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#### Short communication

# Core-shell structured poly(methyl methacrylate)-coated zirconium dioxide nanoparticle and its dispersion stability



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#### ARTICLE INFO

#### ABSTRACT

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Keywords: Core-shell structure Dispersion stability Organic-inorganic nanocomposite To improve dispersion stability of ZrO<sub>2</sub> nanoparticles dispersed in a poly(4-vinylphenol-co-methyl methacrylate) (PVP-co-PMMA)/2-methoxyethanol solution, they were coated with poly(methyl methacrylate) (PMMA) by dispersion polymerization. Cross-linked PMMA-coated ZrO<sub>2</sub> particles were characterized by SEM and TEM. Chemical composition was examined by FT-IR spectroscopy. Mass ratio of PMMA to the ZrO<sub>2</sub> was analyzed by TGA. Density (1.97 g/cm<sup>3</sup>) of the PMMA coated ZrO<sub>2</sub> particles was much lowered compared to 5.46 g/cm<sup>3</sup> of ZrO<sub>2</sub>, and the improved dispersion stability of the coated particles in the PVP-co-PMMA/2-methoxyethanol solution was demonstrated using a Turbiscan.

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#### 1. Introduction

Organic–inorganic hybrid materials including core–shell structured particles have attracted considerable attentions because of their advantageous characteristics resulted from combining properties of organic materials and inorganic materials [1,2]. Organic materials, mainly polymers, possess the flexibility, lightness and plastic mechanical properties with the concomitant ease of processing and structural diversity, while inorganic materials have the potential for wide electronic properties, magnetic and dielectric transitions, as well as thermal stability. Therefore these composite particles have been applied for a range of applications, such as bioseparation [3], gate insulator in electronics [3–6], and electrorheological and magnetorheological materials [6].

Meanwhile, polymeric gate dielectric materials have been developed to apply their mechanical flexibility and low-cost production to low-temperature processible gate dielectric materials for large-area electronics [7–9]. On the other hand, the high threshold voltage due to the low dielectric properties of polymeric insulators is one of the drawbacks that need to be overcome, and a variety of inorganic dielectric materials have been incorporated with polymeric systems for the organic–inorganic composites. Among the suitable inorganic materials, zirconium dioxide (ZrO<sub>2</sub>) nanoparticles have attracted considerable attention because of

\* Corresponding author. Tel.: +82 32 860 7486. *E-mail address:* hjchoi@inha.ac.kr (H.J. Choi). their optical and electrical properties with applications to electrooptic, piezoelectric and insulator [10].

As one of the polymeric gate insulators, the poly(4-vinylphenol) group has been studied as a gate dielectric [11,12] in terms of the solution processability, thin film quality, and surface chemistry. Furthermore, the poly(vinyl acetate)/clay composite increased the dielectric constant because of the high-k of the clay (Na<sup>+</sup>-montmorillonite) [13]. The dispersion stability of inorganic nanoparticles is important for achieving the excellent performance of these composites. The dispersion stability is affected by the particle volume fraction [14], density difference between the particles and medium [15], affinity between the particles and medium, etc. The surfactants can be used to disperse the aggregated nanoparticles uniformly, resulting in serious gateleakage currents in organic thin film transistors to the solution mixture and composite film [16]. On the other hand, the surfactants reduced the dielectric constant of this system. The approach for designing and fabricating in a core-shell model has attracted considerable attention because nanoparticles with a well defined core and shell structures often exhibit improved physical and chemical properties compared to their single component counterparts. A range of methods have been reported for the fabrication of nanoparticles in a core-shell structured model over the past few decades. Some researchers reported that core-shell structured particles could be fabricated by the encapsulation of inorganic particles in a polymer shell [17,18]. The uniformly coated polymer shell provides an improvement in the affinity between the core-shell structured nanoparticles and polymer matrix.

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Furthermore, organic-inorganic hybrid materials in a coreshell structure can be synthesized by dispersion polymerization [19], seeded polymerization [3] and Pickering emulsion polymerization [20] to decrease the inorganic particle density and improve the affinity between the particles and medium. This study examined the improvement in the dispersion stability of organic-inorganic composites.

#### 2. Experimental

Zirconium dioxide particles (ZrO<sub>2</sub>) (Aldrich, USA particle size <100 nm, density:  $5.46 \text{ g/cm}^3$ ) were used as a core inorganic material. Methacrylic acid (MAA) (Junsei, Japan) was used as a pre-treatment for the modification of ZrO<sub>2</sub> surface. Methyl methacrylate (MMA) (Junsei, Japan) and ethylene glycol dimethacrylate (EGDMA) (Aldrich, USA) were used as the monomer and a cross-linking agent, respectively. Polyvinylpyrrolidone (PVP) (Aldrich, USA,  $M_w$  = 55,000 g/mole) was used as a dispersion stabilizer. Azobisisobutyronitrile (AIBN) (Junsei, Japan) was used as the initiator for dispersion polymerization. Methanol (MeOH) (OCI, Korea) was used as the solvent. Poly(4-vinylphenol-co-methyl methacrylate) (PVP-co-PMMA) (Aldrich, USA,  $M_w$ : 9000 g/mole) and 2-methoxyethanol (Aldrich, USA) were used as a matrix polymer and solvent for the medium phase, respectively.

