



# Enhancement in performance of polymer solar cells by introducing solution-processed dipole interlayer



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## ABSTRACT

Metal fluorides (e.g., LiF and BaF<sub>2</sub>) are usually introduced as an electron transport layer in conventional polymer solar cells (PSCs). However, because they are insoluble, vacuum deposition is inevitable, resulting in a high process cost and the difficulty of introduction in PSCs. In this study, we fabricated inverted PSCs with a double interlayer by using solution-processable cesium fluoride and ZnO to improve the electron extraction. The power conversion efficiency was enhanced from 7.5% to 8.2% by a 9.3% increase in the fill factor. In addition, we demonstrated the enhancement of the FF by atomic force microscopy, space charge limited current, and photoluminescence.

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## Introduction

Bulk heterojunction (BHJ) polymer solar cells (PSCs) have attracted considerable attention because of their flexibility, lightness, easy fabrication, and suitability for large-scale and low-cost solution processes. Recently, they achieved a power conversion efficiency (PCE) over 10% in a single structure with the addition of processing additives [1–4] and the introduction of an interlayer [5–7]. However, to commercialize these PSCs, research must be conducted regarding the improvement of their lifespan. In the photoactive layer of the PSCs in a conventional structure, the BHJ is formed by the electron donor ‘D-A type polymer’ and electron acceptor ‘fullerene-derivative’ [8]. This photoactive layer is situated between a poly (3,4-ethylenedioxythiophene):poly(styrene sulfonate)(PEDOT:PSS)-coated indium tin oxide (ITO) anode and a metal cathode with a low work function. However, because of the acidic and hygroscopic characteristics of PEDOT:PSS, ITO electrodes corrode. Moreover, the metal electrode with a low work function easily oxidizes in air, resulting in the decrease of device stability. Therefore, the power conversion efficiency (PCE) decreases [6,9].

To compensate for these disadvantages, there have been many recent studies on inverted-structures. In such a structure, as shown in Fig. 1(a), a photoactive layer is positioned between an

ITO cathode coated with metal oxide (e.g., ZnO or TiO<sub>x</sub>) or polyelectrolyte and a metal anode with a high work function. PEDOT:PSS is not used in this structure. Because of the metal electrode with a high work function, self-encapsulation occurs. Therefore, inverted-structure exhibits superior long-term stability compared with conventional structures [10–14]. Regarding the photoactive layer, which is usually formed by spin-coating, a donor polymer with a relatively low surface energy is situated in the upper part of the layer, and a fullerene acceptor with a relatively high surface energy is positioned in the lower section [15]. This type of vertical phase separation has a direction opposite to the movement of the electric charge in a conventional structure, which is not ideal. In contrast, for an inverted structure, the PCE can be enhanced by taking advantage of this vertical phase separation [14,15]. Using the active layer of poly{thieno[3,4-b]thiophene/benzodithiophene} (PTB7):[6,6]-phenyl C<sub>71</sub> butyric acid methyl ester (PC<sub>71</sub>BM), Cao et al. [14] reported PCEs of 8.24% and 9.2% in conventional and inverted structures, respectively. So et al. [10] reported PCEs of 4.9% and 5.8% in conventional and inverted structures, respectively.

Another typical material for electron-transport layers (ETLs) is metal fluoride (e.g., LiF, BaF<sub>2</sub>). If this material is thin enough for an electron to pass through the gap between the cathode and active layer, band bending occurs. Consequently, the energy barrier between the cathode and active layer decreases, and electrons move efficiently owing to the tunneling effects [16–18]. Lee et al. [16] reported the improvement of the efficiency from 2.1% to 4.0% for a poly(3-hexylthiophene) (P3HT):[6,6]-phenyl C<sub>61</sub> butyric acid

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methyl ester (PC<sub>61</sub>BM)-based device by introducing LiF and BaF<sub>2</sub> between the cathode and active layer in a conventional structure. Sun et al. [18] reported that the efficiency was improved from 2.5% to 3.5% by introducing LiF on the ITO in an inverted structure.

However, because LiF or BaF<sub>2</sub> layer must be formed through vacuum deposition owing to the low solubility, the process becomes complicated, and the unit price increases. Lemmer et al. [19] reported the introduction of CsF with a high solubility in the alcohol-based solvent as an ETL. However, a P3HT:PC<sub>61</sub>BM-based inverted-structure device exhibited a low efficiency (1.5%).

In this study, an inverted-structure device with a double interlayer (sol-gel zinc oxide (ZnO) layer + CsF layer formed through solution process) was fabricated. This device exhibited an efficiency of 8.2% after the electron mobility was enhanced by adding tunneling effects via metal fluoride.

## Experimental

### Materials

In both PTB7 and PC<sub>71</sub>BM, 1-material products were acquired and used. A ZnO precursor was fabricated by performing hydrolysis at 50 °C for 12 h after dissolving 1 g of zinc acetate dihydrate (Zn(CH<sub>3</sub>COO)<sub>2</sub>·2(H<sub>2</sub>O), 99.8%, Aldrich) and 0.28 g of monoethanolamine (HOCH<sub>2</sub>CH<sub>2</sub>NH<sub>2</sub>, 99.5%, Aldrich) in 10 ml of 2-methoxyethanol ((H<sub>3</sub>CO)CH<sub>2</sub>CHOH, 99.8%, Aldrich) [20]. For an ETL, a solution obtained by dissolving CsF (99.9%, Acros) in isopropyl alcohol was used.

