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Enhanced photocatalytic activity by bulk trapping and spatial separation of charge carriers: A case study of defect and facet mediated TiO₂



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ABSTRACT

Revealing the interplay between composition and structure of a given photocatalyst allows in-depth understanding of the photocatalytic mechanism and also presents as an effective way to optimise the photocatalytic activity. In this work, by studying the property and performance of defect and facet mediated rutile TiO_2 synthesized via a facile oxidation reaction, we demonstrated that the abundant self-doped defects can not only render visiblie responce to the reduced TiO_2 but also supress the prompt recombination of the charge carriers through bulk trapping, while the $\{111\}-\{110\}$ facet couples can induce the spatial separation of the e-h pairs. A synergistic effect between the defects must be facets was observed in the co-mediated TiO_2 , and its performance of hydrogen liberation was enhanced by a factor of 18 under simulated solar light compared with referenced commercial rutile, consequently. The present work sheds new light on how the composition and structure of a material can be finely tuned and work synergistically to achieve a much boosted performance.

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1. Introduction

The interplay between the composition and structure on the properties of a certain material has long been a significant theme of material science. While the components lay the foundation for properties, the structure of the material can always dramatically affect its performance. Conversely, rational design of the two factors is also indispensable for specific applications of a material [1,2]. Recent progress in photocatalytic materials has well witnessed this interrelationship, particularly in TiO₂ based photocatalyst [2,3]. In fact, among various materials that have been developed as photocatalyst, titanium dioxide is one of the most promising and popular stuffs owing to its merits like abundance, nontoxicity and high stability. Despite so, the overall efficiency of TiO₂ is restricted to a great extent by the light absorption limitation due to the intrinsic wide band gap and the spontaneous rapid recombination of photogenerated charge carriers within the semiconductor nature [4,5]. Therefore, further works addressing these two barriers are indeed needed for the widespread practical applications of titania.

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In the perspective of composition of TiO₂, Ti³⁺ and/or oxygen vacancy (V_0) self-doping, which is developed with respect to heteroatom (including metal and non-metal atoms) doping, has aroused intense research interest in the past a few years [6-12]. Just resembles that in the heteroatom doping methods, the self-doped defects can also introduce extra electronic states in the forbidden band [6-8]. The concentration of the defects, however, should theoretically be high enough to create a new band that eventually leads to the visible light absorption of the TiO_2 [6,7,10]. Up to now, a series of works [7-12] have reported deliberate introduction of Ti³⁺ and/or V_0 defects into the bulk lattice of TiO₂ based on either reduction or oxidation methods, and the resulting materials exhibit good photoactivity and stability under visible light. Nevertheless, in many cases, the performance under UV light, of which the incident light possesses higher energy and is much more efficient for TiO₂, is always neglected. Actually, the visible light absorption does not necessarily lead to higher overall performance as defects are sometimes reported to act as annihilation sites especially in the bulk of TiO₂ [13,14]. For example, Liu et al. [14] reported that their Ti³⁺ self-doped TiO₂ showed negligible activity under both UV and visible light though the sample exhibited broad visible light absorption. Leshuk et al. [15] found that the performance of their hydrogenated black TiO₂ under simulated solar light was deteriorated by the introduction of bulk V_0 species. Hence, it is still

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Fig. 1. (a) XRD patterns of the hydrothermal products; (b) Raman spectra of different hydrothermal samples as well as commercial rutile. The spectrum intensity of the referenced rutile is magnified.

necessary to explore other facile and effective strategies to prepare defect-mediated TiO₂ with wide spectral response, high activity as well as good stability.

