



# Sn–CeO<sub>2</sub> thin films prepared by rf magnetron sputtering: XPS and SIMS study

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## ARTICLE INFO

### Article history:

Received 17 April 2008

Received in revised form 5 January 2009

Accepted 23 February 2009

Available online 6 March 2009

### PACS:

68.47.Gh

82.80.Pv

68.49.Sf

81.15.Cd

### Keywords:

Cerium oxide

Tin–cerium mixed oxide

SIMS

XPS

Magnetron sputtering

## ABSTRACT

Sn addition in the CeO<sub>2</sub> thin film by simultaneous Sn metal and cerium oxide magnetron sputtering causes growth of Ce<sup>3+</sup> rich films whilst pure cerium oxide sputtering provides stoichiometric CeO<sub>2</sub> layers. Ce<sup>4+</sup> → Ce<sup>3+</sup> conversion is explained by a charge transfer from Sn atoms to unoccupied orbital Ce 4f<sup>0</sup> of cerium oxide by forming Ce 4f<sup>1</sup> state. XPS and SIMS revealed a formation of a new chemical Ce(Sn)<sup>+</sup> state, which belongs to SnCeO<sub>2</sub> species.

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## 1. Introduction

The electronic structure of the CeO<sub>2</sub> oxide is characterized by unoccupied 4f states of Ce<sup>4+</sup> (4f<sup>0</sup>) whilst the Ce<sub>2</sub>O<sub>3</sub> oxide has a Ce<sup>3+</sup> (4f<sup>1</sup>) configuration. Different 4f configurations for Ce<sup>4+</sup> and Ce<sup>3+</sup> result in different core level and valence band structures. One of the important properties of ceria is its oxygen storage capacity, which can provide oxygen to the gas mixture in catalytic contexts. The key factor for this property is the reversible transformation from Ce<sup>4+</sup> to Ce<sup>3+</sup>.

The cerium chemical state may be determined by a photoelectron spectroscopy. Using a conventional laboratory X-ray source, Ce 3d levels can be analysed [1–3]. The Ce 3d spectrum of partially reduced ceria consists of three 3d<sub>3/2</sub>–3d<sub>5/2</sub> spin–orbit–split doublets representing different 4f<sup>0</sup> (Ce<sup>4+</sup>) configurations in the photoemission final state. Two other doublets represent 4f<sup>1</sup> (Ce<sup>3+</sup>) states.

The previous X-ray photoelectron spectroscopy (XPS) and resonance photoelectron spectroscopy (RPES) studies on epitaxial CeO<sub>2</sub> thin films grown on Cu(1 1 1) showed that the Sn deposition on cerium oxide causes Ce<sup>3+</sup> intensity to appear in the Ce 3d

spectra [4,5]. The mixed oxide Sn–Ce–O has been found to be efficient CO oxidation catalyst which was more active than the individual oxides.

The purpose of this study was to contribute to the elucidation of the Sn–Ce interaction mechanism in mixed oxides prepared by simultaneous magnetron sputtering of ceria and tin.

There are several reports concerning the deposition of CeO<sub>2</sub> films using sputtering, mainly for superconducting and microwave applications, e.g., [6–9].

Secondary ion mass spectroscopy (SIMS) study of the chemical nature of ceria thin films showed the possibility to determine depth profiles by analysing different positive ion clusters [10].

We report that the Sn introduction in the CeO<sub>2</sub> thin film causes Ce<sup>3+</sup> concentration enhancement and complete Ce<sup>4+</sup> → Ce<sup>3+</sup> conversion could be caused by a charge transfer from Sn atoms to unoccupied orbital Ce 4f<sup>0</sup> of cerium oxide by forming Ce 4f<sup>1</sup> state.

