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SUSC-20706; No of Pages 14

Surface Science xxx (2015) xxx-xxx



Contents lists available at ScienceDirect

Surface Science



journal homepage: www.elsevier.com/locate/susc

- Q1 Structural and Electronic Transformations of Pt/C, Pd@Pt(1 ML)/C and
- ² Pd@Pt(2 ML)/C Cathode Catalysts in Polymer Electrolyte Fuel Cells
- during Potential-step Operating Processes Characterized by In-situ
- ⁴ Time-resolved XAFS

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13 ARTICLE INFO

15 Available online xxxx

- 16 Keywords:
- 17 Polymer electrolyte fuel cells
- 18 Pt/C and Pd(core)-Pt(shell)/C cathode catalysts
- 19 Nanoparticle surface characterization
- 20 Potential-dependent surface restructuring
- 21 Structural kinetics
- 22 In-situ time-resolved XAFS

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1. Introduction

At Berkeley in 1964 Prof. Gabor A. Somorjai started working with 39 40 platinum in the hope of learning how and why it is so good at carrying out so many chemical surface reactions. It was first used in 1823 to 41 produce flames, instantly aiding the combustion of hydrogen in air 42 $(H_2 + \frac{1}{2}O_2 - \rightarrow H_2O)$. Today, platinum is at the heart of the catalytic 4344converter that cleans the exhaust gases from automobiles, and the metal produces high-octane gasoline from naphtha. Surface science 45 and catalysis developed by the tremendous achievements of Prof. 46 47 Somorjai on Pt single crystal surfaces and nanoparticles have been applied to the systems with more complex surfaces including electro-48 chemistry, biointerfaces corrosion, etc. To probe real catalysts more 4950fully, studies advanced from work on single crystal surfaces of metals 51to work on nanoparticles, Prof. Somorjai says this is the ongoing 52third phase of his research. The outstanding and highly suggestive

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http://dx.doi.org/10.1016/j.susc.2015.10.053 0039-6028/© 2015 Published by Elsevier B.V.

ABSTRACT

The dynamic structural and electronic transformations of Pt/C, Pd@Pt(1 ML)/C, Pd@Pt(2 ML)/C cathode catalysts 23 in polymer electrolyte fuel cells (PEFCs) during the potential-step operating processes between 0.4 and $1.4 V_{RHE}$ 24 (potential vs RHE) were characterized by in-situ (operando) time-resolved Pt L_{III}-edge quick-XAFS at 100 ms 25 time-resolution. Potential-dependent surface structures and oxidation states of Pt, Pd@Pt(1 ML) and Pd@Pt(2 26 ML) nanoparticles on carbon at 0.4 and 1.4 V_{RHE} were also analyzed by in-situ Pt L_{III}-edge and Pd K-edge 27 quick-XAFS. The Pt, Pd@Pt(1 ML) and Pd@Pt(2 ML) nanoparticle surfaces were restructured and disordered at 28 1.4 V_{RHE}, which were induced by strong Pt-O bonds as well as alloying effects. The rate constants for the changes 29 of Pt valence, CN(Pt-Pt), CN(Pt-Pd) and CN(Pt-O) (CN: coordination number) in the potential-step operating 30 processes were also determined and discussed in relation to the origin of oxygen reduction reaction (ORR) activities 31 of the Pt/C, Pd@Pt(1 ML)/C and Pd@Pt(2 ML)/C cathode catalysts. 32

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achievements of Prof. Gabor A. Somorjai have had great effect and 53 big impact on both academic and industrial researches in the world. 54

In autumn 2014 Toyota Motor Corporation launched the world's first 55 commercially available hydrogen fuel-cell vehicle, "Toyota Mirai" by 56 using Pt/C nanoparticle catalysis. Polymer electrolyte fuel cells (PEFCs) 57 (converter of the chemical reaction energy, $H_2 + \frac{1}{2}O_2 - \rightarrow H_2O$, to 58 electrochemical energy (electric power)) have the potential to provide 59 clean energy for powering vehicles, but improved performance and 60 durability are needed for wide-spread commercialization [1-15]. 61 There is still a lot of fundamental research required in order to benefit 62 from the ultimate potential of PEFCs. In particular, our understanding of 63 the chemical reactions and structural behaviors of cathode nanoparticle 64 surfaces governing the generation of electricity from PEFCs and the 65 long-term durability of cathode catalysts under the harsh PEFC conditions 66 is still incomplete [1–15]. Specifically, the active structures, reactions 67 mechanisms, and degradation mechanisms of PEFC cathode catalysts 68 in-situ under the PEFC operations are still "a black box", which must 69 be understood more thoroughly for development of next-generation 70 PEFCs for automobiles. 71

