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## Computational prediction of hydrogen sulfide and methane separation at room temperature by anatase titanium dioxide

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#### ABSTRACT

Removal of hydrogen sulfide  $(H_2S)$  is a key step for biogas purification. Herein, the adsorption of  $H_2S$  and methane  $(CH_4)$  on anatase titanium dioxide  $(TiO_2)$  has been studied by first principle calculations. It is found that  $TiO_2$  offers excellent capacity for the  $H_2S/CH_4$  separation. Using force-field molecular dynamics, this high separation capacity has been examined at room temperature.

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#### 1. Introduction

Biogas, generated from the anaerobic digestion of biological wastes, has been viewed as a promising renewable energy source [1,2]. Generally the major compositions of biogas are methane (CH<sub>4</sub>) and carbon dioxide (CO<sub>2</sub>), together with small amount of impurities, such as hydrogen sulfide (H<sub>2</sub>S), moisture, nitrogen oxides, volatile organic compounds, etc. [1]. Among those impurities, H<sub>2</sub>S is the most problematic contaminant due to its high toxicity and corrosivity. Therefore, efficient removal of H<sub>2</sub>S is critical for the large-scaled applications of biogas.

In the industry, the technology most commonly employed for H2S removal involves the use of active media, such as iron oxides (Fe<sub>2</sub>O<sub>3</sub> and Fe<sub>3</sub>O<sub>4</sub>), zinc oxide (ZnO), alkali solution, etc. [3,4]. Although such treatment is simple and effective, the removal is actually based on the consumption of the media. Using Fe<sub>2</sub>O<sub>3</sub> as an example, Fe<sub>2</sub>O<sub>3</sub> + 3H<sub>2</sub>S $\rightarrow$ Fe<sub>2</sub>S<sub>3</sub> + 3H<sub>2</sub>O; as a result, the used media needs to be renewed and safely disposed from time to time, which gets rise to additional cost and environmental concerns [5]. Another well-established technique is based on the Claus process, H<sub>2</sub>S + 1/2O + 2 $\rightarrow$ S + H<sub>2</sub>O [6]. Apparently, this process recovers elementary sulfur and no additional solid media has to be sacrificed, while the environmental issues associated with its by-products, like SO<sub>x</sub>, are seriously concerned [7]. To address the environmental problems, it is desirable to develop clean technologies for H<sub>2</sub>S-removal.

Photo-induced  $H_2S$  splitting,  $H_2S \rightarrow S + H_2$ , is attractive, because it not only recovers elementary sulfur as Claus process does, but also generates hydrogen fuel with sunlight as the only energy input. Therefore, photocatalytic decomposition  $H_2S$  over semiconductors

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has been proposed as an option to produce solar-hydrogen [8]. Using CdS-based photocatalysts, H2S-splitting has been realized even under visible light [9], which can harvest more sunlight and thus reduce the purification cost. To make this feasible for biogas purification, it is essential to ensure that H<sub>2</sub>S can be efficiently captured by photocatalysts, which underlines the importance of the separation of H<sub>2</sub>S. Under this context, we investigated the adsorption of H<sub>2</sub>S on anatase titanium dioxide (TiO<sub>2</sub>), one of the mostly widely employed photocatalysts. The initial motivation is to investigate whether H<sub>2</sub>S can be efficiently separately from CH<sub>4</sub> and strongly captured by TiO2, which is the basis for photocatalytic H<sub>2</sub>S-splitting. As shown below, anatase TiO<sub>2</sub> is an excellent media for H<sub>2</sub>S/CH<sub>4</sub> separation, especially when they are dominated by minority surface (001). The photocatalytic splitting of H<sub>2</sub>S over TiO<sub>2</sub> photocatalysts is out of the scope of this letter, but will be studied experimentally in the future. In nature, anatase TiO2 crystals are dominated by the majority surface (101) [10], but if special controlling agents (e.g., fluorine acid) are employed, well-defined crystals with high percentage of minority surface (001) can be synthesized readily [11-14].

#### 2. Computational methods

In nature, anatase  $TiO_2$  crystals are dominated by the majority surface (101) [10], but if special controlling agents (e.g., fluorine acid) are employed, well-defined crystals with high percentage of minority surface (001) can be synthesized readily [11–14]. In this letter, both (101) and (001) have been investigated and modeled by slab models, as shown in Figure 1.

