

## Effects of RE<sub>2</sub>O<sub>3</sub> (RE = Tm, Sc, Yb) addition on the superconducting properties of ErBa<sub>2</sub>Cu<sub>3</sub>O<sub>y</sub>

R. Kita<sup>a,\*</sup>, N. Hosoya<sup>a</sup>, N. Otawa<sup>a</sup>, S. Kawabata<sup>a</sup>, T. Nakamura<sup>a</sup>, O. Miura<sup>b</sup>, M. Mukaida<sup>c</sup>, K. Yamada<sup>c</sup>, A. Ichinose<sup>d</sup>, K. Matsumoto<sup>e</sup>, M.S. Horii<sup>f</sup>, Y. Yoshida<sup>g</sup>

<sup>a</sup> Graduate School of Science and Technology, Shizuoka University, Johoku 3-5-1, Naka-ku, Hamamatsu, Shizuoka 432-8561, Japan

<sup>b</sup> Tokyo Metropolitan University, Minamiosawa 1-1, Hachioji, Tokyo 192-0364, Japan

<sup>c</sup> Kyushu University, Hakozaki 6-10-1, Higashi-ku 4-3-16, Fukuoka 992-8510, Japan

<sup>d</sup> CRIEPI, Nagasaki 2-6-1, Yokohama, Kanagawa 240-0916, Japan

<sup>e</sup> Kyusyu Institute of Technology, 1-1 Sensui-cho, Tobata-ku, Kitakyushu, Fukuoka 804-8550, Japan

<sup>f</sup> Kochi University of Technology, Kami-shi, Kochi 782-8502, Japan

<sup>g</sup> Nagoya University, Furo-cho, Chikusa-ku, Nagoya 464-8603, Japan

### ARTICLE INFO

#### Article history:

Accepted 22 January 2009

Available online 30 May 2009

#### PACS:

74.72.Dn

74.72.Jt

#### Keywords:

RE<sub>2</sub>O<sub>3</sub>

Tm<sub>2</sub>O<sub>3</sub>

Sc<sub>2</sub>O<sub>3</sub>

Yb<sub>2</sub>O<sub>3</sub>

Artificial pinning center

### ABSTRACT

We investigated the effects of added Tm<sub>2</sub>O<sub>3</sub>, Sc<sub>2</sub>O<sub>3</sub>, and Yb<sub>2</sub>O<sub>3</sub> on the superconducting properties of sintered Er123 samples. Tm<sub>2</sub>O<sub>3</sub> addition caused the least  $T_c$  degradation, exhibiting a  $T_c$  above 90 K even for 17 vol% addition. Samples with added Sc<sub>2</sub>O<sub>3</sub> maintained a  $T_c$  at above 90 K up to an addition of 7.2 vol%, while Yb<sub>2</sub>O<sub>3</sub>-containing samples showed a monotonic decrease in  $T_c$  with increased vol% of added Yb<sub>2</sub>O<sub>3</sub>. Tm<sub>2</sub>O<sub>3</sub>-containing samples exhibited a slight increase in  $J_c(0.1\text{ T})/J_c(0)$  and had constant  $J_c$  values even for 17 vol% addition. XRD and SEM results indicate that the Tm<sub>2</sub>O<sub>3</sub> is very stable in the superconducting matrix.

© 2009 Elsevier B.V. All rights reserved.

## 1. Introduction

Improvement of the critical current density ( $J_c$ ) of REBa<sub>2</sub>Cu<sub>3</sub>O<sub>y</sub> (RE123) superconductors is essential for the application of RE123 superconducting melt-textured bulk and thin films to high-performance permanent magnets or high-current-carrying wires. Introduction of artificial pinning centers (APC) into RE123 superconducting phases is a very effective technique for enhancing  $J_c$ . Significant improvements in  $J_c$  have been reported for melt-textured REBCO by adding RE<sub>2</sub>BaCuO<sub>5</sub> (RE211) [1], Y<sub>2</sub>Ba<sub>4</sub>CuMO<sub>y</sub> [2], ZrO<sub>2</sub> [3], barium oxides (BaZrO<sub>3</sub> [3], BaCeO<sub>3</sub> [4], or BaSnO<sub>3</sub> [5]) to create APCs. For RE123 thin films, the  $J_c$ - $B$  properties were also enhanced significantly by the introduction of nanometer-sized rod-shaped barium oxide into superconducting matrices [6–9]. Recently, we reported a good correlation between the stability of barium oxides in RE123 films and sintered compounds [10]. This suggests that the study of the addition of various materials into

sintered samples will be of great help in the search for new APC materials suitable for high- $J_c$  thin films in high-magnetic fields. RE<sub>2</sub>O<sub>3</sub> is known to have high thermal stability and to be relatively chemically inert. However, there are few reports on the stability of RE<sub>2</sub>O<sub>3</sub> in the superconducting phase [11].

