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# The electrochemical behavior of Cl<sup>−</sup> assisted Al<sup>3+</sup> insertion into titanium dioxide nanotube arrays in aqueous solution for aluminum ion batteries

Yingying Liu<sup>a,d</sup>, Shangbin Sang<sup>a,∗</sup>, Qiumei Wu<sup>b,∗</sup>, Zhouguang Lu<sup>c</sup>, Kaiyu Liu<sup>a</sup>, Hongtao Liu<sup>a</sup>

<sup>a</sup> College of Chemistry and Chemical Engineering, Central South University,410083

<sup>b</sup> Key State laboratory of powder metallurgy, Central South University,410083

<sup>c</sup> Department of Materials Science & Engineering, South University of Science and Technology of China, 518055

<sup>d</sup> Innovation base of energy and chemical materials for graduate students training

#### a r t i c l e i n f o

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#### a b s t r a c t

The titanium dioxide nanotube array  $(TiO<sub>2</sub>-NTA)$  is prepared by a two-step anodic oxidation method and annealed at 450 ℃ subsequently. The Al-inserted TiO<sub>2</sub>-NTA is prepared by polarizing the TiO<sub>2</sub>-NTAs electrode at 0.4 mA for 2 min in 1 mol/L AlCl<sub>3</sub>. The results show that the Al-inserted sample still remains pure anatase TiO<sub>2</sub> phase (by XRD) and keeps intact nanotube array structure (by FE-SEM). The X-ray photoelectron spectroscopy (XPS) analysis indicates that the insertion of  $Al^{3+}$  into TiO<sub>2</sub>-NTAs facilitates in the reduction of  $Ti^{4+}$  to  $Ti^{3+}$ . Electrochemical investigation on the  $Al^{3+}$  insertion process reveals that presence of Cl<sup>−</sup> ions plays vital role for the effective insertion and de-insertion of Al3+ into/from the TiO<sub>2</sub>-NTA.

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### **1. Introduction**

In recent years, the demand for secondary/rechargeable batteries with higher operating voltages, improved cycling stability, higher power densities, enhanced safety, and lower initial costs has increased to meet new needs for smaller, lighter, more powerful electronic devices [\[1\].](#page--1-0)

Titanium dioxide nanotube array referred as  $TiO<sub>2</sub>$ -NTA hereinafter is one of the most extensively investigated compounds as the electrode material for high performance secondary/rechargeable batteries. The performance using as active materials for dye-sensitized solar cells, photocatalysis, photoelectrochemistry or electrochromic devices has significantly enhanced due to the several advantages such as a high surface area, short solid state diffusion paths of the TiO<sub>2</sub>-NTA  $[2,3]$ . Moreover TiO<sub>2</sub>-NTA is considered as excellent electrode candidate for constructing threedimensional microbatteries that were promising power sources for miniaturized electronics due to large surface area and short

∗ Corresponding authors.

E-mail addresses: [shangbinsang@yahoo.com](mailto:shangbinsang@yahoo.com) (S. Sang), [wuqiumei20073@126.com,](mailto:wuqiumei20073@126.com) [sangsb@126.com](mailto:sangsb@126.com) (Q. Wu).

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diffusion distance, which favors the electron and Li<sup>+</sup> transfer  $[4-7]$ . Nevertheless, the shortage of lithium resources will be further aggravated by the development of the electric vehicle industry with Li-ion batteries as power sources in the near future  $[8]$ . Compared with lithium, aluminum is relatively abundant and inexpensive metal in the earth's crust. An aluminium-based redox couple involves three electron transfers during the electrochemical charge/discharge reactions, which provides competitive storage capacity in comparison to the single-electron Li-ion battery. It has higher electrochemical equivalent, namely, 8.04 Ah/cm<sup>3</sup> (5.98 Ah/cm<sup>3</sup> higher than lithium). The obvious advantages of aluminum make it more attractive for future power source development [\[9\].](#page--1-0)

Furthermore, recent studies showed that  $TiO<sub>2</sub>-NTA$  exhibited superior performance as cathode alternative for aluminum ion batteries since  $Al^{3+}$  ions can be reversibly inserted and extracted inside the TiO<sub>2</sub>-NTA probably due to the small radius of  $Al^{3+}$  (53.5 pm vs 76 pm of Li<sup>+</sup>). Preliminary study showed that  $TiO<sub>2</sub>$ -NTA has a reversible capacity of about 75 mAh/g at  $4 \text{ mA/cm}^2$  in AlCl<sub>3</sub> aqueous electrolyte solution  $[10]$ . The feasibility of a safer aluminum ion battery with aqueous electrolyte encourages us to make further investigation to improve the electrochemical performance. A further study of our group showed that it was difficult for  $Al^{3+}$  to insert into





