

# Role of the oxidation state of cerium on the ceria surfaces for silicate adsorption



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## ABSTRACT

In this study, we have investigated the role of the Ce oxidation state ( $\text{Ce}^{3+}/\text{Ce}^{4+}$ ) on the  $\text{CeO}_2$  surfaces for silicate adsorption. In aqueous medium, the  $\text{Ce}^{3+}$  sites lead to the formation of  $-\text{OH}$  groups at the  $\text{CeO}_2$  surface through  $\text{H}_2\text{O}$  dissociation. Silicate ions can adsorb onto the  $\text{CeO}_2$  surface through interaction with the  $-\text{OH}$  groups ( $-\text{Ce}-\text{OH} + -\text{Si}-\text{O}^- \leftrightarrow -\text{Ce}-\text{O}-\text{Si}- + \text{OH}^-$ ). As the  $\text{Ce}^{3+}$  concentration increased from 19.3 to 27.6%, the surface density of  $-\text{OH}$  group increased from 0.34 to 0.72  $\text{OH}/\text{nm}^2$ . To evaluate the adsorption behaviors of silicate ions onto  $\text{CeO}_2$  NPs, we carried out an adsorption isothermal analysis, and the adsorption isotherm data followed the Freundlich model. The Freundlich constant for the relative adsorption capacity ( $K_F$ ) and adsorption intensity ( $1/n$ ) indicated that  $\text{CeO}_2$  NPs with high  $\text{Ce}^{3+}$  concentration show higher adsorption affinity with silicate ions. As a result, we have demonstrated that the Ce oxidation state ( $\text{Ce}^{3+}/\text{Ce}^{4+}$ ) on the  $\text{CeO}_2$  surface can have a significant influence on the silicate adsorption.

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## 1. Introduction

Ceria ( $\text{CeO}_2$ ) nanoparticles (NPs) have been widely used as a promising material in various applications in the fields of catalysis [1], fuel cells [2], gas sensors [3], and chemical mechanical planarization (CMP) [4,5]. In all these applications, the high performance of  $\text{CeO}_2$  NPs is attributed to their rich O vacancies and low redox potential [6,7]. It is well known that O vacancies are formed due to the highly mobile nature of the surface oxygens. Two excess electrons, left behind by the formation of O vacancies, are localized on the 4f-state of the nearest Ce ions, leading to a valence change in the Ce ions from  $\text{Ce}^{4+}$  to  $\text{Ce}^{3+}$  [8,9].

The  $\text{Ce}^{3+}$  ions at the surface are active sites for reactions in most applications [6,10,11]. In aqueous medium,  $\text{Ce}^{3+}$  ions act as active sites for  $\text{H}_2\text{O}$  dissociation, resulting in the formation of  $-\text{OH}$  groups at the surface [12]. The presence of  $-\text{OH}$  groups on the  $\text{CeO}_2$  surface allows for various interactions (e.g., molecular adsorption, desorption) in their many applications. In CMP applications, the OH groups can form a strong  $\text{Ce}-\text{O}-\text{Si}-\text{O}-$  bonding with silicate ions ( $-\text{Ce}-\text{OH} + -\text{Si}-\text{O}^- \leftrightarrow -\text{Ce}-\text{O}-\text{Si}- + \text{OH}^-$ ) [13]. The formation

of  $\text{Ce}-\text{O}-\text{Si}$  bonding directly corresponds with the polishing efficiency of  $\text{CeO}_2$  NPs [13]. The high affinity and binding energy with silicate ions lead to an increase in the formation of  $\text{Ce}-\text{O}-\text{Si}$  bonding, which can increase the removal rate of  $\text{SiO}_2$  films [14–17].

In our previous study, the surface function effects (e.g.,  $-\text{OH}$ ,  $-\text{NO}_3$  groups) on the interaction between  $\text{CeO}_2$  and silicate ions were studied through adsorption isotherms and theoretical analyses [15]. Besides the surface functionalities, the surface Ce oxidation state ( $\text{Ce}^{3+}/\text{Ce}^{4+}$ ) can have a significant influence on the interactions of  $\text{CeO}_2$  NPs with silicate ions. However, there is still a lack of understanding on the role of the Ce oxidation state ( $\text{Ce}^{3+}/\text{Ce}^{4+}$ ) in the  $\text{CeO}_2$  NPs for the interactions with silicate ions.