## 2. Experimental

Non-reactive rf magnetron sputtering was used to deposit CeO<sub>2</sub> and Sn–CeO<sub>2</sub> thin films on Si(1 0 0) wafer substrates. CeO<sub>2</sub> sputtering was performed using a pure CeO<sub>2</sub> target (Kurt J. Lesker Company, 99.99% purity) at a distance of 90 mm from the Si substrate (Si wafer covered by natural oxide layer) with rf power of

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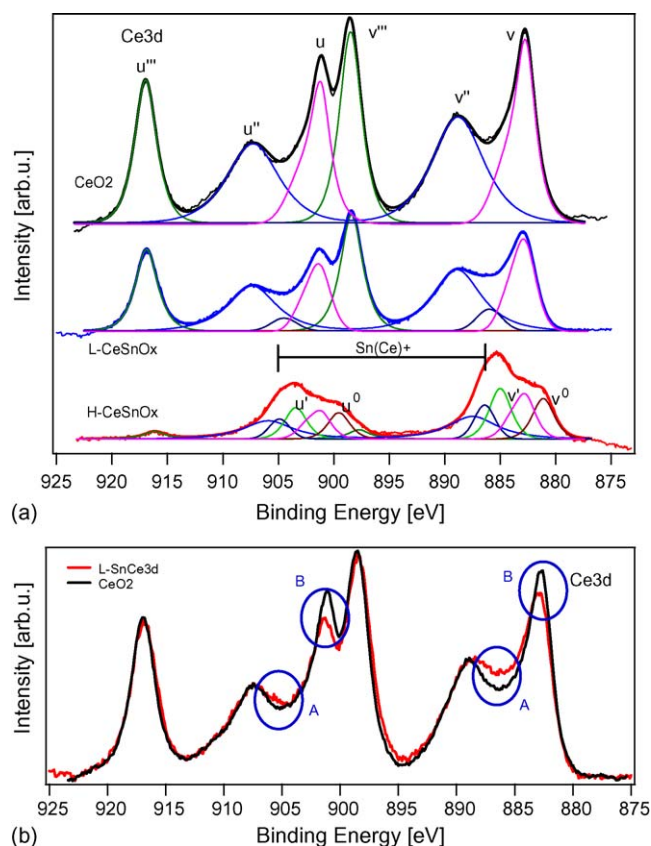
20 W. Sn–CeO<sub>2</sub> thin films sputtering was performed from the composite target prepared by partial covering of the ceria target with a Sn foil (Goodfellow Cambridge Limited, 99.99% purity). Deposition was carried out at room temperature of the substrate in Ar atmosphere keeping the total pressure in the deposition chamber constant at  $6 \times 10^{-1}$  Pa.

XPS and SIMS study were performed ex-situ in two analysis chambers. All photoemission data acquisition was done with the sample at 300 K using K $\alpha$  photons from the Al anode of the X-ray source (1486.6 eV). The spectra were acquired under the emission angle perpendicular to the sample surface. SIMS analysis was performed in PHI 600 series instrument by using a scanned Ar<sup>+</sup> ion beam with following parameters: ion current: 60 nA, spot size 200  $\mu$ m, crater raster area  $1.5 \times 1.5$  mm<sup>2</sup>. As the source of the primary beam the Specs IQE 12/38 ion gun was used and quadrupole mass analyser was used for detection of secondary ion signal. In order to obtain well resolved depth profiles and minimize crater effects, the technique of gating of the secondary ion beam was used together with secondary ion optics focused to area of  $0.5 \times 0.5$  mm<sup>2</sup> [11].

### 3. Results

#### 3.1. Pure cerium oxide

The 3d core level spectrum of continuous CeO<sub>2</sub> film deposited without adding Sn is presented in Fig. 1(a) (upper spectrum). It consists of three 3d<sub>3/2</sub>–3d<sub>5/2</sub> spin–orbit–split doublets (u'''–v''', u''–v'' and asymmetric u–v). The appearance of a high u'''–v''' signal, together with u–v peaks which are less intense than the u''–v'' peaks, is evidence of the formation of CeO<sub>2</sub> oxide.



