



# Optical thermometry based on up-conversion luminescence of $\text{Tm}^{3+}$ doped transparent $\text{Sr}_2\text{YF}_7$ glass ceramics



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## ABSTRACT

$\text{Tm}^{3+}/\text{Yb}^{3+}$  co-doped transparent glass ceramics (GC) containing  $\text{Sr}_2\text{YF}_7$  nanocrystals were successfully manufactured by traditional melt-quenching technique. Their structural and up-conversion luminescent properties were systemically investigated through X-ray diffraction (XRD), transmission electron microscope (TEM) and a series of spectroscopy methods. Tremendously enhanced up-conversion emissions with obvious Stark splitting and prolonged luminescence lifetime were observed after crystallization. These optical spectroscopy results manifest that  $\text{Tm}^{3+}$  ions have incorporated into the  $\text{Sr}_2\text{YF}_7$  crystalline lattice with low phonon energy ( $\sim 400 \text{ cm}^{-1}$ ) after heat-treatment. The  $^3\text{F}_{2,3}$  and  $^3\text{H}_4$  levels of  $\text{Tm}^{3+}$  can be explored as thermally coupled energy levels (TCEL) for temperature sensing because of their befitting energy gap ( $\sim 2000 \text{ cm}^{-1}$ ). At the same time, the population of  $^1\text{G}_4$  state of  $\text{Tm}^{3+}$  is indirectly originated from  $^3\text{H}_4$  one. Therefore the temperature sensing performances of  $\text{Tm}^{3+}$  doped  $\text{Sr}_2\text{YF}_7$  GC were investigated by fluorescence intensity ratio (FIR) between  $^3\text{F}_{2,3} \rightarrow ^3\text{H}_6$  and  $^1\text{G}_4 \rightarrow ^3\text{F}_4$  up-conversion emissions of  $\text{Tm}^{3+}$ . Results show that the theoretical maximum value of relative sensitivity  $S_{R-\text{max}}$  is  $1.16\% \text{ K}^{-1}$  at 428 K, and absolute sensitivity  $S_A$  keeps increasing with temperature. Such  $\text{Tm}^{3+}/\text{Yb}^{3+}$  co-doped  $\text{Sr}_2\text{YF}_7$  GC may be excellent candidate for optical temperature sensors.

## 1. Introduction

Recently, optical temperature sensors, types of non-contact temperature sensors, have been investigated widely due to their potential applications in electrical power station, oil refineries, building fire detection and biocompatible temperature probe [1–13]. Compared to conventional contact temperature sensors, optical temperature sensors have high sensitivity, short response time and can be used in corrosively, electromagnetically and thermally special environment [5,8]. Optical temperature sensors are based on optical techniques of luminescent materials, such as fluorescence lifetime, amplified spontaneous emissions and fluorescence intensity ratio (FIR) [7–9]. Among these optical based techniques, FIR technique has been regarded as a promising approach for temperature sensing because of their intrinsic advantages of high reliability, improved measurement accuracy and broad operating temperature range [5]. This technique is built on temperature dependent emissions from thermally coupled energy levels (TCEL), which makes it can avoid the influences of spectrum losses and fluctuations in excitation intensity [14–16].

Rare earth ( $\text{RE}^{3+}$ ) ions (such as  $\text{Eu}^{3+}$ ,  $\text{Er}^{3+}$ ,  $\text{Ho}^{3+}$ ,  $\text{Tm}^{3+}$ ) were

usually used as active centers for temperature sensing based on FIR technique owing to the different dependences of their electronic transitions on temperature [10,16–18]. Among these  $\text{RE}^{3+}$  ions,  $\text{Er}^{3+}$  ion is one of the most sensitive probe in optical temperature sensing [19–21]. The up-conversion FIR value of  $\text{Er}^{3+}$  TCEL ( $^2\text{H}_{11/2}$  and  $^4\text{S}_{3/2}$ ) varies with the temperature notably, which can be acted as index of temperature [16]. Unfortunately, it is difficult to avoid the spectra overlap of emission bands from  $\text{Er}^{3+}$  TCEL for their small energy gap ( $\sim 720 \text{ cm}^{-1}$ ), which may impact the measurement resolution [22].

$\text{Tm}^{3+}$  ion can overcome the intrinsic shortcoming of  $\text{Er}^{3+}$ . As shown in Fig. 1,  $^3\text{F}_{2,3}$  and  $^3\text{H}_4$  levels of  $\text{Tm}^{3+}$  are chosen as TCEL because their energy gap ( $\sim 2000 \text{ cm}^{-1}$ ) is suitable and larger than that of  $\text{Er}^{3+}$  [22]. In fact, the population of  $^1\text{G}_4$  and  $^1\text{D}_2$  states of  $\text{Tm}^{3+}$  are indirectly originated from  $^3\text{H}_4$  one. Hence, the up-conversion FIR of transitions from  $^3\text{F}_{2,3}$  and other three levels ( $^1\text{D}_2$ ,  $^1\text{G}_4$ ,  $^3\text{H}_4$ ) of  $\text{Tm}^{3+}$  ions can be applied in temperature detecting [4].