However, there are few suitable in-situ spectroscopic techniques to 72 elucidate the structures and chemical states of cathode catalysts with 73

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S. Nagamatsu et al. / Surface Science xxx (2015) xxx-xxx

nanometer dimensions inside membrane electrode assembly (MEA) of 74 75PEFC under the operating conditions, where the wet, heterogeneous 76 and multiphasic reaction field involving catalyst nanoparticles, carbons, 77 ionomers, water and fuel gases prevents in situ measurements of the MEA cathode catalysts by spectroscopic techniques. X-ray absorption 78 79fine structure (XAFS) is an element-selective powerful tool for in-situ 80 investigation of the local coordination structures and oxidation 81 states of supported metal nanoparticle catalysts under the reaction 82 conditions [16-36]. In-situ potential-dependent XAFS analysis at Pt 83 Lui-edge can provide atomic-scale information on the structure and composition and their transformations of Pt surface layers in MEA 84 85 cathode catalysts [29–33]. In-situ time-resolved XAFS for PEFCs has recently been developed to elucidate the dynamic transformations of 86 87 structures and electronic states of Pt/C, Pt₃Co/C and Pt₃Ni/C cathode catalysts and to decide elementary steps and their rate constants for 88 the chemical events at the catalyst surfaces [20,28,30,34-36]. The 89 cathode degradation mechanism has also been studied to improve the 90 91 MEA durability [37–40]. To image the spatially-heterogeneous cathode degradation, recently, two-dimensional and three-dimensional micro 92and nano-scale observations of PEFCs [41-44] and functional materials 93 [45–51] using X-rays have also received much attention. 94

95In comparison to the constant potential hold process, Pt dissolution 96 in MEA Pt/C cathode catalysts is 3-4 times higher in voltage cycling, which is a major problem because automobiles require continual 97 repetition of the power-on/off processes with rapid changes in cell 98 voltages for driving cars [52]. Although the operation potential of PEFCs 99 is usually 0.6-1.0 V_{RHE} (potential vs. RHE), ramping of the potential to 100 101 open circuit voltage (OCV) around 1.0 V_{RHE} during the power-on/off drive, steep ramping to the higher potentials (>1.4V_{RHE}) by start-up 102and shut-down processes and possible undesirable ramping to the higher 103potentials (>1.1 V) at unidentified places of MEA cathode catalysts with 104 105ill-defined heterogeneous environments may promote the Pt leaching 106and detachment, resulting in degradation of catalyst performances.

Among a variety of approaches to reduce Pt loadings and to improve 107performance and durability of cathode catalysts, a promising approach 108 is deposition of thin Pt overlayers on less or non-noble metal substrates 109to make Pt surface-enriched nanoparticles called as skin, skeleton and 110 111 core-shell structures, where Pt can be sufficiently used and hence catalyst cost can be reduced [53–60]. Among examined metal core 112substrates (Ru, Ir, Rh, Au and Pd) for Pt monolayer shell, Pd showed 113 the highest oxygen reduction reaction (ORR) activity. However, the 114 115 Pt surface-specific ORR activity in the bimetallic systems has been demonstrated to depend on a variety of key issues, such as nanoparticle 116 size and shape (plane), surface roughness, compressive strain effect 117 (bond distance), downward Pt d-band center, ligand effect (electron 118 transfer), metal-oxygen bond strength, etc. [28,30,34] Although many in-119120 vestigations on electrochemical surface events have been performed for both model systems and practical PEFCs, the genesis and fundamental is-121 sues for the ORR activity and durability of Pt-based cathode catalysts such 122as the potential-dependent structures and electronic states of cathode 123 catalyst surfaces, the structural kinetics of the dynamic transformations 124125of catalysts in transient potential operating processes, and the kinetic 126 factors for the Pt dissolution are not clearly understood yet.

In this paper, we report the results of in-situ time-resolved Pt 127L_{III}-edge XAFS analysis for Pt/C, Pd(core)-Pt(1 ML shell)/C (denoted 128as Pd@Pt(1 ML)/C) and Pd(core)-Pt(2 ML shell)/C (denoted as Pd@ 129130Pt(2 ML)/C) cathode catalysts in PEFC MEAs during potential-step operating processes between 0.4 V_{RHE} and 1.4 V_{RHE} . Their performances 131 depend on the number of Pt shell layers, but there are controversy reports 132on the activity and the activity order for Pt monolayer shell, Pt two 133 monolayers, and Pt three monolayers compared to a standard Pt/C. It is 134still difficult to calculate and simulate theoretically the property and 135reactivity of the shell surfaces. Thus, direct spectroscopic characterization 136 of the core-shell cathode catalysts have been needed. The in-situ time-137 resolved XANES and EXAFS spectra revealed the transient response 138 139 transformations of structures and electronic states of the Pt/C, Pd@ Pt(1 ML)/C and Pd@Pt(2 ML)/C in the potential-step operations. 140 The in-situ XAFS analysis also revealed the potential-dependent 141 restructuring of the catalyst surfaces. The rate constants for the electronic 142 and structural transformations of the Pt/C, Pd@Pt(1 ML)/C and Pd@Pt(2 143 ML)/C in the potential-step operating processes were determined and compared in relation to the origin of the ORR activity and durability. 145

