



# In-situ growth of LiMnPO<sub>4</sub> on porous LiAlO<sub>2</sub> nanoplates substrates from AAO synthesized by hydrothermal reaction with improved electrochemical performance



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## ABSTRACT

LiAlO<sub>2</sub> composited LiMnPO<sub>4</sub>/C cathode material has been synthesized by in-situ growth hydrothermal method. The coexistence of minor LiAlO<sub>2</sub> in LiMnPO<sub>4</sub> plays an important role in electrochemical properties. The composite is characterized by XRD, SEM, HRTEM, Raman microprobe spectroscopy and their electrochemical properties are also studied. It shows that the LiAlO<sub>2</sub> is composed of pure  $\alpha$ -LiAlO<sub>2</sub> phase with flaky-shaped nanoplates. LiAlO<sub>2</sub> nanoplates porous structure is inherited from anodic aluminum oxide (AAO) structure and serves as substrates to grow LiMnPO<sub>4</sub> nanocrystals, which provide a high surface area with a porous structure. The structure of LiMnPO<sub>4</sub> is not affected by LiAlO<sub>2</sub> nanoplates compositing. Among the investigated samples, the one with 6 wt.% LiAlO<sub>2</sub> exhibits a higher specific discharge capacity of 142.8 mAh/g at 0.05C rate with a high capacity retention of 94.8% after 50 cycles. Electrochemical impedance spectroscopy (EIS) results show that the lithium diffusion constant ( $D_{Li^+}$ ) for 6 wt.% content LiAlO<sub>2</sub>-LiMnPO<sub>4</sub>/C electrode is  $2.07 \times 10^{-14}$  cm<sup>2</sup>/s which is higher than the values for other contents LiAlO<sub>2</sub>-LiMnPO<sub>4</sub>/C electrode. The LiAlO<sub>2</sub> is effective in suppressing the increase of the interfacial resistance between the electrode/electrolyte interface during charge-discharge cycling.

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## 1. Introduction

Being used as potential electrodes for Li-ion batteries, LiMnPO<sub>4</sub> has recently attracted attention due to its high energy capacity (20% higher than LiFePO<sub>4</sub>), nontoxicity, lower cost and better chemical and thermal stability [1–5]. So, LiMnPO<sub>4</sub> becomes a promising material for the application of electric vehicles (EVs) and hybrid electric vehicles (HEVs) [6–8]. However, low intrinsic electronic conductivity ( $<10^{-10}$  S/cm) and ionic conductivity [9,10] makes its practical capacity much lower than the theoretical capacity and inhabits its further use in commercial applications. Consequently, many research efforts have been focused on overcoming these problems and improving the electrochemical performance of LiMnPO<sub>4</sub>.

To date, many efforts such as conductive surface coating [11–13], particle size reduction [14,15] and cations/metal oxides doping have been applied to improve the electronic conductivity and lithium ion mobility of LiMnPO<sub>4</sub>. On increasing the electronic conductivity, it is a common practice to prepare carbon coated materials. A thin carbon layer can provide a path for electrons without blocking the access of Li<sup>+</sup> ions. The effectiveness of surface carbon depends on the type and amount of the carbon source being used. Additionally, other surface modification with metallic particles can increase the electronic conductivity of LiMnPO<sub>4</sub>, but little for ionic conductivity. Recently, coating of Li-contained oxides such as Li<sub>4</sub>Ti<sub>5</sub>O<sub>12</sub>, LiNbO<sub>3</sub>, Li<sub>2</sub>TiO<sub>3</sub>, Li<sub>2</sub>ZrO<sub>3</sub>, Li<sub>2</sub>SiO<sub>3</sub> and LiAlO<sub>2</sub> on LiCoO<sub>2</sub> particles and Li<sub>1.5</sub>Ni<sub>0.25</sub>Mn<sub>0.75</sub>O<sub>2.5</sub> cathode materials have been reported as an effective modification technique [16–23]. For example, Wu et al. [24,25] reported LiMnPO<sub>4</sub>/C with an average size of 75 nm synthesized by sol-gel combined ball milling and quenching method showing a reversible capacity of 131 mAh/g at 0.05C. Recently, Wu and coworkers synthesized xLiMn<sub>0.9</sub>Fe<sub>0.1</sub>PO<sub>4</sub>·yLi<sub>3</sub>V<sub>2</sub>(PO<sub>4</sub>)<sub>3</sub>/C composite cathode

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materials by a ball-milling and post-calcination method. Among them,  $5\text{LiMn}_{0.9}\text{Fe}_{0.1}\text{PO}_4 \cdot \text{Li}_3\text{V}_2(\text{PO}_4)_3/\text{C}$  exhibited the highest reversible capacity of 158.1 mAh/g at 0.05C. The sample also shows good cycling performance at various C-rates. Two modification methods have been employed simultaneously to improve the electrochemical properties of  $\text{LiMnPO}_4$ . And moreover, it can effectively enhance the electrochemical performance due to their good conductivity and internal tunnels for  $\text{Li}^+$  transportation during charge/discharge processes [26,27].