Generally, host materials with lower phonon energy and excellent stability will benefit the performances of optical temperature sensors. Hosts with lower phonon energy will show higher luminescent efficiencies, which will result in higher measurement accuracy [5,15]. The

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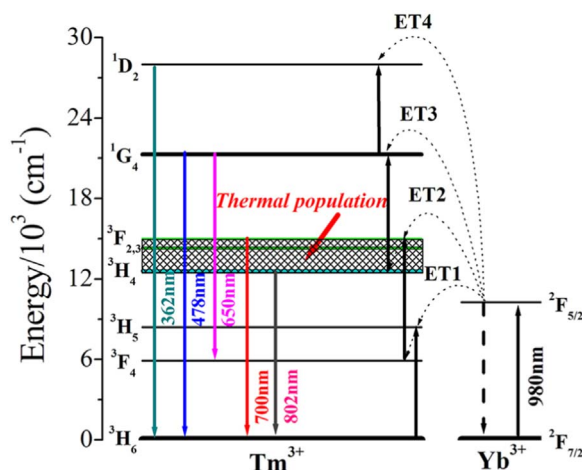


Fig. 1. Energy level diagrams of  $\text{Tm}^{3+}$  and  $\text{Yb}^{3+}$  ions as well as the possible up-conversion mechanisms and temperature detector mechanism.

excellent stability of hosts makes them can be used in high temperature and other special environment [23]. Transparent oxyfluoride glass ceramics (GC) are composite materials that containing fluoride nanocrystals phase and oxide glass phase [24–26]. These materials are generally achieved by controlling crystallization process of precursor glass with specially designed chemical compositions [26–28]. They combine the advantages of good stability of oxide glass and low phonon energy ( $\sim 400 \text{ cm}^{-1}$ ) environment of fluoride nanocrystals, which can be appropriate matrices for optical temperature sensors [5,14,16].

Tetragonal  $\text{Sr}_2\text{YF}_7$  is an important matrix for  $\text{RE}^{3+}$  ions to fabricate up-conversion or downshifting phosphors because of its low phonon energy, similar ionic radius and identical coordination of  $\text{Y}^{3+}$  and  $\text{RE}^{3+}$  ions [29–32]. Recently,  $\text{RE}^{3+}$  doped  $\text{Sr}_2\text{YF}_7$  nanoparticles have attracted much attention [29–31]. Chen et al. reported the syntheses of monodisperse and size-controllable  $\text{Sr}_2\text{YF}_7:\text{RE}^{3+}$  nanoparticles and investigated the up-conversion properties. Their results revealed great potential of  $\text{Sr}_2\text{YF}_7:\text{RE}^{3+}$  nanoprobles in practical bioassays for cancer diagnosis [31]. At the same time, tetragonal  $\text{Sr}_2\text{YbF}_7:\text{Er}^{3+}$  GC present excellent structure, up-conversion and temperature sensing properties [5]. Therefore, we believe that the tetragonal  $\text{Sr}_2\text{YF}_7$  GC doped with  $\text{Tm}^{3+}$  will show excellent temperature sensing performances.

In this work,  $\text{Tm}^{3+}/\text{Yb}^{3+}$  co-doped transparent  $\text{Sr}_2\text{YF}_7$  GC were successfully manufactured. Under 980 nm laser excitation, the emission intensity of  $\text{Tm}^{3+}$  was highly enhanced after heat-treatment. Temperature-dependent up-conversion from  ${}^3\text{F}_{2,3}$  and  ${}^1\text{G}_4$  levels of  $\text{Tm}^{3+}$  were systemically investigated to explore their potential application in temperature detecting. The theoretical max relative sensitivity  $S_{R-\text{max}}$  of  $1.16\% \text{ K}^{-1}$  is achieved at 428 K and the absolute sensitivity  $S_A$  keeps increasing with temperature. All results show that  $\text{Sr}_2\text{YF}_7:\text{Tm}^{3+}/\text{Yb}^{3+}$  GC have greatly possible application in optical temperature sensors.

## 2. Experimental

Glass samples with nominal composition  $70\text{SiO}_2\text{-}7\text{Al}_2\text{O}_3\text{-}16\text{SrF}_2\text{-}7\text{YF}_3\text{-}0.5\text{YbF}_3\text{-}0.1\text{TmF}_3$  (in mol%) were prepared through high temperature melt-quenching route.  $\text{SiO}_2$ ,  $\text{Al}_2\text{O}_3$ ,  $\text{SrF}_2$  (A.R., all from Sinopharm Chemical Reagent Co., Ltd., Shanghai, China) and high purity  $\text{YF}_3$ ,  $\text{YbF}_3$ ,  $\text{TmF}_3$  (99.99%, from AnSheng Inorganic Materials Co., Ltd., Ganzhou, Jiangxi, China) were used as starting materials. The well ground stoichiometric chemicals were put into an alumina crucible and melted at  $1550 \text{ }^\circ\text{C}$  for 1 h in air atmosphere. The melt was poured onto a  $300 \text{ }^\circ\text{C}$  preheated stainless-steel plate and then pressed quickly by another plate to get solid samples. After annealed at  $450 \text{ }^\circ\text{C}$  for 3 h to release internal stress, precursor glass (labeled as PG) were formed. Subsequently, PG was heat-treated for 2 h at  $750 \text{ }^\circ\text{C}$  to elaborate

transparent glass ceramics, which was labeled as GC. Finally, the obtained specimens were cut and polished to plates of about  $15 \times 10 \times 1.8 \text{ mm}^3$  in size for further measurements.

