



Micromagnetics of shape anisotropy based permanent magnets

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ABSTRACT

In the search for rare-earth free permanent magnets, various ideas related to shape anisotropy are being pursued. In this work we assess the limits of shape contributions to the reversal stability using micromagnetic simulations. In a first series of tests we altered the aspect ratio of single phase prolate spheroids from 1 to 16. Starting with a sphere of radius 4.3 times the exchange length L_{ex} we kept the total magnetic volume constant as the aspect ratio was modified. For a ferromagnet with zero magnetocrystalline anisotropy the maximum coercive field reached up to 0.5 times the magnetization M_s . Therefore, in materials with moderate uniaxial magnetocrystalline anisotropy, the addition of shape anisotropy could even double the coercive field. Interestingly due to non-uniform magnetization reversal there is no significant increase of the coercive field for an aspect ratio greater than 5. A similar limit of the maximum aspect ratio was observed in cylinders. The coercive field depends on the wire diameter. By decreasing the wire diameter from $8.7L_{ex}$ to $2.2L_{ex}$ the coercive field increased by 40%. In the cylinders nucleation of a reversed domain starts at the corners at the end. Smoothing the edges can improve the coercive field by about 10%.

In further simulations we compacted soft magnetic cylinders into a bulk-like arrangement. Misalignment and magnetostatic interactions cause a spread of $0.1M_s$ in the switching fields of the rods. Comparing the volume averaged hysteresis loops computed for isolated rods and the hysteresis loop computed for interacting rods, we conclude that magnetostatic interactions reduce the coercive field by up to 20%.

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1. Introduction

Rare earth permanent magnets exhibit the highest maximum energy product $(\mathbf{B} \cdot \mathbf{H})_{\max}$ of all known magnetic materials; a combination of high magnetic moment and high coercivity [1]. For efficient electrical generators and motors this is crucial; the permanent magnets that are contained in either the rotor or stator must provide as strong magnetic fields as possible, without themselves being demagnetized. Recently much of the research into permanent magnets has focussed on reducing their dependence on rare earth elements. A number of approaches to finding new permanent magnets are being pursued, including new hard magnetic compounds and nanostructuring, where a soft phase contributes large magnetization and a hard phase contributes high coercivity [2–4].

In this paper we assess the limits of shape anisotropy effects on the improvement of coercivity in magnets made of a single soft

magnetic phase. The coercive field of an ideal magnetic particle is dependent on size and shape, with various modes of reversal such as coherent rotation, curling and nucleation [5]. At small sizes, where internal magnetization is homogeneous, reversal proceeds by coherent rotation and is described by

$$H_c = \frac{2K_1}{\mu_0 M_s} + \frac{M_s}{2}(1 - 3N) \quad (1)$$

where K_1 is the uniaxial magnetocrystalline anisotropy constant, μ_0 is the vacuum permeability, M_s is the saturation magnetization and N is the demagnetizing factor parallel with the c -axis. The coercive field is dependent on the angle θ between the field and the c -axis of the sample, so that an angle-adjusted coercive field H_c^* is given by [6]

$$H_c^* = \frac{H_c}{(\cos^{2/3} \theta + \sin^{2/3} \theta)^{3/2}} \quad (2)$$

For a prolate spheroid (also known as an “ellipsoid of rotation”), which has equal dimensions along two axes, the demagnetizing factors can be calculated following the work of Osborn [7].

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Previous work [8–10] showed that, above a certain length, the reversal of columnar grains was no longer dependent on grain length but on the nucleation of a reversal volume. The nucleation field then depends on the diameter only, where below a certain diameter related to the reversal volume size the nucleation field increases as an inverse function of diameter. This is a consequence of the spatial confinement of the reversal volume and the increased exchange energy contribution when forming a domain wall.

2. Method

The finite element method is used to numerically solve the Landau–Lifschitz–Gilbert (LLG) equation. At each time step we apply a hybrid finite element/boundary element method to compute the magnetic scalar potential [11]. Finite element models are created with tetrahedral volume elements and triangular surface elements. Prolate spheroids are prepared by defining inner axes a in the x – y plane and b along z , with $b \geq a$ defining the long axis (Fig. 2b). The dimensions are modified to change the aspect ratio, keeping the total volume $V = \frac{4}{3}\pi a^2 b$ constant at $1824L_{\text{ex}}^3$, which corresponds to a sphere with $a = b = 4.35L_{\text{ex}}$ and an aspect ratio of 1.0. L_{ex} is the exchange length of the material. In order to assess the effects of shape alone we reduce the magnetocrystalline anisotropy to zero.

