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# Study on the structural properties and relaxation mechanisms in $\text{LiRb}_{1-x}(\text{NH}_4)_x\text{SO}_4$ (x=0, 0.5, and 1) mixed crystals by <sup>1</sup>H, <sup>7</sup>Li, and <sup>87</sup>Rb nuclear magnetic resonance

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#### ABSTRACT

LiRb<sub>1-x</sub>(NH<sub>4</sub>)<sub>x</sub>SO<sub>4</sub> (*x*=0, 0.5, and 1) mixed crystals were grown using the slow evaporation method, and their lattice constants were determined by X-ray diffraction. The effects of replacing ammonium ions with rubidium on the phase transitions of and dynamics of LiRb<sub>1-x</sub>(NH<sub>4</sub>)<sub>x</sub>SO<sub>4</sub> mixed crystals were studied. We investigated the structural properties and phase transition temperatures of the LiRb<sub>1-x</sub>(NH<sub>4</sub>)<sub>x</sub>SO<sub>4</sub> mixed crystals by performing <sup>1</sup>H, <sup>7</sup>Li, and <sup>87</sup>Rb NMR relaxation and DSC. When a fraction of rubidium ions in the pure LiRbSO<sub>4</sub> crystal is partially replaced by ammonium ions, we compare the structural properties for LiRb<sub>0.5</sub>(NH<sub>4</sub>)<sub>0.5</sub>SO<sub>4</sub> and LiNH<sub>4</sub>SO<sub>4</sub> single crystals. The <sup>1</sup>H, <sup>7</sup>Li, and <sup>87</sup>Rb NMR relaxation mechanisms of LiRb<sub>0.5</sub>(NH<sub>4</sub>)<sub>0.5</sub>SO<sub>4</sub> are different from those of pure LiRbSO<sub>4</sub> and LiNH<sub>4</sub>SO<sub>4</sub>; the sequence of the structural phase transitions of LiRb<sub>0.5</sub>(NH<sub>4</sub>)<sub>0.5</sub>SO<sub>4</sub> is different, which might be related to the dynamics and orientation of the sulfate tetrahedral groups in the crystal structure.

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# 1. Introduction

A large number of studies have examined the physical properties of lithium rubidium sulfate, LiRbSO<sub>4</sub>, and lithium ammonium sulfate, LiNH<sub>4</sub>SO<sub>4</sub>; these materials are interesting because of their ferroelastic and ferroelectric properties. LiRbSO<sub>4</sub> undergoes successive phase transitions at 477 K ( $=T_{C1}$ ), 475 K ( $=T_{C2}$ ), 458 K  $(=T_{C3})$ , and 439 K  $(=T_{C4})$  [1,2]. The successive phases in order of decreasing temperature are referred to as I, II, III, IV, and V. Phases I and V are paraelectric [2]. Phases II and III are ferroelectric and antiferroelectric, respectively [3]. Further, phase III has been found to exhibit both antiferroelectric and ferroelastic characteristics. Microscopic observations have shown that phases IV and V are monoclinic and that the crystal system is orthorhombic above 458 K [4]. Ferroelectricity has been found in phase IV between 439 and 458 K [2,5]. The phase transitions of LiRbSO<sub>4</sub> can be approximately described as arising from changes of the orientations of the tetrahedral SO<sub>4</sub> groups. LiNH<sub>4</sub>SO<sub>4</sub> undergoes several structural phase transitions with changes in temperature. A large number of studies have examined the properties of the different phases of LiNH<sub>4</sub>SO<sub>4</sub>, with the ferroelectric behavior of the room temperature phase attracting particular attention. LiNH<sub>4</sub>SO<sub>4</sub> is an extremely interesting material that undergoes four phase