The adsorption of  $H_2S$  and  $CH_4$  on (001) and (101) surfaces has been studied by density functional theory (DFT) [15]. The calculations were carried out using DMol3 package [16,17]. The exchange and correlation terms were treated under the generalized gradient

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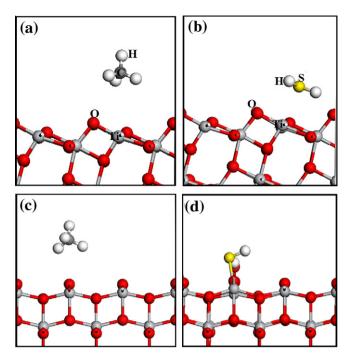


Figure 1. Optimized geometries for (a)  $CH_4$  and (b)  $H_2S$  on  $TiO_2(101)$ , (c)  $CH_4$  and (d)  $H_2S$  on  $TiO_2(001)$ .

approximation (GGA) functional by Perdew et al. [18]. A double numerical quality basis set with polarization function (DNP) were utilized for all geometric optimization and total energy calculations [19]. K-space was sampled by the gamma point due to the large size of the supercells. Along the z-direction, a vacuum space of 20 Å was employed to avoid the interaction between neighboring images. The gas–TiO $_2$  interaction was described by the averaged adsorption energy, *E*ads, which is defined by

$$Eads = E(gas) + E(TiO_2) - E(gas-TiO_2)$$

where E(gas),  $E(TiO_2)$  and  $E(gas-TiO_2)$  are total energies of single gas molecule, clean  $TiO_2$  slab and the interacting gas- $TiO_2$  system. By this definition, positive Eads indicates that the adsorption is stable.

Force-field molecular dynamics (FFMD) has been further employed to simulate the separation of  $H_2S$  from  $CH_4$  based on periodical boundary models, as shown in the Support information (Figure S1). FFMD was performed in the NVT ensemble at 298.0 K, with a timestep of 1.0 fs. The overall simulation time is up to 400 ps (1 ps =  $10^{-12}$  s), which is long enough to show the separation of  $H_2S$  as shown below. The random velocities were generated from the Boltzmann distribution and the temperature was maintained using the Andersen method [20] with a collision ratio of 1.00. The gas–gas and gas– $TiO_2$  interactions were described by the COMPASS force field [21].

#### 3. Results and discussion

Figure 1 shows the optimized geometries of  $H_2S$  and  $CH_4$  over (001) and (101) surfaces. As indicated by the adsorption energies (Eads),  $CH_4$  can only weakly adsorb on both (001) and (101) with Eads = 0.02 and 0.03 eV, respectively, which is typical physisorption, being line with previous report [22]; however,  $H_2S$  can be strongly captured by anatase  $TiO_2$ , with Eads = 0.56 eV via molecular adsorption on (101), agreeing well with the report by Lin and coworkers (Eads = 0.49 eV) [23]. With respect to  $TiO_2(101)$ , (001) is more reactive and Eads is up to 1.43 eV.  $H_2S$  actually spontaneously

dissociates on  $TiO_2(001)$ . On (101), there is a barrier of 0.37 eV for the dissociation of  $H_2S$  [23]. In fact, similar difference of the adsorption capacity between (101) and (001) has been well known for water on  $TiO_2$  [10,24–26]. From the optimized geometries (see Figure S2 in the Support information), it is found that both five-coordinated titanium ( $Ti_{5c}$ ) and two-coordinated oxygen ( $O_{2c}$ ) are involved in  $H_2S$  dissociation on  $TiO_2(001)$ ; moreover, short  $Ti_{5c}-O_{2c}$  distance plays a critical role for the formation of  $H-O_{2c}$  hydrogen bonds (HBs). Based on the above calculated data, it appears that  $TiO_2$  surfaces, especially minority surface (001), can offer strong capacity to separate  $H_2S$  from  $CH_4$  effectively.

