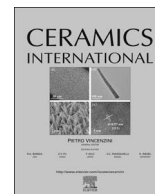




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Fully solution-processed metal oxide thin-film transistors via a low-temperature aqueous route

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ABSTRACT

We report a facile and low-temperature aqueous route for the fabrication of various oxide thin films (Al_2O_3 , In_2O_3 and InZnO). A detail study is carried out to reveal the formation and properties of these sol-gel-derived thin films. The results show that the water-based oxide thin films undergo the decomposition of nitrate group as well as conversion of metal hydroxides to form metal oxide framework. High quality oxide thin film could be achieved at low temperature by this aqueous route. Furthermore, these oxide thin films are integrated to form thin-film transistors (TFTs) and the electrical performance is systematically studied. In particular, we successfully demonstrate $\text{In}_2\text{O}_3/\text{Al}_2\text{O}_3$ TFTs with high mobility of $30.88 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ and low operation voltage of 4 V at a maximum processing temperature of 250 °C.

1. Introduction

Metal oxide semiconductors have attracted enormous attention in the area of active matrix devices such as liquid crystal displays (LCDs) and organic light-emitting diodes (OLEDs) because of their high mobility, good transparency to visible light, good uniformity and reasonable electrical stability [1–4]. Besides, the solution processes have been developed due to the possibility of low-cost and large-area fabrication without using vacuum deposition techniques [5–12].

Despite the substantial benefits and strong potential of the solution-based technique, the high-temperature annealing step is inevitable for oxidation and impurity removal in the solution process. Many researchers have attempted to develop low-temperature annealing processes for solution-processed oxide semiconductors. These approaches include sol-gel on chip [9], combustion process [8], photochemical activation [10], and annealing in an O_2/O_3 atmospheric environment [11]. Unfortunately, among most of the studies, toxic organic solvents such as 2-methoxyethanol (2-ME) were used, which will cause health risks and environmental hazards. Recently, a facile, novel and inexpensive ‘aqueous route’ has been proposed for the fabrication of In_2O_3 -based thin-film transistors (TFTs) at low annealing temperature ($< 250 \text{ °C}$) [12]. This aqueous route makes significant

contributions in terms of environmental protection and low temperature process [13–19].

On the other hand, most of these oxide TFTs were integrated with conventional SiO_2 dielectric and thus operated at high operating voltages ($> 10 \text{ V}$), which inhibit their applications in low power portable electronics. To overcome this bottleneck, various approaches have been explored to achieve large areal-capacitance dielectric [1–4,20]. Among these, the use of solution-processed high- k oxide dielectrics is the most attractive option as it enables the low leakage current, the low-voltage operation and the ease process integration with solution-processed oxide semiconductors. A few promising results on solution-processed oxide dielectrics have been reported, including Al_2O_3 , ZrO_2 , La_2O_3 and Y_2O_3 [20–37]. Among the various high- k dielectrics, Al_2O_3 is an excellent candidate, due to its abundance in nature, low cost, good chemical stability, relatively high dielectric constant, wide energy bandgap, smooth amorphous surface, and low interfacial trap density at the interface between the dielectric and active channel layers of oxide semiconductors [13,19,20,22,24,27–30]. However, the high annealing temperature ($> 350 \text{ °C}$) and multiple spin-coating processes (> 5 times) inhibit their use for low-cost and large-scale flexible electronics. Encouraged by the success of aqueous route on In_2O_3 -based oxide semiconductor, this technique may be

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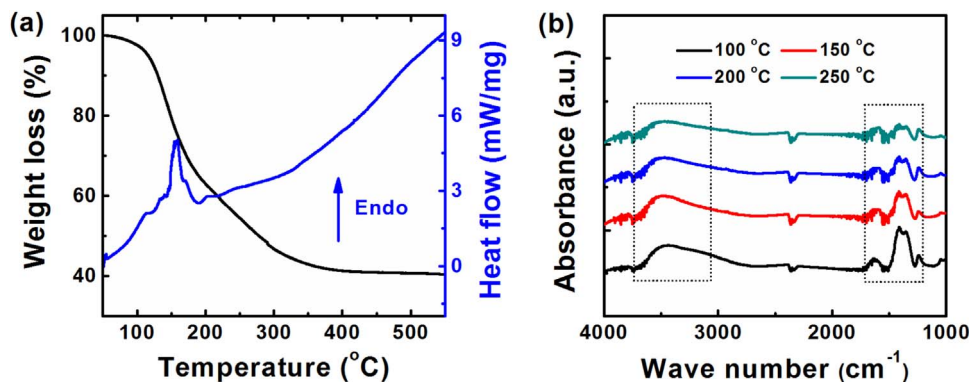


Fig. 1. (a) TG and DSC curves of Al_2O_3 precursor powder from 50 to 550 °C. (b) ATR-FTIR spectra of Al_2O_3 films annealed at indicated temperatures.

applicable to fabricate Al_2O_3 dielectrics at low temperature. Furthermore, the combination of aqueous-deposited high- k Al_2O_3 and In_2O_3 -based oxide semiconductor may achieve fully solution-processed low-temperature oxide TFTs.