Furthermore,  $\text{LiAlO}_2$  particles have a wide application on lithium secondary battery [28]. The compound  $\text{LiAlO}_2$  is known to exist in three possible allotropes, namely,  $\alpha$ - $\text{LiAlO}_2$ ,  $\beta$ - $\text{LiAlO}_2$ ,  $\gamma$ - $\text{LiAlO}_2$ . Compared with  $\beta$  and  $\gamma$  phase, the Li atom in  $\alpha$ - $\text{LiAlO}_2$  is generally the most reactive and solely extractable, which is due to the good Li electrophilicity, a significantly smaller energy for Li binding to its neighboring atoms and a dramatic difference in energetic behavior upon dislocating a Li atom from its equilibrium position in different phases [29]. In the structure of  $\alpha$ - $\text{LiAlO}_2$  there are only octahedral aluminum ions with the same lamellar hexagonal system. The  $\alpha$ - $\text{LiAlO}_2$  phase has the hexagonal lattice structure (R3m,  $a = 2.799\text{Å}$ ,  $c = 14.18\text{Å}$ ) [30] of layered Li-O-Al-O, because the Al-O bonds are very stable. For all of these applications, fine lithium aluminate powders as a superionic conductor material with high surface area and considerable lithium-ion conductivity are desirable for improving the rate capability and cycling stability of  $\text{LiMnPO}_4$ . To the best of our knowledge, however, there have been no reports on  $\text{LiAlO}_2$  modified  $\text{LiMnPO}_4$  composite materials as yet. Inspired by these findings, we try to form  $\text{LiAlO}_2$ - $\text{LiMnPO}_4/\text{C}$  composite particles, and investigate the effects of  $\text{LiAlO}_2$  on electrochemical performances of olivine structure  $\text{LiMnPO}_4$  materials.

## 2. Experimental

All of the chemicals were used as received without any pretreatment or further purification.

### 2.1. Synthesis of AAO and $\text{LiAlO}_2$

Nanoporous AAO templates were fabricated by a two-step anodization process. A high-purity aluminum plate (99.999%,  $8\text{ cm} \times 3\text{ cm} \times 0.3\text{ mm}$ ) was annealed at  $600^\circ\text{C}$  for 5 h in  $\text{N}_2$  atmosphere and degreased in acetone. After electropolishing to a mirror finish in a 1:4 (v/v) perchloric acid/ethanol solution at room temperature for 2~3 min at a current of 15 A, the anodization was carried out in a 0.3 mol/L phosphoric acid solution at 80 V and

$25^\circ\text{C}$  for 3 h. Then the alumina layer was removed in a mixture aqueous solution of phosphoric acid and chromic acid. Subsequently, the anode was anodized for 10 h in the same condition again and became AAO template with high ordered nanoporous arrays ( $\sim 100\text{ nm}$  in diameter). The  $\text{LiAlO}_2$  preparation was on the AAO template substrate. The AAO and  $\text{Li}_2\text{CO}_3$  (at a 1:3 molar ratio) was mixed in the deionized water, the mixture then moved into Teflon-lined autoclave a few hours. The resulting precursors were dried at  $80^\circ\text{C}$  under vacuum and then calcined at  $550^\circ\text{C}$  in air. Two-dimensional nanosturcture porous  $\text{LiAlO}_2$  nanoplates were synthesized.

