

MICROELECTRONIC ENGINEERING

Microelectronic Engineering 84 (2007) 2460-2465

www.elsevier.com/locate/mee

# Materials characterization of $WN_xC_y$ , $WN_x$ and $WC_x$ films for advanced barriers

H. Volders <sup>a,\*</sup>, Z. Tökei <sup>a</sup>, H. Bender <sup>a</sup>, B. Brijs <sup>a</sup>, R. Caluwaerts <sup>a</sup>, L. Carbonell <sup>a</sup>, T. Conard <sup>a</sup>, C. Drijbooms <sup>a</sup>, A. Franquet <sup>a</sup>, S. Garaud <sup>a</sup>, I. Hoflijk <sup>a</sup>, A. Moussa <sup>a</sup>, F. Sinapi <sup>a</sup>, Y. Travaly <sup>a</sup>, D. Vanhaeren <sup>a</sup>, G. Vereecke <sup>a</sup>, C. Zhao <sup>a</sup>, W.-M. Li <sup>b</sup>, H. Sprey <sup>c</sup>, A.M. Jonas <sup>d</sup>

<sup>a</sup> IMEC vzw, Kapeldreef 75, 3001 Leuven, Belgium
<sup>b</sup> ASM Microchemistry Oy, Väinö Auerin katu 12 A, 00560 Helsinki, Finland
<sup>c</sup> ASM Belgium N.V., Kapeldreef 75, 3001 Leuven, Belgium
<sup>d</sup> Unité de Physique et de Chimie des Hauts Polymères, Université Catholique de Louvain, Belgium

Received 14 May 2007; accepted 21 May 2007 Available online 26 May 2007

#### **Abstract**

A ternary  $WN_xC_y$  system was deposited in a thermal ALD (atomic layer deposition) reactor from ASM at 300 °C in a process sequence using tungsten hexafluoride (WF<sub>6</sub>), triethyl borane (TEB) and ammonia (NH<sub>3</sub>) as precursors. The WC<sub>x</sub> layers were deposited by a novel ALD process at a process temperature of 250 °C. The WN<sub>x</sub> layers were deposited at 375 °C using bis(*tert*-butylimido)-bis-(dimethylamido)tungsten ('BuN)<sub>2</sub>(Me<sub>2</sub>N)<sub>2</sub>W (imido–amido) and NH<sub>3</sub> as precursors. WN<sub>x</sub> grows faster on plasma enhanced chemical vapor deposition (PECVD) oxide than WC<sub>x</sub> does on chemical oxide. WN<sub>x</sub>C<sub>y</sub> grows better on PECVD oxide than on thermal oxide, which is opposite of what is seen for WN<sub>x</sub>. In the case of the ternary WN<sub>x</sub>C<sub>y</sub> system, the scalability towards thinner layers and galvanic corrosion behavior are disadvantages for the incorporation of the layer into Cu interconnects. ALD WC<sub>x</sub> based barriers have a low resistivity, but galvanic corrosion in a model slurry solution of 15% peroxide (H<sub>2</sub>O<sub>2</sub>) is a potential problem. Higher resistivity values are determined for the binary WN<sub>x</sub> layers. WN<sub>x</sub> shows a constant composition and density throughout the layer.

© 2007 Published by Elsevier B.V.

Keywords: Barrier; Atomic layer deposition; ALD;  $WN_xC_y$ ;  $WN_x$ ;  $WC_x$ 

#### 1. Introduction

The deposition of a barrier layer for future technologies in Cu interconnects is inextricably linked to scaling of the barrier thickness. Scaling on the other hand is difficult to maintain with the traditional deposition techniques such as physical vapor deposition (PVD). Traditional PVD cannot fulfill the stringent demands for future barrier layers such as the deposition of a thin (<5 nm), conformal layer with excellent step coverage and minimal overhang. ALD

is a possible alternative to the traditional PVD technique since it promises to satisfy these requirements. Pure metal ALD however is difficult, so therefore binary and ternary systems attract a lot of attention, Ref. [1].

In search of a new possible barrier candidate for the 32 nm technology node and beyond three different ALD processes, as developed by ASM Microchemistry, are compared: a ternary  $WN_xC_y$  system and two binary  $(WN_x, WC_x)$  systems. The processes are compared in terms of growth characteristics, film properties (e.g. homogeneity, crystallinity, roughness, density and sheet resistance) as well as galvanic corrosion behavior in a model solution of 15%  $H_2O_2$  in distilled (DI) water.

<sup>\*</sup> Corresponding author. Tel.: +32 16 28 14 96; fax: +32 16 28 12 14. E-mail address: Henny.Volders@imec.be (H. Volders).

#### 2. Experimental details

The deposition of the ternary system  $WN_xC_y$  was performed in a thermal ALD reactor from ASM at 300 °C in a process sequence using WF<sub>6</sub>, TEB and NH<sub>3</sub> as precursors. The WC<sub>x</sub> layers were deposited by a novel ALD process at a process temperature of 250 °C. Cycle numbers varied between 27 cycles corresponding to nominally 2 nm WC<sub>x</sub> and 267 cycles corresponding to nominally 20 nm WC<sub>x</sub> on a substrate of 1.1 nm chemical oxide. The WN<sub>x</sub> layers were deposited at 375 °C using imido—amido and NH<sub>3</sub> as precursors Refs. [2,3]. The binary layer was deposited on to either 20 nm thermal oxide or 100 nm PECVD oxide with cycle numbers ranging between 30 cycles for 1 nm nominally and 1000 cycles for 20 nm nominally.

