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## Impact of $\text{In}_2\text{S}_3$ shells thickness on the electrochemical and optical properties of oriented $\text{ZnO}/\text{In}_2\text{S}_3$ core/shell nanowires

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### ABSTRACT

In this paper, we address the effect of shell thickness and core/shell structure on the electrochemical and photoelectrochemical cell (PEC) measurements properties of  $\text{ZnO}/\text{In}_2\text{S}_3$  core/shell nanowires (NWs). The objective is to elucidate the mechanisms responsible for the extended photoresponse of  $\text{ZnO}/\text{In}_2\text{S}_3$  core/shell NWs to solar radiation. Well aligned  $\text{ZnO}/\text{In}_2\text{S}_3$  core/shell NWs were fabricated on indium tin oxide substrates using electrochemically grown  $\text{ZnO}$  NWs as the cores and electrodeposited  $\text{In}_2\text{S}_3$  as the shells. The samples structure was characterized by X-ray diffraction, revealing the mixed wurtzite and tetragonal structures of both  $\text{ZnO}$  cores and  $\text{In}_2\text{S}_3$  shells, and the improvement in the structure with the increases of  $\text{In}_2\text{S}_3$  shell thickness. The optical properties were studied through optical absorbance and photoluminescence measurements, showing the optical properties featured with type-II heterogeneous nanostructures constructed from  $\text{ZnO}$  and  $\text{In}_2\text{S}_3$ . Electrochemical impedance spectroscopy (EIS) is employed and an equivalent circuit model is designed suggesting that the cell performances were affected by increasing  $\text{In}_2\text{S}_3$  shell deposition times. From Mott–Schottky plots, several parameters such as flat-band potential and free carrier concentration were determined. Next, from (PEC) measurements, the highest photocurrent density produced by the  $\text{In}_2\text{S}_3$  shell electrode prepared at deposition time of 5 min reached  $9.30 \text{ mA cm}^{-2}$  at an applied potential of 0.8 V vs. Ag/AgCl. This value was about 6.8 times as much as more than that measured on  $\text{ZnO}$  NWs one and provided the highest value of solar-to-hydrogen energy efficiency  $\eta$  (%). These results are very encouraging and suggest that  $\text{In}_2\text{S}_3$  covered  $\text{ZnO}$  NWs nanostructures are valuable photo-anodes in order to build cheap devices for solar energy-to-hydrogen generation devices.

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## Introduction

Recently, photoelectrochemical cell (PEC) water splitting has been extensively studied as a promising candidate for hydrogen energy production and renewable energy [1–4]. In fact, many investigations have demonstrated the feasibility of semiconductor metal oxides application in photoelectrochemical cells, including photocatalysis, solar water splitting cells and dye sensitized solar cells due to their high activity, environment-friendly feature and low cost [5–8]. Among the various oxide 1D nanostructured materials, zinc oxide nanowires (ZnO NWs) [9] have also attracted much attention due to their cheap fabrication, high specific surface area and wide applications in PEC water splitting [10–15]. However, the disadvantage of ZnO is that its large band gap ( $E_g = 3.37$  eV). Consequently, only a small portion of the solar spectrum is absorbed since the UV spectrum represents only ca. 5% of sunlight. This greatly hinders the use of ZnO materials for solar energy harvesting [16]. To overcome this difficulty, many efforts have been devoted to create a heterojunction between ZnO NWs and another semiconductor which has spread visible light range. In the cases where the two materials recombine firmly, the absorption edge shifts towards the visible light region. As a result, it may be an interesting approach to improve the visible light driven PEC performances [16,17]. We have identified  $\text{In}_2\text{S}_3$  as a valuable candidate for visible decorated material combined to ZnO since it offers good stability and high carrier mobility [6,18,19]. In addition, the indium sulfide is a convention n-type binary chalcogenide semiconductor which appears highly desirable for use in solar energy conversion. Actually, the low band gap value ( $E_g = 2.1$  eV) favors the utilization of a significant portion of the visible solar spectrum. Despite these advantages, the core/shell structure involving ZnO/ $\text{In}_2\text{S}_3$  nanowires does not widely studied.

