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Hydrogen evolution reaction performance of the molybdenum disulfide/nickel–phosphorus composites in alkaline solution

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ABSTRACT

The low phosphorus (LP), medium phosphorus (MP), and high phosphorus (HP) three nickel–phosphorus (Ni–P) powders were employed as supports of molybdenum disulfide (MoS₂) to fabricate MoS₂/Ni–P composites (MoS₂/Ni–LP, MoS₂/Ni–MP, and MoS₂/Ni–HP) via a hydrothermal technique. Crystal structures, morphologies, components and hydrogen evolution reaction (HER) performances of them were characterized and evaluated. During the incorporation between MoS₂ and Ni–P, the crystallization of Ni–P powders and the presence of metastable nickel phosphides can be validated; also, the 2H-MoS₂ tends to transfer to the phase of 1T-MoS₂. Compared with MoS₂ and Ni–P samples, MoS₂/Ni–P composites show a higher HER property. Prominent HER behaviors of the composites can be due to the presence of monolayer 1T-MoS₂ on the surface, the excellent conductivity of Ni–P powder at the bottom, as well as the coexistence of nickel phosphides. Among the three composites, MoS₂/Ni–MP exhibits the remarkably electrocatalytic activity.

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Introduction

The clean and efficient energy carrier of hydrogen derived from the electrocatalytic hydrogen evolution reaction (HER), has been intensively explored to alleviate the severe energy shortage and environmental deterioration caused by the excessive consumption of coal and petroleum fuels [1]. It is well known that platinum (Pt) and Pt-based electrocatalysts are competent for the HER applications, whereas high costs and scarcities of these metals hinder their widespread commercializations [2–4]. Thus, the ongoing exploitation of cost-effective and earth-abundant substitutes employed for HER

to replace the Pt-based metals has attracted enormous attention. Various non-noble materials including nickel-based alloys [5–9], Fe-based alloys [10–12], Co-based alloys [13,14], transition/alkaline earth metal oxide [15], metal sulfides [16–19], metal carbides/carbon materials [20–24], nitrides [25,26] as well as phosphides [27–29] have been tested for catalyzing the electrochemical HER.

Among various candidates, molybdenum disulfide (MoS₂) in the form of crystalline [30,31] and amorphous states [32,33] has received tremendous attention due to its abundance and high activity. However, HER performance of MoS₂ has been limited because of the insufficient active edge sites and the

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poor electrical transport. To unravel the conundrum of the inferior HER performance of MoS₂, three main issues have been proposed [34–37]: (1) the increase in the electrocatalytic activity of active sites, (2) the improvement in the electrical contact to the active sites, and (3) the enhancement in the number of active sites of this catalyst. Thus, a great deal of efforts have been made for modifying the surface structures of MoS₂-based materials, aiming at enhancing the HER properties of these materials. Among the reported efforts, the phase transformation from 2H-MoS₂ to the metallic 1T-MoS₂ [38,39], gentle oxidation [40,41], functional structural design [42,43], and stabilizing the edge layers with organic molecules [39] have been extensively attempted. Up to now, many reports have focused on the reduction in the stacked layers of the as-prepared MoS₂ to increase the number of edge sites for enhancing the electrocatalytic performance of this catalyst. Unquestionably, the presence of abundant active sites will be valuable for the increase in HER performance. But, a high stacked height along the (002) basal plane of MoS₂ is required to gain a maximum value of edge/basal ratio, which is crucial for the enhancement in HER property.

It should be pointed out that the drawbacks of poor bulk conduction and anisotropic electrical transport of MoS₂ restrict its overall HER efficiency. For this nano-structured material, the poor conductivity within the interparticle and interdomain can reduce the total HER activity. The improvement in the electrical contact between the active sites deserves to be considered to increase the intrinsic activity. Some promoters such as gold [44], core-shell MoO₃ [45], macroporous and mesoporous carbon materials [36,46–48], graphene [37,49], and graphene oxide [42,50] have been employed as electrical conduction supports to improve the electrical contact between the active sites of this catalyst. Compared with the above-mentioned supports, the electroless nickel-phosphorus (Ni-P) coating or powder exhibits the advantages of facile fabrication, anticorrosion, excellent wear resistance, prominently electrical conductivity, and the promisingly electrocatalytic activity of HERs [51–55]; thus, this material may be competent for the candidate employed as a support to enhance the HER performance of MoS₂. However, to the best of our knowledge, the electroless Ni-P coating or powder applied as the supporting substrate to enhance the HER property of MoS₂ has not yet been reported. We are sure that this work will offer a significant guidance for the exploitation of HER catalysts.

