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# NH<sub>4</sub>-doped anodic WO<sub>3</sub> prepared through anodization and subsequent NH<sub>4</sub>OH treatment for water splitting



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#### ABSTRACT

Tungsten trioxide (WO<sub>3</sub>) prepared by anodization of a W foil was doped with  $NH_4$  through  $NH_4OH$  treatment at  $450\,^{\circ}C$ . Since aqueous  $NH_4OH$  was used during doping instead of  $NH_3$  gas, the treatment step does not require complicated annealing facilities. Moreover, the state of doped N is a form of  $NH_3$ -W instead of  $W_2N$ , which lowers the bandgap but increases photocorrosion. We found that incorporation of  $NH_4$  into  $WO_3$  leads to reduction of the bandgap from 2.9 eV to 2.2 eV, regardless of the amount of  $NH_4OH$  treatment, lowering the onset potential and increasing the current density at fixed potential for oxygen evolution reaction under illumination. Scanning electron microscopy, X-ray diffraction and X-ray photoelectron spectroscopy were employed to investigate the surface morphologies, crystallinities of tungsten oxides and existence of  $NH_4$  doping, respectively. The bandgap energy was determined by UV-V is spectroscopy to measure the transmittance and refraction. The water splitting performance of each sample was measured by electrochemical linear sweep voltammetry in a 3-electrode configuration under illumination.

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#### 1. Introduction

Tungsten trioxide ( $WO_3$ ) has long been studied due to its versatile applications in gas sensors [1], electrochromic devices [2], and electrodes for photocatalysis [3]. Since  $WO_3$  is a reliable and stable material that protects against photocorrosion during water splitting reactions [4–6], synthesis and characterization of nanostructured tungsten oxide as photoanode electrodes has been intensively investigated with the goal of readily reaching saturated photocurrent under external bias with the smallest overpotential possible [7].

The nanostructure of WO<sub>3</sub>, which is typically used as an n-type semiconductor, has a relatively far longer minority carrier lifetime than hematite ( $\alpha$ -Fe<sub>2</sub>O<sub>3</sub>), indicating that it is relatively stable in acidic media and has a less positive onset potential [8–11]. However, only a small portion of the solar spectrum is absorbed by WO<sub>3</sub> because of its large indirect band-gap energy (2.5–3.0 eV) [12–15].

Sol-gel [16], hydrothermal [17], solvothermal [18], anodization [13,14,19,20], sputtering [15,21], and chemical vapor deposition

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(CVD) [22] have been reported as methods for preparing well-controlled nanostructured WO<sub>3</sub>.

Doping with foreign elements has also been attempted to customize the bandgap of  $WO_3$  to produce efficient water splitting electrodes. High quantum efficiency has been reported to enable the poor spectral absorption to be overcome [23]. Different strategies for increasing quantum efficiency such as transition metal doping [24–26], noble metal deposition [27–29], dye sensitization [30], and organic material absorption [31] have been studied.

Although nitrogen doping is an effective method of reducing the bandgap of WO<sub>3</sub>, the effects of N-doping on the photoelectrochemical (PEC) performance of WO<sub>3</sub> films have been controversial [14,15,22].

Nah et al. reported that nanoporous WO<sub>3</sub> could be prepared by simple anodization and subsequent annealing at 300 °C under NH<sub>3</sub> gas. As a result, N-doped WO<sub>3</sub> showing significant enhancement of photocurrent in the visible light region was prepared [13]. Cole et al. reported that WO<sub>3</sub> film with a doping of N was fabricated by RF sputtering. In their study, the optical bandgap of WO<sub>3</sub> was reduced to 2.0 eV; however, the photocurrent was reduced due to degradation of electron transport caused by the defective lattice. Formation of W<sub>2</sub>N via conversion from WO<sub>3</sub> by NH<sub>3</sub> gas significantly reduced the bandgap from 2.9 eV to 2.2 eV, but increased instability of the photoelectrode due to photocorrosion and defects. In addition, electrons were easily recombined since grain size and

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crystallinity decreased in response to N-doping of defective sites [15]. It is possible that these opposing effects are responsible for the controversial results.

In this study, nanoporous WO $_3$  films prepared by anodization in non-fluoride based electrolytes were doped by a wet-based method using ammonium hydroxide (NH $_4$ OH) instead of NH $_3$  gas; thus, complex gas facilities were not required for annealing. As a result of using NH $_4$ OH, new phases (NH $_4$ ) $_{10}$ W $_{12}$ O $_{41}$  were generated, leading to the reduced bandgap. In addition, the PEC performance in 0.33 M H $_3$ PO $_4$  was comparable to that of the existing NH $_3$  treatment.

