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Two-dimensional nanocomposites based on tungsten oxide nanoplates and graphene nanosheets for high-performance lithium ion batteries

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1. Introduction

Lithium ion batteries (LIBs) have been considered as the most important energy storage devices for use in portable equipment such as cellular phones and lap-top computers as well as in highcapacity devices such as hybrid electrical vehicles. However, the rapidly advanced applications demand high capacity, fast charging/discharging rate, and improved cycle life [1–5]. For the advanced electrochemical characteristics in LIBs, both the composition and structure of negative and positive active materials for better LIBs characteristics have been investigated over the past few decades. In particular, electrode materials of LIBs require a high electrical conductivity specific surface area, low cost, and longterm electrochemical stability [6,7].

Typically, carbon-based materials have been used as a promising anode, even showing lower theoretical capacity in LIBs [8–12]. Furthermore, transition metal oxides have been investigated to replace graphite and have exhibited high reversible capacities, resulting from their unique electrochemical properties. However, most transition metal oxide suffers from poor rate performance leading to drastic capacity drop due to low electrical conductivity and large volume change during charge-discharge

http://dx.doi.org/10.1016/j.electacta.2015.02.121 0013-4686/© 2015 Elsevier Ltd. All rights reserved. cycling [13–17]. Accordingly, many efforts have been made to overcome the critical problems using composite structures with carbon-based or other electrical conductive materials. Among the composites consisting of transition metal oxides and carbon-based materials, much interest has been given to two-dimensional structure composites containing graphene nanosheet (GNS), leading to higher specific capacity and better high-rate stability [18–27]. Graphene nanocomposites show good potential in LIBs due to their low energy barrier for Li-ion, large surface area, and excellent electrical conductivity. Tungsten oxide (WO₃) may be one of the powerful candidates in the transition metal oxides because of its environmental friendliness, and high theoretical capacity of 693 mAh g⁻¹ [28–36]. The charge-discharge mechanism of WO₃ is based on conversion reaction process, which requires the formation of a metal and lithium oxide, as follows:

WO₃ (crystalline) + 6 Li⁺ + 6 e⁻ \rightarrow W + 3 Li₂O (1)

W+3 $Li_2O \rightarrow WO_3$ (crystalline)+6 Li^+ +6 e⁻ (2)

Herein, we prepared the two-dimensional nanocomposites (WO_3/GNS) containing WO_3 nanoplates and GNS with varying amount of GNS synthesized using a hydrothermal method and heating process. The structure and morphology of the nanocomposites were characterized using XRD, SEM, and TEM analysis. The capacity, capacity retention, and high rate cycling performance





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ABSTRACT

Well-defined nanocomposite electrodes based on active materials and graphene have been known to hold improved lithium ion reaction properties such as the lithium ion insertion/desertion reaction, cycle life, and high rate performance. Here, we report two-dimensional nanocomposites consisting of WO₃ nanoplates and graphene nanosheets (GNS) using a hydrothermal method and heating process for use in high-performance lithium ion batteries. A compilation of the data from XRD, SEM, and TEM suggests that the nanocomposites consist of high crystalline WO₃ nanoplates and GNS. The capacity and high rate cycling performance of the nanocomposites as an anode for lithium ion batteries are evaluated using coin cells. The nanocomposite electrode with an optimized amount of GNS exhibits high reversible capacity, good capacity retention, and excellent high rate cycling performance.

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Fig. 1. (a) Wide scan and (b) small range $(26^{\circ}-30^{\circ})$ XRD patterns of the as-prepared WO₃/GNS-0, WO₃/GNS-5, WO₃/GNS-10, and WO₃/GNS-20. The red bars indicate XRD patterns of the bulk WO₃ (JCPDS No. 43-1035).

of the nanocomposites as an anode for lithium ion batteries were evaluated using coin cells.

