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Enhanced thermoelectric power factor and low thermal conductivity in onedimensional Te/Ag₂Te composites

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ARTICLE INFO Keywords: Thermoelectric Te nanorod Ag₂Te nanorod 1D nanostructure ABSTRACT We reports the synthesis and characterization of Te/Ag₂Te nanorod composites with various Ag₂Te contents. The composite samples were prepared by mixing Te and Ag₂Te nanorods synthesized by polyvinylpyrrolidone (PVP)-assisted solution-phase mixing. The thermoelectric properties of the prepared composites vary according to the Ag2Te content. Furthermore, the samples exhibit an enhancement in electrical conductivity and a reduction in the Seebeck coefficient with increasing Ag₂Te content. The maximum power factor (350.71 μ V/K at room temperature) is observed for the sample containing 30% Ag₂Te. The samples were assembled as 1D nanostructures, which led to a decrease in thermal conductivity owing to the strong phonon scattering effect. The maximum thermoelectric figure of merit (ZT) at room temperature, 0.34 was obtained for the sample with

1. Introduction

The rapid increase in the demand for energy in the past few decades requires the development of alternative energy sources. Some renewable energy conversion devices such as piezoelectric and thermoelectric (TE) devices, fuel cells, and solar cells have been developed. Among these alternative energy sources, TE materials are of particular interest, because they can convert heat into electricity, and vice versa. The performance of TE materials is determined by the dimensionless figure of merit $ZT = S^2σT/\kappa$, where σ, κ, S, and T are the electrical conductivity, thermal conductivity, Seebeck coefficient, and absolute temperature, respectively. So far considerable efforts have been made forward enhancing the figure of merit values in other existing TE materials classes, including tellurides $[1-3]$ $[1-3]$, half-Heuslers $[4]$, and silicides [\[5\]](#page--1-2). To achieve high ZT values, materials with a high power factor (PF = $S^2\sigma$) and low thermal conductivity are needed. A high power factor is generally achieved by enhancing the Seebeck coefficient or electrical conductivity of the material.

One simple way to achieve a high Power factor involves using materials with a high Seebeck coefficient. Tellurium (Te) [\[6,7\]](#page--1-3) and its alloys [\[8,9\]](#page--1-4) have attracted significant attention for this purpose. Te is a p-type narrow-band-gap semiconductor, with a direct band gap of approximately 0.35 eV at room temperature [\[10\]](#page--1-5) and a particularly high Seebeck coefficient ($\sim 400 \mu$ V/K at room temperature [\[11,12\]\)](#page--1-6). However, Te does not represent the best option for achieving a high power factor because of its low electrical conductivity $\left(\sim 10 \text{ S/m at}\right)$ room temperature [\[12\]](#page--1-7)) compared to other Te-based materials.

30% Ag2Te content. This value is 480% larger than that obtained for the pristine Te nanorod samples.

This limitation has prompted many studies aimed to determine methods of enhancing the electrical conductivity of Te [\[13\]](#page--1-8). One of the simplest ways to achieve this goal involves preparing composites with other materials $[14]$. Silver telluride (Ag₂Te) is an n-type narrow band gap semiconductor material with a high electrical conductivity and very high Seebeck coefficient [\[15,16\].](#page--1-10) Ag₂Te exists in two phases, α and β-Ag2Te. The α-Ag2Te phase, a face-centered cubic structure, is stable at temperatures below 473 K and exhibits better thermoelectric proper-ties than to the monoclinic β Ag₂Te phase [\[17\]](#page--1-11). The electrical conductivity of Ag₂Te (\sim 20,000 S/m at room temperature [\[15,16\]](#page--1-10)) is higher than that of Te. Furthermore, $Ag₂Te$ shows an outstanding Seebeck coefficient, which makes it a suitable material for enhancing the power factor of Te.

Another important requirement for achieving a high ZT is a low thermal conductivity. The unique performance and functionalities of one-dimensional (1D) nanostructures have attracted considerable interest [\[18,19\].](#page--1-12) The 1D nanostructure of a material can increase its lattice scattering of phonons [\[20\],](#page--1-13) reducing the thermal conductivity. Therefore, a low thermal conductivity can be achieved by nanostructuring a thermoelectric material. Introducing nanostructures in bulk thermoelectric materials generates a large number of boundaries that can scatter phonons [\[21\].](#page--1-14) Previous studies showed that Te-based thermoelectric materials with 1D nanostructure had lower thermal conductivity than their bulk counterparts [\[22,23\]](#page--1-15). This suggests that nanostructured Te and Ag2Te materials could exhibit low thermal

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conductivity, leading to high-performance thermoelectric composites. Yang et al. [\[24\]](#page--1-16) reported a reduction in the thermal conductivity of Ag2Te nanowires fabricated by solution-phase synthesis.

