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Large adiabatic temperature rise above the water ice point of a minor Fe substituted $Gd_{50}Co_{50}$ amorphous alloy

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ABSTRACT

In this study, we have successfully improved the Curie temperature of $Gd_{50}Co_{50}$ amorphous alloy from 267 K to 277 K by a minor Fe addition as a replacement for Co. The $Gd_{50}Co_{48}Fe_2$ as-spun ribbons exhibit typical characteristics of a soft magnetic amorphous alloy. The magnetic entropy change peak for this amorphous alloy was slightly decreased by the minor Fe addition, but the adiabatic temperature rise (ΔT_{ad}) of the $Gd_{50}Co_{48}Fe_2$ amorphous alloy is comparable to that of the $Gd_{50}Co_{50}$ amorphous alloy, both of which are larger than those of other metallic glasses near the ice point of water. The maximum ΔT_{ad} of the $Gd_{50}Co_{48}Fe_2$ amorphous ribbon at 277.5 K is about 1.44 K under 1 T, 2.44 K under 2 T, 3.31 K under 3 T, 4.1 K under 4 T and about 4.84 K under 5 T. The large maximum value of ΔT_{ad} above the ice point indicates that the $Gd_{50}Co_{48}Fe_2$ amorphous alloy could be an ideal candidate for the high efficient magnetic refrigerant in a household refrigerator.

1. Introduction

Magnetic refrigeration (MR) based on the magneto-caloric effect (MCE) of magnetic materials has attracted increasing interests because it is more compact, more effective, safes for the environment and has lower energy consumption than the traditional vapor-cycle refrigeration [1–5]. In the last two decades, MR working materials have been studied intensively and numerous MCE alloys have been developed [5–24]. Among these MCE materials, some intermetallic compounds such as $Gd_5(Si_2Ge_2)$, $MnFeP_{0.45}As_{0.55}$, $MnAs_{1-x}Sb_x$ and $Ni_{52.6}Mn_{23.1}Ga_{24.3}$, show a sharp magnetic entropy change ($-\Delta S_m$) peak due to their first order magnetic phase transition [5–9]; in contrast, metallic glasses and some crystalline alloys (*e.g.*, Gd, $Gd_6Co_2Si_3$ and so on) exhibit a broadened -ΔS_m peak because they undergo a second order magnetic phase transition [10–24]. Except for the relatively lower peak values of -ΔS_m ($-\Delta S_m^{peak}$), almost all the features of metallic glasses are superior to crystalline alloys: ultrahigh refrigeration capacity (*RC*), which is several times higher than that in crystalline alloys; tunable Curie temperature (T_c) without dramatic deterioration of MCE within a large compositional range; low hysteresis loss and low current eddy loss; excellent mechanical properties and good corrosion resistance [12–24]. It is therefore important to improve the -ΔS_m^{peak} values of amorphous MCE alloys, especially near room temperature.

Although some of the Gd-based metallic glasses exhibit higher -ΔS_m^{peak} and even much higher *RC* than those of the pure Gd, their T_c values are far from room temperature [15–20]. The -ΔS_m^{peak} values of Fe-based amorphous alloys, however, are not high enough for use as magnetic refrigerants even though they exhibit a T_c around room temperature [12–14]. Currently, we have prepared $Gd_{50}Co_{50}$ binary amorphous ribbons with excellent magneto-caloric properties near the freezing temperature of water [21]. This binary metallic glass exhibits a large -ΔS_m^{peak} and adiabatic temperature rise (ΔT_{ad}) peak at about 267 K. On the other hand, the $Gd_{48}Co_{52}$ binary amorphous ribbons exhibit a ΔT_{ad} peak comparable to that of $Gd_{50}Co_{50}$ metallic glass above the ice point of water, but is hard to be fabricated due to its poor glass forming ability (GFA) [22]. Although the T_c of binary $Gd_{50}Co_{50}$ amorphous alloy has been successfully improved to nearly 290 K by adding 5% (at. %) Fe as a replacement of Co in the binary glass forming alloy, the -ΔS_m^{peak} of the $Gd_{50}Co_{45}Fe_5$ decreased dramatically [23]. Considering the application of metallic glasses as magnetic refrigerants for a household refrigerator, it is more important to develop amorphous MCE alloys with a high -ΔS_m^{peak} value above the freezing temperature of water. In the present work, we add small amount of Fe as a replacement for Co in the $Gd_{50}Co_{50}$ binary amorphous alloy in an attempt to improve the T_c to the temperature to above the ice point of water, and at the same time keep the ΔT_{ad} of the $Gd_{50}Co_{48}Fe_2$ metallic glass comparable to the that of the $Gd_{50}Co_{50}$ binary amorphous alloy. The magnetic properties as well as the magneto-caloric behavior of the $Gd_{50}Co_{48}Fe_2$ amorphous alloy were studied in detail.