The cross-linked PMMA coated ZrO<sub>2</sub> particles were synthesized by dispersion polymerization, as shown in Fig. 1. The surface of the ZrO<sub>2</sub> nanoparticles was initially modified with MAA by sonication for 30 min to improve the affinity between MMA and ZrO<sub>2</sub> nanoparticles. The treated ZrO<sub>2</sub> was washed with methanol by centrifuge. The treated ZrO<sub>2</sub> was added to 100 g of methanol containing 3 g of PVP, and stirred for 30 min. At the same time, MMA and EGDMA were mixed with AIBN, and the mixture was then added to the dispersion. Dispersion polymerization was continued for 24 h at 60 °C at 300 rpm. After the reaction, the synthesized particles were washed sequentially with methanol and di-water, and freeze-dried. To prepare the thin films for the surface analysis of ZrO<sub>2</sub> and the cross-linked PMMA coated ZrO<sub>2</sub> suspensions, PVP-co-PMMA solution was prepared by dissolving the PVP-co-PMMA in 2-methoxyethanol at a rate of 5 wt% for the medium phase. To prepare the ZrO<sub>2</sub> suspensions and PMMAcoated ZrO<sub>2</sub> suspensions, the individual particles were dispersed in a PVP-co-PMMA solution with a fraction of 1 vol%. For higher dispersity, the suspensions were ultrasonicated for 1 h and homogenized for 3 min at 10,000 rpm using a homogenizer. Both pure ZrO<sub>2</sub> suspension and PMMA-coated ZrO<sub>2</sub> suspension were spin coated onto a cleaned silicon wafer at 1000 rpm for 30 s and then baked at 100 °C for 5 min. Subsequently, the sample was baked at 200 °C in a dry oven for 1 h to cure the polymer.

The synthesized ZrO<sub>2</sub> particles coated with cross-linked PMMA were characterized by scanning electron microscopy (SEM) (S-4300, Hitachi, Japan) and transmission electron microscopy

(TEM) (Philips CM200). In addition, average particle size and size distribution were analyzed by dynamic light scattering apparatus (DLS, ELS-8000, Otsuka, Japan). The prepared cross-linked PMMAcoated ZrO<sub>2</sub> particles' structure was analyzed by Fourier transform infrared (FT-IR) (Perkin Elmer System 2000) spectroscopy in the range from 400 to 4000 cm<sup>-1</sup>. The amount of the polymer coating was measured by thermogravimetric analysis (TGA) (TA instrument O50, USA) in nitrogen gas. The dispersion stability of pure ZrO<sub>2</sub> and the particles obtained in the PVP-co-PMMA solution was analyzed using a Turbiscan (Classic, MA 2000). The surface morphology of the spin coated film of pure ZrO<sub>2</sub> and cross-linked PMMA coated ZrO<sub>2</sub> dispersed in PVP-co-PMMA solution was analyzed by SEM and atomic force microscopy (AFM) (NS4A). The rheological properties of the cross-linked PMMA-coated ZrO<sub>2</sub> particles and pure ZrO<sub>2</sub> dispersed in a PVP-co-PMMA solution in 2methoxyethanol at 1 vol% were observed using a rotational rheometer (MCR 300, Physica, Germany) with CC17 geometry.

#### 3. Results and discussion

Fig. 2 shows SEM images of both pure ZrO<sub>2</sub> particles (a) and synthesized cross-linked PMMA coated ZrO<sub>2</sub> particles (b). In Fig. 2(a), pure ZrO<sub>2</sub> particles exhibited a fairly smooth surface and polydispersed size distribution. After encapsulating the ZrO<sub>2</sub> surface with PMMA, the surface morphology became rough, as shown in Fig. 2(b). The synthesized PMMA-coated ZrO<sub>2</sub> particles were larger than the pure  $ZrO_2$  particles. Fig. 2(c) and (d) presents TEM images of the pure ZrO<sub>2</sub> particles and synthesized crosslinked PMMA-coated ZrO<sub>2</sub> particles, respectively, with a spherical shape. After the coating process of PMMA, the core and shell of the synthesized particles were ZrO<sub>2</sub>, which was more opaque, and PMMA, which was more transparent, respectively, as shown in Fig. 2(b). The density of the synthesized PMMA-coated ZrO<sub>2</sub> particles was 1.97 g/cm<sup>3</sup>, which was lower than that of pure ZrO<sub>2</sub> (5.46 g/cm<sup>3</sup>). As shown in Fig. 2(b), the product was not coated as a single particle because of the aggregation force of the pure particles. Therefore, the particle size distribution was large.

Average particle size and size distribution of cross-linked PMMA coated  $ZrO_2$  particles are indicated in Fig. 3. Fabricated PMMA coated  $ZrO_2$  particles are widely distributed over a range of 430 nm–2  $\mu$ m, with the maximum intensity positioned at about 998 nm which is regarded as an average particle size. This wide range size distribution depends on how many particles are aggregated in the process of coating with PMMA and coating thickness of PMMA layer.

Fig. 4 shows the FT-IR spectra of pure  $ZrO_2$  particles (a) and cross-linked PMMA coated  $ZrO_2$  particles (b). The two peaks at 2997 cm<sup>-1</sup> and 2952 cm<sup>-1</sup> can be assigned to the C–H bond stretching vibrations of the –CH<sub>3</sub> and –CH<sub>2</sub>– groups, respectively, while the peak at 1444 cm<sup>-1</sup> can be attributed to the bending vibration of the C–H bond of the –CH<sub>3</sub> group. The two peaks at



Fig. 1. Schematic diagram for the synthesis of crosslinked PMMA coated ZrO<sub>2</sub> particles.

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