INTERNATIONAL JOURNAL OF HYDROGEN ENERGY XXX (2017) 1-6



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# Significant improvement in the dehydriding properties of perovskite hydrides, NaMgH<sub>3</sub>, by doping with K<sub>2</sub>TiF<sub>6</sub>

## Zhongmin Wang<sup>\*</sup>, Song Tao, Jianqiu Deng, Huaiying Zhou, Qingrong Yao

School of Materials Science and Engineering, Guilin University of Electronic Technology, Guilin 541004, China

#### ARTICLE INFO

Article history: Received 23 October 2016 Received in revised form 8 December 2016 Accepted 16 December 2016 Available online xxx

Keywords: NaMgH<sub>3</sub> Perovskite structure K<sub>2</sub>TiF<sub>6</sub> dopant Lattice distortion Dehydriding property

#### ABSTRACT

In this study, perovskite hydrides (NaMgH<sub>3</sub>) were synthesized in the presence and absence of a dopant, e.g.,  $K_2$ TiF<sub>6</sub>, by high-energy ball milling. Clear lattice distortion was observed in  $K_2$ TiF<sub>6</sub>-doped NaMgH<sub>3</sub> as compared to pristine NaMgH<sub>3</sub>. The doped sample exhibited better dehydriding properties, and the onset desorption temperature significantly decreased from 580 K to 328 K. The maximum amount of desorbed hydrogen was approximately 3.8 wt% at 638 K, and approximately 90% of hydrogen was desorbed within 20 min. Differential scanning calorimetry results indicated that the activation energy for the two steps in the decomposition of NaMgH<sub>3</sub> decreases by doping with  $K_2$ TiF<sub>6</sub>. This result suggested that the lattice distortion and low decomposition energy of the NaMgH<sub>3</sub> phase, caused by the introduction of the  $K_2$ TiF<sub>6</sub> dopant, result in improved dehydriding properties for NaMgH<sub>3</sub>. © 2016 Hydrogen Energy Publications LLC. Published by Elsevier Ltd. All rights reserved.

## Introduction

NaMgH<sub>3</sub> has attracted considerable attention because of its high gravimetric and volumetric densities (6 wt% and 88 kg/ m<sup>3</sup>, respectively), as well as reversible hydriding and dehydriding properties [1–6]. NaMgH<sub>3</sub> is characterized by an orthorhombic perovskite structure comprising [MgH<sub>6</sub>] octahedra and [NaH<sub>12</sub>] cubo-octahedra, which is analogous to GdFeO<sub>3</sub>-type perovskite (space group Pnma). This space group is typical for low-tolerance-factor oxide perovskites, where the singly charged Na cation occupies eight-fold coordinated voids [7]. The desorption of NaMgH<sub>3</sub> is a two-step reaction: NaMgH<sub>3</sub>  $\rightarrow$  NaH + Mg + H<sub>2</sub>  $\rightarrow$  Na + Mg + 3/2H<sub>2</sub>. Sheppard et al. [8] have measured the kinetic and thermodynamic data for the decomposition of NaMgH<sub>3</sub> into NaH and Mg in the first desorption step. Desorption enthalpy and entropy are  $86.6 \pm 1.0 \text{ kJ/(mol H_2)}$  and  $132.2 \pm 1.3 \text{ kJ/(mol H_2 K)}$ , respectively, indicating that NaMgH<sub>3</sub> is thermodynamically more stable than MgH<sub>2</sub>. Some studies have reported that the kinetic properties of NaMgH<sub>3</sub> can be modified by doping with catalytic additives [9–11]. For example, Li et al. [10] have revealed that the *in situ* introduction of Mg<sub>2</sub>NiH<sub>4</sub> and YH<sub>3</sub> nanoparticles into NaMgH<sub>3</sub> leads to significantly enhanced kinetics for hydrogen absorption and desorption cycling. Komiya et al. [12] have reported that MMgH<sub>3</sub> (M = Na, K, and Rb) hydrides, synthesized by ball milling, are decomposed at temperatures between 673 and 723 K by several pathways depending on M. Chaudhary et al. [11] have reported that the onset decomposition temperature decreases with the introduction of a Si

E-mail addresses: zmwang@guet.edu.cn, ghab1987@163.com (Z. Wang). http://dx.doi.org/10.1016/j.ijhydene.2016.12.078

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Please cite this article in press as: Wang Z, et al., Significant improvement in the dehydriding properties of perovskite hydrides, NaMgH<sub>3</sub>, by doping with  $K_2$ TiF<sub>6</sub>, International Journal of Hydrogen Energy (2017), http://dx.doi.org/10.1016/j.ijhydene.2016.12.078



