



## Damage onset modeling in woven composites based on a coupled stress and energy criterion



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### ABSTRACT

The crack onset configuration at damage onset in a four-layer plain weave glass fiber/epoxy matrix composite is studied at the mesoscopic scale using a coupled criterion based on both a stress and an energy condition. The possible crack shapes are selected based on optical microscope observations of damage mechanisms on a specimen edge during a tensile test. The crack location, length and orientation, the decohesion length and the strain at damage onset are determined. The damage onset strain is underestimated compared to the experimental value determined by acoustic emission if only a stress criterion is considered. The coupled stress and energy criterion leads to a more reasonable estimate of strain at damage onset.

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## 1. Introduction

Textile composites are increasingly used for aeronautical and automotive advanced structural applications. An advantage of woven composites is the large variety of the fiber reinforcements, whose architecture can be varied throughout the structure. Therefore, fewer assembly operations are required compared to classical laminated composite structures, which reduces both production costs and the number of weak points in the structure.

Woven composites offer a high potential for material design, which can be optimized using design tools able to describe the evolution of the mechanical behavior from damage onset to final failure of the materials. Macroscopic phenomenological models have been developed for the prediction of damage evolution and failure in 2D [1,2] and 3D [3–7] textile composites. However, expensive and time consuming experimental identifications of the model parameters are required. Moreover, these parameters have to be re-identified each time the fiber architecture or the constituents change. The number of tests may be reduced by using more predictive models at the mesoscopic scale, which take into account the fiber reinforcement architecture.

Different meso-scale approaches have been used to model the evolution of the material behavior with growing damage [8–12]. An analytical model using a damage mosaic laminate model was proposed by Gao et al. [8]. It allows the calculation of the effective Young's modulus of the damaged composite. However, yarn undulation, which has a significant influence on the damage location [11,13], is not described by such a model. In most published studies, a Finite Element (FE) approach based on continuum damage mechanics (CDM) is used (e.g., [9,11,14]). It consists in detecting damaged elements using a

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## Nomenclature

### Energy criterion

$\Delta W_k$	change in kinetic energy
$\Delta W$	change in potential energy
$W(0)$	potential energy of the undamaged state
$W(d)$	potential energy of the cracked state
$G^c$	critical energy release rate
$\Delta S$	crack surface
$h$	specimen thickness
$E^{eq}$	equivalent elastic modulus
$A(d)$	normalized incremental energy release rate

### Stress criterion

$f$	stress based failure criterion
$f_c$	criterion for transverse cracking
$f_d$	criterion for decohesions
$Y_t$	transverse ply strength
$Z_t$	out-of-plane ply strength
$S_{12}^R$	in-plane shear strength
$S_{13}^R, S_{23}^R$	out-of-plane shear strengths
$p_{12}, p_{13}, p_{23}$	shape parameters
$Y_c$	transverse compressive ply strength

### Crack shape

$d = (d_1, \dots, d_n)$	parameters describing the crack shape
$(x, y)$	coordinates of the crack center
$(x_c, y_c, z_c)$	coordinates of the crack center localized using the stress criterion
$(x^*, y^*)$	crack location at damage onset determined using the energy criterion
$d_c$	crack length
$d_s^*$	crack length at damage onset for a crack localized using the stress criterion
$d_c^*$	crack length at damage onset determined using the energy criterion
$\theta$	crack orientation
$\theta^*$	crack orientation at damage onset
$\mu$	decohesion length
$\mu^*$	decohesion length at damage onset

### Strains

$\epsilon_e$	strain measured experimentally
$\epsilon_e^c$	strain at damage onset determined experimentally
$\epsilon$	applied global strain
$\epsilon^{energy}$	strain calculated using the energy criterion
$\epsilon^{energy,c}$	strain at damage onset calculated using the energy criterion
$\epsilon^{energy,s}$	strain at damage onset for a crack localized using the stress criterion
$\epsilon_s$	strain at damage onset obtained using the stress criterion only

### Material properties

$E_m$	Young's modulus of the matrix
$\nu_m$	Poisson's ratio of the matrix
$E_f$	Young's modulus of the fibers
$\nu_f$	Poisson's ratio of the fibers
$E_l$	longitudinal Young's modulus of the yarns
$E_t$	transverse Young's modulus of the yarns
$\nu_{tt}, \nu_{lt}$	Poisson's ratios of the yarns
$G_{lt}$	shear modulus of the yarns

stress based failure criterion and reducing the local stiffness of these damaged elements with increasing loading. An advantage of this method is the ease of implementation in FE codes. However, erroneous damage propagation directions can be predicted with these models [14,15]. Moreover, regularization methods [16,17] are required in order to avoid damage pattern dependence on FE mesh, which leads to a non-local damage zone. In this case, a very small mesh size compared to the

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