The aim of this study is to reveal the effect of Ni-P addition with different phosphorus content on the structure and HER activity of the fabricated MoS₂/Ni-P composite. Three Ni-P powders, i.e. low phosphorus- (Ni-LP), medium phosphorus- (Ni-MP), and high phosphorus-type Ni-P (Ni-HP) were employed as the supports for the fabrications of MoS₂/Ni-P composites via a hydrothermal synthesis process. Crystal structures, morphologies, components, and porosities of the three MoS₂/Ni-P composites (MoS₂/Ni-LP, MoS₂/Ni-MP, and MoS₂/Ni-HP) were characterized using X-ray diffraction (XRD), scanning electron microscope (SEM), Raman spectroscopy, energy dispersive X-ray spectrometry (EDS), X-ray photoelectron spectroscopy (XPS), and the nitrogen gas adsorption/desorption test. The electrocatalytic performances in HER of these composites were evaluated by techniques of linear

sweep voltammetry (LSV), Tafel polarization, cyclic voltammetry (CV), and electrochemical impedance spectroscopy (EIS).

Material and methods

Material

Analytical grade reagents of ammonium molybdate tetrahydrate ((NH₄)₆Mo₇O₂₄·4H₂O, 99 wt%), thiourea (CH₄N₂S, 99 wt%), nickel sulfate hexahydrate (NiSO₄·6H₂O), citric acid monohydrate (C₆H₈O₇·H₂O), sodium hypophosphite monohydrate (NaH₂PO₂·H₂O), succinic acid (C₄H₆O₄), ammonium hydrogen fluoride (NH₄HF), sodium acetate trihydrate (NaAc·3H₂O), ammonia solution (25 wt%), sodium hydroxide (NaOH), hydrochloric acid (36.5 wt%), and sulfuric acid (H₂SO₄, 98 wt%) were purchased from Jingchun Scientific Co. Ltd. (Shanghai, China). The 5 wt% of Nafion solution was provided by Alfa Aesar Chemicals Co. Ltd. (Shanghai, China). The above mentioned reagents were used as received and without further purification.

Synthesis of Ni-P powders

A kind of stainless steel slice was employed as the substrate for the fabrication of three nickel-phosphorous (Ni-P) powders (Ni-LP, Ni-MP, and Ni-HP). The chemical composition in weight content (wt%) of this slice was as follows: 0.08% for carbon, 0.03% for sulfur, 0.045% for phosphorous, 1.0% for silicon, 2.0% for manganese, 8.0% for nickel, 18.0% for chromium, and 70.845% for iron. First of all, the stainless steel slices with a dimension of 5 cm × 5 cm × 0.5 cm were polished mechanically to mirror finish and cleaned with absolute ethanol and distilled water successively. Then, they were immersed into 3 mol/L of HCl solution at room temperature for 3 min, followed by cleaned with distilled water and placed into the electroless plating bath for the preparation of Ni-P coatings.

The compositions of the electroless Ni-P bath were described as follows: 30 g/L of NiSO₄·6H₂O, 20 g/L of C₆H₈O₇·H₂O, 15 g/L of NaAc, 5 g/L of C₄H₆O₄, 5 g/L of NH₄HF, and 7.5 g/L of NaH₂PO₂·H₂O for Ni-LP coating (15 g/L for Ni-MP coating, and 30 g/L for Ni-HP coating). The pH value of this bath was adjusted to 4.6 by adding ammonia solution. The temperature of this solution during the electroless plating process was kept at 85 ± 2 °C; the plating time was controlled for 1 h. After the plating, the stainless steel slices loading Ni-P coatings removed from the bath were cleaned using deionized water adequately. Then, the deposited Ni-P coatings were scraped away from the substrate and milled in the agate-vibrating mill with the agate balls for 3 h. The size of the obtained Ni-P powders ranged from 1.5 to 15 μm. The determined weight content of phosphorus element for the three powders was that: 3.2 wt% of P for Ni-LP, 7.9 wt% of P for Ni-MP, and 11.1 wt% of P for Ni-HP.

Synthesis of the MoS₂/Ni-P composites

The preparation processes of the three MoS₂/Ni-P composites (denoted as MoS₂/Ni-LP, MoS₂/Ni-MP, and MoS₂/Ni-HP) were

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