X-ray diffraction (XRD) patterns were performed on a Philips X'Pert PRO SUPER X-ray diffraction apparatus (Almelo, the Netherlands) with CuK $\alpha$  radiation ( $\lambda = 0.154056 \text{ nm}$ ) over the angular range  $10^\circ \leq 2\theta \leq 80^\circ$  in a step size of  $0.0167^\circ$ . PG and GC samples were ground into powder for XRD measurement. Transmittance spectra were measured on a Hitachi U-3900 Ultraviolet-Visible (UV-vis) spectrophotometer (Tokyo, Japan). The microstructure of GC sample was analyzed by a JOEL JEM-2010 transmission electron microscopy (TEM) (Tokyo, Japan). Up-conversion spectra were measured with an Edinburgh FS920 spectrofluorometer (Livingston, UK) with a 980 nm laser as excitation source. The spot size of laser was about  $1 \times 6 \text{ mm}^2$ . And the pump power was adjusted through neutral density filters. Lifetime measurements were recorded on an Edinburgh FS5 spectrofluorometer equipped with a tunable PSU-111-LED 980 nm pulse laser as the excitation source. Temperature of GC sample fixed on a copper post was controlled over the range of 303–663 K by a temperature controller (FOTEK MT48-V-E, Taibei, Taiwan) with a type-K thermocouple and a heating tube.

## 3. Results and discussion

### 3.1. Structural analyses of samples

The XRD patterns of PG and GC samples are shown in Fig. 2(a). Obviously, PG exhibits a very weak diffraction peak over diffuse humps and the diffuse humps are attributed to amorphous glass phase. Several intense diffraction peaks are observed in XRD patterns of GC sample. These diffraction peaks are corresponding to tetragonal  $\text{Sr}_2\text{YF}_7$  nanocrystals (JCPDS no. 53-0675), which indicates the formation of  $\text{Sr}_2\text{YF}_7$  nanocrystals in GC sample. And the doping of  $\text{RE}^{3+}$  ions does not change the lattice structure of  $\text{Sr}_2\text{YF}_7$  nanocrystals due to their similar ionic radius and identical valence state. The mean size of  $\text{Sr}_2\text{YF}_7$  nanocrystals in GC can be estimated by Scherer's equation [14],

$$D = k\lambda/\beta\cos\theta \quad (1)$$

where  $k = 0.89$ ,  $\lambda (= 0.154056 \text{ nm})$  represents the wavelength of CuK $\alpha$  radiation,  $\theta$  is Bragg angle and  $\beta$  represents the corrected half-width of diffraction peak. The mean size of  $\text{Sr}_2\text{YF}_7$  nanocrystals in GC is estimated to be 16 nm.

The transmittance of GC is dependent on the size of nanocrystals. Generally, the smaller size of nanocrystals in GC, the higher transparency of GC is [33]. The size of  $\text{Sr}_2\text{YF}_7$  nanocrystals in present GC is about 16 nm, which is much smaller than the wavelength of VIS light. Therefore, PG and GC samples both hold a high transparency up to 80% in VIS light region, as shown in Fig. 2(b). There are two evident absorption peaks located at 680 and 790 nm, which are corresponding to the transitions of  $\text{Tm}^{3+}$  ions from  ${}^3\text{H}_6$  ground state to  ${}^3\text{F}_{2,3}$  and  ${}^3\text{H}_4$  excited states, respectively [34].

The TEM and HRTEM images of GC sample are displayed in Fig. 2(c) and (d), respectively. TEM bright-field micrograph clearly shows that 10–20 nm sized nanocrystals homogeneously distributed in the glass matrix, which is in agreement with the result estimated by Scherer's equation. Selected area electron diffraction (SAED) patterns (inset of Fig. 2(c)) prove that the obtained GC is composite material that containing polycrystalline phase and glass phase. HRTEM image displays well-defined lattice fringes of  $\text{Sr}_2\text{YF}_7$  nanocrystals. The associated interplanar distance  $d$  value calculated through two separated crystal planes is about 0.333 nm, which can be indexed to the (302) crystal plane of tetragonal  $\text{Sr}_2\text{YF}_7$  ( $d_{(302)} = 0.330 \text{ nm}$ ).

### 3.2. Up-conversion luminescence and mechanism

Fig. 3(a) exhibits the up-conversion spectra ( $\lambda_{\text{ex}} = 980 \text{ nm}$ ) of PG

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