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High density copper nanowire arrays deposition inside ordered titania pores by electrodeposition

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ARTICLE INFO

Article history: Received 21 January 2009 Received in revised form 12 February 2009 Accepted 13 February 2009 Available online 26 February 2009

Keywords: Titanium dioxide Nanotube Electrodeposition Membrane Nanowire

ABSTRACT

Large area and free-standing TiO_2 films was prepared by ultrasonic splitting and a chemical etching step was used to open the closed bottom end of TiO_2 films and yields a high aspect-ratio anodic titanium oxide membrane open at both ends. Ordered Cu nanowire structures were fabricated by a simple electroplating method inside high aspect-ratio anodic titanium oxide membrane. Scanning electron microscopy (SEM), transmission electron microscopy (TEM), and the X-ray diffraction (XRD) were employed to characterize the resulting samples. Detailed results and the possible mechanism are presented.

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1. Introduction

Nanostructured titanium dioxide (TiO2) crystalline material is widely studied due to its potential technological applications as solar cells, rechargeable lithium batteries, gas sensors [1-3]. Onedimensional TiO2 nanostructures have attracted significant scientific interests in the aspect of nanotubular synthesis. Self-organized electrochemical growth of TiO2 nanotube on Ti has been given particular interest. In 2001, Gong et al. [4] reported the fabrication of vertically oriented, high ordered TiO₂ nanotube arrays by potentiostatic anodization of titanium in a HF aqueous electrolyte. Later, TiO₂ nanotube arrays with the length of 134 μm were successfully produced in a non-aqueous electrolyte [5]. Recently, free-standing TiO₂ nanotube membrane was fabricated by dissolving the titanium in CH₃OH/Br₂ solution, and the bottom of the wet TiO₂ nanotube membrane can be opened by exposing in HF vapors [6], which can also be obtained by using solvent-evaporation-induced delamination [7] and ultrasonic splitting [8].

Self-organized TiO₂ nanotubes can be used as channels to vertically grow nano-crystallites and which may have new potential applications. Macak et al. [9] reported that Pt/Ru nanoparticles were on the top of the nanotubular TiO₂ by electrodepositing. Fe₂O₃ nanorods were synthesized inside a TiO₂ nanotubular array template by pulsed electrochemical deposition [10], while the Fe₂O₃ nanorods were also found on the surface of the membrane from their scanning electron micrograph (SEM) and transmission

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electron microscopy (TEM) images. CdTe can be electrodeposited into the ${\rm TiO_2}$ nanotube [11], whereas a high quality of CdTe/TiO₂ junction was difficultly achieved due to the equal conductivity of the ${\rm TiO_2}$ tubes in the surface and the bottom [11–13]. Therefore, Macak et al. [14] switched the pore bottoms to a higher conductivity by electrochemical self-doping and then the pore was filled by electrodeposition.

In the present study, a free-standing TiO_2 nanotube arrays are firstly attained by using ultrasonication, after that, two-end-opened TiO_2 membranes are achieved by chemical etching, and at last Cu nanowire arrays are electrodeposited into the channels of the self-organized TiO_2 nanotubes.

2. Experimental

2.1. Materials preparation

The anodic titania templates with pore size of about 120 nm were grown on high purity titanium plates (0.30 mm thick, 99.5% purity) by potentiostatic anodization with the voltage of 60 V in glycol at 20 °C for about 48 h. After that, the whole sample was transferred into 95% of ethanol solution and ultrasonic agitation was applied at room temperature for several minutes. During the ultrasonication, the titania membrane will be detached from the Ti substrate, then the titania membrane was taken out from the solution and rinsed with anhydrous ethanol for three times. The barrier layer of the titania membrane was hereafter dissolved by using a 5 wt% NH₄F–1 M H₂SO₄ solution. Finally, an Ag sol was

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coated onto one of the surfaces of the membrane and dried at $100\,^{\circ}\text{C}$ for 12 h.

Cu was deposited using a constant current (-1 mA) in 0.5 M CuSO₄ solution at room temperature with magnetic stirring at 500 rpm, and Pt gauze was used as the counter electrode.

