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# Photoanode characteristics of multi-layer composite BiVO<sub>4</sub> thin film in a concentrated carbonate electrolyte solution for water splitting

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The improvement of the solar energy conversion efficiency is important for the solar hydrogen production using semiconductor photoelectrodes. In this paper, the photoelectrochemical properties of multi-layer composite photoelectrodes of thin film BiVO<sub>4</sub> in various electrolyte solutions were investigated in detail. The improvement of photocurrent and the decrease of onset potential were observed on BiVO<sub>4</sub> composite electrodes in carbonate electrolyte solution as well as bare BiVO<sub>4</sub> electrode. The LHE (light-harvesting efficiency) and photocurrent were significantly improved by the light trapping structure of the double stacked photoelectrodes. The photocurrent was increased by insertion of an optimum SnO<sub>2</sub> intermediate layer. The decrease of resistance at the BiVO<sub>4</sub> composite electrodes was observed comparison with the bare BiVO<sub>4</sub>. In the BiVO<sub>4</sub>/SnO<sub>2</sub>/WO<sub>3</sub> photoelectrode, the highest IPCE (incident photon to current efficiency) was 53% at 420 nm. The H<sub>2</sub> and O<sub>2</sub> were evolved stoichiometrically. The maximum value of the applied bias photon-to-current efficiency (ABPE) was 1.35%. The reaction mechanism of carbonate anions, mainly affecting the BiVO<sub>4</sub> layer, was discussed.

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#### 1. Introduction

The solar splitting of water into H<sub>2</sub> and O<sub>2</sub> using a photoelectrochemical cell composed of an n-type semiconductor photoanode and H<sup>+</sup>-reducing cathode has been widely investigated for application on solar energy conversion and storage [1–3]. Some nano-crystalline oxide semiconductor photoelectrodes with narrow band-gap energy  $(E_g)$  [2,4,5], such as Fe<sub>2</sub>O<sub>3</sub>  $(E_g = 2.1 \text{ eV})$ , WO<sub>3</sub>  $(E_g = 2.7 \text{ eV})$  and BiVO<sub>4</sub>  $(E_g = 2.4 \text{ eV})$  on a conductive glass substrate, are easily prepared by wet coating process and calcination under an air atmosphere. These photoelectrodes have significant advantages for the practical production of solar hydrogen, including simple preparation, H<sub>2</sub> gas accumulation and large area production. These nano-crystalline photoelectrodes offer an improved photocurrent. Augstynski et al. reports the ca. 2.7 mA cm<sup>-2</sup> (at 1.23  $V_{RHE}$ ) using the nano-crystalline WO<sub>3</sub> photoelectrode [6]. Grätzel et al. reported that in the case of a modified Fe<sub>2</sub>O<sub>3</sub> film photoelectrode consisting of a perpendicularly oriented dendritic nanostructure [7–10], the best photocurrent reached ca.  $3.2 \text{ mA cm}^{-2}$  (at  $1.23 \text{ V}_{\text{RHE}}$ ) [9]. Moreover, Pt-doped Fe<sub>2</sub>O<sub>3</sub>/Pt nanorod arrays on a gold substrate photoelectrode was developed by Park et al., and the photocurrent at  $1.23 V_{RHE}$  was very high (ca.  $7.0 \text{ mA cm}^{-2}$ ) [11]. On the other hand, the solar energy conversion efficiency  $(\eta_{sun})$  value is still low

and the improvement of  $\eta_{sun}$  poses the greatest challenge. Several equations have been used to calculate the value of  $\eta_{sun}$  [5,12,13], and a  $\eta_{sun}$  considering the energy loss resulting from the external bias in a two-electrode system, that is, applied bias photon-to-current efficiency (ABPE or  $\eta_{sun}^{ex}$ ), is calculated by the following equation [5].

ABPE (%) = 
$$\left[\frac{J_{\text{opt}} \times (1.23 - E_{\text{opt}})}{Int}\right] \times 100$$
(1)

where  $J_{opt}/mA \text{ cm}^{-2}$  is the photocurrent density at  $E_{opt}$ ;  $E_{opt}/V$  is the applied voltage at the optimal operating conditions between the working and counter electrodes;  $Int/mW \text{ cm}^{-2}$  is the intensity of incident solar light under A.M. 1.5, 1 Sun condition; and 1.23 V vs. reversible hydrogen electrode (RHE) is the standard electrode potential of H<sub>2</sub>O.

