrystalline material

ABSTRACT

We have studied the effect of Cr substitution on magnetic and magnetocaloric properties in nanocrystalline $\text{La}_{0.65}\text{Eu}_{0.05}\text{Sr}_{0.3}\text{Mn}_{1-x}\text{Cr}_x\text{O}_3$ ($x=0.05, 0.1$ and 0.15). The materials were prepared using the Pechini sol–gel method. All the studied samples were crystallized into a single phase rhombohedral structure with $R\bar{3}C$ space group. Magnetic measurements indicate that the ferromagnetic double exchange interaction is weakened with increasing Cr concentration, resulting in a shift in T_C from 338 K to 278 K as x varied between 0.05 and 0.15. Detailed analyzes in the vicinity of the ferromagnetic (FM)–paramagnetic (PM) phase-transition temperature prove the samples undergoing a second-order phase transition. The magnetocaloric effect is calculated from the measurement of initial isothermal magnetization versus magnetic field at various temperatures. The maximum magnetic entropy change $|\Delta S_M^{\text{max}}|$ is found to decrease with increasing of Cr content from 4.04 J/Kg K for $x=0.05$ –0.78 J/KgK for $x=0.15$ upon 5 T applied field change.

The relative cooling power (RCP) of $\text{La}_{0.65}\text{Eu}_{0.05}\text{Sr}_{0.3}\text{Mn}_{1-x}\text{Cr}_x\text{O}_3$ series is nearly 54% of pure Gd, which will be an interesting system for application in room temperature refrigeration.

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1. Introduction

In recent years, room-temperature magnetic refrigeration, based on the magnetocaloric effect (MCE), raised a great attention because of its potential advantages over the conventional vapor-cycle refrigeration [1–4]. Recently, many of the hole-doped manganites of the formula $\text{R}_{1-x}\text{A}_x\text{MnO}_3$ ($\text{R}=\text{La}, \text{Pr}$, etc. and $\text{A}=\text{Ca}, \text{Sr}$, etc.) concerning second order magnetic transition, have been studied in an attempt to achieve a large MCE [5–8]. Furthermore, the manganites have additional advantages such as low cost, good chemical stability, easy preparation and more importantly the ability to control their magnetic transition temperatures (T_C) close to room temperature by R-site or Mn-site substitutions. One recent example of this chemical substitution for tuning T_C in manganites, is in the $\text{La}_{0.7}\text{Sr}_{0.3}\text{MnO}_3$ (LSMO) system, which is a colossal magnetoresistive (CMR) ferromagnetic manganite with $T_C \sim 370$ K, and by substitution of the La ion by the rare-earths ions Eu, T_C is lowered [9].

Many theories have been proposed to understand the ferromagnetism and CMR of manganites such as double-exchange (DE) interaction, Jahn–Teller effect, and phase separation [10,11]. In hole-doped manganites, manganese ion exist in two different oxidation states, Mn^{3+} and Mn^{4+} , results in a $\text{Mn}^{3+}/\text{Mn}^{4+}$ mixed-valence state, which creates mobile charge carriers and ferromagnetic ordering of Mn spins [12]. This behavior is usually interpreted with the help of DE mechanism, where the magnetic coupling between Mn^{3+} and Mn^{4+} ions results from the motion of an e_g electron between the two partially filled d-orbitals with strong on-site Hund's coupling [13]. Since the magnetic and MCE properties are dependent on the strength of DE interaction between Mn^{3+} and Mn^{4+} through oxygen atom, the doping at Mn-sites in manganites has a pronounced effect on the physical properties [14–16]. On the other hand, the ceramic preparation route, such as the conventional solid-state reaction method, the sol–gel method and combustion thermal spray techniques play an important role on the quality and the magnetocaloric properties of materials [17].

In this paper, we attempt to synthesize single phase, nanocrystalline $\text{La}_{0.65}\text{Eu}_{0.05}\text{Sr}_{0.3}\text{Mn}_{1-x}\text{Cr}_x\text{O}_3$ samples with $x=0.05, 0.1, 0.15$

