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Orientation imaging of macro-sized polysilicon grains on wafers using spatially resolved acoustic spectroscopy $\stackrel{\sim}{\succ}$



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ABSTRACT

Due to its economical production process polysilicon, or multicrystalline silicon, is widely used to produce solar cell wafers. However, the conversion efficiencies are often lower than equivalent monocrystalline or thin film cells, with the structure and orientation of the silicon grains strongly linked to the efficiency. We present a non-destructive laser ultrasonic inspection technique, capable of characterising large ($52 \times 76 \text{ mm}^2$) photocell's microstructure – measurement times, sample surface preparation and system upgrades for silicon scanning are discussed. This system, known as spatially resolved acoustic spectroscopy (SRAS) could be used to optimise the polysilicon wafer production process and potentially improve efficiency.

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1. Introduction

Most manufactured crystalline silicon solar cells available can be categorised as either mono-crystalline (i.e. single crystal) or multicrystalline (mc-Si), also known as polysilicon. Whilst single crystal solar cells are usually more efficient, most processes commonly used in creating them (e.g. the Czochralski process) can be expensive, time consuming and energy intensive. It has been reported that polysilicon wafers account for around 65% of the solar cell market due to their economical production costs [1]. An important challenge for developers is in increasing polysilicon solar cell efficiency - currently, figures of > 21% have been achieved [2,3]. A well known method of improving the conversion efficiency involves controlling the microstructure of the polysilicon ingots during the directional solidification (DS) process [4-6]. High dislocation densities result in increased recombination rates (affecting the cell lifetime and efficiency), with nearby grain sizes a key factor determining the density [7,8]. Some studies also report on the dislocation densities in relation to the predominant orientation of nearby grains [6,9,10]. In order to optimise the quality of manufactured silicon cells, measuring the grain structure of the material becomes an integral part of the process control. Requirements of such a polysilicon measurement

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technique could include for it to be non-destructive, economical, safe for workshop environments, and free from measurement size restrictions.

Many methods exist to determine material grain sizes, including nanoscale measurements using scanning electron microscopy (SEM) [11] or X-ray based methods [12], and rapid optical imaging techniques [13,14]. However, only a limited number of inspection techniques exist capable of accurately determining material grain orientation. The most well known is characterisation using electron back scatter diffraction (EBSD) [15,16] - the major advantage of using EBSD is its resolution, both spatial (<10 nm) and angular $(<1^{\circ})$ [17]. However, there is a heavy expense associated with the system, operation and maintenance. Additionally, the measured samples are subject to extreme restrictions on size and surface finishing. Another well used method of grain characterisation is Xray diffraction (XRD), such as one based on Laue back scattered diffraction [18]. In work presented by Lehmann et al., a system was developed that determined grain boundaries optically, with a Laue scanner measuring a single point in each grain to determine the overall sample microstructure. The scan time was 5 h on a 156×156 mm² silicon wafer, with the sample size limited by the motorised translation stage.

In this paper, we present a laser ultrasonic inspection method capable of determining the grain orientation of silicon wafers. The presented method is known as spatially resolved acoustic spectroscopy, or SRAS [19], and has in the past been used to characterised the microstructure of metals and alloys [20,21]. SRAS is capable of



rapid scanning, and is an all-optical measurement system - we have adapted this technique to produce microstructure images on silicon.

2. Spatially resolved acoustic spectroscopy

SRAS is a non-contact, non-destructive evaluation (NDE) technique which probes the surface of materials using surface acoustic waves (SAW). The technique has been described in detail previously [19–21] but to-date it has not been used to image silicon - a brief system description is given in this section.

A surface acoustic wave is generated using a pulsed laser, patterned by an optical mask, and imaged onto the sample surface. The patterned absorption of laser energy through the thermo-elastic effect creates a surface wave, the frequency of which, f_s , is determined by the period (i.e. separation) of the grating fringes, or grating wavelength, λ_g , and the SAW velocity, v_s , which is dictated by the material properties, such that;

$$v_s = f_s \lambda_g \tag{1}$$

A second laser is used to measure the surface perturbation due to the propagating SAW - the frequency, f_s , is measured and the SAW velocity, v_s , is calculated. The frequency within the wave packet does not change once the wave begins to propagate - the frequency measured is determined by the surface properties *where the SAW is generated*, and can be detected anywhere along the propagation path, making this a very robust measurement as it does not rely on time of flight and is immune to acoustic aberrations [19]. Fig. 1 illustrates a simple schematic of the SRAS system.