The growth characteristics of the different films were analyzed by Rutherford back scattering (RBS). The RBSmeasurements were executed on an RBS-400 system, which is installed around a 2 MV tandem-accelerator. The film resistivity was calculated using four point probe sheet resistance measurements and film thickness. X-ray photoelectron spectroscopy (XPS) was deployed to determine the layer composition at the surface and inside the layer on a Thermo VG scientific Theta300. The chemical bonds that are formed between the different elements were also characterized. In order to determine the density of the barrier layers, samples were analyzed with a high-resolution specular X-ray reflectometer (XRR) using a  $\Theta/2\Theta$  configuration with a rotating copper anode as the radiation source and a Siemens D5000 two-circle goniometer. X-rays of 1.5418 Å wavelength (Cu Kα) were selected by a graphite secondary monochromator, complemented with electronic discrimination (scintillation counter). The X-ray beam was collimated by a set of adjustable slits with micrometer precision. The reflected intensity was measured as a function of detector angle  $(\Theta)$  and subsequently plotted versus kz0, the momentum transferred perpendicularly to the sample, which is defined by  $(2\pi/\lambda)\sin\Theta$ . By means of Auger electron spectroscopy (AES), the surface contamination and film purity were analyzed on a Thermo VG scientific Microlab 350. Surface smoothness of the deposited films was evaluated with atomic force microscopy (AFM) on a Nanoscope IV Dimension 3100. Grazing angle X-ray diffraction (XRD) analysis using Cu Kα radiation  $(\lambda = 1.54 \text{ A})$  was conducted to determine the crystal structure of the nominally 20 nm thick WN<sub>x</sub> sample. The nominally 20 nm thick WCx layer was analyzed by high temperature XRD with a temperature range between room temperature and 1100 °C. A sequence of XRD spectra were taken at temperatures with an interval of 100 °C and a ramp rate of 50 °C/min. The high temperature X-ray diffraction set-up used is a 3003 TT diffractometer (Seifert, Ahrensburg, Germany) equipped with a parabolic multilayer mirror, which provides a wide parallel incident beam and a furnace in vacuum for heating the sample to the temperature of interest. Cu Kα radiation is used as the X-ray source. Static etch rates of the different barrier layers and of Cu were determined at room temperature in a controlled solution of 15% H<sub>2</sub>O<sub>2</sub> in DI water by means of four point probe sheet resistance measurements. In the same medium the open circuit potential (OCP) was measured for the three materials with galvanic series measurements under static conditions. To model the galvanic corrosion behavior in the controlled solution, galvanic currents for the Cu/W-containing barrier couples (area ratio 7:1) were measured.

#### 3. Results

On 100 nm PECVD oxide,  $WN_x$  grows well after an incubation period of  $\pm 30$  cycles but the growth per cycle (Figs. 1 and 2) is even faster on thermal oxide as derived from RBS measurements. On chemical oxide,  $WC_x$  has an incubation period of around 20 cycles after which the layer grows in a linear mode (Fig. 3).  $WN_x$  grows faster on PECVD oxide than  $WC_x$  does on chemical oxide.  $WN_xC_y$  grows better on PECVD oxide than on thermal oxide which is opposite of what is seen for  $WN_x$ , Ref. [4].

The bulk resistivity among the three materials differs, both  $WN_xC_y$  and  $WC_x$  have low resistivity of about 300–400  $\mu\Omega$  cm beneficial for barrier application, but the bulk resistivity of  $WN_x$  goes up to 4000  $\mu\Omega$  cm.

The layer composition of the different ALD layers was analyzed with in depth XPS. A constant composition throughout a layer is important when scaling down the barrier thickness, because changes in composition imply changes in density and thereby changes in barrier properties. Surface composition of a layer is important because good adhesion to Cu has to be ensured however it falls out of the scope of this paper to investigate the adhesion between the barrier layer and the dielectricum on one hand or the Cu on the other hand.

XPS gives a ratio of W:N:O:C of 60:20:10:10 at.% throughout a very homogeneous layer of nominally  $20 \text{ nm WN}_x$  layer on 20 nm thermal oxide (Fig. 4). The spectra recorded at half the sputter time, show that the main bonds are W-N bonds together with some minor oxygen

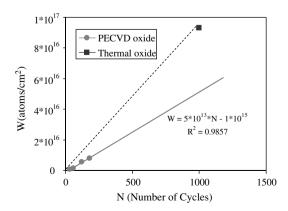


Fig. 1. Growth curve for  $WN_x$  on  $100\,\mathrm{nm}$  PECVD oxide and  $20\,\mathrm{nm}$  thermal oxide.

### Download English Version:

## https://daneshyari.com/en/article/540766

Download Persian Version:

https://daneshyari.com/article/540766

<u>Daneshyari.com</u>