In this article, we reported on the fabrication of Al_2O_3 , In_2O_3 and InZnO thin films using water as solvent by single-step spin-coated process. The formation and properties of aqueous solution-based oxide thin films were systematically investigated. Besides, the electrical performance of fully solution-processed oxide TFTs was studied in detail. In particular, the $\text{In}_2\text{O}_3/\text{Al}_2\text{O}_3$ TFTs annealed at 200 and 250 °C showed good mobilities of 2.04 and 30.88 $\text{cm}^2 \text{V}^{-1} \text{s}^{-1}$ at low operation voltage of 4 V.

2. Experimental section

2.1. Precursor preparation

All chemicals were purchased from Sigma-Aldrich and used as received. The Al_2O_3 precursor solution was prepared by dissolving about 2.5 M aluminum nitrate hydrate ($\text{Al}(\text{NO}_3)_3 \cdot x\text{H}_2\text{O}$) in water. For the In_2O_3 solution, 0.15 M indium nitrate hydrate ($\text{In}(\text{NO}_3)_3 \cdot x\text{H}_2\text{O}$) was dissolved in water. For the InZnO solution, 0.1 M indium nitrate hydrate ($\text{In}(\text{NO}_3)_3 \cdot x\text{H}_2\text{O}$) and 0.05 M zinc nitrate hydrate ($\text{Zn}(\text{NO}_3)_2 \cdot x\text{H}_2\text{O}$) were mixed with water. All the precursors are fully dissolved in water since the nitrate salts show very good solubility in water. The picture of the solutions could be seen in Fig. S1. The solutions were filtered through 0.45 μm polyethersulfone (PES) syringe filters before spin coating.

2.2. Devices fabrications

Heavily doped Si substrates were exposed to oxygen plasma for 10 min to enhance the hydrophilicity before spin coating. The Al_2O_3 precursor solution was spun onto the substrate at 4500 rpm for 40 s and annealed at a selected temperature (100, 150, 200 and 250 °C) under ambient atmosphere for 1 h. For the metal-insulator-metal (MIM) devices, an Al electrode (100 nm) was deposited on the dielectric layer by thermal evaporation. The area of the circular Al electrode was 0.03 mm^2 . To fabricate the TFTs with a bottom-gate top-contact configuration, the In_2O_3 and InZnO films were then spun onto the Al_2O_3 gate dielectric at 3000 rpm for 20 s and annealed at desired temperature (150, 200, and 250 °C) for 1 h to achieve thickness of about 10 nm. Subsequently, the Al source and drain top electrodes (100 nm) were deposited by thermal evaporation through the shadow mask. The channel width (W) and length (L) were 1500 and 100 μm , respectively. The large W/L ratio over 15 in this work could efficiently limit mobility overestimation as suggested by previous studies [20,28].

2.3. Film and device characterization

The thermal behavior of the precursor powders, which were dried at 105 °C for 12 h, were measured by thermogravimetric analyzer (PerkinElmer, TGA 6) and differential scanning calorimeter (PerkinElmer, DSC 7) at a heating rate of 10 °C/min from 50 to 550 °C. The crystallization and structural information of the sol-gel derived thin films were obtained using x-ray diffraction (XRD, Siemens D5005) with Cu K_α radiation. The chemical characteristics of the thin films were investigated by attenuated total reflectance Fourier transform infrared spectroscopy (ATR-FTIR, Bruker Tensor 27) and x-ray photoelectron spectroscopy (XPS, VG Scientific ESCALAB 250). The thicknesses of the thin films were measured via a variable-angle spectroscopic ellipsometer (SE, J. A. Woollam Co., Inc.). The surface morphologies of the solution-processed films were observed by atomic force microscopy (AFM, Veeco Dimension V). The frequency-dependent capacitance characteristics of the dielectrics were performed using a HP 4284 A in a frequency range of 20–1 MHz. The leakage of the oxide dielectric films and the electrical characteristics of the TFTs were measured with a precision semiconductor analyzer (Keithley 4200) in the dark at room temperature. Threshold voltage (V_{th}) was extracted from measurements in the saturation region by plotting $(I_{\text{DS}})^{1/2}$ vs. V_{GS} and extrapolating to $I_{\text{DS}}=0$ plots. The mobility (μ) and subthreshold swing (S) were calculated by the following formulas:

$$I_{\text{DS}} = \left(\frac{\mu C_i W}{2L} \right) (V_{\text{GS}} - V_{\text{th}})^2$$

$$S = \left(\frac{d(\log_{10} I_{\text{DS}})}{dV_{\text{GS}}} \right)^{-1}$$

where C_i , W and L are the capacitance of the gate dielectrics per unit area, channel width and channel length, respectively.

3. Results and discussion

3.1. Fabrication and characterization of Al_2O_3 film

Fig. 1(a) displays the TG-DSC of Al_2O_3 precursor powder from 50 to 550 °C. An endothermic peak accompanying a small weight loss was observed at 113 °C, which could be attributed to the evaporation of solvent and/or the hydrolysis of the metal precursor [23]. The peak at 160 °C indicated the dehydroxylation behavior of the hydrolyzed metal hydroxide, which reacted with adjacent metal hydroxide molecules and formed metal oxide lattices [14,23,30]. Another peak at 204 °C could be related to the decomposition of residual nitrate [14,23,30].

To investigate the formation of the solution-deposited Al_2O_3 , ATR-FTIR measurements were carried out, and the results are shown in Fig. 1(b). The broad peaks in the range of 3000–3500 cm^{-1} could be

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