Fig. 1 – XRD patterns of as-synthesized NaMgH<sub>3</sub> samples. (a) NaMgH<sub>3</sub>, (b) NaMgH<sub>3</sub>-0.042K<sub>2</sub>TiF<sub>6</sub>, (c) Pristine  $K_2TiF_6$ .

dopant into NaMgH<sub>3</sub> by mechanical milling. Martínez-Coronado et al. [13] have reported that the hydrogen desorption temperature clearly decreases in Na<sub>1-x</sub>Li<sub>x</sub>MgH<sub>3</sub> (x = 0, 0.25, 0.5) because of the low structural stability of the Li-containing perovskite-type structure. Xiao et al. [14] have predicted that the substitution of Li in NaMgH<sub>3</sub> is favorable for onboard hydrogen storage applications based on density functional theory calculations. Pottmaier et al. [15] have also comprehensively investigated the structural and thermodynamic properties of NaMgH<sub>3</sub>.

Recently, Liu et al. [16] have reported that the combination of K, Ti, and F exerts a synergistic catalytic effect on the improvement of the reversible hydrogen storage properties of NaAlH<sub>4</sub> via the introduction of  $K_2TiF_6$  as a catalyst precursor: Approximately 4.4 wt% of hydrogen is released from NaAlH<sub>4</sub>–0.025  $K_2TiF_6$  within 40 min at 413 K. Li et al. [17] have investigated the effect of  $K_2TiF_6$  on the dehydrogenation properties of LiAlH<sub>4</sub> and suggested that TiH<sub>2</sub>, Al<sub>3</sub>Ti, KH, and LiF, formed *in situ*, play a crucial role in the improvement of the dehydrogenation properties of LiAlH<sub>4</sub>. Mustafa et al. [18] have reported that the addition of 10 wt%  $K_2TiF_6$  to MgH<sub>2</sub> leads to a lower onset desorption temperature of 518 K, corresponding to a decrease of approximately 105 K and 205 K as compared with those observed for as-milled and as-received MgH<sub>2</sub>, respectively. With this background,  $K_2TiF_6$  exhibited superior catalytic effects. Hence, some interesting results can be obtained using  $K_2 TiF_6$  as the catalyst while it is introduced into AMgH<sub>3</sub> perovskite-type hydrides (where A = alkaline earth metal).

To improve dehydriding properties of NaMgH<sub>3</sub>, NaMgH<sub>3</sub> samples with and without the  $K_2 TiF_6$  dopant were synthesized by high-energy ball milling. The structure and dehydriding properties of NaMgH<sub>3</sub> hydride with and without the  $K_2 TiF_6$  dopant were investigated and compared by X-ray diffraction (XRD), temperature-programmed desorption (TPD), and differential scanning calorimetry (DSC) analysis. Experimental results indicated that the addition of  $K_2 TiF_6$  leads to clear improvement in the dehydriding kinetic properties of NaMgH<sub>3</sub>.

### Materials and methods

NaMgH<sub>3</sub> samples with and without the  $K_2TiF_6$  dopant were prepared from stoichiometric mixtures of MgH<sub>2</sub> (>98% purity, Alfa Aesar), NaH (>95% purity, Sigma Aldrich), and  $K_2TiF_6$  (97% purity, Sigma Aldrich). The mixing molar ratio of NaH:MgH<sub>2</sub>: $K_2TiF_6$  was 0.75:1:0.042. All chemicals were directly used without further purification. The mixed powders were milled with stainless steel balls under H<sub>2</sub> (0.8 MPa) for 45 h at 350 rpm using a QM-3SP2 planetary mill at ambient temperatures. The weight ratio of ball to powder was 80:1. All material operations were conducted in an argon-filled glove box to eliminate the effect of oxygen and water moisture.

X-ray diffraction (XRD) measurements were conducted on an Empyrean PIXcel 3D (Cu  $K_{\alpha}$  radiation) diffractometer. Powder samples were sealed in a special cell, which was coated with a Kapton film to prevent exposure to air. To determine the phase structure, samples were annealed at 473 K for 4 h under H<sub>2</sub> (1.0 MPa) to eliminate the internal strain resulting from ball milling. The collected XRD data were evaluated by the Rietveld profile refinement method with Jade 6.5 (XRD analysis software). The amount of desorbed hydrogen and dehydriding kinetic properties of samples were measured on an automatic Sievert-type apparatus (PCTpro2000, Setaram Co.). For TPD experiments, the sample was heated from 298 K to 673 K in a vacuum chamber (10<sup>-3</sup> Pa) at a heating rate of 5 K/min. Dehydriding kinetics measurements were conducted at 638 K and 593 K in vacuum (10<sup>-3</sup> Pa),



Fig. 2 – Refined XRD patterns for pristine NaMgH<sub>3</sub> (Left) and NaMgH<sub>3</sub>-0.042K<sub>2</sub>TiF<sub>6</sub> (Right).

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