transitions, at 459.5 K ( $=T_{C1}$ ), 284.5 K ( $=T_{C2}$ ), 133 K ( $=T_{C3}$ ), and 27 K ( $=T_{C4}$ ) [6–9]. These four transitions of LiNH<sub>4</sub>SO<sub>4</sub> single crystals separate five phases denoted by I, II, III, IV, and V in order of decreasing temperature. The symmetry of the high temperature paraelectric phase above 459 K has been identified as orthorhombic. At room temperature the crystal displays orthorhombic symmetry, which differs from the space group of phase I, and has ferroelectric characteristics. The structure below 284.5 K has been described as monoclinic with both antiferroelectric and ferroelectric characteristics. Below 27 K, the space group is C1c<sub>1</sub> [10]. Thus, the crystal structures of LiRbSO<sub>4</sub> and LiNH<sub>4</sub>SO<sub>4</sub> in phase I are isomorphous and described by the space group *Pmcn*, whereas the crystal structures of all the remaining phases show no mutual correspondence. LiRbSO4 is pseudoisostructural to LiNH<sub>4</sub>SO<sub>4</sub> and the main difference between the two structures is in the orientations of the LiO<sub>4</sub> and SO<sub>4</sub> tetrahedra [11], as shown in Fig. 1.

Kawamura et al. [11] has studied the *x*-*T* phase diagrams of  $\text{LiRb}_{1-x}(\text{NH}_4)_x\text{SO}_4$  mixed crystals in detail using X-ray diffraction. They investigated the influence of low concentrations of ammonium on the phase transitions in  $\text{LiRb}_{1-x}(\text{NH}_4)_x\text{SO}_4$  ( $0.0 \le x \le 0.2$ ) and the influence of low concentrations of rubidium on the phase transitions in  $\text{LiRb}_{1-x}(\text{NH}_4)_x\text{SO}_4$  ( $0.9 \le x \le 1.2$ ). The effects of replacing an ammonium ion with rubidium on the *x*-*T* phase diagram in the concentration range  $0.5 \le x \le 1.0$  have been investigated using X-ray and neutron powder diffraction over a wide temperature range by Smirnov et al. [12]. It was shown that

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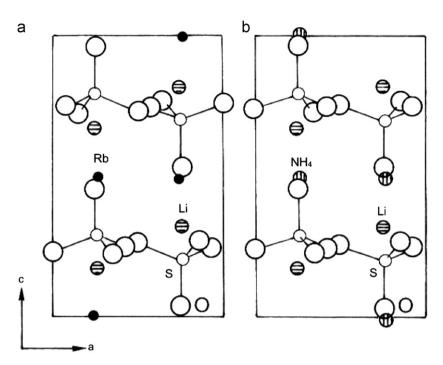


Fig. 1. Projection of the structures of (a) LiRbSO<sub>4</sub> and (b) LiNH<sub>4</sub>SO<sub>4</sub> along [0 1 0] at room temperature.

a decrease in the ammonium concentration is accompanied by an increase in the temperature of the II–I phase transition and by the stabilization of phase II up to low temperatures.

In the present study,  $LiRb_{1-x}(NH_4)_xSO_4$  (x=0, 0.5, and 1) mixed crystals were grown using the slow evaporation of aqueous solutions. The structures and the phase transition temperatures of these crystals were determined using X-ray diffraction and differential scanning calorimetry (DSC), respectively. Our aim was to study the phase transitions and the structural dynamics of  $LiRb_{1-x}(NH_4)_xSO_4$  mixed crystals. Further, this study examined the effects of the random presence of electronically different cations of similar size. To probe the structural properties of the three single crystals, LiRbSO<sub>4</sub>, LiRb<sub>0.5</sub>(NH<sub>4</sub>)<sub>0.5</sub>SO<sub>4</sub>, and LiNH<sub>4</sub>SO<sub>4</sub>, the measurement of the <sup>1</sup>H, <sup>7</sup>Li, and <sup>87</sup>Rb relaxation times was preferred, because these relaxation times are likely be very sensitive to changes in the symmetry of these crystals. The nuclear magnetic resonance (NMR) spectrum and spin-lattice relaxation times,  $T_1$ , for <sup>1</sup>H, <sup>7</sup>Li, and <sup>87</sup>Rb in LiRbSO<sub>4</sub>, LiRb<sub>0.5</sub>(NH<sub>4</sub>)<sub>0.5</sub>SO<sub>4</sub>, and LiNH<sub>4</sub>SO<sub>4</sub> single crystals were obtained. We use these results to compare the structural properties and relaxation mechanisms of the three single crystals.

