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# Peculiarities of cascade photon emission and energy storage in $M_{1-x}Pr_xF_{2+x}$ (M=Ca, Sr, Ba, $x \approx 0.35$ ) crystals

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### ABSTRACT

Peculiarities of cascade photon emission (CPE) and energy storage in  $M_{1-x}Pr_xF_{2+x}$  (M=Ca, Sr, Ba,  $x \approx 0.35$ ) crystals were studied. The investigation of lattice parameters revealed that these solid solutions belong to the fluorite structure type with the lattice constant noticeably different from that of MF<sub>2</sub> crystals. Absorption, emission and excitation spectra of  $M_{0.65}Pr_{0.35}F_{2.35}$  were measured at LHeT and RT. As it turned out the typical for  $4f^2 \rightarrow 4f^2$  transition in  $Pr^{3+}$  emission lines are broadened as compared with the PrF<sub>3</sub> crystal. The analysis of the excitation spectra broadening does not allow bringing out the type of the superlattice, which is inherent to the material, but it indicates clearly the simultaneous presence of different types of the Pr centers in mixed crystals. Yet another specific feature is the higher radiation sensitivity of these fluorides relatively PrF<sub>3</sub>, MF<sub>2</sub> and Pr-doped MF<sub>2</sub> crystals. Coloration efficiency enhances in direction Ca  $\rightarrow$  Sr  $\rightarrow$  Ba, and the positions of induced absorption band depend on composition of the solid solution. Colorization, thermo-stimulated luminescence and afterglow of the  $M_{1-x}Pr_xF_{2+x}$  crystals denote high radiation sensitivity as compared with  $M_{1-x}Ce_xF_{2+x}$ .

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# 1. Introduction

Fluorite-structured MF<sub>2</sub> (M=Ca, Sr, Ba) crystals are transparent in a wide spectral range and being doped by rare earths (RE) are applied as lasers and radiation detectors. Surplus charge of RE<sup>3+</sup> ions is compensated by interstitial fluorine ions. Contrary to MF<sub>2</sub>:RE, in highly concentrated M<sub>1-x</sub>RE<sub>x</sub>F<sub>2+x</sub> solid solutions RE<sup>3+</sup> ions are not distributed statistically and form nano-sized inclusions with a certain atomic composition [1]. It is supposed that the clusters and fluoride phases in nonstoichiometric crystals form nanostructures. EPR and neutron diffraction studies of M<sub>1-x</sub>RE<sub>x</sub>F<sub>2+x</sub> systems have shown the presence of extensive disorder in the form of vacancies in the anion sublattice and a large excess of interstitial atoms over dopant trivalent cations [2,3].

Fluorite-structured solid solutions are promising superionic conductors for medium temperatures [4,5]. Large regions of heterovalent solid solutions (up to x=0.5) and a high concentration of mobile charge compensating defects result in a high ionic conductivity already near the room temperature.

Generally, highly RE concentrated fluoride solid solutions represent a new class of materials promising for the design of wide-gap converters of VUV and ionizing radiation to be used for medical, scientific and industrial applications. For the first time these attempts produced a positive result for Ca<sub>0.65</sub>Eu<sub>0.35</sub>F<sub>2.35</sub> crystals [6]. As it was found this compound has good prospects to be applied as a red phosphor and display material due to bright trivalent Eu<sup>3+</sup> ion luminescence under UV, VUV, X- and  $\gamma$ -excitation.

Recently, Pr-doped crystals have become an object of new scintillation media research. The cascade photon emission (CPE) of Pr-doped fluorides is yet another argument in favour of the research, since this phenomenon enables to produce two photons per one absorbed quantum. CPE is typical of broad gap dielectrics, allowing initiating two low-energy photons from an excited state, in other words, making it possible to increase the light yield of Pr-doped materials [7,8]. The first results concerning the luminescence characteristics of  $M_{0.65}Pr_{0.35}F_{2.35}$ , as compared with MF<sub>2</sub> and PrF<sub>3</sub> crystals, were presented at SCINT'07 Conference [9].

The goal of this research was to study the effects of concentration and structure state of praseodymium ions on peculiarities of the luminescence, as well as on implementation of the photon cascade emission and energy storage in  $M_{1-x}$   $Pr_xF_{2+x}$  crystals.