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2. Material and methods

2.1. Catalyst fabrications

The Pd core-Pt shell (1 ML and 2 ML) catalysts were prepared by a 148 similar method to that reported by Zhang et al. [57], where Pt skin layers 149 were deposited on Pd/C by the galvanic displacement of Cu monolayer 150 underpotentially deposited on Pd core surfaces by Pt. 151

2.1.1. Preparation of Pd@Pt(1 ML)/C (Pt 15wt%, Pd 25wt%) 152

0.1 M CuSO₄ was dissolved into 0.1 M H₂SO₄ (100 mL), to which 153 solution 1.00 g of 30wt% Pd/C purchased from N. E. CHEMCAT was 154 dispersed under stirring for 30 min. The solution was applied to current 155 until the whole quantity of 175 Coulomb(C) at 0.32 V vs RHE. The slurry 156 solution was filtered and the precipitates were washed with deionized 157 water 10 times to remove Cu^{2+} ions from the solution. The obtained 158 sample was added to 0.1 mM H₂PtCl₄ in H₂SO₄ and stirred for 1 h in a 159 N₂ atmosphere to replace the Cu monolayer by Pt monolayer. The slurry 160 solution was filtered and washed with deionized water 10 times. The 161 obtained catalyst was dried at 313 K for 1 day.

2.1.2. Preparation of Pd@Pt(2 ML)/C (Pt 20wt%, Pd 20wt%)

1.00 g of 25wt% Pd/C purchased from N. E. CHEMCAT was dispersed 164 into 100 mL of 0.1 M CuSO₄/0.1 M H₂SO₄ under stirring for 30 min. The 165 slurry solution was applied to current until the whole quantity of 100 C 166 at 0.32 V vs RHE and filtered in a N₂ atmosphere. The obtained precipi- 167 tates were washed with deionized water 10 times. The obtained sample 168 was added to 0.1 mM H₂PtCl₄ in H₂SO₄ and stirred for 1 h in a N₂ atmo- 169 sphere to replace the Cu monolayer by Pt monolayer. The slurry solution 170 was filtered and washed with deionized water 10 times. The obtained 171 sample was dispersed into 100 mL of 0.1 M CuSO₄/0.1 M H₂SO₄ under 172 stirring for 30 min. The slurry solution was applied to current until the 173 whole quantity of 150 C at 0.32 V vs RHE and filtered in a N_2 atmo- 174 sphere. The obtained precipitates were washed with deionized water 175 10 times. The obtained sample was added to 0.1 mM H₂PtCl₄ in H₂SO₄ 176 and stirred for 1 h in a N₂ atmosphere to replace the Cu monolayer by 177 Pt monolayer again. The slurry solution was filtered and washed with 178 deionized water 10 times. The obtained catalyst was dried at 413 K for 179 1 day. 180

2.2. Cyclic voltammetry measurements

The cyclic voltammograms (CVs) were recorded in H₂ (anode) 182 and N₂ (cathode) flows at 5 mV/s in the range 0.05–0.9 V at 353 K. 183 The electrochemical surface areas (ECSAs) of the cathode catalysts 184 were calculated by charge density of hydrogen adsorption on a Pt 185 surface (210 μ C/cm²_{-Pt}) in the hydrogen underpotential deposition 186 region (0.05 V– ca. 0.35 V of the onset of the double-layer region). 187

2.3. MEA and RDE operations

We used KB(Ketchenblack)-supported Ru (0.6 mg cm⁻²) anode 189 (Ru/C (43.1 wt%) to avoid interference against XAFS measurements 190 of Pt/C and Pd@Pt/C cathodes. Nafion NR-212 was used as electrolyte 191 membrane in MEAs. The electrode catalyst areas in the MEAs were 192 3×3 cm². The MEAs were conditioned by 150 aging cycles of galvanostatic current steps every 6 s from open circuit voltage (OCV) to a potential 194 near 0.3 V_{RHE} in H₂ and air operating atmospheres. CV measurements 195

Please cite this article as: S. Nagamatsu, et al., Structural and Electronic Transformations of Pt/C, Pd@Pt(1 ML)/C and Pd@Pt(2 ML)/C Cathode Catalysts in Polymer Electrolyte Fuel C..., Surf. Sci. (2015), http://dx.doi.org/10.1016/j.susc.2015.10.053

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