2.2. Materials characterization

The morphological characterization of the samples was carried out by scanning electron microscopy (JSM 6700F; JESL, Tokyo, Japan). For transmission electron microscopy, the template with Cu nanowire was separated by ultrasonically dispersion in 1 ml of water, then a drop of the solution was placed on a Cu grid covered with carbon film. A transmission electron microscope (JEM-3010; JESL, Tokyo, Japan) was employed to study the morphology of the array structure. X-ray diffraction measurement was carried out on a MXPAHF X-ray diffractometer with Cu K α radiation (λ = 1.54056 Å).

3. Results and discussion

A schematic method employed to fabricate Cu nanowire arrays inside the nanoporous titania template is shown in Fig. 1. Ti is firstly anodized and ultrasonic agitated, the free-standing nano-

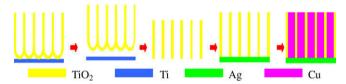


Fig. 1. Schematic of the process employed to produce the Cu nanowire arrays into the porous anodic titania template.

porous titania membrane will be gained. The nanoporous titania membrane is thereafter soaked in 0.5 wt% NH₄F-1 M H₂SO₄ solution and then two-end-opened TiO₂ membranes are achieved. After drying, one side of the membrane is daubed by Ag sol and the nanoporous titania template is achieved. The template is transferred into 0.5 M of CuSO₄ aqueous solution and the Cu nanowire is grown inside the template.

Fig. 2a shows the digital image of the free-standing TiO_2 nanotube membrane and it is displayed that the dimension of the membrane is about $2.5 \times 2.0 \, \mathrm{cm}^2$. The top surface of the membrane by SEM is shown in Fig. 2b and it is evident that the array is consisted from very regular tubes with a diameter of about 120 nm and a wall thickness of about 10 nm. The layer (in Fig. 2c) is about 15.9 μ m and has very smooth walls typical of nanotubes grown in organic electrolytes [5]. Fig. 2d shows the back layer side and the bottom of the nanotube is closed.

Fig. 3a and b present the SEM of the free-standing TiO_2 nanotube membrane after soaking in 5 wt% NH_4F-1 M H_2SO_4 solution for 5 and 15 min, respectively, it can be seen that the pores are partly opened after 5 min soaking and completely opened after 15 min. Fig. 3c is SEM of the top side of the membrane after 15 min soaking and it shows the nanotube becomes nanowire. To overcome this problem, we have designed a new device (a homemade box), which can prohibit the top side of TiO_2 membrane immerging the $NH_4F-H_2SO_4$ solution (in Fig. 3d). The bottom side and four walls of the box are composed of organic glass, the top side of the box is composed of steel nets, and TiO_2 membrane is laid on steel nets. During the experiment, the box is filled with 5 wt% NH_4F-1 M H_2SO_4 solution. The result is similar with Fig. 2b.

The images of a highly ordered free-standing TiO₂ nanotube filled with Cu are shown in Fig. 4a, b by SEM and Fig. 4c by TEM, and it is seen that the TiO₂ nanotube is filled with Cu. The array/ template composite is also assessed by X-ray diffraction (XRD), as seen in Fig. 4d. The arrays are dominated by the 111 Cu peak,

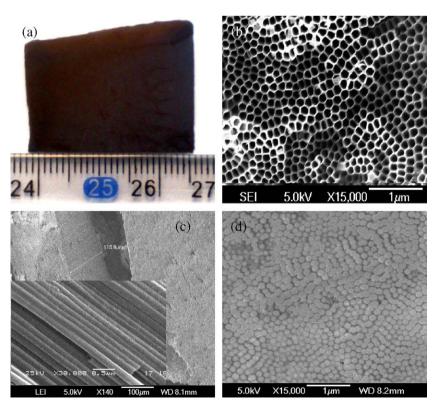


Fig. 2. The digital image (a) of the titania membrane after anodization at 60 V for 48 h, and the SEM image of the titania membrane after anodization: top view (b), cross-sectional view (c), bottom view (d).

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