



Effective CO₂ detection based on LaOCl-doped SnO₂ nanofibers: Insight into the role of oxygen in carrier gas

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ABSTRACT

In this paper, undoped and LaOCl-doped SnO₂ nanofibers were prepared by a simple one-step electrospinning technique and their responses upon exposure to CO₂ gas in different oxygen containing backgrounds were systematically investigated. It was observed that the obtained nanofibers were hollow porous structures that gave rise to excellent performance. The sensor based on 8 at.% LaOCl-SnO₂ nanofibers exhibited optimal response ($R_{\text{gas}}/R_{\text{air}} = 3.7$) toward 1000 ppm CO₂ at 300 °C with response/recovery times of 24 s/92 s, and didn't show any saturation over a wide range of CO₂ concentrations (100–20000 ppm). In terms of the sensing behavior of these sensors, their sensing mechanisms are proposed as follows: (1) In low oxygen concentration background, CO₂ primarily reacts with $V_{\text{O}}^{\bullet\bullet}$, which needs high activation energy to occur, so only a slight number of CO₂ can take part in the reaction. (2) In high oxygen concentration background, CO₂ mainly reacts with $\text{O}_2^{\times}$. Since this reaction conducts easily, CO₂ can react with $\text{O}_2^{\times}$ sufficiently. The proposed sensing mechanisms can help readers better understand the role oxygen plays in CO₂ gas sensing process.

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1. Introduction

The detection and control of CO₂ concentration in ambient environment are becoming more and more necessary owing to the global warming and climate change. Besides, CO₂ gas measurements are also demanded in various fields such as indoor air quality monitor, beverage industry and smoke systems. Due to these applications, there is an urgent need to fabricate reliable, cost effective and highly sensitive CO₂ gas sensors.

For monitoring CO₂, the prevailing technology today relies on optical [1,2], electrochemical [3–5], capacitive [6–8] and work function based [9–11] sensors. The disadvantages of these sensors span from high costs to complicated operation conditions. Therefore developing inexpensive and facilely manipulated CO₂ sensors is eagerly required. Resistive sensors based on metal oxide semiconductors (MOS) provide a promising alternative since they are cost effective, easily implemented and highly-reliable [12]. Among all the candidates of MOS, SnO₂, as an n-type wide band gap semi-

conductor ($E_g = 3.6$ eV, at 300 K), has been widely used due to easy fabrication, high gas response and good stability [13].

However, unlike general gases such as H₂, CO and NO_x, which can be easily monitored by means of redox reactions, CO₂ is rather inert, so the detection of CO₂ with pristine SnO₂ is much more difficult. On this account, several approaches to functionalize SnO₂ with a second phase for the enhancement of CO₂ sensing properties come into being. For instance, modifying the surface of SnO₂ material by rare-earth based oxides [14–18]. In addition, forming composites with other n-type semiconductors such as WO₃ [19] or with p-type semiconductors like CuO [20] is also a common way. Among these researches, much attention has been devoted to the synthesis of SnO₂ modified with La because La doping can manipulate the acid-based properties and increase the amount of active sites on the surface of SnO₂ [21].

Many CO₂ gas sensors based on undoped and La doped SnO₂ nanostructured materials have been investigated (Table 1). To our best knowledge, up to now almost all literatures mainly report CO₂ measurements in air background, very few works are related to CO₂ detection under various oxygen concentrations in carrier gas. Kim et al. [14] fabricated La-SnO₂ thick films for the detection of CO₂ at 400 °C in N₂, synthetic air and pure O₂ atmosphere. The La-SnO₂ exhibited the highest response of 1.59 to 2000 ppm CO₂ in pure O₂

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Table 1A comparison of the performance of CO₂ sensors based on SnO₂ nanostructured materials.

Material	Synthesis route	T (°C)	response/C(ppm)	Ref.
SnO ₂	Coprecipitation	240	1.71/2000	[26]
2.2 mol%La ₂ O ₃ -SnO ₂	Mechanical milling	350	1.6/1000	[16]
4 mol%La ₂ O ₃ -SnO ₂	Hydrothermal	250	1.42/500	[18]
2at.%LaOCl-SnO ₂	Electrostatic spray pyrolysis	425	1.38/2000	[15]
LaOCl-SnO ₂ NWs	Thermal evaporation & dropping	400	5.6/2000	[17]
LaOCl-SnO ₂	Impregnation	350	1.6/2000	[27]
8 mol%LaOCl-SnO ₂	Electrospinning	300	3.7/1000	This work

atmosphere. Zhou et al. [22] prepared layered reduced graphene oxide thin films for room temperature CO₂ detection using dry air and N₂ as carrier gases, respectively. The sensor showed better response toward 5000 ppm CO₂ in dry air. Conversely, Hafiz et al. [23] reported the CO₂ sensor based on hydrogen plasma reduced graphene oxide, which showed higher response in N₂ (3.4) than that in air atmosphere (1.18) on exposure to 1500 ppm CO₂. Thus, whether the existence of oxygen in the background has a positive or negative effect on CO₂ gas sensing properties is still controversial. In order to fundamentally understand the CO₂ sensing mechanism within the reaction process, thorough research on this subject needs to be carried out. In addition to this primary interest, the detection of CO₂ gas in different oxygen concentration backgrounds is very necessary in several fields such as automotive applications and certain military projects.