Single phase rods are modeled as regular cylinders with diameter D and length L . For comparison, rounded rods are created where the total length remains the same but both ends are rounded spherically. For these rods a moderate magnetocrystalline anisotropy is used. Reversal loops are calculated by applying a slowly increasing external field opposing the initial magnetization, where the ramp speed is much slower than the Larmor precession, so that dynamic effects may be safely ignored [12]. A small field angle of 2° is used. The nucleation field H_{nuc} is defined as the field strength required to reduce M_z to $0.9M_s$. The coercive field H_c is defined as the field required to reduce M_z to zero. For all simulations the c -axis is oriented parallel to the cartesian z -axis, except for particle ensembles where the c -axis

follows the rotation of the geometry, staying with the shape-defined long axis.

3. Results and discussion

Numerically calculated hysteresis loops for prolate spheroids with identical volumes but aspect ratios ranging from 1.0 up to 15.6 are presented in Fig. 1a. The theoretical values of H_c are calculated with Eq. (1) and adjusted for the 2° field angle with Eq. (2). The curling mode of reversal is not considered, and images of the changing magnetization configurations during reversal for aspect ratios of 1.95 (Fig. 2a) and 15.6 (Fig. 2b) show that curling is not present in this size regime.

H_{nuc} approaches zero for an aspect ratio of 1.0, which corresponds to the theoretical coherent mode anisotropy field $H_A = 2K/M_s = 0$, where for spheres with $N = 1/3$ [13]. For larger aspect ratios reversal begins with nucleation of a reversal domain at the ends of the wire, followed by domain wall propagation until the whole wire is switched. The value of H_{nuc} converges at a value of $b/a = 5$, consistent with previous investigations [8–10]. For an aspect ratio of 15.6 the nucleation field is $0.42M_s$.

The simulation results match closely with the theoretical model, even at large aspect ratios where reversal begins by nucleation. This is consistent with the idea of nucleation of a reversal volume at the ends of the sample, since the reversal volume forms through localized rotation from the fully saturated sample (Fig. 2b), therefore the Stoner–Wohlfarth model remains valid for predicting the nucleation fields. This kind of behavior has been observed before in the switching of patterned elements in recording media, where non-uniform switching is observed but the switching field follows the Stoner–Wohlfarth angle dependence [14].

Similar results are presented for cylindrical and rounded rods (Fig. 3), where for a discrete set of rod diameters D the length L is modified. The nucleation field H_{nuc} is plotted as a function of L/D , where each symbol represents a different diameter D . H_{nuc} values reach a plateau when the aspect ratio is greater than 5. In this regime switching occurs by nucleation of a reversed domain and wall motion and the nucleation field increases with decreasing

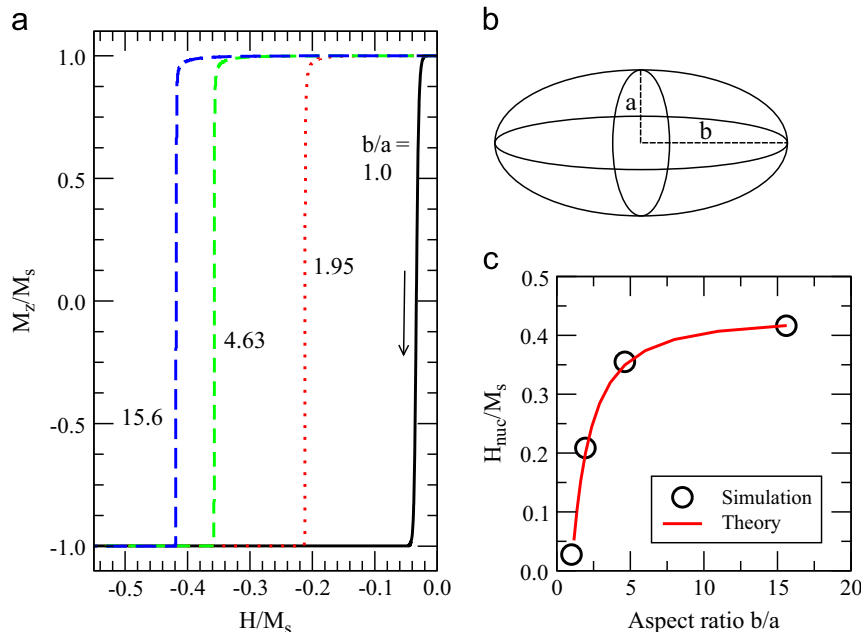


Fig. 1. (a) Computed reversal curves for prolate spheroids with zero magnetocrystalline anisotropy and varying aspect ratio between 1.0 and 15.6, (b) schematic of the spheroid geometry, with a being the radius in the xy plane and b along the long z -axis and (c) the respective nucleation fields for different aspect ratios, with the theoretical plot for coherent reversal.

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