## 2. Crystal structures

LiRbSO<sub>4</sub> has a monoclinic structure with four molecular formula units per unit cell. The space group is  $P2_1/n$  and the lattice parameters of the monoclinic cell are a=5.288 Å, b=9.105 Å, c=8.731 Å, and  $\gamma=90.09^{\circ}$  at room temperature [13]. The projection of the structure of LiRbSO<sub>4</sub> along [0 1 0] is shown in Fig. 1(a). The Rb<sup>+</sup> ions are shown as dark circles. LiRbSO<sub>4</sub> has a pseudohexagonal network of six-membered rings of SO<sub>4</sub> and LiO<sub>4</sub>, and the structure of LiRbSO<sub>4</sub> is characterized by the alternate rotation in opposite directions of each tetrahedron, layer by layer. The SO<sub>4</sub><sup>2–</sup> anion is an almost regular tetrahedron, and each SO<sub>4</sub> tetrahedron shares all of its corners with distorted LiO<sub>4</sub> tetrahedra.

The structure of LiNH<sub>4</sub>SO<sub>4</sub> is orthorhombic with the space group  $P2_1cn$  and the lattice constants are a=5.280 Å, b=9.140 Å,

and c=8.786 Å in the temperature range 283 < T < 459 K (Phase II) [14–16]. This structure consists essentially of six-membered rings typical of a tridymite framework that is made up of tetrahedral LiO<sub>4</sub> and SO<sub>4</sub> groups. In each ring, three adjacent tetrahedra have their vertices pointing up and down alternately around the ring. The six-membered rings are connected along the *c*-axis by an almost linear Li–O–S bond. The layers of the six-membered rings form channels along the axis in which the NH<sub>4</sub> groups are located. The NH<sub>4</sub> groups lie approximately at the centers of the large cavities in the tetrahedral framework. The structure of LiNH<sub>4</sub>SO<sub>4</sub> is characterized by the alternate rotation in same directions of each tetrahedron, layer by layer, as shown in Fig. 1(b).

### 3. Experimental procedure

 $LiRb_{1-x}(NH_4)_xSO_4$  (x=0, 0.5, and 1) mixed crystals were obtained by slow evaporation of aqueous solutions of Li<sub>2</sub>SO<sub>4</sub>,  $Rb_2SO_4$ , and  $(NH_4)_2SO_4$  with the appropriate molar ratios at room temperature. The compositions of the resulting crystals differ from the initial compositions of the respective aqueous solutions. The compositions of the grown crystals were determined using chemical analysis. Three types of compounds were obtained: LiRbSO<sub>4</sub>, LiRb<sub>0.5</sub>(NH<sub>4</sub>)<sub>0.5</sub>SO<sub>4</sub>, and LiNH<sub>4</sub>SO<sub>4</sub>. The crystals were transparent and colorless. The structures of the Li  $Rb_{1-x}(NH_4)_xSO_4$ (x=0, 0.5, and 1) single crystals at room temperature were determined with an X-ray diffractometer system (Bruker AXS GMBH) at the Korea Basic Science Institute. The structures and the lattice constants of the LiRbSO<sub>4</sub> and LiNH<sub>4</sub>SO<sub>4</sub> crystals are consistent with previous results [13-16]. The LiRb<sub>0.5</sub>(NH<sub>4</sub>)<sub>0.5</sub>SO<sub>4</sub> crystal has an orthorhombic structure with cell parameters a=5.286 Å, b=8.755 Å, and c=9.110 Å. These results are consistent with those of Smirnov et al. [12]. The phase transition temperatures were determined using a DuPont 2010 DSC instrument. Measurements were made at heating rate of 10 K/min. The phase transition temperatures of LiRbSO<sub>4</sub> and LiNH<sub>4</sub>SO<sub>4</sub> are consistent with those previously reported [1,2,6-9], and only

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