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 $TiO<sub>2</sub>-NTA$  in  $Al<sub>2</sub>(SO<sub>4</sub>)<sub>3</sub>$  aqueous solution. The same phenomenon was also observed in an aprotic electrolyte system by N. Jayaprakash et al. [\[1\],](#page--1-0) by using two different electrolytes of 1:1 v/v of Al triflate in PC/THF and 1.1:1 molar ratio of AlCl<sub>3</sub> in [EMIm]Cl at room temperature. No redox anodic and cathodic peaks was observed in the CV curves for the Al-ion battery using  $V_2O_5$  nanowire cathode and Al triflate in PC/THF as the electrolyte in the measured voltage range of -0.75-4.2V, however, a pair of obvious reversible redox peaks were detected for the Al-ion battery with AlCl<sub>3</sub> in [EMIm]Cl electrolyte. It is anticipated that chloride ion (Cl<sup>−</sup>) plays a vital role in the insertion process of Al<sup>3+</sup>. The presence of Cl<sup>−</sup> promotes the reversible and effective insertion and de-insertion of  $Al^{3+}$  into and from perhaps which would be an important theory basis in the research of aluminum storage behavior of transitional metal nanomaterials (such as anatase  $TiO<sub>2</sub>$  nanotube arrays and  $V<sub>2</sub>O<sub>5</sub>$  nano-wire), which might be the fundamentals governing the improved electrochemical performance of in aqueous solution for aluminum ion batteries in the future [\[10\].](#page--1-0) In this paper, cyclic voltammetry (CV), electrochemical impedance spectra (EIS) and galvanostatic electrochemical charge-discharge method are performed to investigate the effect of Cl<sup>−</sup> on Al3+ insertion/extraction process into the  $TiO<sub>2</sub>$ -NTA in a series of aqueous solution. Virtually perfect hexagonally ordered arrays of  $TiO<sub>2</sub>$  were synthesized by a two-step anodic oxidation method followed by calcinations at 450  $\rm ^{\circ}$ C [\[11,12\].](#page--1-0) The effect on Al<sup>3+</sup> insertion/extraction process into the TiO<sub>2</sub>-NTA of Cl<sup>−</sup> is discussed.

#### **2. Experimental**

#### 2.1. Preparation of the TiO<sub>2</sub> nanotube arrays

All chemical agents used were purchased from Changsha Shenghua Chemical Agents Corporation limitd, China). Anatase  $TiO<sub>2</sub>$ -NTA were prepared by a two-step anodic oxidation method. Prior to the anodization, the titanium foils (99.7% purity, thickness of 0.1mm, Baoji Queen Titanium Corporation limited, Shanxi Province, China) with dimensions of  $1.5 \times 2.5$  cm<sup>2</sup> were polished with 800 mesh abrasive paper, then ultrasonically cleaned in distilled water, acetone (99.5% AR.), alcohol (99.7% AR.) and the mixture of 15 wt% hydrogen peroxide (30% AR.) and 0.06 mol/L sodium hydroxide (96.0% AR.) respectively for 15 min. Finally, a chemical polish was carried out in 4 wt% HF (40.0% AR.) for 10 min. This was the key step in our pre-experiments, which determined the surface status of the titanium foils, and all foils were dried in a  $N_2$  stream. During anodization process, a twoelectrode system (a Pt electrode as cathode electrode, titanium foils as anode electrode) was performed, and temperature-controlled water bath (DF-101S) was used to control temperature. A stirring apparatus was used to make sure that the reaction occurred in a uniform electrolyte. A regulated DC power supply (DAZHENG PS-6050) was used to ensure a uniform voltage. Anodization was carried out in ethylene glycol (99.7% AR.)/deionized water (24:1) electrolyte containing 0.3 wt% ammonium fluoride (99.6% AR.) at 60V for 4h to form a  $TiO<sub>2</sub>$  nanotube layer at ambient temperatures [\[13,14\].](#page--1-0) After the anodization, the oxide layer would be ultrasonically peeled off the titanium foils. Next, we used the remaining titanium foils as a new anode electrode. The following anodization operation was the same as before, however, the second anodization was performed for 1 h. Highly ordered  $TiO<sub>2</sub>$ -NTAs were fabricated through this two-step anodizing electrochemical procedure. After two-step oxidation the samples were washed with distilled water and dried in an  $N_2$  stream. Thermal annealing was performed with muffle (SK-5-12K) at 450 ◦C for 2 h at a ramping rate of  $1 °C$ /min to obtain an anatase structure.