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2. Experimental procedure

A $Gd_{50}Co_{48}Fe_2$ ingot was prepared by arc-melting a mixture of Gd, Co and Fe metals with purities above 99.9% (at. %) and re-melting for least four times in a water cooled copper crucible under a Ti-gettered argon atmosphere. $Gd_{50}Co_{48}Fe_2$ as-spun ribbons were prepared by melt-spinning on a single copper wheel with a linear speed of about 30 m/s under a pure argon atmosphere. The amorphous structure of the $Gd_{50}Co_{48}Fe_2$ as-spun ribbon was ascertained by a Rigaku D\max-2550 X-ray diffractometer (XRD) using Cu K_α radiation. The differential scanning calorimetry (DSC) curve of the $Gd_{50}Co_{48}Fe_2$ amorphous ribbon was measured at a heating rate of 20 K/min using a Perkin-Elmer DIAMOND DSC under a purified argon atmosphere. The heat capacity and magnetic properties of the $Gd_{50}Co_{48}Fe_2$ amorphous ribbon were measured by a Physical Properties Measurement System (Quantum Design PPMS 6000).

3. Results and discussion

Fig. 1 shows the XRD pattern of the $Gd_{50}Co_{48}Fe_2$ and $Gd_{50}Co_{50}$ as-spun ribbons. The $Gd_{50}Co_{50}$ as-spun ribbon was also prepared at a surface speed of 30 m/s. The ribbons show the typical amorphous structures of a broadened hump without any sharp peaks of crystalline phases on the XRD patterns. The glass transition and crystallization behavior, as typical characteristics of amorphous alloys, are also found in the DSC trace of the as-spun $Gd_{50}Co_{48}Fe_2$ ribbon, as shown in the inset of Fig. 1. The $Gd_{50}Co_{48}Fe_2$ alloy can easily be fabricated in ribbon or wire shape, which can achieve a larger heat exchange efficiency and a lower eddy current loss than their bulk counterparts [21–22,25].

Fig. 2 shows the temperature dependence of zero field cooled (ZFC) and field cooled (FCC) magnetization (M-T) curves of the $Gd_{50}Co_{48}Fe_2$ amorphous ribbon under a field of 0.03 T, and the FCC M-T curve of the $Gd_{50}Co_{50}$ amorphous ribbon under the same magnetic field for comparison purposes. For the $Gd_{50}Co_{48}Fe_2$ amorphous alloy, the ZFC M-T curve is almost the same as the FCC M-T curve. The T_c of the $Gd_{50}Co_{48}Fe_2$ amorphous ribbon, marked clearly on the *M-T* curve, is about 277 K. Clearly, the T_c of the $Gd_{50}Co_{50}$ amorphous alloy can be enhanced by a small Fe addition, and in particular, the T_c of the $Gd_{50}Co_{48}Fe_2$ metallic glass is about 4 K higher than the ice point of water.

Fig. 3 shows the isothermal magnetization (*M-H*) curves of the $Gd_{50}Co_{48}Fe_2$ amorphous ribbon at different temperatures ranging from 150 K to 330 K. Therefore, the temperature dependence of $-\Delta S_m$ ($(-\Delta S_m)$-T) curves of the $Gd_{50}Co_{48}Fe_2$ amorphous ribbon under various magnetic fields can be derived from their *M-H* curves, and are shown in Fig. 4. The $-\Delta S_m^{peak}$ value of the $Gd_{50}Co_{48}Fe_2$ amorphous ribbon is about 1.32 $Jkg^{-1}\ K^{-1}$ under 1 T, 2.24 $Jkg^{-1}\ K^{-1}$ under 2 T, 3.04 $Jkg^{-1}\ K^{-1}$ under 3 T, 3.76 $Jkg^{-1}\ K^{-1}$ under 4 T and 4.44 $Jkg^{-1}\ K^{-1}$ under 5 T at 277.5 K. Compared to the Gd-Co-based amorphous alloys

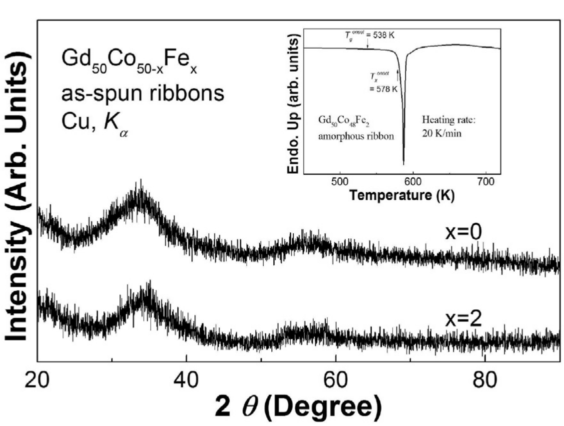

Fig. 1. XRD patterns of the $Gd_{50}Co_{48}Fe_2$ and $Gd_{50}Co_{50}$ as-spun ribbons.