To improve ABPE, decreases of  $E_{opt}$  and onset potential, and an improvement of *I*–*V* curve shape are very important factors of the research as well as an increase in the photocurrent. The applied bias and the onset potential are influenced by the conduction band potential ( $E_{CB}$ ) of semiconductors. In the case of the Fe<sub>2</sub>O<sub>3</sub> photoelectrodes, it was reported that large applied bias was needed to get high photocurrent due to the positive  $E_{CB}$ . Recently, the BiVO<sub>4</sub> photoelectrode has been attracting considerable attention. It is noteworthy that the  $E_{CB}$  of BiVO<sub>4</sub> (-0.4 V vs. NHE, pH = 7 [14]) is higher than that of WO<sub>3</sub>, and is close to that of TiO<sub>2</sub>. The photocurrent of BiVO<sub>4</sub> was significantly increased by the addition of an under-layer coating of WO<sub>3</sub> [14–16] or SnO<sub>2</sub> [17,18]

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on a conducting glass. However, the photocurrent of theses  $BiVO_4$  multi composite photoelectrodes is still low. Lee et al. reported that the photocurrent at  $1.23 V_{RHE}$  of heterojunction  $BiVO_4/WO_3$  electrodes reached ca.  $1.4 \text{ mA cm}^{-2}$  [14]. Moreover, various surface treated and modified  $BiVO_4$  photoelectrodes were developed [19–25] and Choi et al. reported that the photocurrent at  $1.23 V_{RHE}$  of FeOOH/BiVO<sub>4</sub> photoelectrodes reached ca.  $2.3 \text{ mA cm}^{-2}$  [24].

We have investigated the improvement of the photoelectrochemical properties of bare BiVO<sub>4</sub> photoelectrode in a carbonate electrolyte [26], as opposed to the sulfate electrolyte that has generally been used. Recently, we reported BiVO<sub>4</sub> composite electrodes and the effect of introducing a SnO<sub>2</sub> intermediate layer between the BiVO<sub>4</sub> and WO<sub>3</sub> layers (i.e., BiVO<sub>4</sub>/SnO<sub>2</sub>/WO<sub>3</sub> photoelectrode) in a carbonate electrolyte [27]. The photocurrent and ABPE were found to increase by the insertion of a SnO<sub>2</sub> intermediate layer and by carbonate ion effect.

In this study of a multi-layer composite thin-film  $BiVO_4$  photoelectrode, we investigated the effects of the  $BiVO_4$  and  $SnO_2$ film thickness, photoelectrochemical properties on composite electrodes in various electrolyte solution, and light trapping structure on the photoelectrochemical and optical properties in detail. The photocurrent was prominently affected by the film thickness of the  $BiVO_4$  upper layer and the  $SnO_2$  intermediate layer. Regarding the effect of the light trapping structure, it was found that the incident photon-to-current conversion efficiency (IPCE) was improved in the visible light region. The reaction mechanism of effect of the carbonate anions was also discussed.

#### 2. Experimental

#### 2.1. Photoelectrodes preparations

BiVO<sub>4</sub>/WO<sub>3</sub> and BiVO<sub>4</sub>/SnO<sub>2</sub>/WO<sub>3</sub> photoelectrodes were prepared as follows. The precursor solutions of each oxide semiconductor were coated on a F-doped SnO<sub>2</sub> (FTO) conductive glass substrate (surface resistance 10  $\Omega$ /sq, Nippon Sheet Glass Co. Ltd.) using a spin coater (1000 rpm, 30 s) and then calcinated at 500 °C for 30 min for each coating. Initially, a WO<sub>3</sub> layer was coated on a FTO glass with subsequent multiple coatings of the SnO<sub>2</sub> film (in the case of the inserted intermediate layer) and the BiVO<sub>4</sub> upper layer. The precursor solution used the following. In the case of the WO<sub>3</sub>, it was 1.4 M peroxotungstic acid, as reported previously [28]. In the case of the BiVO<sub>4</sub>, bismuth oxide and vanadium oxide of EMOD (Enhanced Metal Organic Decomposition) materials provided by Symetrix Co., USA were mixed with Bi:V = 1:1 and then these were diluted with butyl acetate. The SnO<sub>2</sub> precursor solution was EMOD materials of tin oxide provided by Symetrix Co. (USA) and diluted with xylene.

