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# Opportunity of dipole layer formation at non-SiO<sub>2</sub> dielectric interfaces in two cases: multi-cation systems and multi-anion systems

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**Abstract:** This paper demonstrates possible dipole layer formation at non-SiO<sub>2</sub> interfaces in two cases: Al<sub>2</sub>O<sub>3</sub>/AlF<sub>x</sub>O<sub>y</sub> as an example of multi-anion systems and Al<sub>2</sub>O<sub>3</sub>/MgO as that of multi-cation systems. While unbalanced anion migration is considered as the origin at Al<sub>2</sub>O<sub>3</sub>/AlF<sub>x</sub>O<sub>y</sub> interface, cation migration could play a dominant role in the charge separation at Al<sub>2</sub>O<sub>3</sub>/MgO interface annealed at higher temperature.

**Keywords:** dipole layer; gate dielectric interfaces; flatband voltage shift

## 1. Introduction

As threshold voltage tuning is important for advance metal-oxide-semiconductor field-effect transistors, dipole layer formed at gate dielectric interface has been recognized as a promising way to intentionally control the threshold voltage [1-4]. Extensive studies [5-10] have demonstrated dipole layer formation at high-k oxide/SiO<sub>2</sub> interfaces which could change the band alignment and shift the flatband voltage ( $V_{fb}$ ) of MOS structures. However, reports on the dipole layer formation at the interfaces other than high-k/SiO<sub>2</sub> are quite limited. There are even reports suggesting dipole layer formation could be unique at high-k/SiO<sub>2</sub> interfaces [7] possibly because of the different structural characteristics between high-k oxides and SiO<sub>2</sub>. In order to develop a strategy to design a MOS capacitors with desirable  $V_{fb}$ , understanding on the origin of dipole layer formation at general dielectric interfaces is necessary.

In this work, we studied dipole layer induced  $V_{fb}$  shift at non-SiO<sub>2</sub> interfaces, in the cases of Al<sub>2</sub>O<sub>3</sub>/AlF<sub>x</sub>O<sub>y</sub> and Al<sub>2</sub>O<sub>3</sub>/MgO, to demonstrate that even a non-SiO<sub>2</sub> dielectric interface can form an interface dipole layer. The oxide/fluoride interface was studied to clarify the effect of different kinds of anions since oxygen migration has been considered as the origin of dipole layer formation at high-k/SiO<sub>2</sub> interfaces [7]. In addition, the effect of F incorporation on  $V_{fb}$  needs to be understood because F has been used to improve the interface quality of gate stacks [11]. Al<sub>2</sub>O<sub>3</sub>/MgO interface was investigated to study the possible cation effect which is suggested as the origin of dipole layer formation at MgO/SiO<sub>2</sub> interface [10]. Molecular dynamics (MD) simulation was also employed to reproduce the charge separation at Al<sub>2</sub>O<sub>3</sub>/AlF<sub>x</sub>O<sub>y</sub> and Al<sub>2</sub>O<sub>3</sub>/MgO. Further, the possible driving force of interface dipole layer formation at the simulated interfaces was discussed based on the interface atomic migration.

## 2. Experimental and Calculation Methods

The MOS capacitors with tri-layer dielectrics consisting of thermally-grown SiO<sub>2</sub>, dielectric 1 and dielectric 2 were fabricated. Dielectrics 1 and 2 as shown in TABLE I were

sequentially deposited by sputtering on p-type Si substrates covered with thermal oxides. Note that ~0.5 nm Al<sub>2</sub>O<sub>3</sub> was capped on MgO to prevent moisture effect for Sample 3. Post deposition annealing (PDA) 1 and 2 were conducted in 0.1% O<sub>2</sub> ambient after the deposition of each layer. The final step was deposition of Au gate electrodes and Al back-side electrodes to form MOS capacitors. The flatband voltages were determined from the capacitance-voltage characteristics measured at a frequency of 1 MHz.

The molecular dynamics (MD) simulation was performed using a commercial simulation package Fujitsu SCIGRESS ME 2.1. Born-Mayer-Huggins potential, which includes point-to-point Coulombic interaction between two atoms, van der Waals potential, and short-range repulsion, was employed,

$$V(r_{ij}) = \frac{q_i q_j}{4\pi \epsilon_0 r_{ij}} - \frac{C_i C_j}{r_{ij}^6} + f(B_i + B_j) \left( \frac{A_i + A_j - r_{ij}}{B_i + B_j} \right),$$

where  $r_{ij}$  is the interatomic distance between the  $i$ -th and  $j$ -th atoms,  $f$  is a standard force of 1.154 kJ Å<sup>-1</sup> mol<sup>-1</sup>,  $q$  is the amount of ionic charge,  $A$  is the ionic radii,  $B$  is the softness parameter, and  $C$  is the van der Waals coefficient. In this simulation, amorphous structures without defects have been studied. The completely ionic model (CIM) parameters of Al, O and F determined by Kawamura [12] were adopted in which the states of the ions are Al<sup>3+</sup>, O<sup>2-</sup>, and F<sup>-</sup>. This simulation has been demonstrated to be effective in reproducing dipole layer formation at high-k/SiO<sub>2</sub> interfaces [13, 14]. The typical simulation process is as follows: first, amorphous Al<sub>2</sub>O<sub>3</sub>, AlF<sub>3</sub>, and MgO blocks were obtained by melting the random structures with sizes of  $\sim 7 \times 7 \times 5$  nm<sup>3</sup> at 4000K. The interfaces were obtained by stacking the amorphous blocks and annealed by the isothermal-isobaric MD calculation for 10 ps, thermostated at 1000 K by speed scaling with keeping the pressure at 1 atm; finally, the structure was cooled from 1000 to 300 K within 30 ps and maintained at 300 K for another 10 ps.

TABLE I. List of dielectrics and PDA temperatures. ~0.5 nm Al<sub>2</sub>O<sub>3</sub> was capped on MgO to prevent moisture effect for Sample 3.

	Layer 1 (bottom)	PDA 1 T (°C)	Layer 2 (top)	PDA 2 T (°C)
Sample 1	Al <sub>2</sub> FO <sub>2.5</sub>	400	Al <sub>2</sub> O <sub>3</sub>	500
Reference 1			-	
Sample 2	Al <sub>2</sub> O <sub>3</sub>	600	AlFO	400
Reference 2			-	
Sample 3	Al <sub>2</sub> O <sub>3</sub>	w/o, 600,	MgO(cap)	600
Reference 3		700, or 800	-	or 800

## 3. Results and Discussions

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