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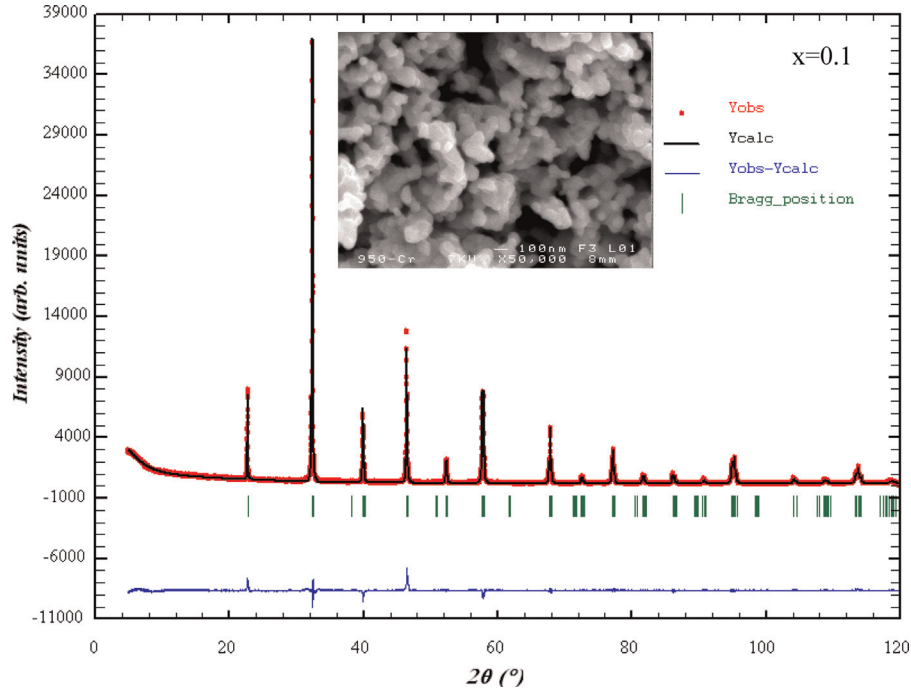


Fig. 1. Rietveld refinement profile for $\text{La}_{0.65}\text{Eu}_{0.05}\text{Sr}_{0.3}\text{Mn}_{0.9}\text{Cr}_{0.1}\text{O}_3$ sample performed using FULLPROF. Open circles correspond to experimental data and the lines are fits. Vertical bars are the Bragg reflections for the space group $R-3C$. The difference pattern between the observed data and fits is shown at the bottom. The inset represents a SEM micrograph for the sample.

Table 1

Refined structural parameters of $\text{La}_{0.65}\text{Eu}_{0.05}\text{Sr}_{0.3}\text{Mn}_{1-x}\text{Cr}_x\text{O}_3$ ($0.05 \leq x \leq 0.15$) samples at room temperature. Space group $R-3C$. V is the cell volume; B_{iso} is the overall isotropic thermal parameter, CS is the crystallite size evaluated from Rietveld refinements, R_{wp} , R_p and R_F are the agreement factors for the weighted profiles, the profiles and the structure factors; χ^2 is the goodness of fit. The numbers in parentheses are estimated standard deviations to the last significant digit.

Sample	0.05	0.10	0.15
Structure type	Rhombohedral	Rhombohedral	Rhombohedral
Space group	$R-3C$	$R-3C$	$R-3C$
Lattice parameter			
$a(\text{\AA})$	5.5144 (2)	5.5123(5)	5.5108 (2)
$c(\text{\AA})$	13.3779 (7)	13.3770 (1)	13.3704 (3)
$V (\text{\AA}^3)$	352.182 (2)	352.011 (1)	351.653 (1)
$d_{(\text{Mn}, \text{Cr})-\text{O}} (\text{\AA})$	1.9574(7)	1.9563 (2)	1.9552(3)
$\theta_{(\text{Mn}, \text{Cr}-\text{O}-\text{Mn}, \text{Cr})} (^\circ)$	166.97(3)	166.23 (3)	165.32(1)
W	0.09469	0.09480	0.09489
La/Eu/Sr			
x	0.000	0.000	0.000
y	0.000	0.000	0.000
z	0.25	0.25	0.25
$B_{\text{iso}} (\text{\AA}^2)$	0.82(2)	0.75 (3)	0.59 (3)
Mn/Cr			
x	0.000	0.000	0.000
y	0.000	0.000	0.000
z	0.000	0.000	0.000
$B_{\text{iso}} (\text{\AA}^2)$	0.61(4)	0.47(4)	0.34 (4)
O			
x	0.457(1)	0.456(4)	0.454(1)
y	0.000	0.000	0.000
z	0.25	0.25	0.25
$B_{\text{iso}} (\text{\AA}^2)$	1.28 (3)	1.33 (5)	1.15 (2)
$CS (\text{nm})$	74	68	63
Discrepancy factors (%)			
$R_{\text{wp}} (\%)$	7.83	8.18	8.55
$R_p (\%)$	6.02	6.28	7.34
$R_F (\%)$	2.70	6.53	5.92
$\chi^2 (\%)$	2.32	3.47	3.48

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