Herein, we prepared the  $\text{CeO}_2$  NPs at different  $\text{Ce}^{3+}$  concentrations, and the corresponding adsorption properties of silicate ions onto  $\text{CeO}_2$  surfaces were studied. The  $\text{Ce}^{3+}$  concentration on the surface depends on the size of the  $\text{CeO}_2$  NPs. The  $\text{Ce}^{3+}$  concentration increased with decreasing NP size due to the increase of the surface to volume ratio of the NPs [18–21]. To prepare the  $\text{CeO}_2$  NPs at different  $\text{Ce}^{3+}$  concentration, three different-sized  $\text{CeO}_2$  NPs (small, mid, and large  $\text{CeO}_2$ , hereafter noted as S- $\text{CeO}_2$ , M- $\text{CeO}_2$ , and L- $\text{CeO}_2$ , respectively) were synthesized in supercritical water (SCW) [22]. The adsorption properties of silicate ions onto three different-sized  $\text{CeO}_2$  NPs were investigated through adsorption isothermal

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analysis. Adsorption isotherm data for the silicate ions onto CeO<sub>2</sub> surfaces were fitted using the Langmuir and Freundlich models.

## 2. Experimental section

### 2.1. Synthesis of CeO<sub>2</sub> NPs

CeO<sub>2</sub> NPs were synthesized as follows. Precursor solutions: To prepare S-CeO<sub>2</sub> NPs, ammonium cerium nitrate was dissolved at 20.0 wt% in de-ionized water (DIW). To obtain M-CeO<sub>2</sub> and L-CeO<sub>2</sub> NPs, 37.5 wt% cerium nitrate solution was prepared in DIW. SCW: DIW was preheated to 400, 470 and 350 °C for the S-CeO<sub>2</sub>, M-CeO<sub>2</sub> and L-CeO<sub>2</sub> NPs, respectively. Precursor solutions and SCW were simultaneously delivered at a rate of 40 and 160 mL min<sup>-1</sup>, respectively, into the reactor. The reactors were controlled at the same temperature using SCW. The obtained CeO<sub>2</sub> NPs were rapidly and continuously collected. They were filtered, washed and dried in an oven at 100 °C for 12 h. Before the characterization, CeO<sub>2</sub> NPs were further dried at 100 °C in a vacuum oven for 12 h to remove residual water.

### 2.2. Material characterization

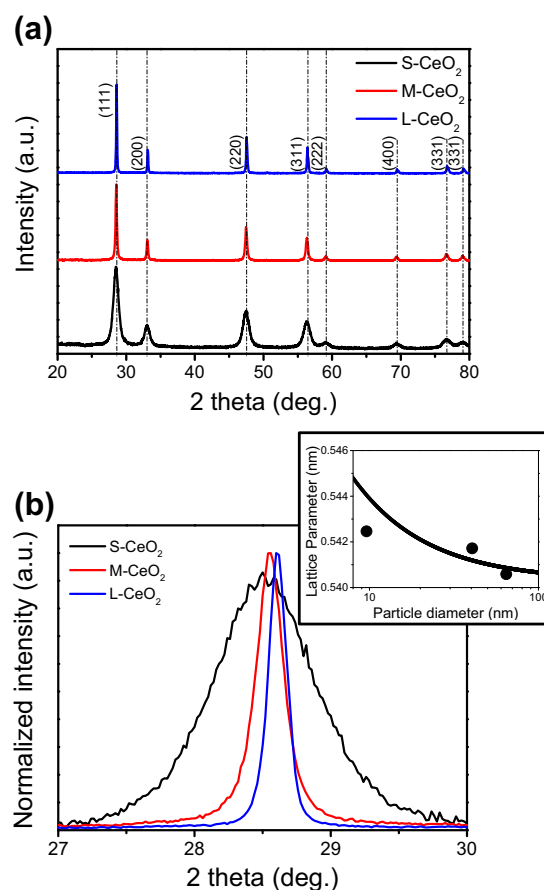
The CeO<sub>2</sub> NPs were characterized using various tools as follows. X-ray diffraction patterns of CeO<sub>2</sub> NPs were estimated using an X-ray diffraction analyzer (XRD, Bruker, New D8 Advance). The specific surface areas were measured by the Brunauer–Emmett–Teller (BET) method using N<sub>2</sub> gas adsorption at 77 K (BET, Quantachrome, Autosorb-1). Electron microscope images were obtained by a transmission electron microscope (TEM, JEOL, JEM-2100F). The average sizes ( $d_{TEM}$ ) of the CeO<sub>2</sub> NPs were calculated from the TEM images using the ImageJ analysis software (average 200 particles counted). The Ce<sup>3+</sup> concentrations in CeO<sub>2</sub> NPs were calculated using an X-ray photoelectron spectrometer (XPS, Thermo Fisher Scientific Co., theta probe base system). The surface characteristics of CeO<sub>2</sub> NPs were analyzed using Fourier transform infrared spectroscopy (FT-IR, Nicolet 5700, ThermoElectron) using an attenuated transmission reflectance (ATR, Smart Miracle, Pike Tech.) accessory with a ZnSe crystal. The weight loss of CeO<sub>2</sub> NPs was estimated using a thermo gravimetric analyzer with a mass spectrometer (TGA-MS, SDT Q600, TA Instruments) in the temperature range of 40–600 °C at a ramping rate of 10 °C/min in N<sub>2</sub>. It was used to calculate the number of –OH groups per area (#OH/nm<sup>2</sup>) of CeO<sub>2</sub> NPs.