Recently, electrospun nanofibers (NFs) have been demonstrated to display extraordinary sensing characteristics ascribed to their open porosity, rich defect sites [24] and easy production on a large scale [25]. Herein, CO₂ sensors based on undoped and LaOCl-doped SnO₂ are fabricated via this technique and their sensing properties in various oxygen containing backgrounds are intensively investigated. In addition, the detection mechanism is also proposed, paving the way for further development of other CO₂ gas sensors with high-quality.

2. Experimental

2.1. Materials

All chemicals including tin (II) chloride dehydrate (SnCl₂·2H₂O, Sigma-Aldrich Co., UK), lanthanum chloride (LaCl₃, Sigma-Aldrich Co., UK), N, N-dimethylformamide (DMF, Sinopharm, China), poly(vinyl pyrrolidone) (PVP, MW = 1,300,000, Sinopharm, China) and ethanol (Sinopharm, China) were analytical grade reagents and used as received without any further purifications.

2.2. Synthesis and characterization of porous SnO₂ based nanofibers

The LaOCl-SnO₂ NFs were synthesized by an economical one-step electrospinning technique. And then, a heat-treatment process was performed [28]. Firstly, the precursor solution was prepared by mixing 0.6 g SnCl₂·2H₂O with LaCl₃ with different ratios of La and Sn (0, 4, 8, 12, and 16 at.%), followed by the addition of 4.42 g DMF and 4.42 g ethanol, and then under magnetic stirring for 30 mins. Secondly, 0.8 g of PVP powders were dissolved in the above solution, and stirred vigorously at room temperature for at least 12 h. Thirdly, the obtained compound was transferred into a 5 mL syringe with a capillary tip for spinning. A high voltage of 20 kV was applied at the spinneret by a direct-current power supply. The solution was pushed out of the spinneret by a syringe pump at the rate of 1 mL/h and the NFs were collected using aluminum foil with a thickness of 0.5 mm wrapped on cylindrically rotating collector. After that, the NFs were annealed at 600 °C for 2 h to obtain the final SnO₂ based NFs.

Thermal analysis was carried out by using a thermogravimetric and differential thermal analysis apparatus (TG-DTA; Netzsch, model STA 449C). The analyses were carried out with a heating rate of 10 °C min⁻¹ in static air from room temperature up to 1200 °C. X-ray diffraction (XRD) patterns were recorded on a Philips X'Pert diffractometer from 2θ = 10–90° using Cu-Kα radiation (λ = 1.5406 Å). The general surface morphology of the NFs was observed by field-emission gun scanning electron microscopy (FE-SEM) using a Hitachi S-4800 field-emission at a voltage of 10 kV. Energy dispersive X-ray spectrum (EDS) was also carried out with it. High-resolution transmission electron microscope (HRTEM) observations were performed with a JEM-2100 UHR microscope. XPS studies were analyzed with a PHI5000 Versa Probe employing Al Kα radiation (ULVAC-PHI, Japan). Bonding energy was calibrated with reference to C1s peak (285.0 eV).

2.3. Fabrication and measurements of SnO₂-based sensors

The SnO₂-based NFs were mixed with an organic binder in the weight ratio of 7:3 to form SnO₂-based paste, which was then subsequently printed on alumina substrate with Pt-interdigitated electrodes and Pt-resistive-type heater whom were previously printed by screen-printing technique. Eventually, the SnO₂-based films were sintered firstly at 350 °C and then at 550 °C both for 2 h in air to evaporate the organic solvent and ensure the adhesion of the samples on the alumina substrate. Before any measurements the samples were heated at 600 °C for 8 h in N₂ gas flow to create more oxygen vacancy defects. Then, the as-prepared sensors were stored at room temperature. All the prepared LaOCl-SnO₂ films with different doping ratios of LaOCl are named as 4 at.%, 8 at.%, 12 at.% and 16 at.% La-SnO₂, respectively. The undoped sample is named as pure SnO₂. The simplified schematic diagram of the sensor fabrication is presented in Fig. 1.

Since operating temperatures higher than 400 °C are not practical for application, here, gas sensing tests are performed at temperatures ranging from 100 to 400 °C. The sensor response is defined as $S = R/R_0$. Where R is the resistance of the material in background atmosphere with different oxygen concentrations and R₀ is the resistance of the material when exposed to target test gas CO₂. Sensor elements are tested with a gas detection apparatus (Huachuang Ruike Science and Technology Co. Ltd). Gas concentration is controlled by four mass flow controllers and the total gas flow rate was kept at 2 L/min. The measurement range of resistance is 100 Ω–1 GΩ with an error less than 5%. The response time (t_{res}) is defined as the time taken to attain 90% of the final value, and recovery time (t_{rec}) is the time required until 90% of the original baseline is recovered.

3. Results and discussion

3.1. Characterization of the samples

The phase-transformation of La compound was investigated by TG/DTA analyse. Fig. 2 shows the TG/DTA curves of starting materials LaCl₃ powders. In the TG curve, there are three discrete weight

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