In the present study, we have investigated the effects of Tm<sub>2</sub>O<sub>3</sub>, Sc<sub>2</sub>O<sub>3</sub> and Yb<sub>2</sub>O<sub>3</sub> additions on the superconducting properties of Er123 and compared their stability in the RE123 matrix.

## 2. Experimental

Er123 samples were prepared from Er<sub>2</sub>O<sub>3</sub> (99.9%), BaCO<sub>3</sub> (99.95%), and CuO (99.99%) using a standard solid-phase reaction technique. Appropriate amounts of the reagents were thoroughly ground and calcined at 1173 K for two periods of 12 h in air, with intermediate regrinding. The resultant Er123 powder was pressed into pellets and sintered at 1233 K in air for two periods of 12 h, with intermediate regrinding. High-purity (99.9%) Tm<sub>2</sub>O<sub>3</sub>, Sc<sub>2</sub>O<sub>3</sub>, and Yb<sub>2</sub>O<sub>3</sub> powders were then added to the pulverized Er123 pellets in concentrations of 1–17 vol%. The resulting powders were

\* Corresponding author. Tel./fax: +81 53 478 1129.

E-mail address: [terkita@ipc.shizuoka.ac.jp](mailto:terkita@ipc.shizuoka.ac.jp) (R. Kita).

thoroughly ground, pressed into pellets and then sintered at 1233 K in air for 12 h. All samples were subsequently annealed in flowing oxygen at 1183 K for 12 h, cooled to 773 K with a 12 h stay, then cooled to room temperature in a furnace. The electrical resistivity of the samples was measured by a standard four-probe technique to determine their critical temperatures ( $T_c$ ).  $J_c$  of the samples at 77 K was calculated from  $B$ – $M$  curves measured by a superconducting quantum interference device (SQUID) at magnetic fields of 0–1.0 T. X-ray diffractometry (XRD) was employed to identify the phases present in the samples and determine lattice constants. The sample surfaces were characterized using a scanning electron microscope (SEM). The chemical compositions of the precipitates on the sample surfaces were determined by energy-dispersive X-ray spectroscopy (EDX).

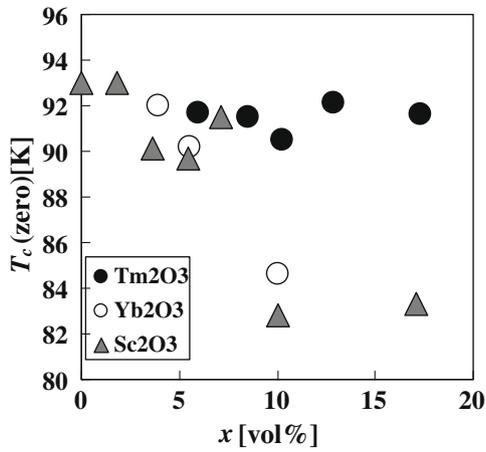


Fig. 1. Dependence of  $T_c$  on the amount of RE<sub>2</sub>O<sub>3</sub> added to ErBCO sintered samples.

tivity of the samples was measured by a standard four-probe technique to determine their critical temperatures ( $T_c$ ).  $J_c$  of the samples at 77 K was calculated from  $B$ – $M$  curves measured by a superconducting quantum interference device (SQUID) at magnetic fields of 0–1.0 T. X-ray diffractometry (XRD) was employed to identify the phases present in the samples and determine lattice constants. The sample surfaces were characterized using a scanning electron microscope (SEM). The chemical compositions of the precipitates on the sample surfaces were determined by energy-dispersive X-ray spectroscopy (EDX).

### 3. Results and discussion

The  $T_c$  of the RE<sub>2</sub>O<sub>3</sub>-containing Er123 sintered samples are shown as a function of the amount added in Fig. 1. Samples with Tm<sub>2</sub>O<sub>3</sub> added showed the least degradation of  $T_c$ , and maintained  $T_c$  above 90 K even for 17.3 vol% addition. Maintaining high- $T_c$  is comparable to BaSnO<sub>3</sub> with high stability in the superconducting phase [10]. Sc<sub>2</sub>O<sub>3</sub>-containing samples maintained a  $T_c$  above 90 K up to 7.2 vol% added, but showed degradation of  $T_c$  above 10 vol% added. Yb<sub>2</sub>O<sub>3</sub>-containing samples showed a monotonic decrease in  $T_c$  with increasing Yb<sub>2</sub>O<sub>3</sub> content. These results indicate that Tm<sub>2</sub>O<sub>3</sub> oxides are relatively stable in the superconducting matrix.