### 2.3. Adsorption isotherm

The adsorption behaviors of silicate ions on the CeO<sub>2</sub> NPs were determined through the solution-depletion method using inductively coupled plasma atomic emission spectroscopy (ICP-AES, Optima 7300 DV, Perkin–Elmer). CeO<sub>2</sub> suspensions were prepared as a function of the silicate ions. The pH was adjusted to 7.0 by the addition of a HNO<sub>3</sub> and NH<sub>4</sub>OH solutions. Suspensions were aged for 12 h at room temperature under mixing. Then, they were centrifuged at 30,000 rpm for 20 min to obtain the unadsorbed silicate ions. Their supernatants were filtered using a 0.02 μm Anotop 25 syringe filter, and then measured using ICP. The adsorption isotherms of silicate ions on the CeO<sub>2</sub> NPs were derived from the difference between the added and remaining amount of silicate ions in the supernatants [20].

## 3. Results and discussion

We prepared three samples (S-CeO<sub>2</sub>, M-CeO<sub>2</sub>, and L-CeO<sub>2</sub>) at different Ce<sup>3+</sup> concentrations to understand the role of the Ce oxidation state (Ce<sup>3+</sup>/Ce<sup>4+</sup>) of CeO<sub>2</sub> NPs for interactions with silicate



**Fig. 1.** (a) XRD patterns of CeO<sub>2</sub> NPs. (b) Magnified view of the (111) peak in XRD. The inset presents the lattice parameter of CeO<sub>2</sub> NPs as a function of particle size. The curve is obtained by the equation ( $\alpha = \alpha_{\text{bulk}} + 0.036/D$ ) in ref 22.

ions. The particle sizes ( $d_{BET}$ ) of the CeO<sub>2</sub> NPs were calculated from  $d_{BET} = 6000/(SSA_{BET} \rho)$ , where  $SSA_{BET}$  is the specific surface area (m<sup>2</sup>/g) and  $\rho$  is the density of CeO<sub>2</sub> (7.2 g/cm<sup>3</sup>). The resulting  $d_{BET}$  values were 11.8 (S-CeO<sub>2</sub>), 63.1 (M-CeO<sub>2</sub>), and 320.5 nm (L-CeO<sub>2</sub>) (Table 1). Fig. 1 shows the XRD patterns of the CeO<sub>2</sub> NPs. All the CeO<sub>2</sub> NPs have a well crystalline cubic fluorite structure (JCPDS 65-5923). The crystallite sizes ( $d_{XRD}$ ) were calculated using the full-width at half-maximum of the (111) peak from the Scherrer equation, and the lattice parameters were obtained from the (111) diffraction peak position. The  $d_{XRD}$  values of the CeO<sub>2</sub> NPs are summarized in Table 1. Fig. 1 (b) shows a change in the (111) peak in the XRD patterns of the CeO<sub>2</sub> NPs. The (111) peak is shifted toward a lower 2 theta value as the size of the CeO<sub>2</sub> NPs decreased. As the particle size  $d_{XRD}$  of CeO<sub>2</sub> decreased from 64.4 to 9.6 nm, the lattice parameter increased from 0.5406 to 0.5425 nm, a 3.5% increase (inset in Fig. 1). The lattice parameter of S-CeO<sub>2</sub> (0.5425 nm) was higher than that of bulk-CeO<sub>2</sub> (0.5403 nm) (JCPDS 65-5923), which is attributed to the increase of O vacancies with an increasing surface to volume ratio [19–21,23]. The formation of O vacancies left two free electrons on the Ce ions at the surface, leading to a reduction of Ce<sup>4+</sup> to Ce<sup>3+</sup>. This change in the oxidation state of the Ce ions leads to a lattice expansion of the CeO<sub>2</sub> structure (cubic fluorite) because the ionic radius of Ce<sup>3+</sup> (1.143 Å) is bigger than that of Ce<sup>4+</sup> (0.970 Å) [19].

The TEM images and fast Fourier transformed (FFT) patterns of the CeO<sub>2</sub> NPs are shown in Fig. 2. The size distributions of the CeO<sub>2</sub> NPs were measured from the TEM images (insets in Fig. 2). Both M-CeO<sub>2</sub> and L-CeO<sub>2</sub> NPs have a faceted shape whereas S-CeO<sub>2</sub> NPs have a rounded shape. The corresponding FFT patterns

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