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Adsorption of thioether molecules on an alumina thin film

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ABSTRACT

Low-temperature scanning tunneling microscopy has been employed to study the adsorption of (bis(3-18 phenylthio)-phenyl)sulfane (BPPS) molecules on an aluminum-oxide film grown on NiAl(110). Large 19 variations in the molecular coverage on incompletely oxidized samples indicate substantial differences in the 20 binding strength of BPPS to metallic (NiAl) versus dielectric (alumina) surfaces. From atomically resolved 21 images, we obtain possible BPPS adsorption geometries on the oxide, in which the sulfur centers and not the 22 phenyl rings of the molecule govern the interaction. A local hexagonal ordering of BPPS, as deduced from pair 23 correlation functions, suggests a preferential binding of the BPPS sulfur atoms to Al ions with distorted pyramidal 24 coordination in the oxide surface. Our work provides insight into rarely explored binding schemes of organic 25 molecules on wide-gap oxide materials.

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32 1. Introduction

The interplay between organic molecules and metal surfaces has 33 been already in the focus of research for several decades [1,2]. This 34continued interest is driven by various applications of molecular-35 metal interfaces, e.g. for new electronic, light-emitting and photovoltaic 36 devices based on organic materials. The interaction of organic molecules 37 and non-metallic surfaces, on the other hand, has attracted much less 38 39 attention, although the technological relevance of this combination is equally large [3]. Interfaces between organic matter and oxide supports, 40 for example, are of crucial importance for liquid-solid solar cells, in 41which the optical excitation occurs in molecular sensitizers, while sepa-4243 ration and transport of the photo-generated carriers are realized by an oxide [4]. The most prominent example in this regard is the Grätzel 44 cell, consisting of a TiO₂ powder covered with Dye-molecules, which 45 46 has reached conversion efficiencies of 15% to date [5]. Molecule-oxide interfaces are of interest also in the widespread field of heterogeneous 47 catalysis, where oxides are typically used as cheap and robust support 48 49materials [6–8].

50 The significance of molecule–oxide interfaces is in contrast to the 51 limited research activities in the field. Despite a vital technological inter-52 est, fundamental studies on such systems are rare, a discrepancy that

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http://dx.doi.org/10.1016/j.susc.2014.05.021 0039-6028/© 2014 Elsevier B.V. All rights reserved. arises from several experimental difficulties. First of all, the binding be- 53 tween wide-gap insulators and common organic molecules is weak, as 54 the interaction arises mostly from van-der-Waals and Coulomb forces 55 and not from chemical bonds [9,10]. As a consequence, molecule-oxide 56 junctions exhibit high structural flexibility at room temperature and de- 57 fined interfaces only develop at cryogenic conditions. More relevant is 58 the poor electrical conductivity of most molecule-oxide systems, 59 which limits the applicability of electron-mediated spectroscopic and 60 microscopic techniques [11]. Especially, the use of scanning tunneling 61 microscopy (STM) for structural characterization at the nanometer 62 scale is challenging, as the tip easily perturbs the weakly bound adsor- 63 bates [12-15]. Closely related to the issue of finite conductivity are the 64 unusual imaging properties of molecules in the STM. In many cases, mo- 65 lecular orbitals contribute only weakly to the STM image contrast, as 66 they poorly overlap with the substrate electronic states and cannot 67 serve as initial or final state in a tunneling process. 68

In this study, we have overcome these restrictions by using an oxide 69 film, thin enough to maintain a finite conductivity, to support the organ-70 ic molecules. Our model system is alumina, an archetypical ionic insula-71 tor with 8 eV band gap, grown on a NiAl(110) substrate [16,17]. Our 72 probe molecule, a thioether compound consisting of four phenyl rings 73 linked by sulfur atoms, has recently attracted attention as it was found 74 to act as re-dispersion agent for metal nanoparticles [18,19]. On the 75 basis of atomically resolved STM images, we will determine the binding 76 behavior of the molecules on the oxide surface and analyze general 77 aspects of molecule–oxide interactions. 78