On the other hand, the fate of the photogenerated charge carriers can be remarkably affected by the architecture of a given photocatalyst [3]. Facet, as a typical micro structure aspect of crystal, has become a research hotspot since distinct atom arrangements and electronic band structures of different facets will influence the transfer pattern of carriers [3,16-20]. Specifically, photoinduced electron and hole can be separately driven to different crystal facets due to the subtle distinction of coupled facets, which thus facilitate the charge separation [16,18]. In this regard, some recent studies have been well conducted in BiVO₄ and anatase TiO₂ systems [18,19]. For rutile, the presence of high energy facets like {111} and {001} can boost the photocatalytic activity, and it is demonstrated that the $\{111\}/\{001\}$ and $\{110\}$ facets of rutile can potentially act as oxidative and reductive sites, respectively [16,21]. Recently, Zuo et al. reported that, in comparison to irregular powder TiO₂ polymorphs, the rutile with $\{1\,1\,1\}-\{1\,1\,0\}$ facet couples and bulk Ti³⁺ species showed enhanced performance in hydrogen liberation from water under visible light [10]. Notwithstanding, so far there is still the dearth of the interplay between the defect and facet, their separate and cooperative roles in boosting the photocatalytic activity of TiO₂, especially under full solar light irradiation.

In the present work, we studied the underlying interplay of the defect and facet by studying the properties of Ti³⁺ self-doped rutile which has explicit facets. A facile oxidation based synthesis process was rationally adopted in hoping of stabilizing Ti³⁺ ions by thermodynamically spontaneous oxidation of Ti²⁺ species from TiH₂ in HCl solution. The products showed a distinct absorption within the whole visible light region as well as some near-infrared light. By simply varying the hydrothermal temperature, Ti^{3+} self-doped rutile with explicit {111} and {110} facets was obtained. It was demonstrated that the abundant defects can not only render visible light response to the reduced TiO₂ but also suppress the prompt recombination of the charge carriers, while the $\{111\}-\{110\}$ facet couples can induce the spatial separation of the e-h pairs. A synergistic effect between the defects and the facets was observed and as a result, the performance of hydrogen liberation of the optimised sample was enhanced by a factor of 18 under simulated solar light compared with commercial rutile. The results of this work suggest that the defect composition and the facet structure of a photocatalyst can work cooperatively and be rationally tuned for a much boosted overall activity.

2. Experimental section

2.1. Synthesis of reduced TiO₂

Typically, 400 mg TiH₂ (Alfa Aesar, 325 mesh) powders were dispersed into 25 mL 2 M HCl aqueous solution. The suspension was then transferred into a 50 mL Teflon autoclave and kept at different temperatures (180, 205 and 215 °C) for 12 h. After cooling naturally to room temperature, the precipitate was collected and washed with deionized water for several times until the ionic strength was less than 10 ppm. The clean samples were dried under vacuum overnight. The samples were denoted as TH-*x*, where *x* stands for the corresponding hydrothermal temperature.

2.2. Photodeposition of Pt and PbO₂ particles

The selective depositions of Pt and PbO₂ species were carried out to determine the preferred transfer routes of the charge carriers. For Pt deposition, 100 mg of TH-215 sample was dispersed into 100 mL deionized water, and then a certain volume of H₂PtCl₆ was added to ensure the nominal weight ration of Pt-TiO₂ being 3 wt%. The solution was irradiated by full-solar light for 1 h. The powders were then centrifuged and dried at 60 °C in vacuum. This Pt-deposited TiO₂ was then dispersed into 100 mL water again, and 150 mg Pb(NO₃)₂ was dissolved into the mixture followed by irradiation for another 3 h under solar simulator. The precipitate was then separated out and dried for further analyses.

2.3. Characterization

Crystal structure identification was performed using Bruker D8 X-ray diffractometer (XRD) with Cu K α radiation (λ = 0.15418 nm) operating at 40 kV and 40 mA. Raman scattering measurements were performed with a multichannel modular triple Raman system (Renishaw Co.) with confocal microscopy at room temperature using the 532 nm laser. A 50× microscope objective lens was used for focusing the laser beam and collection of the scattered light. Scanning electron microscope (SEM) images was obtained with an FEI Nova NanoSEM 230 field-emission scanning electron microscope. Microstructures and morphologies were investigated using TecnaiG2 F20 S-TWIN (FEI company) Transmission electron microscopy (TEM) with a field emission gun at 200 kV and the accessary Energy dispersive X-ray spectroscopy (EDS) was used to determine to composite elements. Diffuse reflection spectra (DRS) of the samples were recorded on a Varian Cary-500 spectrophotometer and BaSO₄ was used as a reference. X-ray Download English Version:

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