The present study describes the preparation of Te and $Ag₂Te$ nanorods and Te/Ag_2Te composites, with the final goal of obtaining materials with enhanced the thermoelectric properties. Te nanorods were fabricated from a Te precursor solution by a polyvinylpyrrolidone (PVP)-assisted solution-phase synthetic process. Ag2Te nanorods were also fabricated from the Te nanorods using a solution-phase synthetic route. After the synthesis of these materials, Te/Ag₂Te composites with various Ag2Te contents were fabricated by ball milling. Thermoelectric properties such as S, σ, к, the power factor and ZT were determined for each sample as a function of its $A\epsilon_2T$ e nanorod content. The combination of Te and Ag₂Te nanorods is expected to lead to enhanced thermoelectric properties in the composite materials.

2. Experimental

2.1. Materials

PVP [molecular weight $(MW) = \sim 40,000$] was purchased from Sigma Aldrich (St. Louis, USA). Tellurium (IV) oxide (TeO₂, 99%), ethylene glycol (EG, C₂H₆O₂, 99.5%), sodium hydroxide (NaOH, 98%), silver (I) nitrate (AgNO₃, 99.8%), and hydrazine monohydrate (N₂H₄ \cdot H2O, 80%) were purchased from Daejung Chemicals & Metals Co (Seoul, Korea).

2.2. Synthesis of Te nanorods

Te nanorods were prepared by mixing $0.96g$ of TeO₂ (MW = 159.6), 0.4g of PVP, 1.2g of NaOH (MW = 40) and 40 m l of ethylene glycol in a 250 ml round-bottom flask under magnetic stirring. This solution was heated to 120 °C, and then 2.45 ml of $N_2H_4 \cdot H_2O$ was injected in the solution. The color of the mixture became white within few seconds, indicating the formation of tellurium oxide colloids. The color of the tellurium oxide gradually turned to dark gray. After a reaction time of 90 min, a dark blue solution of Te nanorods was obtained. The resulting solution was poured into a deionized (DI) water solution containing 10 vol% $N_2H_4·H_2O$ and vigorously stirred. Subsequently, the resulting solution was centrifuged with the addition of volumetric water twice, and the supernatant was decanted. Finally, the resulting solution was dried in a vacuum oven at 60 °C for 24 h.

2.3. Synthesis of Ag_2Te nanorods

The Ag precursor solution was prepared by dissolving 0.05g of AgNO₃ and $0.1g$ of PVP in 30 ml EG. The collected Te nanorods $(0.02g)$ were redispersed in 100 ml EG. After the Ag precursor solution was poured into the EG, the reaction was carried out under magnetic stirring. The temperature was increased to 120 °C and maintained for 90 min. The resulting solution was poured into DI water, centrifuged and then dried in a vacuum oven at 60 °C for 24 h.

2.4. Fabrication of Te/Ag₂Te composite samples

Samples consisting of round disks of the Te/Ag_2Te composites were prepared by ball milling. The Te nanorods were ball-milled with various amounts of Ag₂Te nanorods using zirconia balls in an inert atmosphere for 2 h. The final product was loaded into a Fe mold and pressed at 200 °C under a pressure of 50 MPa for 10 min.