Recently, in our previous work [6], we have developed novel core/shell ZnO/ $\text{In}_2\text{S}_3$  nanowires by a simple route using two steps electrodeposition process. The typical type II band alignment and uniform  $\text{In}_2\text{S}_3$  distribution deduced from optical measurements and morphological study contributed to the effective separation of the photogenerated electron–hole pairs and transportation of the electrons to the electrode surface. These effects lead to a significant enhancement of the photoelectrochemical performance. In addition, an excellent stability of the core/shell nanowires was also demonstrated. We have also shown in our other studies [20,21], that the deposition time or thicknesses of thin film have an influence on the physical and particularly electrical properties of a semiconductor.

On the basis of our recent study [6], in the present paper we will address the impact of the deposition time, which is closely related to the  $\text{In}_2\text{S}_3$  shell thickness, on the photoelectrochemical properties as well as the electrochemical behavior taking place at the ZnO/ $\text{In}_2\text{S}_3$  electrode –  $\text{Na}_2\text{SO}_4$  electrolyte solution interface. A special attention will be put on the electrochemical impedance spectroscopy analysis.

## Experimental sections

### Sample preparation

All chemicals used in this study were of analytical reagent grade and were used as received without further purification. All aqueous solutions were prepared using deionized water.

### Synthesis of the ZnO nanowires

The electrodeposition procedure was performed on a glass sheets coated with Sn-doped polycrystalline  $\text{In}_2\text{O}_3$  (ITO, resistance of  $10 \Omega/\text{sq}$ ). Prior to the electrodeposition, the ITO/glass substrates were cleaned sequentially with an acetone solution and isopropanol (for 15 min each) in an ultrasonic bath, and then thoroughly rinsed with abundant water. The sheets were immersed in an aqueous solution of  $\text{ZnCl}_2$  ( $5.0 \times 10^{-4}$  M) and KCl ( $1.0 \times 10^{-1}$  M), under continuous bubbling of oxygen inside a classical three-electrode electrochemical cell with optimized potentiostatic conditions, as described recently by our group [22]. The electrodeposition of ZnO nanowires was performed at a fixed potential of  $-1$  V versus Ag/AgCl, while the reaction temperature was kept at  $80^\circ\text{C}$  [23].

### Synthesis of the ZnO/ $\text{In}_2\text{S}_3$ core/shell structure

ZnO/ $\text{In}_2\text{S}_3$  core/shell nanowires arrays were prepared via an electrodeposition process. The previously deposited ZnO nanowires arrays were employed as working electrode. The plating was conducted in an aqueous solution containing 1 mM  $\text{InCl}_3$  and 2 mM  $\text{Na}_2\text{S}_2\text{O}_3$ , as indium and sulfur sources, 0.1 M KCl supporting electrolyte and 2 mM  $\text{Na}_2$  EDTA chelating agent. More details were given in our recent work [6]. The electrodeposition of  $\text{In}_2\text{S}_3$  was carried out at a fixed potential of  $-1.2$  V at room temperature. To modulate the shell thickness, the deposition time was varied between 1 min and 10 min. Finally, the as-synthesized samples were rinsed with distilled water and dried in air at room temperature. The produced samples will be called sample B, C, D and E respectively, in the further section, while bare ZnO/ITO one will be called sample A.

### Samples characterization

The crystal phase of the obtained samples was identified by X-ray diffraction technique (XRD). The apparatus was an analytical Empyrean equipped with a multichannel detector (Pixel3D) using  $\text{CuK}\alpha$  radiation in the  $20$ – $70^\circ$   $2\theta$  range, with a scan step of  $0.026^\circ$  for 297s. The morphology of the prepared samples was investigated by scanning electron microscopy (SEM) equipped with a Silicon Drift Diode (SDD) detector and operating at 10 kV. The elemental composition was obtained using an energy-dispersive X-ray spectrometer (EDX). The UV–visible absorption spectra of all samples were recorded using Perkin-Elmer-Lambda 1050 spectrophotometer equipped with a PTFE coated integration sphere within the range of 300–1500 nm. Finally, photoluminescence spectra (PL) were performed by exciting with the frequency of a pulsed  $\text{Ar}^+$  laser at an excitation wavelength of 340 nm.

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