Experimentally, the adsorption of H<sub>2</sub>S on TiO<sub>2</sub> has been studied for many years [27-29]. Using temperature-programmed desorption, both molecular and dissociative adsorptions have been reported over rutile surfaces [30,31], and the dissociation is believed to be resulted by defects, like oxygen vacancies [30]. Over anatase TiO2, however, only molecular adsorption has been identified [30]. It should be clear that typical anatase TiO<sub>2</sub> crystals are dominated by the majority surface (101). As shown in Figure 1b, H<sub>2</sub>S does adsorb molecularly on (101). While in the case of anatase  $TiO_2(001)$ , which is the minority surface with a low percentage in most TiO<sub>2</sub> samples and has been rarely studied before, spontaneous dissociation of H<sub>2</sub>S is obtainable even there is no defect, as predicted in Figure 1d. This may introduce new reaction routes for the degradation of H<sub>2</sub>S over anatase TiO<sub>2</sub>. For instance, HS<sub>ads</sub> generated by H<sub>2</sub>S dissociation may be attacked by photo-induced holes, and form  $S^{\bullet}$  radicals via  $HS_{ads} + h^{+} \rightarrow HS^{+}_{ads} \rightarrow S^{+}_{ads} + H^{+}$ , with an analogy to the general reaction pathways proposed by Portela et al. [28]. S radicals may further react with other S<sup>-</sup> radicals to form elementary sulfur or be directly oxidized by adsorbed O<sub>2</sub> or lattice oxygen to form SO<sub>2</sub>. The above speculations is beyond the scope of this work, but will be investigated in our following work.

Giving that gas adsorption and desorption on surfaces are strongly affected by gas–gas and gas–TiO<sub>2</sub> collisions, it is essential to take more gas molecules involved in the simulation, which can be explored by large-scaled FFMD simulation. At the initial state, 72 H<sub>2</sub>S molecules are homogeneously mixed with 288 CH<sub>4</sub> molecules and 6 H<sub>2</sub>S molecules adsorb on TiO<sub>2</sub> based on energy minimization, as shown in Figure 2a. Only after 50 ps, half of the H<sub>2</sub>S molecules are captured by TiO<sub>2</sub> (see Figure S3 in the Support information) and 220 ps later all H<sub>2</sub>S molecules are separated from CH<sub>4</sub> and strongly adsorb on TiO<sub>2</sub>, as shown in Figure 2b. In the following 180 ps (t = 220–400 ps), no H<sub>2</sub>S desorption has been observed, indicating H<sub>2</sub>S molecules are strongly captured by TiO<sub>2</sub> surface, being consistent with the prediction by DFT calculations.

To minimize the cost for renewing the TiO<sub>2</sub> media, it is expected that TiO<sub>2</sub> surfaces can keep the high working efficiency even when they are partially covered by H<sub>2</sub>S molecules. As observed in FFMD, the adsorption rate of H<sub>2</sub>S has no notable change when more and more H<sub>2</sub>S molecules adsorb on TiO<sub>2</sub> surfaces. For instance, there are 162 Ti<sub>5c</sub>-sites on the two surfaces of the slab shown in Figure 2, and at t = 210 ps, 70 sites have been occupied by H<sub>2</sub>S and all the left by CH<sub>4</sub>; however, the adsorption of the last two H<sub>2</sub>S molecules only takes 10 ps. Such stable performance has been contributed by two factors: (i) CH<sub>4</sub> desorption occurs almost at the same rate with its adsorption, obviously due to its weak adsorption (Eads = 0.03 eV, comparable to the thermal energy at room temperature 0.026 eV), which is critically important since the adsorption sites occupied by CH<sub>4</sub> can be quickly released for H<sub>2</sub>S adsorption; and (ii) H<sub>2</sub>S molecules adsorbed on TiO<sub>2</sub> surface are linked with H<sub>2</sub>S molecules in the H<sub>2</sub>S/CH<sub>4</sub> stream by HBs (similar with the HB network in water) and thus promote the separation of H<sub>2</sub>S from CH<sub>4</sub>. In fact, almost all H<sub>2</sub>S molecules can be captured by TiO<sub>2</sub> surfaces immediately when the H<sub>2</sub>S-TiO<sub>2</sub> distance is smaller than 3.50 Å (Ti–S bond: 2.46 Å) based on the snapshots collected in this simulation.

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