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# 2. Experimental

Pure and Pr-doped MF<sub>2</sub> crystals as well as PrF<sub>3</sub> and  $M_{1-x}Pr_xF_{2+x}$  (M=Ca, Sr, Ba,  $x \approx 0.35$ ) were grown by Czochralski technique in CF<sub>4</sub> atmosphere. Lattice parameters were determined by the X-ray diffraction method (XRD) at Oxford Diffraction Xcalibur-3 single crystal diffractometer (MoK $\alpha$  radiation, graphite monochromator, Sapphire 3 CCD detector,  $7^{\circ} \le 2\theta \le 87^{\circ}$ ). The phase structure tests with material powder were performed using the Siemens D500 powder diffractometer (CuK $\alpha$ , graphite monochromator, scan step of 0.01° within 20°  $\le 2\theta \le 155^{\circ}$ , a dwell time 40 s) (see details in Ref. [6]). Luminescence spectra were recorded using time-resolved FLS 920 spectrometer (Edinburgh Instruments) and synchrotron radiation at the SUPERLUMI station, HASYLAB at DESY [10]. Absorption was measured using SPECORD 40 spectrophotometer. Irradiation was performed by an X-ray tube (160 kV, 10 Gy/min).

## 3. Results

The investigations of lattice parameters by the XRD method have confirmed that  $M_{1-x}Pr_xF_{2+x}$  (M=Ca, Sr, Ba,  $x \approx 0.35$ ) crystals belong to the fluorite structure type (the fcc lattice, space group Fm3m) with the lattice constant (*a*) noticeably differ from that of MF<sub>2</sub> (Table 1). A partial deficiency in the fluorine positions and excessive interstitial fluorine ions (F<sub>i</sub><sup>-</sup>) were revealed. This feature most likely determined the change in the crystal lattice ( $\Delta a$ ) of MF<sub>2</sub> and  $M_{1-x}Pr_xF_{2+x}$ . The similar arrangement was found earlier in  $M_{1-x}RE_xF_{2+x}$  structures containing different RE<sup>3+</sup> ions in the positions of divalent alkali earth ones [1]. It should be noted that any clusters and phases were not revealed in studied compounds.

A priori, the optical properties of mixed crystals would be similar to the crystal-forming compound, i.e. MF<sub>2</sub> or PrF<sub>3</sub>. From this standpoint, all of the absorption and luminescence characteristics are compared with similar data of the base compounds.

Absorption spectra of PrF<sub>3</sub> and  $M_{0.65}Pr_{0.35}F_{2.35}$  (M=Ca, Sr, Ba) crystals are shown in Fig. 1. The lines belonging to  ${}^{3}H_{4} \rightarrow {}^{3}P_{J}$  (J=0,1,2) transitions in Pr<sup>3+</sup> ions are distinctly revealed in all solid solutions.

The synchrotron radiation excited emission spectra of  $M_{0.65}Pr_{0.35}F_{2.35}$  crystals measured at 8 K are presented in Fig. 2. Intensity of typical  $Pr^{3+}$  ion emission in the 220–280 nm range (5d 4f bands and  ${}^{1}S_{0} \rightarrow {}^{3}H_{6}$ ,  ${}^{3}F_{4}$ ,  ${}^{1}G^{4}$ ,  ${}^{1}D_{2}$  lines) decreases in a row  $Ca \rightarrow Sr \rightarrow Ba$ . It is remarkable that the energy gap ( $E_{g}$ ) of  $CaF_{2}$ ,  $SrF_{2}$  and  $BaF_{2}$  is close to ~ 12.1, 11.2 and 10.57 eV, respectively. Long wave  $4f^{2} \rightarrow 4f^{2}$  luminescence at 482 nm ( ${}^{3}P_{0} \rightarrow {}^{3}H_{4}$ ), 553 nm ( ${}^{3}P_{0} \rightarrow {}^{3}H_{5}$ ), ~ 609 nm ( ${}^{3}P_{0} \rightarrow {}^{3}H_{6}$ ), 638 nm ( ${}^{3}P_{0} \rightarrow {}^{3}F_{2}$ ), 721 nm ( ${}^{3}P_{0} \rightarrow {}^{3}F_{4}$ ) reveals an opposite tendency. The set of overlapped bands between 5.0 and 9.0 eV in excitation spectra of 5d  $4f \rightarrow 4f^{2}$  and  $4f^{2} \rightarrow 4f^{2}$  emissions are assigned to  $4f \rightarrow 4f$  5d transitions in  $Pr^{3+}$  ion (Fig. 3). At lower energies, E < 5.0 eV only narrow  $4f^{2} \rightarrow 4f^{2}$  lines typical of  $Pr^{3+}$  ions are observed (not presented in Fig. 3). The weak excitation line at 5.8 eV, which is well resolved

