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Manganese dioxide nanosheet arrays grown on graphene oxide as an advanced electrode material for supercapacitors



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ABSTRACT

A simple hydrothermal method has been provided for synthesis of the MnO_2 -GO hybrid with birnessite-type manganese dioxide (MnO_2) nanosheet arrays on graphene oxide (GO) flakes. To assess the properties of the as-synthesized hybrid for use in supercapacitors, cyclic voltammetry (CV) and galvanostatic charging/discharging (GCD) measurements are performed. The MnO_2 -GO hybrid exhibits enhanced specific capacitance (213 F g⁻¹ at current density of 0.1 A g⁻¹), rate capability (even 80 F g⁻¹ at 10 A g⁻¹) and capacitance retention ratio (98.1% at 1 A g⁻¹ after 1,000 cycles) compared with the GO and MnO_2 . Furthermore, the electrochemical impedance spectroscopy (EIS) demonstrates the electrochemical resistance of MnO_2 nanosheet which directly grows on EIS0 is reduced, indicating easier access for intercalation/deintercalation of charges in EIS1.

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1. Introduction

Over the past few years, considerable effort has been devoted to the development of alternative energy storage/conversion devices with high power and energy densities because of the everincreasing environmental problems and the up-coming depletion of fossil fuels [1]. As an intermediate system between dielectric capacitors and batteries, supercapacitors have attracted a great deal of attention owing to their higher power densities relative to secondary batteries [2,3]. Nevertheless, the disadvantages of them, such as low energy density and high production cost, have been identified as major challenges for the furtherance of supercapacitor technologies. To overcome the obstacle of low energy density, active electrode materials with high capacity performance are indispensable [4].

As one of the green supercapacitor electrode materials, MnO₂ shows great potential to replace RuO₂ due to its high specific capacitance, environmental compatibility, low cost, and abundance in nature [5,6]. It is known that, for an electrode to perform effectively in a supercapacitor cell, access of the electrolyte to as much of the internal and external electrode surface as possible is

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desirable. Attempts have often been made to tailor MnO_2 synthesis to give the highest specific surface area and pore volume [7–9]. Lately, the birnessite-type MnO_2 nanostructures, in which exhibit specific capacitances of over $110 \ F \ g^{-1}$ with surface areas of only $17 \ m^2 \ g^{-1}$, are attracting great interest due to the small thickness of birnessite platelets allows the access of the electrolyte to almost all of the MnO_2 [6,10]. However, the birnessite MnO_2 nanomaterials are suffered from some disadvantages, such as poor electronic conductivity, easy aggregation and partial dissolution, thus less utilization of MnO_2 in most cases [11]. To improve the electrochemical performance, the combination of birnessite MnO_2 nanostructures with other materials such as graphene, carbon nanotubes, carbon fibers and mesoporous carbons has been investigated extensively [4,5].

On the other hand, GO has tremendous attraction, because of its unique structure, amazing electrical and mechanical properties. It has applications in wide areas including catalysts, solar cell, gas sensor, fuel cell, battery, supercapacitor, antibacterial study, and so on [12–14]. Recently, GO has been investigated as a suitable support for manganese oxide/oxyhydroxide loadings in electroactive materials for supercapacitors [11]. Chen et al. reported needle-like MnO₂-GO nanocomposite achieved the excellent capacitive behavior with the specific capacitance of 197.2 F g⁻¹ at 0.2 A g⁻¹ current density [15]. Wang et al. investigated MnOOH nanowire-GO and MnO₂ nanowire-GO hybrids for supercapacitors, maximum specific capacitances of 76 F g⁻¹ and 155 F g⁻¹ at current density of 0.1 A g⁻¹ were obtained, respectively [16]. Gund et al. studied

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the capacitive performance of hybrid Mn_3O_4/GO composite and shown the specific capacitance up to $180~F~g^{-1}$ at scan rate of $20~mV~s^{-1}$ [17]. However, research on the syntheses and supercapacitive behaviors of materials combining the GO with birnessite MnO_2 nanostructures is merely carried. Seredych et al. fabricated MnO_2/GO composites containing GO of different oxidation levels and low crystallization birnessite type MnO_2 , and they showed specific capacitances of $125~Fg^{-1}$ and $63~Fg^{-1}$ at current density of $0.1~A~g^{-1}$, respectively [13]. Nevertheless, the structure of these composites had not been rationally designed to ensure making the best use of each ingredient's advantages and the synergetic effect between them.

Herein, we present a MnO₂-GO hybrid with birnessite-type MnO₂ nanosheet arrays on GO substrate which serves as an electroactive material for supercapacitors. Structurally, the MnO₂ nanosheet arrays work effectively as a pseudosupercapacitor material for energy storage, whereas the GO is selected as a substrate because (i) GO can possess a high surface area for the deposition of MnO₂, accordingly inhibits the aggregation of MnO₂ nanosheets; (ii) the oxygen-containing functionalities (epoxide, hydroxyl, carbonyl and carboxyl groups), acting as anchor sites, enable the subsequent in situ formation of MnO₂ nanostructures attaching on the surfaces of GO sheets [12]; (iii) the high hydrophilicity of GO makes MnO⁴⁻ ions react with every graphite layer, leading to simultaneous MnO₂ precipitation on each GO sheets in aqueous solution [15]. Here, the hybrid has been synthesized by a simple hydrothermal approach. The structure, morphology, electrochemical performances of the hybrid are fully investigated and discussed. In addition, the electrochemical properties of the pristine GO and birnessite MnO₂ nanosheet have also been investigated for comparison.

