

Investigation of the elastic, hardness, and thermodynamic properties of actinide oxides



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ABSTRACT

The elastic and thermodynamic properties of actinide oxides (AO₂) compounds have been investigated by using the first-principle density functional theory (DFT) within the generalized gradient approximation (GGA). The calculated lattice constants of AO₂ are in agreement with the available experiments data. The calculated elastic constants reveal that all AO₂ compounds are mechanically stable. The shear modulus, Young's modulus, Poisson's ratio σ , the ratio B/G and the anisotropy factor are also calculated. Finally, the Vickers hardness, Debye temperature, melting point and thermal conductivity have been predicted.

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1. Introduction

Actinide and their oxides show many intricate physical behaviors due to the complex electronic structure properties of 5f states, and have attracted considerable attention [1,2]. Another reason for a thorough study of actinide oxides (AO₂) is their important role played in all stages of the nuclear fuel cycle. Fission yield of a uranium based conventional fission nuclear reactors can generate more than 20 fission products such as Np, Pu, Am, and Cm in the actinides [3]. The actinides affect the thermo-physical and mechanical properties, thermal conductivity, ionic diffusion and phase stability of the fuel. Therefore, thorough understandings of the physical properties of AO₂ are of great significance.

Over the past few decades, extensive studies of AO₂ have been carried out. Lu et al. [4] systematically studied the electronic, mechanical, tensile and thermodynamic properties of AmO₂ by performing density functional theory (DFT)+U calculations. Behera et al. [5] investigated the structural and mechanical properties of UO₂ using atomic level simulations. Boettger [6] predicted transition pressure of ThO₂ around 27.5 GPa by using the relativistic linear combinations of the Gaussian type orbital-fitting function technique. Wang et al. [7] studied the thermodynamic properties and the phase transition of ThO₂ from the cubic structure to the orthorhombic structure using the projector-augmented wave method. Wang et al. [8] studied the mechanical properties, electronic structure and phonon dispersion of ground

state ThO₂ as well as the structure behavior up to 240 GPa. Chu et al. [9] investigated the thermal expansion coefficient, entropy, heat capacity and enthalpy of PuO₂ and α -Pu₂O₃ between 298 K and 1500 K using BMH empirical potential and shell potential. Shi et al. [10] investigated the electronic structure and optical properties of UO₂ and PuO₂ based on the DFT using the generalized gradient approximation (GGA)+U scheme. However, systematical studies on the elastic properties, Debye temperature, Vickers hardness, melting point and thermal conductivity of AO₂ are still lacking. In this paper, we present a detailed study of these physical properties of AO₂ by using the first-principle plane-wave pseudo-potential (PWPP) method.

2. Calculation methods

The first-principle calculations were carried out by using the PWPP method within DFT [11]. For the exchange and correlation terms, the revised Perdew–Burke–Ernzerhof (rPBE) [12] was used within the GGA. Using the PWPP method, 2s²2p⁴ for O, 6d¹7s² for Ac, 6s²6p⁶6d²7s² for Th, 5f²6s²6p⁶6d¹7s² for Pa, 5f³6s²6p⁶6d¹7s² for U, 5f⁴6s²6p⁶6d¹7s² for Np, 5f⁶6s²6p⁶7s² for Pu, 5f⁷6s²6p⁶7s² for Am, 5f⁷6s²6p⁶6d¹7s² for Cm, 5f⁹6s²6p⁶7s² for Bk, 5f¹⁰6s²6p⁶7s² for Cf, 5f¹¹6s²6p⁶7s² for Es, 5f¹²6s²6p⁶7s² for Fm, 5f¹³6s²6p⁶7s² for Md and 5f¹⁴6s²6p⁶7s² for No were treated explicitly as valence electrons. Plane wave cut-off energy of 800 eV and an 8 × 8 × 8 grid of Monkhorst–Pack points have been employed in this study to ensure well convergence of the computed structures and energies. The structural parameters of AO₂ were calculated

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by using the Broyden–Fletcher–Goldfarb–Shanno (BFGS) [13–16] method.

3. Results and discussions

3.1. Structural properties

At ground state, all AO_2 crystallize in face-centered cubic CaF_2 -like (space group Fm-3m) structure. Their cubic unit cells are composed of four AO_2 formula units with the actinide atom and the O atom in $4a$ and $8c$ sites, respectively. In this arrangement, each actinide atom is located at the face-centered positions, together with eight O atoms which fill tetrahedral sites (Fig. 1). The lattice constants of AO_2 are summarized in Table 1. The

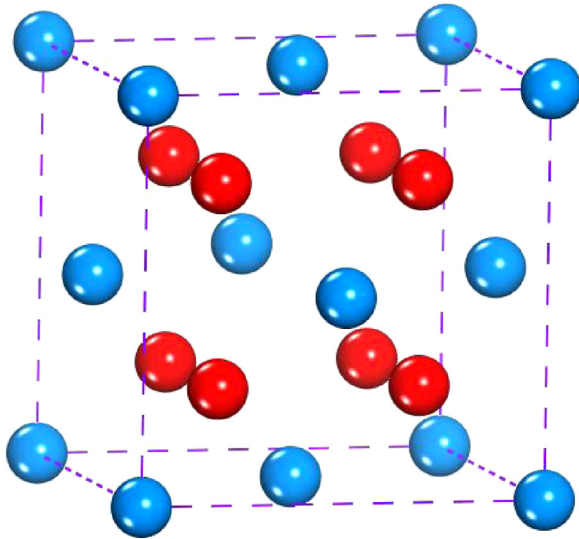


Fig. 1. The cubic structures of AO_2 ; A atoms are shown in blue, and O atoms in red. (For interpretation of the references to color in this figure legend, the reader is referred to the web version of this article.)