2. Experimental Section

WO₃/GNS composite electrodes were synthesized using a hydrothermal method and heating process. To synthesize WO₃/GNS, the pre-oxidized GNS (ENanoTec $4 \sim 5$ layer) powders with intended amounts of 5, 10, and 20 wt% were dispersed in 70 mL of 5 M hydrochloric acid solution (Samchun, 35–37%) with continuous sonication for 90 min at room temperature. With vigorous stirring, 1 g ammonium tungstate ((NH₄)₁₀H₂(W₂O₇)₆, Aldrich, 99.99%) was dissolved in the mixture solution. After sonication, the mixtures of tungsten salt and GNS powder in 5 M hydrochloric acid solution were heated 140 °C for 6 h. The resulting dispersion was cooled to room temperature, and the precipitates were then filtered and washed with di-water and ethanol several times. After drying at 50 °C in an oven, the temperature for the obtained powders was increased to 10 °C min⁻¹ at 450 °C and kept in a box furnace at 450 °C for 2 h in an air atmosphere [37–40].

The size and morphology of the as-prepared samples were observed on a field-emission scanning electron microscope (FE-SEM, JSM-6700F, JEOL Ltd.) and a field-emission transmission electron microscope (FE-TEM, Tecnai G2 F30 system operating at 300 kV). The crystal structure was studied using an X-ray diffractometer (XRD, D2 PHASER, Bruker AXS) with a Cu K_{α} (λ = 0.15418 nm) source with a nickel filter. The source was operated at 30 kV and 10 mA. The 2 θ angular scan from 10° to 80° was explored at a scan rate of 0.5° min⁻¹. Thermogravimetric analysis (TGA) curves were obtained in a thermal analyzer (SDTA851, Mettler Toledo) in the range of 25–900 °C at a heating rate of 5 °C min⁻¹ under an air flow of 60 cm³ min⁻¹.

To assemble the coin cells (size 2032, Hohsen Corporation), the nanocomposite electrodes were used as a working electrode and evaluated with respect to a lithium foil (FMC Corporation) as a counter/reference electrode. The as-synthesized WO₃/G composite as an active material, Ketjen black as a conducting material (Alfa Aesar), and polyvinylidene difluoride (Alfa Aesar) as a binder with a weight ratio of 70:10:20 were mixed in 1-methyl-2-pyrrolidinone solvent (Aldrich, 99%). The well-mixed slurry was coated onto a Cu foil as a current collector using a doctor blade technique and then dried in an air atmosphere at 100 °C for 12 h. The average loading amount of active materials for all electrodes was $0.5 \sim 0.6$ mg cm⁻². All of the WO₃/GNS electrodes were coated uniformly on 15 mm

thick copper foil substrates and the thickness of the electrodes was $40 \sim 45 \ \mu m$ after pressing. The coin cells were assembled inside an argon-filled glove box (<5 ppm, H₂O and O₂). The electrodes of the cells were separated by a porous polypropylene membrane (Celgard 2400) and an electrolyte solution consisting of 1.1 M LiPF₆ in (ethylene carbonate:dimethyl carbonate)=(1:1) as a solvent mixture (Techno Semichem Co., Ltd). The charge–discharge cycling was performed within the voltage range of 0–3.0 V vs. Li/Li⁺ on a battery test instrument (PEBC 50.2 PnE Solution Co., Ltd) under constant temperature of 25 °C. Electrochemical impedance spectroscopy measurements (EIS) were performed on the AUTOLAB electrochemical workstation by applying an AC voltage of 5 mV amplitude over the frequency range 100 kHz–0.01Hz at open circuit potential. All the measurements were performed at room temperature.

3. Results and discussion

The XRD patterns of the as-prepared WO₃/GNS nanocomposites (denoted as WO₃/GNS-0, WO₃/GNS-5, WO₃/GNS-10, and WO₃/GNS-20, respectively) with varying amounts of graphene (0, 5, 10, and 20 wt%, respectively) are shown in Fig. 1(a). The unit cell parameters of WO₃ for the as-prepared samples were determined to be a = 7.297 Å, b = 7.539 Å, and c = 7.688 Å in agreement with those of the bulk WO₃ material with a space



Fig. 2. TGA data of WO₃/GNS-5, WO₃/GNS-10, and WO₃/GNS-20.

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