2.5. Sample characterization

The morphology and structural properties of the as-prepared products were analyzed by field-emission scanning electron microscopy (FE-SEM, SIGMA). Elemental mappings of the samples were analyzed by energy-dispersion X-ray diffraction (EDS, NORAN System 7, Thermo Scientific), X-ray diffraction (XRD, New D8 Advance, Bruker AXS) analyses and X-ray photoelectron spectroscopy (XPS, VG-Microtech, ESCA2000). X-ray diffraction analysis was performed at 40 mA, 40 kV, and a scan rate of $1^{\circ}/s$, with 2 θ ranging from 5 to 70 $^{\circ}$ using Cu Kα radiation ($\lambda = 0.154056$ nm). The electrical conductivity of the samples was measured using the 4-point-probe method and the thickness of the sample was measured using a digital micrometer. The Seebeck coefficient S was analyzed using a custom-made device containing a pair of thermocouples and voltmeters, according to the linear relationship between the thermal electromotive force (ΔV) and the through-plane temperature difference (ΔT) of the composite films $(S = \Delta V/\Delta T)$. The thermal conductivity of the samples was determined using the formula κ = $\alpha \rho C_p$, where α is the thermal diffusivity, $ρ$ is the bulk density, and C_p is the specific heat of the materials. The α values were measured using an LFA 447 NanoFlash (NETZSCH), whereas the C_p values were measured using differential scanning calorimetry (DSC) 131 Evo, SETARAM Instrumentation) under temperatures between −20 °C and 120 °C at a rate of 10 K/min under N₂ atmosphere.

3. Results and discussion

Te and Ag2Te nanorods were prepared following a solution-phase synthetic route. The overall scheme for the synthesis of the Te/Ag_2Te composite is shown in [Fig. 1](#page--1-17). TeO₂ and NaOH were mixed with PVP in EG. $N_2H_4 \cdot H_2O$, was injected into the solution after raising the temperature to 120 °C. In this reaction, PVP plays a key role as a surfactant. Previous studies have reported that linear polymers can react with inorganic ions to form a chain-shaped intermediate [\[25,26\].](#page--1-18) PVP induces 1D growth and controls the growth rate of Te and Ag2Te nanorods. Nucleation of Te^{2} ions took place at this stage. Te^{2} ions were reduced to elemental Te and solid crystal nuclei were formed. The reduction accelerated the growth of Te crystal nuclei into Te nanorods, and the Te²⁻ concentration in solution decreased. PVP also plays an important role in the Ag2Te synthesis, where it protects the Te nanorods so that Ag₂Te nanorods can be produced in the second step. The following reactions occur during the $Ag₂Te$ synthesis:

$$
2HO(CH_2)OH \rightarrow CH_3CHO + 2H_2O
$$
 (1)

$$
CH_3CHO + 2Ag^+ \rightarrow 2Ag + CH_3COCOCH_3
$$
 (2)

$$
2Ag + Te \rightarrow Ag_2Te \tag{3}
$$

The synthesis occurs in three stages: [\(1\)](#page-1-0) EG is dehydrated into acetaldehyde. [\(2\)](#page-1-1) The aldehyde group of the acetaldehyde reduces the $Ag⁺ ions of the Ag precursor to elemental Ag. (3) The newly formed Ag$ $Ag⁺ ions of the Ag precursor to elemental Ag. (3) The newly formed Ag$ $Ag⁺ ions of the Ag precursor to elemental Ag. (3) The newly formed Ag$ reacts with Te, producing Ag₂Te nanorods.

In order to confirm the formation of Te and Ag₂Te nanorods, XRD patterns were recorded. [Fig. 2](#page--1-19)a shows the XRD patterns of Te and Ag2Te nanorods. All peaks in the XRD pattern of the Te nanorods can be indexed to the hexagonal phase of Te (JCPDS no. 36-1452), whereas the XRD peaks observed for the Ag2Te nanorods, also shown in [Fig. 2a](#page--1-19), can be indexed to Ag2Te (JCPDS no. 34-0142). The absence of other peaks denotes the successful formation of pure Te and $Ag₂Te$.

XPS analysis was used to further confirm the successful synthesis of Te and Ag2Te samples. The XPS data for the Te sample are shown in [Fig. 2](#page--1-19)b and c. The two peaks at 572.5 and 582.9 eV correspond to the Te $3d_{5/2}$ and Te $3d_{3/2}$ states, respectively. The other two peaks at 586.4 and 576.1 eV can be assigned to the binding energy of Te(Ⅳ) 3d states indicating the oxidation of Te. This finding demonstrates that the outer surface of the synthesized Te sample is easily oxidized in air, because of its nanostructure; in fact, previous studies showed that nanomaterials are could be oxidized more easily in air [\[27\].](#page--1-20) The XPS spectra of the Ag2Te nanorods are shown is [Fig. 2d](#page--1-19)-f. The peaks at 368.3 and 572.5 eV correspond to Ag $3d_{5/2}$ and Te $3d_{5/2}$ states, suggesting that

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