#### 2.2. Morphological characterization

For surface morphological characterization of the  $TiO<sub>2</sub>$ -NTAs layers, top view and cross-sectional were analyzed by fieldemission scanning electron microscopy (Nova NanoSEM 230). The crystal structure of sample was determined by X-ray diffraction (XRD, Phillips Xpert-MPD PW3040 diffractometer) using Cu K $\alpha$ radiation ( $\lambda$ =1.5406 Å). The chemical composition of the layers was characterized by X-ray photoelectron spectroscopy (XPS, K-Alpha 1063) using Al Ka radiation of 1468.6 eV.

#### 2.3. Electrochemical performance measurement

All electrochemical experiments were carried out by threeelectrode system. The self-standing  $TiO<sub>2</sub>$ -NTA film was used as the working electrode, silver-silver chloride electrode as the reference electrode, and a platinum foil with dimensions of 2 cm  $\times$  2 cm as the counter electrode. Cyclic voltammetry (CHI 660D) was performed with a series of electrolyte, namely,  $0.50 \text{ mol/L Al}_2(\text{SO}_4)_3$  (99.0%) AR.), 1.00 mol/L AlCl<sub>3</sub> (97.0% AR.) and two groups of mixed solution (0.25 mol/L  $Al<sub>2</sub>(SO<sub>4</sub>)<sub>3</sub>$  mixed with different concentrations of NaCl solution; 1.50 mol/L NaCl mixed with different concentrations of  $Al_2(SO_4)_3$  solution). CVs at different scan rates were conducted in the mixed solution (0.25 mol/L  $\text{Al}_2(\text{SO}_4)_3$  and 1.50 mol/L NaCl). Electrochemical impedance spectra (PARSTAT2273 Advanced Electrochemical System) were carried out in a frequency range from 0.1 Hz to 100 kHz. The apparent area of the TiO<sub>2</sub> working electrode for CV and EIS tests were about 0.25  $\text{cm}^2$  after coating epoxy resin on the edge and back of the metallic titanium foil. In the galvanostatic electrochemical charge-discharge test, the working electrode (TiO<sub>2</sub>-NTAs) with total area of 1 cm<sup>2</sup> was charged for 1 min and discharged to -0.7 V (vs. Ag/AgCl) at a current density of  $4 \text{ mA/cm}^2$  in the mixed solution 0.25 mol/L  $Al<sub>2</sub>(SO<sub>4</sub>)<sub>3</sub>$  and 1.5 mol/L NaCl. The Alinserted sample used for XRD, SEM and XPS spectra analysis were prepared by polarizing the TiO<sub>2</sub>-NTAs electrode at 0.4 mA for 2 min in 1 mol/L AlCl<sub>3</sub>, followed by washing with deionized water and drying in vacuum at ambient temperature. All the electrochemical tests were performed at ambient temperature.

#### **3. Results and discussion**

#### 3.1. Morphology and Crystal structure

[Fig.](#page--1-0) 1a and Fig. 1b show the top-view and side-view FE-SEM image of the as-prepared  $TiO<sub>2</sub>$ -NTAs, respectively. As we can see from the images, the nanotubes with smooth walls and uniform dimensions are formed on the Ti substrate. The average outside and inside diameters of the nanotubes are about 160 nm and 130 nm, respectively, and the obtained thickness is approximately  $3 \mu m$ as shown in [Fig.](#page--1-0) 1b. The highly ordered vertically oriented tubelike nanostructures with the short ion diffusion path through walls of nanotubes and large surfare area between active materials and electrolyte are in favor of the transfer of electron and  $Al^{3+}$ . The topview and side-view FE-SEM images of Al-inserted TiO<sub>2</sub>-NTA are shown in [Fig.](#page--1-0) 1c and [Fig.](#page--1-0) 1d, as shown in the figure, the nanotube array structure remains relatively intact with  $Al^{3+}$  inserted into lattice, which adequately proves that the  $TiO<sub>2</sub>$ -NTA film is potential electrode material in the aluminum-ion battery due to the special structure and stability.

X-ray diffraction analysis was conducted to investigate the crys-tal structure of the TiO<sub>2</sub>-NTAs, and the results are shown in [Fig.](#page--1-0) 2, a set of reflection centered at  $2\theta$  degree of  $25.3^\circ$ ,  $37.8^\circ$ ,  $48.0^\circ$ ,  $53.9^\circ$ , and 55.1◦ can be indexed to the (101), (004), (200), (105), and (211) peaks, respectively, of tetragonal anatase TiO<sub>2</sub> (JCPDS no. 84-1285), coexisting with the metallic Ti peak form the titanium foil substrate.

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