### 2.2. Synthesis of $\text{LiMnPO}_4/\text{C}$

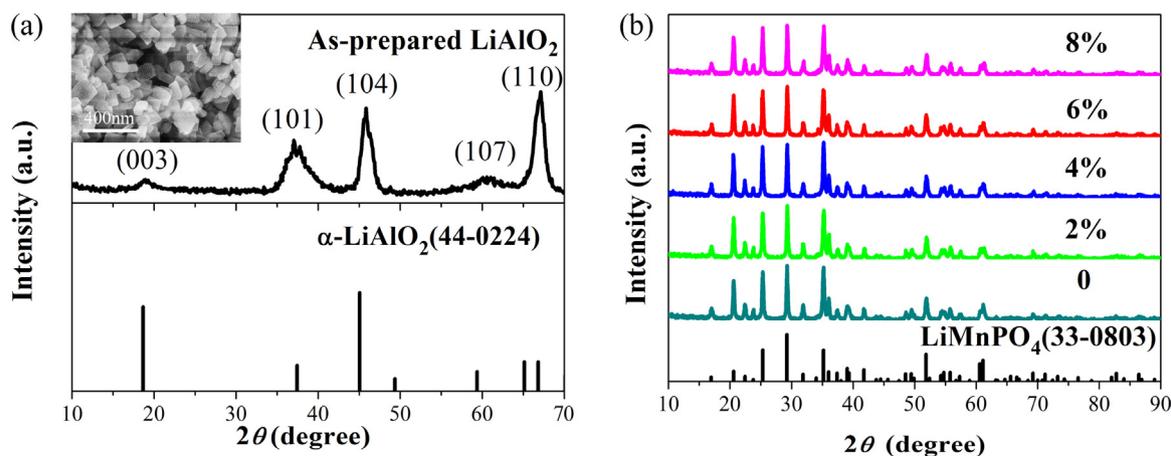
The  $\text{Li}_3\text{PO}_4$  precursor was precipitated by dropping 1.7 mol/L  $\text{H}_3\text{PO}_4$  (85%) into a 2.0 mol/L  $\text{LiOH}$  (99%) aqueous solution using the peristaltic pump (the flow rate was 3.3 ml/min). The resulting white precipitates were washed with distilled water and dried in air at  $120^\circ\text{C}$ . For all syntheses, the reactants such as  $\text{MnSO}_4 \cdot \text{H}_2\text{O}$  (99%) and the above  $\text{Li}_3\text{PO}_4$  (in a 3:1 molar ratio) dissolved in 1:2 volume ratio PEG400- $\text{H}_2\text{O}$  mixed solvents and stirred for 0.5 h. Then, the slurry was transferred into a Teflon-lined autoclave, and heated at  $160^\circ\text{C}$  for 9 h. Finally, the products were filtered off and washed several times with deionized water and anhydrous alcohol thoroughly and dried at  $80^\circ\text{C}$  for 3 h to get  $\text{LiMnPO}_4$  precursor. The as-prepared  $\text{LiMnPO}_4$  precursor was mixed with ascorbic acid at 1:0.25 weight ratio within alcohol solution, mixed homogeneously and dried for 12 h. Then the dried composite powder was calcined at  $550^\circ\text{C}$  for 3 h under  $\text{N}_2$  atmosphere to get  $\text{LiMnPO}_4/\text{C}$  material.

### 2.3. Synthesis of $\text{LiAlO}_2$ - $\text{LiMnPO}_4/\text{C}$ composite

The procedures for different contents of  $\text{LiAlO}_2$ - $\text{LiMnPO}_4/\text{C}$  composite materials were prepared by in-situ growth hydrothermal method.  $\text{Li}_3\text{PO}_4$  and  $\text{MnSO}_4 \cdot \text{H}_2\text{O}$  (99%) (in a 1:3 molar ratio) and a certain contents  $\text{LiAlO}_2$  substrates were dissolved in 1:2 volume ratio PEG400- $\text{H}_2\text{O}$  mixed solvents and stirred for 0.5 h. Then we put it under vacuum treatment to ensure the subsequent immersion. The subsequent procedures were the same with the preparation of  $\text{LiMnPO}_4/\text{C}$  mentioned above. Finally, the  $\text{LiAlO}_2$ - $\text{LiMnPO}_4/\text{C}$  composite material was obtained.

### 2.4. Characterization

The samples were studied by X-ray diffraction (XRD, DX-2500) using  $\text{Cu K}\alpha$  radiation ( $\lambda = 1.5406\text{Å}$ ) from  $2\theta = 10$ – $90^\circ$  at a scan



**Fig 1.** X-ray diffraction patterns of (a)  $\text{LiAlO}_2$  nanoplates after heat treatment at  $550^\circ\text{C}$ , inset: SEM of  $\text{LiAlO}_2$ , and (b) prepared with different amounts of  $\text{LiAlO}_2$ .

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