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79 2. Experiment

The experiments have been performed with a custom-built ultra-80 81 high vacuum STM operated at 4.5 K. The alumina films were prepared by oxidizing a sputtered and annealed NiAl(110) single crystal in 5 82 \times 10⁻⁶ mbar of oxygen at 550 K, followed by a vacuum-annealing 83 step at 1000 K [17]. The film quality was routinely checked with low-84 85 energy-electron-diffraction, exhibiting a complex yet sharp spot 86 pattern, and STM measurements revealing wide, atomically-flat oxide 87 terraces. The main properties of the NiAl-supported alumina films have been discussed in several papers before and shall only be sketched 88 at this point [20,21]. The film comprises four atomic layers; (i) an inter-89 facial Al_i plane consisting of atoms in pentagonal and heptagonal config-90 uration, (ii) a hexagonal O_i plane, (iii) an iso-structural Al_s plane and 91(iv) a terminating O_s plane made of triangular and square units 92 (Fig. 1a). While each interface Al_i ion has three O neighbors, the Al_s 93 coordination number varies between four and five due to ions in tetra-94 hedral and pyramidal sites [20]. The oxide lattice is composed of rectan-95gular unit cells (10.7 \times 17.9 Å²), being arranged in two reflection 96 domains rotated by $\pm 24^{\circ}$ against the lattice of the NiAl support [22]. 97 The oxide has a commensurate relationship with the substrate only 98 along the NiAl $[1\overline{10}]$ direction, while no commensurability exists along 99 100 the orthogonal NiAl[001]. In order to release misfit strain in the commensurate direction, a periodic network of dislocation lines is inserted 101 into the film, comprising both antiphase and reflection boundaries 102between equal and mirrored surface domains, respectively [23]. The 103 boundaries appear as protruding lines in empty-state STM images, 104 105as they possess a set of unoccupied defect states in the oxide band [24]. 106

The adsorption experiments were carried out with (bis(3-107phenylthio)-phenyl)sulfane (referred to as BPPS in the following), syn-108 109thesized in the group of Prof. Blechert at the Technical University Berlin [18,25]. The compound consists of four phenyl rings linked via three sul-110fur atoms in meta-positions. Due to facile rotation about the C-S-C axes, 111 a variety of configurations of the molecule can be realized upon adsorp-112 tion, e.g. the chain-like and folded conformers shown in Fig. 1b-d. The 113 molecules were purified by extensive degassing at 430 K and dosed 114 from an alumina crucible onto the fresh oxide film at 300 K. Immediate-115 ly after deposition, the samples were transferred into the cryogenic mi-116 croscope to avoid contamination. 117

3. Results and discussion

On incompletely oxidized surfaces, the BPPS molecules show a high 119 preference for binding to the metallic NiAl, while the coexisting oxide 120 patches remain essentially empty (Fig. 2a). This difference in sticking 121 indicates a much stronger BPPS adhesion onto metallic than dielectric 122 surfaces, reflecting the chemical inertness of the latter. The observed 123 variations in local coverage demonstrate also the high mobility of the 124 molecules that are always able to reach a nearby NiAl patch at 300 K. 125 By evaluating the mean distance between adjacent metal areas on 126 samples with different metal–oxide surface ratio, we have determined 127 the lower bound for the BPPS diffusion length on the oxide film to be 128 of the order of 500 nm [26]. A detailed discussion of thioether molecules 129 interacting with a NiAl surface can be found in the literature [18]. 130

On fully oxidized samples, a molecular fingerprint becomes detect- 131 able also on the alumina film (Fig. 2b). Atom-sized protrusions of ~0.5 132 Å height and 5–7 Å diameter are found on defect-free oxide domains 133 and, less abundant, along the domain boundaries (bright lines in 134 Fig. 2b). Already a crude image inspection reveals that the maxima are 135 not randomly distributed, but form a hexagonal pattern on the surface. 136 Typical distances are derived from pair-correlation functions, calculated 137 on the basis of STM images as shown in Figs. 2b. Three characteristic 138 separations between the maxima are revealed, namely (10 ± 1) , 139 (22 ± 5) and (45 ± 5) Å (Fig. 3). Also the connecting lines between 140 the protrusions have distinct orientations, following the diagonals of 141 the unit cells in the two oxide domains (see arrows in Fig. 2b for one do- 142 main). Note that molecules located directly on the domain boundaries 143 have been excluded from this statistics, as their positions might be 144 altered by the line defects. 145

The adsorption pattern observed on the alumina film seems 146 incompatible with the structure of the BPPS molecules at first glance. 147 Assuming that either the phenyl rings or the sulfur atoms become 148 visible in the STM, we would expect to find either four or three maxima 149 in linear, triangular or rhomboidal configuration to represent a single 150 molecule (Fig. 1b–d). In contrast, most protrusions appear in pairs 151 with either 10 or 22 Å distance, suggesting that only some of the molec-152 ular entities actually contribute to the contrast. In a possible scenario, 153 two sulfur atoms might be detected in the STM, while the third one re-154 mains invisible. A separation of 22 Å between adjacent maxima would 155 then correspond to the first and last sulfur atom in linear BPPS 156



Fig. 1. (a) Ball model of the Al_s and O_s planes of the alumina film grown on NiAl(110). The Al_s ions in distorted (regular) pyramidal configuration are highlighted by blue balls (zig-zag lines), as they dominate the image contrast seen in STM. (b-d) Three planar conformations of the (bis(3-phenylthio)-phenyl)sulfane molecule. Note that various other 2D and 3D conformers exist and our selection is not exhaustive. (For interpretation of the references to color in this figure legend, the reader is referred to the web version of this article.).

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