2. Methods

2.1. Synthesis of materials

All the reagents were analytical grade and used without further purification, and GO was prepared from powdered flake graphite (400 mesh) by a modified Hummers method as described previously [18,19]. In the typical synthesis procedure of the MnO₂-GO hybrid, GO powder (10 mg) was dispersed in deionized (DI) water (30 mL) by ultrasonic vibration for 2 h, then KMnO₄ powder (30 mg) was added into above GO suspension and stirred by magnetic bar for 15 min. Subsequently, the total solution was transferred into a stainless-steel autoclave with a capacity of 40 mL, sealed and heated at 100 °C for 12 h. Finally, the autoclave was cooled to room temperature naturally. The resultant product was collected by centrifugation and repeatedly washed with deionized water. The final product was dried in a vacuum at 60 °C overnight. For comparison, MnO₂-GO hybrids were similarly prepared by hydrothermal reaction for 4 h and 8 h, respectively, and the birnessite MnO_2 nanosheet was synthesized in the absence of GO sheets at 120 °C for 48 h (see supporting information I).

2.2. Characterizations

The crystallographic information of the products was investigated by X-ray diffraction (XRD, Shimadzu X-ray diffractometer 6000, Cu Ka radiation) with a scan rate of 10°/min. Raman spectra were measured on a Laser Raman spectroscopy (HORIBA Jobin Yvon LabRAM HR800) with 532 nm line of an Ar ion laser as an excitation source. Morphologies of as-obtained materials were observed on a field emission scanning electron microscopy (FESEM, Hitachi S4800) at an accelerating voltage of 20 kV and transmission

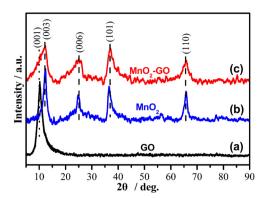


Fig. 1. XRD patterns of the (a) GO, (b) MnO2 and (c) MnO2-GO hybrid.

electron microscope (TEM, Tecnai-G2-F30) with an accelerating applied potential of 20 kV, respectively.

2.3. Electrochemical Measurements

The electrochemical properties of the materials were investigated by the CV, GCD and EIS techniques on a CHI 660E electrochemical workstation (Shanghai CH Instrument Company, China) at room temperature. The working electrode was fabricated by mixing the as-synthesized material (the MnO₂-GO hybrids, GO or MnO₂) with 15 wt% acetylene black and 5 wt% polytetrafluoreneethylene (PTFE) binder. A small amount of ethanol was added to the mixture to produce a homogeneous paste. Then the mixture was pressed onto nickel foam current-collectors (1 cm \times 1 cm) and dried at 80 °C for 12 h. The mass of the active material was about 6 mg. Before the electrochemical tests, the prepared electrode was soaked overnight in a 1 M Na₂SO₄ solution. Electrochemical measurements were carried out in a conventional three-electrode cell with 1 M Na₂SO₄ aqueous solution as the electrolyte. Platinum foil and a saturated calomel electrode (SCE) were used as the counter electrode and the reference electrode, respectively.

3. Results and discussion

3.1. Microstructure characterizations

Fig. 1 shows the powder X-ray diffraction (XRD) patterns of GO, MnO₂ and MnO₂-GO hybrid. As shown in Fig. 1a, the most intensive peak of GO at around 2θ =10.2° corresponds to the (001) reflection, and the interlayer spacing (0.86 nm) is much larger than that of pristine graphite (0.34 nm) due to the introduction of oxygencontaining functional groups on the graphite sheets [16,19]. The MnO₂ obtained by self-limiting deposition of KMnO₄ in hydrothermal treatment shows four peaks at 12.32°, 24.78°, 36.63°, and 65.47° (Fig. 1b), which can be indexed to (003), (006), (101), and (110) reflections of K-birnessite MnO₂ (JCPDS NO. 86-0666) [20,21]. In contrast, the diffraction peaks of the MnO₂-GO hybrid (Fig. 1c) are similar to those of the as-synthesized birnessite MnO₂ but the (001) reflection peak of layered GO has disappeared, indicating the GO sheets are covered by MnO₂ layer.

Raman spectroscopy can be used to gain more information about the structure of MnO_2 and GO sheets in MnO_2 -GO hybrid. Raman spectra of the GO, MnO_2 and MnO_2 -GO hybrid are shown in Fig. 2. It can be seen from Fig. 2a that there are two diagnostic peaks of GO centered around 1367 and 1590 cm⁻¹, which correspond to the breathing mode of κ -point phonons of A_{1g} symmetry and the first-order scattering of the E_{2g} phonons, respectively [16,22]. For the Raman spectrum of MnO_2 (Fig. 2b), three peaks at 638, 571, and 491 cm⁻¹ are observed and can be assigned to the presence of birnessite-type MnO_2 phase [23]. These Raman bands are

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