#### 2.2. Photoelectrochemical measurements and characterizations

The main photoelectrochemical properties were measured by an electrochemical analyzer (BAS. Inc. ALS660B). The *I–V* and the time dependence of the photocurrent measurements for the photoanode characteristics were assessed using a three-electrode cell with an Ag/AgCl reference electrode and a Pt coil counter electrode. The *I–V* curves of the forward and backward scans of potential should overlap in the steady state. The scan rate was slow (50 mV s<sup>-1</sup>). The gas measurement for water splitting and the *I–V* measurement for the ABPE calculation were performed using a two-electrode cell without a reference electrode. These electrodes were soaked in 80 mL of an electrolyte aqueous solution containing KHCO<sub>3</sub> with CO<sub>2</sub> gas bubbling in a Pyrex glass cell. Previously, we reported that the photocurrent of the BiVO<sub>4</sub> photoelectrode was increased with bubbling CO<sub>2</sub> gas into the carbonate electrolyte [26]. The electrochemical impedance spectroscopy (EIS) used CIMPS-system 1 (ZAHNER/ZENNIUM, XPOT, Germany). The Mott-Schottky plots of the photoelectrode was analyzed by result of the impedance-potential. In the main case of the I-V measurement, the working electrode was irradiated through a black mask/aperture  $(0.31 \text{ cm}^{-2})$  from the side of BiVO<sub>4</sub> film (front side irradiation) using a solar simulator (IIS-A-class, SAN-EI ELECTRIC Co.) through a light chopper. Sometimes, photoanode properties were measured under the light irradiation from the side of a glass substrate (back side irradiation) for the comparative experiment. The light intensity of the solar simulator was calibrated to A.M. 1.5 (1 Sun, Supporting Information, Fig. S1) using a solar simulator spectroradiometer (SOMA Optics, Ltd.), immediately after calibration using a standard light source (certification body, Japan Electric Meters Inspection Corporation, JEMIC). The mismatch between the A.M. 1.5 (1 Sun) and the solar simulator spectra was precisely corrected by the spectral mismatch factor (MMF) method [29]. The IPCE was automatically measured by a quantum efficiency-IPCE system (EKO Instruments Co. Ltd.) with a Xe lamp (USHIO INC). The photon flux of the monochromatic light was measured by a Si photodiode detector (Hamamatsu Photonics K. K. and this Si photodiode detector was calibrated by National Metrology Institute of Japan, NMIJ). The IPCE was calculated from Eq. (2).

$$IPCE = \frac{1240 \times \text{photocurrent density} (\text{mA cm}^{-2})}{\text{wavelength} (\text{nm}) \times \text{photon flux} (\text{mW cm}^{-2})}$$
(2)

The LHE (light-harvesting efficiency) of the electrode was calculated from transmittance (T) and reflectance (R) using an integrating sphere (Jasco, V-570, ISN-470) (Eq. (3)).

$$LHE = 1 - R - T \tag{3}$$

The amount of  $H_2$  and  $O_2$  evolved from the photoelectrochemical cell, which consisted of a photoanode and a Pt cathode, was investigated by a closed gas-circulating system with on-line gas chromatography (Shimadzu Co., GC-8A, TCD, 5A molecular sieves, Ar carrier).

The characterization of samples was investigated by a scanning electron microscopy (SEM, Hitachi S-800, SE mode), transmission electron microscopy (TEM, Hitachi High Technologies HD-2700), X-ray fluorescence spectroscopy (XRF, Rigaku ZSXmini), and X-ray photoelectron spectroscopy (XPS, Ulvac-Phi XPS-1800). A stylustype step measuring instrument (Surfcorder ET-3000, Kosaka Laboratory Ltd.) was used to the evaluation of film thickness.

The solar energy conversion efficiency has various derivations [5,12,13]. We used the ABPE by Eq. (1) [5]. The point, where size of photocurrent (at E) × potential (E) of obtained I–V curves became to the maximum, was chosen as the  $E_{opt}$  and  $J_{opt}$ .

#### 3. Results and discussion

## 3.1. Morphology and optical properties of the BiVO<sub>4</sub>/WO<sub>3</sub> photoelectrode

To investigate the effect of BiVO<sub>4</sub>-upper layer film thickness on photoelectrochemical properties, we prepared multi-layer composite BiVO<sub>4</sub> photoelectrodes with several different BiVO<sub>4</sub>-upper-layer film thicknesses maintaining the WO<sub>3</sub>-under-layer film thickness (190 nm). The adjustment of the BiVO<sub>4</sub>-upper-layer film thickness was carried out by changing the coating number of times of the BiVO<sub>4</sub> precursor solution. The BiVO<sub>4</sub> film thickness was ca. 60, 100, 150 and 200 nm. Fig. 1 shows the SEM photographs of the WO<sub>3</sub> and the BiVO<sub>4</sub>/WO<sub>3</sub> photoelectrode surface. The nanocrystalline WO<sub>3</sub> film surface was very smooth. With an increase in the film thickness of the BiVO<sub>4</sub>, the BiVO<sub>4</sub> particle grew and the interparticle crevice widths became narrower. The amounts of Bi

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