# Table 1

Lattice constant (a) of MF<sub>2</sub>, M<sub>0.65</sub>Pr<sub>0.35</sub>F<sub>2.35</sub> and PrF<sub>3</sub> crystals.

Crystal	a (Å)	$\Delta a$ (Å)
CaF <sub>2</sub> Ca <sub>0.65</sub> Pr <sub>0.35</sub> F <sub>2.35</sub> SrF <sub>2</sub> SroceProceFactor	5.46305 5.61359(4) 5.79960 5.81578(2)	+0.15055
BaF <sub>2</sub> Ba <sub>0.65</sub> Pr <sub>0.35</sub> F <sub>2.35</sub> PrF <sub>3</sub>	6.20000 6.03744(6) 7.2390	-0.16255



**Fig. 1.** Absorption spectra of  $PrF_3$  and  $Me_{1-x}Pr_xF_{2+x}$  crystals.

in Ca<sub>0.65</sub>Pr<sub>0.35</sub>F<sub>2.35</sub> crystals corresponds to  ${}^{3}H_{4} \rightarrow {}^{1}S_{0}$  transition. While excitation of Ca<sub>0.65</sub>Pr<sub>0.35</sub>F<sub>2.35</sub> in the high-energy region (8.0 eV) gives rise to emissions mainly from  $4f^{2} \rightarrow 4f^{2}$  transitions, stimulation at low-energy band (5.6 eV) results in strong 5d  $4f \rightarrow 4f^{2}$  emission as well as long wave  $4f^{2} \rightarrow 4f^{2}$  luminescence. This fact points to the presence of several types of emission centers with different 5d 4f levels location. In Sr<sub>0.65</sub>Pr<sub>0.35</sub>F<sub>2.35</sub> crystals the redistribution between 5d  $4f \rightarrow 4f^{2}$  and  $4f^{2} \rightarrow 4f^{2}$  transitions upon selective excitation is less pronounced.

The weak broad excitation bands in the range 9.0–12.0 eV may be connected with CT transitions to  $Pr^{3+}$  ion [11,12] and perturbed excitonic states near  $Pr^{3+}$ . Light yield is low at high excitation energies ( $E > E_g$ ) due to large energy losses in the process of energy transfer.

X-ray luminescence spectra of  $M_{0.65}$  Pr<sub>0.35</sub> F<sub>2.35</sub> crystals measured at 77 K are similar to photoluminescence spectra under the excitation within the range 6.0–12.0 eV. This fact points to energy transfer from the host to the emission centers.

Typical decay curves under photo- and X-ray excitation for Ca<sub>0.65</sub> Pr<sub>0.35</sub> F<sub>2.35</sub> crystal is shown in Fig. 4. UV luminescence with the decay component of 14 ns corresponded to 5d 4f transitions in Pr<sup>3+</sup> whereas for emission originating from <sup>1</sup>S<sub>0</sub> level the decay time  $\tau_2 \ge 160$  ns is typical. Slow component (> 2 µs) was observed for long-wave emission (482–608 nm) connected with (<sup>3</sup>P<sub>0</sub> $\rightarrow$ <sup>3</sup>H<sub>4.5,6</sub>) transitions. In addition, the ultrafast decay component (2 <  $\tau$  < 5 ns) was found which may be due to the energy transfer between 5d 4f and <sup>1</sup>S<sub>0</sub> levels of different emission centers.

Luminescence of mixed compounds under photo- and X-ray excitation decreases with temperature rise from 8 to 297 K. At RT 5d  $4f \rightarrow 4f^2$  and  ${}^{1}S_{0} \rightarrow {}^{3}H_6$ ,  ${}^{3}F_4$ ,  ${}^{1}G^4$ ,  ${}^{1}D_2$  transitions are dominant while emission originating from  ${}^{3}P_0$  level is quenched.

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