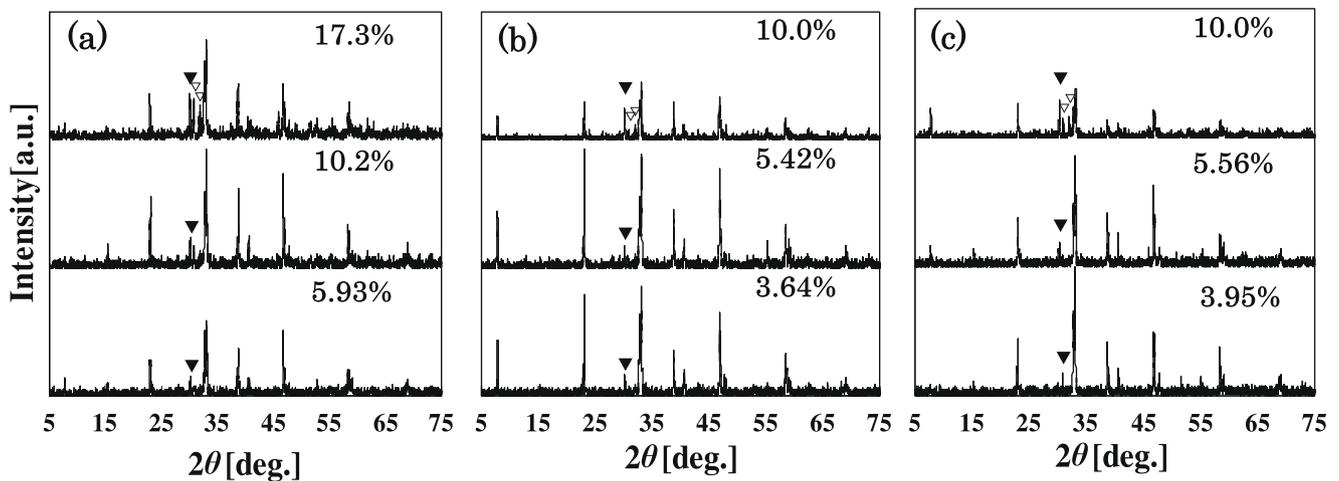


Fig. 2. XRD  $\theta$ – $2\theta$  spectra for (a) Tm<sub>2</sub>O<sub>3</sub>, (b) Sc<sub>2</sub>O<sub>3</sub>, and (c) Yb<sub>2</sub>O<sub>3</sub>-containing ErBCO sintered samples.

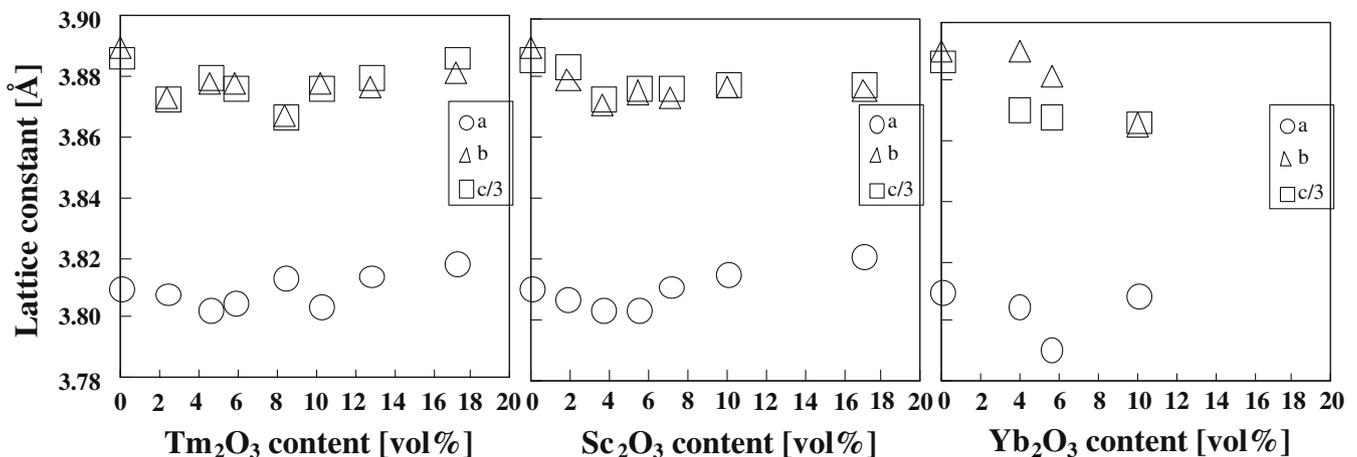


Fig. 3. Lattice constants of the RE<sub>2</sub>O<sub>3</sub>-containing samples calculated from the XRD patterns as a function of the amount of RE<sub>2</sub>O<sub>3</sub> added.

Download English Version:

<https://daneshyari.com/en/article/1819375>

Download Persian Version:

<https://daneshyari.com/article/1819375>

[Daneshyari.com](https://daneshyari.com)