In the current SRAS system, the generation laser is an AOT IR (1064 nm) Q-switched laser (AOT-YAG-10Q) with pulse widths of 1-2 ns and energies of 50–150 µl, with a repetition rate of 2 kHz. The detection laser is a Laser Quantum (532 nm) continuous wave laser (Torus 532) with an output of 200 mW, which is fibre coupled into the optical setup. The sample is scanned using x-y stages (Physik Instrumente motorised translation stages), and their travel dictates the output images' pixel size - the measurement scan area can be up to $300 \times 300 \text{ mm}^2$. For the work presented in this paper, the detection spot was focused down to $\tilde{\rho}8 \ \mu m$ and the generation patch was $\tilde{\beta}$ 200 μm on the sample. The grating wavelength, λ_{g} , used was 24 μm. An optical intensity return is also obtained from each scan point, producing a similar output to that of a scanning optical microscope. The scan speed is limited by the generation laser repetition rate, ADC unit acquisition speed, and translation stage speed - with measurement overheads, the scan speed is typically 1000-1500 points s depending on the detector used.

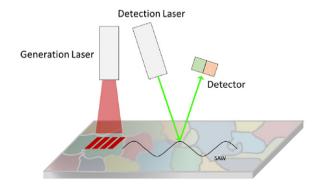


Fig. 1. A schematic of the SRAS operation – a surface acoustic wave is generated and detected using separate lasers, with the perturbation due to the wave picked up using a detector.

3. Finding the orientation

A material's crystallographic grain orientation, the material's elastic constants, and the velocity of a propagating acoustic wave on the material are all intimately related; knowledge of any two of these allows the determination of the third. Generally, the elastic constants for a material are obtained from the literature, or by complementary techniques, and the velocity is measured experimentally (here by the SRAS technique) - this information is sufficient to determine the grain orientation. The process of directly determining the orientation (the inverse problem) is difficult to solve, so fitting is used [20].

Briefly, the expected velocity for each possible orientation is calculated using the material's known elastic constants and density. This data is available in literature - for silicon, these are $C_{11} = 165$ GPa, $C_{12} = 64$ GPa, $C_{44} = 79.2$ GPa, and $\rho = 2.328$ g cm⁻³ [22]. The calculation takes into account the possible wave modes that could be present and applies weightings related to the instrument's ability to detect them (i.e. by the amount of out of plane motion) - this creates a database of dominant velocities for each orientation. The measured SAW velocity is then compared with this database of velocities using a cross correlation algorithm to find the orientation of best fit. This is repeated for each spatial position and the resultant orientation map (presented as an inverse pole figure) is output. The accuracy of the fitting process depends upon the number of directions in which the velocity is measured, the anisotropic ratio for the material, and the SNR of the velocity measurements.

4. Methodology and results

The SRAS instrument in its current form was designed for use on predominantly aerospace materials. The optical wavelengths of the generation and detection lasers are not suitable for direct measurements on silicon. Additionally, the detection system requires an optically smooth surface finish (this is a limitation of the available machine only, and not the SRAS technique in general).

The silicon cells used in this study were commercially available polysilicon panels and required some preparation. The panels were disassembled, the rear screen print coating was removed, and the samples manually polished down to reveal the silicon layer and to give a smooth surface for inspection. For compatibility with the lasers, a thin layer of chromium (~150 nm) was sputter coated on top of the polished silicon. The layer thickness is only a small fraction of the sensing depth of the measurement, which is approximately one acoustic wavelength ($\lambda_a = 24 \,\mu$ m), and therefore will have a negligible impact on the velocity measurement.

SRAS orientation maps of two sets of silicon cells are presented in this paper. The first (i) is of a fragmented silicon cell, mounted to fit on a $\tilde{\wp}2.54$ mm EBSD sample holder. The second (ii) is of two larger 52mm × 38 mm cells. The intention behind presenting the scan data from these samples is to a) compare results obtained using SRAS against an established orientation characterisation technique and to b) display SRAS's ability to scan a large area of material effectively.

The acoustic wavelength used was 24 μ m and the samples were scanned with steps of 50 μ m and 20 μ m along the x-axis and y-axis respectively, creating a 480 × 1200 scan of sample (i), and a 1100 × 4000 scan of sample (ii). To create the orientation maps, shown in Fig. 2 (a) for sample (i) and Fig. 3 for sample (ii), data was taken at each point with the SAW propagating in 9 directions - the time taken to scan each direction was 14 min for sample (i) and 32 min for sample (ii). Both orientation maps show identifiable and contrasting grain shapes and orientations.

To evaluate the measurements collected by SRAS on silicon, a section of sample (i) (highlighted in Fig. 2 (a)) had been analysed using an EBSD instrument (JEOL 7100F FEG-SEM). For comparison, the image generated using data collected by SRAS in this section of sample (i) is shown in Fig. 2 (b) with the EBSD image obtained shown

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