### Device fabrication

First, the patterned ITO was sonicated in a detergent solution, isopropyl alcohol, and deionized water for 10 min each and cleaned. For dehydration, it was baked at 120 °C for 10 min. The ITO surface was treated for 20 min using an ultraviolet-ozone cleaner. A ZnO layer was formed after spin-coating the ZnO precursor solution on the cleaned ITO glass and annealing it at 150 °C. Then, an ETL was placed on the ZnO layer by spin-coating the CsF solution. To form a

photoactive layer on the CsF layer, the devices were transported to a N<sub>2</sub>-filled glove box. Then, a photoactive layer was formed by spin-coating a PTB7:PC<sub>71</sub>BM (1:1.5, w/w) in a chlorobenzene/1,8-diiodooctane (97:3, v/v) solution. Lastly, the anode was acquired by depositing 5 nm of MoO<sub>3</sub> and 100 nm of Ag in a high-vacuum chamber. The CsF layer was optimized by adjusting the concentration from 0.2 to 2 mg ml<sup>-1</sup>.

### Measurements

The current density–voltage (*J*–*V*) characteristics of the fabricated devices were measured using a Keithley 2400 source measurement unit and an AM 1.5G solar simulator (Oriel 96000 150 W solar simulator). In addition, the incident photon to current conversion efficiency (IPCE) was measured with respect to the optimized device and a reference. The thin films' surface morphology was measured using atomic force microscopy (AFM, PSIA XE-100, non-contact mode). To determine the changes in electron mobility through the CsF layer, an electron-only device was fabricated, and the electron mobility was measured using the space charge limited current (SCLC) method. Furthermore, to calculate the quenching rate of devices, the photoluminescence (PL) was measured using PerkinElmer LS55.

## Results and discussion

Fig. 1 shows the structure of inverted devices (Fig. 1a), including the molecular structures of the donor PTB7 (Fig. 1b) and the acceptor PC<sub>71</sub>BM (Fig. 1c). In the device, a ZnO layer spin-coated with a ZnO precursor was formed [20]. Then, the ETL was placed on the ZnO layer by spin-coating the CsF solution. In the N<sub>2</sub>-filled glove box, to form the photoactive layer, the mixed solution comprising PTB7 and PC<sub>71</sub>BM (1:1.5, w/w) was spin-coated. Finally, MoO<sub>3</sub> and Ag anodes were vacuum-deposited. The ETL was optimized by changing the concentration from 0.2 to 2 mg ml<sup>-1</sup>.

Fig. 2 shows the *J*–*V* curves and IPCE spectra of the devices, and Table 1 summarizes the results. As Table 1 shows, a device with

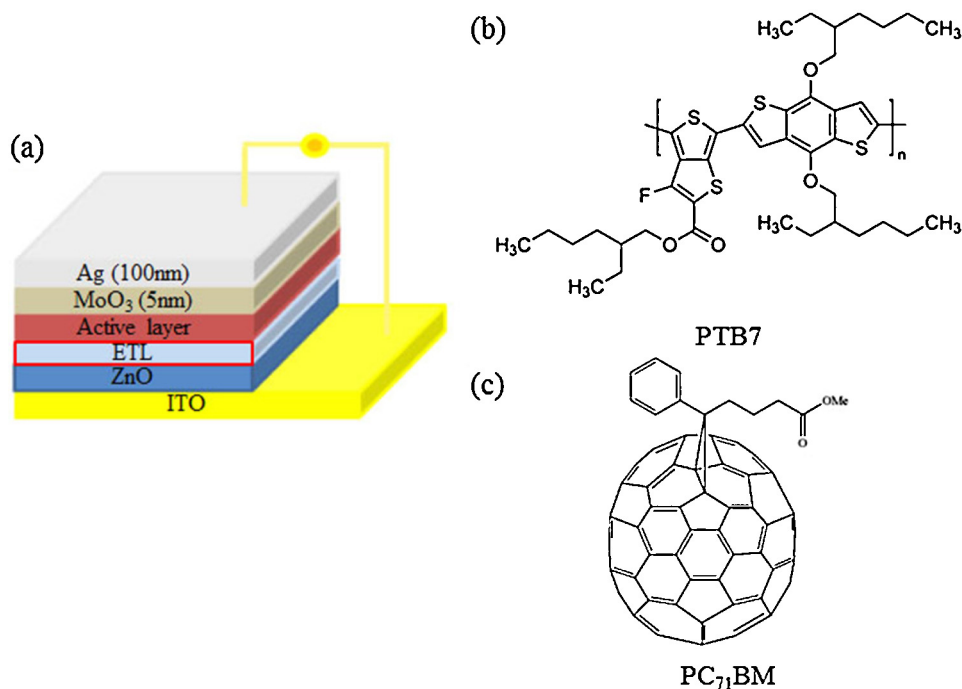


Fig. 1. (a) Structure of device introducing CsF layer and structures of (b) PTB7 and (c) PC<sub>71</sub>BM.

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