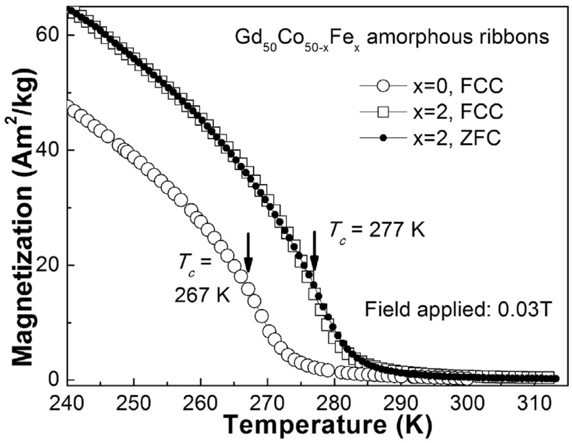

Fig. 2. The FCC and ZFC M-T curves for the $Gd_{50}Co_{48}Fe_2$ amorphous ribbon, and the FCC and ZFC M-T curve for the $Gd_{50}Co_{50}$ amorphous ribbon under a field of 0.03 T.

with a Gd concentration around 50% (at. %), the $-\Delta S_m^{peak}$ of the $Gd_{50}Co_{48}Fe_2$ amorphous ribbon is lower than those of the $Gd_{50}Co_{48}Zn_2$ amorphous ribbon (5.04 $Jkg^{-1}\ K^{-1}$ under 5 T at 260 K), the $Gd_{48}Co_{50}Zn_2$ amorphous ribbon (5.02 $Jkg^{-1}\ K^{-1}$ under 5 T at 262 K) [24], the $Gd_{50}Co_{48}Mn_2$ amorphous ribbon (5.24 $Jkg^{-1}\ K^{-1}$ under 5 T at 258 K, unpublished data), the $Gd_{50}Co_{45}Mn_5$ amorphous ribbon (5.49 $Jkg^{-1}\ K^{-1}$ under 5 T at 245 K, unpublished data) and the $Gd_{50}Co_{50}$ amorphous ribbon (4.6 $Jkg^{-1}\ K^{-1}$ under 5 T at 267 K) [21], but is higher than the $-\Delta S_m^{peak}$ of the $Gd_{48}Co_{52}$ amorphous ribbon (4.23 $Jkg^{-1}\ K^{-1}$ under 5 T at 280 K) and the $Gd_{50}Co_{45}Fe_5$ amorphous ribbon (3.8 $Jkg^{-1}\ K^{-1}$ under 5 T at 289.5 K) [22–23]. It can be noticed that the peak values of $-\Delta S_m$ of these Gd-Co-based amorphous alloys decrease obviously with the increase of their T_c. According to the relationship between $-\Delta S_m^{peak}$ and T_c proposed from the mean field theory [26], we constructed the $-\Delta S_m^{peak}$-$T_c^{-2/3}$ plots for the $Gd_{50}Co_{50}$, $Gd_{48}Co_{52}$, $Gd_{50}Co_{48}Zn_2$, $Gd_{48}Co_{50}Zn_2$, $Gd_{50}Co_{48}Mn_2$, $Gd_{50}Co_{45}Mn_5$, $Gd_{50}Co_{48}Fe_2$ and $Gd_{50}Co_{45}Fe_5$ amorphous ribbons, as shown in Fig. 5. The nearly linear fitting of the $-\Delta S_m^{peak}$-$T_c^{-2/3}$ plots (the dash line in Fig. 5) indicates that

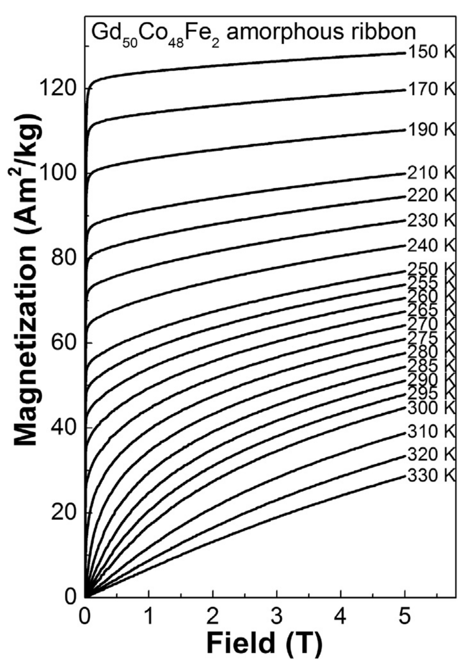

Fig. 3. The isothermal magnetization (*M-H*) curves of the $Gd_{50}Co_{48}Fe_2$ amorphous ribbon at different temperatures ranging from 150 K to 330 K.