#### 2. Experimental

Prior to anodization, tungsten foils (Goodfellow, England, 99.95%, 0.33 mm) were cleaned with acetone, isopropanol and deionized (DI) water for 10 min in air at room temperature. The nanostructured WO<sub>3</sub> films were prepared in a 2-electrode configuration cell by applying constant potential at 50 V with a SourceMeter (Keithley 2400, US) at 95 °C for 20 h. During the process, 10 wt.%  $K_2HPO_4/glycerin$  anhydrous solution was used as the electrolyte, which was heated at 195 °C for 3 h to eliminate the moisture. After anodization, the fabricated WO<sub>3</sub> films were rinsed with isopropanol and DI water and then annealed for 1 h at 450 °C with different amounts (1, 2, and 3 g, corresponding to 0.029, 0.058, and 0.087 M, respectively) of NH<sub>4</sub>OH solution for NH<sub>4</sub>-doping.

Field emission scanning electron microscopy (FE-SEM, S4300, Hitachi, Japan) was employed to characterize the surface morphology and thickness. The crystal phase and chemical surface state were determined by X-ray diffraction (XRD, Rigaku D/max-RB, Japan) and X-ray photoelectron spectroscopy (XPS, K-Alpha, Thermo Scientific, USA) with an Al-K  $\alpha$  X-ray source. The indirect

bandgap energy was calculated from a Tauc plot determined using a UV/Vis/NIR spectrophotometer (LAMBDA 750, PerKin Elmer, USA).

A 3-electrode configuration cell composed of an anode (metal oxide), cathode (Pt-mesh) and reference electrode (Ag/AgCl 3 M KCl) was employed to measure the photoelectrochemical performance in potential ranges of 0-2 V at a scanning rate of 10 mV/s in 0.33 M  $H_3$ PO<sub>4</sub> electrolyte under AM 1.5G illumination.

#### 3. Results and discussion

Fig. 1 shows the top of the FE-SEM images of  $NH_4$ -doped anodic nanoporous tungsten trioxides, which were fabricated by annealing of anodic  $WO_3$  under different amounts of  $NH_4OH$ . The thickness of metal oxide has been determined based on the applied anodizing voltage and post annealing temperature [32]. In this study, the anodizing voltage (50 V) and post annealing temperature (450 °C) were fixed to investigate the effects of the amount of  $NH_4OH$  on  $NH_4$ -doping in the structure. The  $NH_4$  doped  $WO_3$  layers had a thickness higher than that of the pure  $WO_3$  layer because of the additional oxidation step.

X-ray diffraction and X-ray photoelectron spectroscopy were employed to determine the nitrogen state in WO<sub>3</sub> films. Fig. 2 shows the XRD patterns of nanoporous WO<sub>3</sub> prepared by anodization and annealing at  $450\,^{\circ}\text{C}$  with or without NH<sub>4</sub>OH treatment. Monoclinic WO<sub>3</sub> with preferential crystalline orientations of (002), (020), and (200) planes was confirmed at 23.119, 23.586, and 24.380°, respectively (JCPDS 43-1035). The intensity of WO<sub>3</sub> peaks has been reported to decrease as annealing temperature increased above  $450\,^{\circ}\text{C}$  with NH<sub>3</sub> gas [13,14]. As shown in Fig. 2, NH<sub>4</sub>OH treatment of samples at  $450\,^{\circ}\text{C}$  led to formation of stable crystalline WO<sub>3</sub> layers, prohibiting the formation of W<sub>2</sub>N phases. Note that peaks of

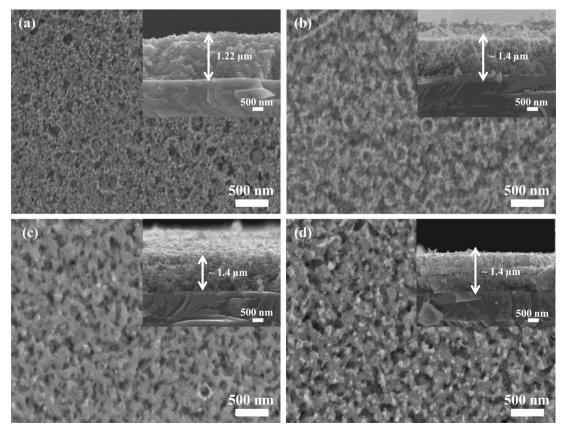


Fig. 1. FE-SEM images of the WO<sub>3</sub> films prepared by NH<sub>4</sub>-doping with (b) 0.029 M, (c) 0.058 M, and (d) 0.087 M NH<sub>4</sub>OH, respectively. Fig. 1 (a) is the SEM image of the as-prepared anodic WO<sub>3</sub> film.

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