**Fig. 1.** (a) XPS Ce 3d core level spectra of continuous pure cerium oxide (CeO<sub>2</sub>), low Sn content (L-SnCeO<sub>x</sub>) and high Sn content (H-SnCeO<sub>x</sub>) cerium oxide films including their deconvolution. (b) Comparison of XPS Ce 3d spectra of pure (CeO<sub>2</sub>) and low Sn content (L-SnCeO<sub>x</sub>) cerium oxide.

The thickness of this layer was estimated to be 2.3 nm taking into account a very low intensity of Si 2p XPS signal and a value of inelastic mean free path of signal electrons in ceria (2.4 nm for Si 2p electrons and Al K $\alpha$  X-ray source [12]. The Ce 3d emission peaks were fitted by pseudo-Voigt functions (combination Gaussian and Lorentzian functions). The asymmetry of u–v doublet was modelled using two components with energy shift 1.6 eV. Further details and resulting fitting parameters are introduced in Table 1.

#### 3.2. Sn–cerium oxide – XPS

Similarly to the Sn deposition on CeO<sub>2</sub>(1 1 1) [4,5], the interaction of the added metal resulting in a partial reduction of CeO<sub>2</sub> to Ce<sub>2</sub>O<sub>3</sub> was observed. Examples of the Ce 3d spectra for two Sn contents are plotted in Fig. 1(a) (L stands for a low Sn content, H for higher Sn content). The spectra were fitted by using the same

**Table 1**

List of resulting fitting parameters of Ce 3d core level spectrum for all samples.

	H-CeO <sub>x</sub>	L-CeO <sub>x</sub>	CeO <sub>2</sub>
<b>u'''–v''' Ce 3d doublet</b>			
Position	897.74	898.40	898.50
Amplitude	375.80	4738.85	7797.00
Width	2.30	2.41	2.42
Shape	1.00	0.80	0.76
Split	18.43	18.49	18.48
Ratio	1.24	1.42	1.34
<b>u''–v'' Ce 3d doublet</b>			
Position	887.49	888.89	888.82
Amplitude	919.99	2497.86	4315.57
Width	5.30	5.23	5.66
Shape	0.90	0.82	0.69
Split	18.43	18.49	18.48
Ratio	1.24	1.36	1.34
<b>u–v Ce 3d asymmetrical doublet</b>			
<b>1st component</b>			
Position	882.92	882.80	882.69
Amplitude	1455.64	3513.60	6803.41
Width	2.26	2.48	2.17
Shape	0.60	0.44	0.85
Split	18.43	18.49	18.48
Ratio	1.46	1.36	1.30
<b>2nd component</b>			
Position	884.32	884.40	884.49
Amplitude	393.02	948.67	1836.92
Width	2.00	2.26	3.11
Shape	0.60	0.60	0.60
Split	18.43	18.49	18.48
Ratio	1.21	1.35	1.21
<b>u'–v' Ce 3d doublet</b>			
Position	885.00	xxxx	xxxx
Amplitude	2061.17	xxxx	xxxx
Width	2.40	xxxx	xxxx
Shape	0.60	xxxx	xxxx
Split	18.43	xxxx	xxxx
Ratio	1.61	xxxx	xxxx
<b>u<sup>0</sup>–v<sup>0</sup> Ce 3d doublet</b>			
Position	881.12	xxxx	xxxx
Amplitude	1648.94	xxxx	xxxx
Width	2.38	xxxx	xxxx
Shape	0.60	xxxx	xxxx
Split	18.43	xxxx	xxxx
Ratio	1.54	xxxx	xxxx
<b>Ce–Sn + Ce 3d doublet</b>			
Position	886.40	885.99	xxxx
Amplitude	1386.36	883.98	xxxx
Width	2.26	2.70	xxxx
Shape	0.26	0.00	xxxx
Split	18.51	18.49	xxxx
Ratio	1.70	1.70	xxxx

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