Table 1
Calculated lattice constants together with experimental data.

| Compounds | Pearson symbol | a (Å) | | ρ (g/cm ³) | |
|----------------|----------------|---------|---------------------|-----------------------------|---------|
| | | Cal. | Exp. | Cal. | Exp. |
| AcO_2 | $cF12$ | 5.20752 | | 12.1818 | |
| ThO_2 | $cF12$ | 5.62802 | 5.695 ^a | 9.83795 | 9.49492 |
| PaO_2 | $cF12$ | 5.41799 | 5.505 ^b | 10.9837 | 10.4482 |
| UO_2 | $cF12$ | 5.4204 | 5.444 ^c | 11.2622 | 11.1164 |
| NpO_2 | $cF12$ | 5.36279 | 5.434 ^d | 11.5847 | 11.1475 |
| PuO_2 | $cF12$ | 5.34571 | 5.3955 ^e | 12.0005 | 11.6713 |
| AmO_2 | $cF12$ | 5.32492 | 5.376 ^f | 12.0976 | 11.7652 |
| CmO_2 | $cF12$ | 5.31354 | 5.357 ^g | 12.3526 | 12.0544 |
| BkO_2 | $cF12$ | 5.30981 | 5.332 ^h | 12.3787 | 12.2385 |
| CfO_2 | $cF12$ | 5.32380 | 5.31 ⁱ | 12.4574 | 12.5548 |
| EsO_2 | $cF12$ | 5.35226 | | 12.3031 | |
| FmO_2 | $cF12$ | 5.39806 | | 12.2037 | |
| MdO_2 | $cF12$ | 5.46707 | | 11.7880 | |
| NoO_2 | $cF12$ | 5.54886 | | 11.3132 | |

^a Ref. [17].

^b Ref. [18].

^c Ref. [19].

^d Ref. [20].

^e Ref. [21].

^f Ref. [22].

^g Ref. [23].

^h Ref. [24].

ⁱ Ref. [25].

calculated lattice constants of AO_2 are in good agreement with the available experiments data. The deviations from the experimental data are underestimated by less than 3.3%. These results show and confirm that the method used in this study is reliable, thereby the optimized lattice constants can be used for future calculations of other parameters.

3.2. Single-crystal elastic constants

Elastic constants of crystals provide a link between mechanical and dynamical behaviors. Also, they give important information concerning the elastic response of a crystal to an external pressure. To calculate the elastic constants, we have applied the non-volume-conserving method.

The mechanical stability of AO_2 is studied by the calculated elastic constants. For the cubic crystals, the mechanical stability criteria are given by [26]

$$\begin{cases} C_{11} > 0; C_{44} > 0 \\ C_{11} - C_{12} > 0 \\ C_{11} + 2C_{12} > 0 \end{cases} \quad (1)$$

The calculated elastic constants of AO_2 at 0 GPa are shown in Table 2. For comparison, the available experimental data and theoretical results by others are also presented. According to Eq. (1), it should be noted that AO_2 are mechanically stable. In Table 2, it can be noted that the calculated values of C_{11} and C_{12} for UO_2 are slightly lower than the corresponding experimental ones [27] with deviations of 6.6% and 28.9%, respectively. For ThO_2 , the calculated values are in good agreement with the experimental results with the maximum deviation of 13.7% [28]. We also found the calculative value of C_{44} is larger than the experimental value, presumably because oxygen relaxation was not included.

3.3. Polycrystalline elastic moduli

In the particular case of randomly oriented polycrystals, one may evaluate aggregate average elastic properties based on additional hypotheses such as isostress named as Reuss or Voigt states. For the cubic structure, the shear modulus G and the bulk modulus B are given for each approximation by [26]

$$\begin{cases} G_V = \frac{1}{5}(C_{11} - C_{12} + 3C_{44}) \\ B_V = B_R = \frac{1}{3}(C_{11} + 2C_{12}) \\ G_R = \frac{5C_{44}(C_{11} - C_{12})}{4C_{44} + 3(C_{11} - C_{12})} \end{cases} \quad (2)$$

To make further conclusions on the mechanical properties, polycrystalline bulk modulus B_H , shear modulus G_H , Young's modulus E and Poisson's ratio σ have been calculated by the Voigt–Reuss–Hill approximation [37,38]:

$$\begin{cases} G_H = \frac{1}{2}(G_V + G_R) \\ B_H = \frac{1}{2}(B_V + B_R) \\ E = \frac{9B_H G_H}{3B_H + G_H} \\ \sigma = \frac{3B_H - 2G_H}{2(3B_H + G_H)} \end{cases} \quad (3)$$

Using Eqs. (2) and (3) the calculated bulk modulus B , shear modulus G and Young's modulus E for AO_2 are presented in Table 3. The experimental data and calculated results are included for comparison. The polycrystalline elastic modulus obtained in our work is found to in good agreement with the previous results. The deviations are partly due to temperature, volume effects and different DFT approximation (LDA or GGA). Bulk modulus measures the resistance that material offers to change in its volume. From Table 3, we can see that the bulk modulus of AcO_2 is larger than the others, indicating that AcO_2 is less compressible than the

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